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# A cascade energy band structure enhances the carrier energy in organic vertical-type triodes

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#### 46 1. Introduction

The study of organic thin film transistors (OTFTs) has 47 attracted much interest for their potential use in high-48 value, low-cost electronics, such as displays, sensors, radio 49 frequency identifications, and e-papers [1-4]. Recently, or-50 ganic semiconductors with mobilities comparable with 51 52 that of amorphous silicon have been realized [5-8]; nevertheless, OTFTs possessing the typical metal-oxide-53 semiconductor field effect transistor structure exhibit low 54 55 output currents and low frequencies of operation because 56 of their high resistivities and low carrier mobilities, which 57 restrict their practical applications. Although shortening

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ABSTRACT

Organic vertical-type triodes (OVTs) based on the cascade energy band structure as emitter 32 layer are studied. The electric characteristics were dramatically enhanced while incorpo-33 rating the cascade energy under current driving and voltage driving modes. The improve-34 35 ment is attributed to that injection carriers can obtain higher energy through a stepwise energy level. When the device has a layered structure of F<sub>16</sub>CuPC (10 nm)/PTCDI 36 37 (10 nm)/pentacene (100 nm) in emitter, it exhibits a common-base transport factor of 0.99 and a common-emitter current gain of 225 under current driving mode and exhibits 38 a high current modulation-exceeding  $-520 \,\mu$ A for a low collector voltage of  $-5 \,V$  and a 39 base voltage of -5 V and the current on/off ratio of  $10^3$  under voltage driving mode. Fur-40 thermore, we realized first organic current mirror that exhibited out/in current ratio of 41 0.75 and output resistance of  $10^5 \Omega$  by using the OVTs. 42

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the channel length can further improve the device performance, this approach requires that state-of-art lithographic techniques, which are not practical for fabricating low-cost flexible electronics, be performed prior to organic semiconductor growth; in addition, the contact resistance between the source/drain electrodes and the organic semiconductor will dominate the device performance [9,10]. To realize the incorporation of OTFTs in most proposed applications, it will be necessary to improve not only the electrical properties of the organic materials but also the device structures. Organic vertical-type transistors are a promising technological option for improving device performance [11–16] because their channel lengths can be controlled precisely by varying the thickness of the active layer.

In the early days of inorganic semiconductor development, bipolar-junction transistors (BJTs)—the most important inorganic vertical-type transistors—were formed using n–p–n or p–n–p double homojunctions [17]. Due to their 75

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larger minority-carrier diffusion and base resistance, the 76 77 operation speeds of BJTs are limited by the transition time 78 [18]. To overcome this obstacle, unipolar hot-carrier transis-79 tors (HCTs) were developed, featuring a thin metallic base 80 sandwiched between two semiconductors to serve as an 81 emitter and a collector. If the depletion region at the metal-82 semiconductor junction of a Schottky diode is thinner than 83 the junction capacitance of a pn diode, it results the lower 84 resistance. To further improve their performance, cascade 85 energy band structures, formed by the different element con-86 centration ratios, have been incorporated in HCTs. Such 87 structures enhance the carrier energy in emitter-base (EB) diodes; therefore, the carriers can be inhibited from contrib-88 89 uting to the recombination current at the base electrode and effectively approach the collector electrode to increase the 90 91 output current. To control the element concentration ratio accurately, the devices must be grown using expensive fabri-92 93 cation processes (e.g., molecular beam epitaxy, metal organic 94 chemical vapor deposition). In addition, the carrier transport 95 mechanism in such devices is based on band-type conduction and the carriers are affected by the acoustic phonons. 96 97 Therefore, it is difficult to operate these devices at room 98 temperature [19,20]. In contrast, organic thin films can be 99 fabricated using simple processing techniques. In their corre-100 sponding devices, carrier transport occurs through tunneling between the localized states caused by defects and/or disor-101 der states, with a conduction theory associated with phonon-102 activated hopping. Therefore, organic vertical transistors are 103 104 readily operated at room-temperature and can be applied in low cost, flexible electronics [21,15,22]. Due to lower carrier 105 mobility of the materials and the shorter mean free path in 106 the base layer, the most of the carriers are recombined at 107 the base electrode. Which lead to a lower common-emitter 108 109 current gain and lack the apparent saturation region. Several approaches have been developed to improve these draw-110 111 backs (e.g.,) by inserting a hole injection enhancement layer (LiF or Al<sub>2</sub>O<sub>3</sub>) at the emitter-base junction or using a metal-112 113 grid base electrode [23-25]. When inserting a hole injection 114 layer the tunneling barrier are formed, as a result of the emit-115 ter current being dominated by the tunneling current, with 116 exhibiting a saturation region and enhanced current gain.

In this paper, we describe organic vertical triodes 117 118 (OVTs) incorporating a cascade-type energy band structure 119 that operate at pronounced saturation with high gain. The 120 most injection carriers can obtain higher energy through a stepwise energy level, more carriers can surmount the thin 121 122 metal base electrode and diffuse into the collector layer. 123 Therefore, this device exhibits a larger transport factor and a current gain when operated in the current-driving 124 mode. On the other hand, the device displays a larger cur-125 126 rent on/off ratio and a smaller offset voltage when operated in the voltage-driving mode. The device exhibits a 127 128 sufficiently large current gain; a current mirror operated 129 at a greater out/in current ratio  $(I_{out}/I_{in})$  and a greater output resistance  $(r_0)$  is achieved by integrating two p-channel 130 OVTs with a load resistor. 131

#### 132 2. Experimental

Prior to deposition, the glass substrates were cleaned sequentially with detergent, acetone, and isopropyl alcohol followed by treating in an ultraviolet (UV) ozone cleaner 135 136 for 15 min. The gold (Au) (30 nm) layer was deposited on the glass substrate to serve as a collector electrode. The 137 copper phthalocyanine (CuPC) (50 nm) layer (Lumines-138 cence Technology) was thermally deposited on the Au 139 layer to smooth the surface morphology and then the 140 pentacene (270 nm) layer (Luminescence Technology), 141 serving as the collector of the p-channel triode, was ther-142 mally evaporated. The aluminum (Al) (10 nm) strip was 143 thermally evaporated onto the pentacene layer to function 144 as the base electrode. Next, a thin Al film (15 nm), operat-145 ing as the base electrode, was deposited on the strip. To 146 study the effects of the materials with different highest 147 occupied molecular orbital (HOMO) energy levels on the 148 device performance, different organic semiconductor lay-149 ers were thermally evaporated onto the thin Al film as 150 shown in Table 1. Finally, films of molybdenum(VI) oxide 151  $(MoO_3)$  (30 nm) and Al (30 nm) were deposited onto the 152 emitter layers of the five OVTs mentioned above to func-153 tion as emitter electrodes. This process was performed 154 with patterning through a metal mask. All organic materi-155 als and metal electrodes were deposited in a thermal evap-156 oration chamber at a base pressure of 10<sup>-6</sup> torr. The active 157 area of the device  $(0.04 \text{ cm}^2)$  was defined by the intersec-158 tion of the emitter and collector electrodes. The current-159 voltage (I–V) characteristics of the devices were measured 160 using a Keithley 4200 semiconductor parameter analyzer. 161 All of the electrical characteristics of these devices were 162 measured in dark environments. 163

#### 3. Results and discussion

Fig. 1a and b presents the device configuration and the 165 energy diagram of the emitter-base diodes, respectively 166 [26–29]. The device, fabricated from two Schottky diodes, 167 168 comprises a collector-base diode as the bottom diode and an emitter-base diode as the top diode. To form the 169 Schottky diode, the active layer was sandwiched between 170 Au and Al. To reduce the leakage current and the energy 171 barrier between the metal and the organics, a CuPC film 172 was used as the buffer layer in the collector-base diodes 173 and MoO<sub>3</sub> was inserted as the carrier injection layer in 174 the emitter-base diodes [30-33]. Several organic semicon-175 ductors-pentacene, *N,N'*-bis(naphth-1-yl)-*N,N'*-bis(phe-176 N,N8-dioctyl-3,4,9,10-perylene 177 nyl)-benzidine(NPB), tetracarboxylic diimide(PTCDI), and copper(II) 1,2,3,4,8, 178 9,10,11,15,16,17,18,22,23,24,25-hexadecafluoro-29H,31H-179 phthalocyanine (F<sub>16</sub>CuPC)-with different energy levels 180 were employed to form the cascade-type structure in the 181 emitter-base diodes. To investigate the influence of the 182 cascade-type energy structure in the emitter-base diodes, 183 we fabricated five different devices by incorporating the 184 various materials in the emitter-base diodes as shown in 185 Table 1. All five different devices configuration were fabri-186 cated and characterized with same material in the collec-187 tor-base diodes ((Au (30 nm)/CuPC (50 nm)/pentacene 188 (270 nm)/Al (30 nm)/thin Al layer (10 nm)). Fig. 2a and c 189 displays the common-base (CB) electrical characteristics 190 of devices A and E, respectively; Fig. 2b and d present the 191 common-emitter (CE) electrical characteristics of devices 192

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