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# Controllable growth and characterizations of hybrid spiral-like atomically thin molybdenum disulfide

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## ABSTRACT

Monolayer MoS<sub>2</sub> is an emerging two-dimensional semiconductor with wide-ranging potential applications in novel electronic and optoelectronic devices. Here, we reported controlled vapor phase growth of hybrid spiral-like MoS<sub>2</sub> crystals investigated by multiple means of X-Ray photoemission spectroscopy, scanning electron microscopy, atomic force microscopy, kelvin probe force microscopy, Raman and Photoluminescence techniques. Morphological characterizations reveal an intriguing hybrid spiral-like MoS<sub>2</sub> feature whose lower planes are AB Bernal stacking and upper structure is spiral. We ascribe the hybrid spiral-like structure to a screw dislocation drive growth mechanism owing to lower supersaturation and layer-by-layer growth mode. In addition, the electrostatic properties of MoS<sub>2</sub> microflakes with hybrid spiral structures are obvious inhomogeneous and dependent on morphology manifested by kelvin probe force microscopy. Our work deepens the understanding of growth mechanisms of CVD-grown MoS<sub>2</sub>, which is also adoptable to other TMDC materials.

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## 1. Introduction

Following the boom of graphene, two-dimensional (2D) transition metal dichalcogenides materials (TMDCs) have emerged as an important layered crystals, exhibiting intrinsic semiconducting, metallic and even superconducting behaviors [1–3]. Generally, TMDCs have the formula MX<sub>2</sub>, where M represents a transition metal from groups IV–VI (e.g. Mo, W, Ti, or Nb), while X is a chalcogen atom (S, Se, or Te). More than 40 types of TMDCs can occur depending on the combination of chalcogen and metal atoms [4–6]. Among the variety of TMDCs being investigated, MoS<sub>2</sub> is a representative one with good chemical stability, excellent optical and electronic properties. Monolayer MoS<sub>2</sub> has been fabricated to many prototype devices with advanced functionalities, such as field-effect transistors (FETs) and photo detectors [7,8]. Photoelectric devices fabricated from monolayer MoS<sub>2</sub> have recently been shown to have excellent photoresponsivity up to 880 A/W at a wavelength of 561 nm, which are superior to graphene-based devices with a similar geometry [8,9].

For realizing the high-performance of MoS<sub>2</sub>-based devices, exploring an approach to high-crystalline MoS<sub>2</sub> with atomically

flat surface and interfaces is necessary. The chemical vapor deposition (CVD) method, a powerful way to large-scale graphene, allows the growth of single-crystalline MoS<sub>2</sub> microflakes directly on arbitrary substrates [10–13]. Although CVD growth of MoS<sub>2</sub> has been well developed, controlled synthesis of MoS<sub>2</sub> with different morphologies and stacking styles is still not achieved and the mechanism is not clear absolutely. For example, Y. J. Cao et al. reported that the dendritic triangle shape MoS<sub>2</sub> microflakes can be synthesized by controlling the carrier gas flow rate [14]. Recently, spiral-like WSe<sub>2</sub> was synthesized by L. Chen et al. and the transistors based on it show the room-temperature hole mobility over 44 cm<sup>2</sup> v<sup>-1</sup> s<sup>-1</sup> and a large current on/off ratio of 1 × 10<sup>6</sup> [15]. The controlled growth and characterizations of spiral-like MoS<sub>2</sub> structures are essential to realize the applications in electronic and optoelectronic devices.

In this contribution, we controlled synthesized novel hybrid spiral-like MoS<sub>2</sub> crystals through increasing the growth temperature to 1000 °C. The combined atomic force microscopy and scanning electron microscopy were used to investigate the hybrid spiral structure with clear steps and helical fringes. The underlying growth mechanism is proposed to be combined layer-by-layer growth and screw dislocation drive mode. Moreover, electrostatic properties of MoS<sub>2</sub> with hybrid spiral structures have been shown to exhibit distinctive surface potential and charge distributions,

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which may have important implications on the applications of MoS<sub>2</sub>-based devices.

## 2. Materials and methods

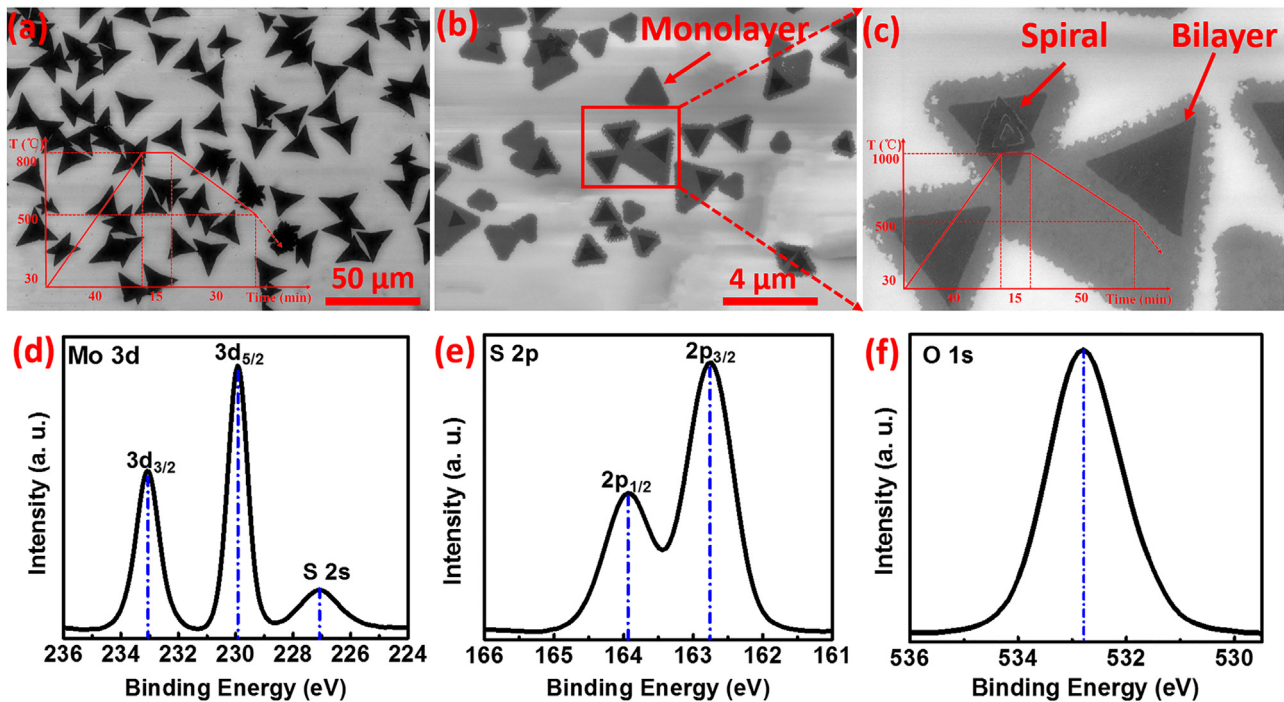
The experiments were carried out in a commercial CVD furnace (KJMTI Corporation, OTF-1200) with a 3 in.-diameter quartz tube and double heating zones in the length of 20 cm separated at border to border distance of 5 cm. The growth temperature and carrier gas flow rate were controlled with a proportion integration differentiation (PID) temperature controller and a mass flow controller, respectively. A quartz boat loaded MoO<sub>3</sub> powders (4N purity, Aladdin) was placed at the 2<sup>nd</sup> heating zone while another loaded sulfur powders (5N purity, Aladdin) was placed at the center of the 1st heating zone of the furnace and at ~15 cm away at the upstream side. A third quartz boat with clean SiO<sub>2</sub> (300 nm thick)/Si substrates in 1 × cm<sup>2</sup> size was placed in the 2nd heating zone at the downstream side. The SiO<sub>2</sub>/Si substrates were initially cleaned in DI water, acetone and isopropanol, following 3 h bath in H<sub>2</sub>SO<sub>4</sub>/H<sub>2</sub>O<sub>2</sub> (volume ratio: 3:1) and then 5 min air plasma spurring to remove the possible organic contaminations. Upon roughly pumping, nitrogen (N<sub>2</sub>, 5N purity) gas flow was introduced to purge the quartz tube at 150 sccm (standard cubic centimeters per minute) for 20 min. Then, the 2<sup>nd</sup> heating zone was rapidly heated up to 800 or 1000 °C and maintained for 15 min with a decreased N<sub>2</sub> gas flow of 100 sccm. Sulfur powders were maintained at the temperature being melting point of ~120 °C during the reaction process, so that sulfur vapor could be slowly carried by N<sub>2</sub> gas flow to the reaction zone. The system was maintained at pressure of 1 kPa during the entire growth run. Subsequently, the furnace was switched off following the growth process for cooling down to room temperature undisturbedly with N<sub>2</sub> gas flow.

The as-synthesized sample was systematically characterized by

Scanning Electron Microscopy (SEM, FEI Helios 600i), X-Ray Photoemission Spectroscopy (XPS, thermo fisher ESCALAB 250xi), Atomic Force Microscopy and Kelvin Probe Force Microscopy (AFM/KPFM, Agilent 5500) and confocal Raman and photoluminescence techniques (Raman/PL, WITec alpha 300R). The XPS investigations were performed using monochromatic Aluminium K<sub>α</sub> X-Ray with light spot size of approximately 1 mm<sup>2</sup>. The binding energy was calibrated by assigning the corresponding C 1s peak at 284.5 eV. The topography and corresponding height profile of sample were characterized using AFM in tapping mode. In our experiment, we used a platinum-iridium (Pt/Ir)-coated silicon cantilever (SCM-PIT from Bruker, USA) in tapping mode to perform KPFM measurements. The topography and surface potential signals of sample were simultaneously measured with a mechanical drive frequency of ~75 KHz and an Ac modulation of 0.2–0.3 V at ~10 KHz. Raman and PL spectra/mapping, were carried out under a 532.0 nm laser light and Si-based CCD detector at ambient conditions. The laser power was maintained below 1 mW to avoid local heating and oxidation to obtain a satisfactory signal-to-noise ratio while maintaining acceptable data acquisition duration and avoiding drift. The 520.0 cm<sup>-1</sup> phonon mode from the silicon substrate was used for calibration of the Raman shift. The emitted signal was collected by a Zeiss 100 × objective (N.A=0.9) and dispersed by a 1800 lines/mm grating for Raman measurements and a 600 lines/mm grating for PL measurements.

## 3. Results and Discussion

The SEM image of sample 1 (Fig. 1a) shows many equilateral and merged triangle shaped crystals in lateral size of ~30 μm which are assigned to single-crystalline MoS<sub>2</sub> microflakes. The inserted plot of temperature vs time in Fig. 1a reveals that the first sample is synthesized at accustomed growth temperature of 800 °C and no additional layer is observed, in consistent with



**Fig. 1.** (a) The SEM image of monolayer MoS<sub>2</sub> crystals and the corresponding plot of temperature vs time for sample growth (Inset). (b) The representative SEM image exhibiting monolayer and stacked few-layer MoS<sub>2</sub> microflakes. (c) The zoomed-in high magnification SEM image showing hybrid spiral-like and stacking bilayer MoS<sub>2</sub> crystals. The corresponding plot of temperature vs time for sample growth is inserted. (d-f) The corresponding XPS narrow scans of Mo 3d, S 2p and O 1s core level spectra for hybrid spiral-like sample.

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