



# Properties prediction of ultra high performance concrete using blended cement hydration model



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## HIGHLIGHTS

- Proposed a hydration model of ultra high performance concrete (UHPC).
- Considered both cement hydration and silica fume reaction.
- Predicted degree hydration of cement, CH contents and mechanical properties.
- Proposed model is valid for both ordinary strength concrete and UHPC.

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## ABSTRACT

Ultra high performance concrete (UHPC) consists of cement, silica fume (SF), sand, fibers, water and superplasticizer. Typical water/binder-ratios are 0.15–0.20 with 20–30% of silica fume. The development of properties of hardening UHPC relates with both hydration of cement and pozzolanic reaction of silica fume. In this paper, by considering the production of calcium hydroxide in cement hydration and its consumption in the pozzolanic reaction, a numerical model is proposed to simulate the hydration of UHPC. The degree of hydration of cement and degree of reaction of silica fume are obtained as accompanied results from the proposed hydration model. The properties of hardening UHPC, such as degree of hydration of cement, calcium hydroxide contents, and compressive strength, are predicted from the contribution of cement hydration and pozzolanic reaction. The proposed model is verified through experimental data on concrete with different water-to-binder ratios and silica fume substitution ratios.

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## 1. Introduction

Concrete or cementitious composites with compressive strength over 150 MPa are generally described as ultra-high performance concrete (UHPC); if fibers are added in order to decrease brittleness and increase energy absorption capacity the term ultra-high performance fiber reinforced concrete (UHP-FRC) is used. UHPC's high compressive strength, obtained through dense particle packing, implies high durability, improved resistance against freeze–thaw cycles and various chemicals as well as higher penetration resistance [1,2].

Many experimental studies have been done on the development of properties and proportioning mixtures of UHPC. Morin et al. [3] found that for reactive powder concrete (RPC), a long dormant period of about 30 h after water addition which was attributed to the high amount of superplasticizer. Then, the hydration started and

was characterized by a strong heat release, which lasted for about 12 h. Bentz et al. [4] and Chung [5] reported that silica fume (SF) fulfils some functions in UHPC: it fills voids between cement grains, it enhances the rheological characteristics and it forms hydration products by pozzolanic activity. Another important effect of the silica fume is the improvement of the interfacial transition zones between binder and steel fibers. Thus, the mechanical strengths are increased and microstructure and packing efficiency of the UHPC are enhanced. Loukili et al. [6] investigated the hydration kinetics, change of relative humidity, and autogenous shrinkage of ultra-high-strength concrete. They found that the degree of hydration of cement is only 0.58 at 28 days due to the lower water to binder ratio in ultra high strength concrete, and the autogenous shrinkage of ultra high strength concrete were much higher than ordinary strength concrete because of high self-desiccation at an early age due to both a very low w/c ratio and a high silica fume content.

Besides experimental investigations on chemical and physical properties of UHPC [3–6], there are some theoretical models for

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## Nomenclature

$\alpha$	degree of cement hydration	CH	calcium hydroxide
$\nu$	stoichiometric ratio by mass of water to cement	$\alpha_{silica}$	degree of reaction of the glass (active) phase of silica fume
$w_g$	physically bound water in C–S–H gel	$m_{silica0}$	silica fume mass in mixing proportion
$\rho_w$	density of water	$\gamma_s$	mass percentage of glass silica in silica fume
$S$	Blaine surface area	$W_{cap}$ and $W_{chem}$	masses of capillary water and chemically bound water, respectively
$\rho_c$	density of the cement	$m_{CH}(t)$	mass of calcium hydroxide in hydrating cement–silica fume blends
$r_0$	radius of unhydrated cement particles	$\nu_{si}$	stoichiometry ratio by mass of CH to silica fume
$S_w$	effective surface area of the cement particles in contact with water	$r_{si0}$	radius of silica fume particle
$S_0$	total surface area if the surface area develops unconstrained	$D_{esi}$	reaction rate coefficient of silica fume in the diffusion period
$B$	the parameter controls the rate of the initial shell formation	$D_{esi0}$	initial value of $D_{esi}$
$C$	the parameter controls the rate of the initial shell decay	$k_{rsi}$	reaction rate coefficient of silica fume during the phase boundary reaction period
$k_d$	cement hydration rate coefficient in the initial dormant period	$D_{esi20}$ and $k_{rsi20}$	values of $D_{esi0}$ , and $k_{rsi}$ at 293 K, respectively
$k_r$	cement hydration rate coefficient in the phase boundary reaction period	$\beta_{esi}$ and $E_{sil}/R$	temperature sensitivity coefficients of $D_{esi0}$ , and $k_{rsi}$ , respectively
$D_e$	effective diffusion coefficient of water in cement hydration products	$\varphi$	capillary porosity in hydrating silica fume–cement paste
$D_{e0}$	initial value of $D_e$	$\alpha_{CEC}$ and $\alpha_{SFC}$	critical reaction degrees of cement and silica fume corresponding to percolation threshold of capillary porosity, respectively
$B_{20}$ , $C_{20}$ , $k_{r20}$ , and $D_{e20}$	values of $B$ , $C$ , $k_r$ , and $D_{e0}$ at 20 °C, respectively	$x_{fc}$	gel/space ratio of blended cement pastes
$\beta_1$ , $\beta_2$ , $E/R$ , and $\beta_3$	temperature sensitivity coefficients of $B$ , $C$ , $k_r$ , and $D_{e0}$ , respectively	$f_c$	compressive strength of blended concrete
$C_0$	mass of cement in the mix proportion	$A$	intrinsic strength of cement and silica fume blends
$W_0$	mass of water in the mix proportion	$a$ and $b$	contributions of cement and silica fume to the intrinsic strength, respectively
$C_{w-free}$	amount of water at the exterior of the C–S–H gel	$n$	strength exponent
$r$	empirical parameter considering the accessibility of water into an inner anhydrous part	$c$ and $d$	contributions of cement and silica fume to the strength exponent, respectively
CHCE( $t$ )	mass of produced calcium hydroxide from the hydration of cement		
SF	silica fume		

predicting properties of silica fume blended concrete or UHPC. Based on a simplified scheme describing the activity of silica fume and fly ash in terms of chemical reactions, Papadakis [7] evaluated the final chemical and volumetric composition of supplementary cementing materials (SCM) concrete. Furthermore, carbonation of the SCM concrete was predicted. Ishida et al. [8] proposed a model to predict the micro–hygro–physical properties of high strength blended concrete and ordinary strength concrete. Mechanical properties, water desorption, moisture loss and drying shrinkage behaviors were determined by an enhanced intrinsic porosity model. Zelic et al. [9] developed a mathematical model to predict the development of compressive strength in cement–silica fume blends. Knudsen's dispersion models were applied in fitting both the degree of hydration and the compressive strength experimental data as a function of time. It was found that the degree of hydration—the compressive strength dependence, for replacement levels varying from 0% to 15% mass of silica fume, indicates a linear mathematical function. Using semi-adiabatic heat of hydration tests, Habel et al. [10] proposed a kinetic hydration model for ultra high performance concrete, and evaluated mechanical properties of hardening UHPC. Compressive strength, tensile strength, fracture energy and secant modulus were evaluated from degree of reaction. It was observed that for the UHPC, the rate of development of mechanical properties was highest for the secant modulus, followed by the compressive and then the tensile strength.

As shown in Refs. [8–10], the development of properties of silica fume blended concrete or UHPC relates with degree of hydration. On the other hand, Papadakis [7] proposed that the development of properties of silica fume blended concrete relates with both

cement hydration and silica fume pozzolanic reaction. In this paper, a numerical model is proposed to simulate the hydration of UHPC. The degree of hydration of cement and degree of reaction of silica fume are obtained as accompanied results from the proposed hydration model. The properties of hardening UHPC are predicted from the contribution of cement hydration and pozzolanic reaction.

## 2. Hydration model of Portland cement

The shrinking-core model, which was originally developed by Tomosawa [11], is used in this study to simulate the development of cement hydration. This model is expressed as a single equation consisting of three coefficients:  $k_d$  the reaction coefficient in the induction period;  $D_e$  the effective diffusion coefficient of water through the C–S–H gel; and  $k_r$  a coefficient of the reaction rate of cement as shown in Eq. (1). These coefficients determine the rate of mass transport through the initial shell layer, the rate of phase boundary reaction process, and the rate of diffusion controlled process. The modeled cement particles are assumed to be spheres surrounded by hydration product. Based on this theory, the rate of cement hydration is derived as follows:

$$\frac{d\alpha}{dt} = \frac{3(S_w/S_0)\rho_w C_{w-free}}{(\nu + w_g)r_0\rho_c} \frac{1}{\left(\frac{1}{k_d} - \frac{r_0}{D_e}\right) + \frac{r_0}{D_e}(1-\alpha)^{\frac{1}{3}} + \frac{1}{k_r}(1-\alpha)^{\frac{2}{3}}} \quad (1)$$

where  $\alpha$  is the degree of cement hydration;  $\nu$  is the stoichiometric ratio by mass of water to cement (= 0.25);  $w_g$  is the physically bound water in C–S–H gel (= 0.15);  $\rho_w$  is the density of water;

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