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Modified silicon carbide whisker reinforced polybenzimidazole used for high temperature proton exchange membrane

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ABSTRACT

Polybenzimidazole containing ether bond (OPBI) was reinforced with silicon carbide whisker (mSiC) modified by 3-aminopropyltriethoxysilane (KH550), and then doped with phosphoric acid (PA) to obtain OPBI/mSiC/PA membranes. These OPBI/mSiC/PA membranes have excellent mechanical strength and oxidative stability and can be used for high temperature proton exchange membrane (HT-PEM). The tensile strength of OPBI/mSiC/PA membranes ranges from 27.3 to 36.8 MPa, and it increases at first and then decreases with the increase of mSiC content. The high mSiC content and PA doping level contribute to improve the proton conductivity of membranes. The proton conductivity of PBI/mSiC-10/PA membrane is 27.1 mS cm⁻¹ at 170 °C without humidity, with an increase of 55.7% compared with that of OPBI/PA membrane. These excellent properties make OPBI/mSiC/PA membranes promising membrane materials for HT-PEM applications.

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1 1. Introduction

2 Proton exchange membrane fuel cell (PEMFC) has received in-3 creasing attention as a power generation device for both stationary and transportation applications due to its high efficiency and 4 low emission of pollutants [1]. There are many advantages of op-5 erating PEMFCs at high temperatures (100-200 °C), including the 6 reduction or elimination of humidification requirements, increased 7 tolerance to CO, wider fuel choices, improved electrode kinetics, 8 9 and higher conductivities [2]. Many efforts have been devoted to the development of low cost and high performance proton ex-10 change membranes (PEMs) for high temperature proton exchange 11 membrane fuel cells (HT-PEMFCs) [3], among which phosphoric 12 acid (PA) doped polybenzimidazole (PBI) membrane [4] might be 13 the most promising one because of its high ionic conductivity at 14 high temperatures, excellent thermal and oxidative stability [5]. 15 16 However, a high PA doping level (PADL) may lead to poor mechanical properties of PA doped PBI membranes [6]. This problem 17 can be overcome by modifying the polymer structure via copoly-18 19 merization [7], N-substitution [8] or cross-linking of the polymer [9–13] at the reactive benzimidazole N–H sites; or by incorporating 20

* Corresponding author at: Key Laboratory for Ultrafine Materials of Ministry of Education, School of Materials Science and Engineering, East China University of Science and Technology, Shanghai 200237, China. *E-mail address:* saxu@ecust.edu.cn (S. Xu). inorganic particles such as hygroscopic oxides (i.e., SiO_2 and ZrO_2) 21 [14–18], perovskite-type oxides (i.e., strontium cerate and barium 22 titanate) [19,20], heteropolyacids [21], zirconium phosphates (ZrP) 23 [22], carbides (i.e., silicon carbide (SiC)) [23], and graphene oxide 24 (GO) [24], which can improve the mechanical properties and acid 25 doping levels of composite membranes because of hydrogen-bond 26 interaction between the oxygen atoms of hydroxyl groups and acid 27 [20]. 28

SiC has many desirable properties, such as high strength, high 29 stiffness, excellent thermal and electrochemical stability [25], mak-30 ing it suitable for the reinforcement of polymer composites for HT-31 PEMFC applications [26]. Joel [27] reported the effects of SiC on 32 the physical and mechanical properties of PBI membranes. In this 33 report, SiC was introduced to provide additional mechanical sta-34 bility to the PBI membranes. The results showed that PBI mem-35 brane containing 5% SiC was more stable than that containing 1%, 36 3% and 7% SiC, and it also met the expected conductivity stan-37 dards (about 0.145 S cm^{-1} in a temperature range of 22-160 °C). 38 SiC whisker grown by single crystal can also improve various prop-39 erties of the polymer matrix due to the special fiber-shaped struc-40 ture. Liepins et al. [28] first introduced β -SiC whisker into PBI to 41 prepare PBI/ β -SiC composite by solution-casting method, which, 42 however, was not used for PEM. To our knowledge, there has been 43 no study on SiC whisker for HT-PEMs. 44

In this study, modified SiC whisker (mSiC) was prepared using 45 a silane coupling agent of 3-aminopropyltriethoxysilane (KH550) 46

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and then blended with OPBI to obtain a series of OPBI/mSiC mem-47 48 branes. With the aid of the alkaline imidazole ring, silicon hydroxyl, and amino groups, these OPBI/mSiC membranes were fur-49 50 ther doped with PA to obtain OPBI/mSiC/PA membranes used for HT-PEM. The results clearly show that the introduction of mSiC 51 has a significant effect on the PADL, mechanical properties, mi-52 crostructure, oxidative stability and proton conductivity of compos-53 ite membranes. 54

2. Experimental 55

2.1. Materials 56

OPBI polymers were prepared by polycondensation of 57 3,3-diaminobenzidine (DAB Chemical Technology Co., Ltd., Shang-58 hai) and 4,4-oxybis (benzoic acid) (OBA, Aladdin Industrial Co., 59 Shanghai) in methanesulfonic acid (MSA, Aidie Industrial Co., Ltd., 60 Shanghai) [29]. SiC whiskers were purchased from Alfa Aesar 61 Chemical Ltd. (Tianjin, China) and used without further purifi-62 cation. 3-Aminopropyltriethoxysilane (KH550, Resin Factory Co., 63 Ltd., Shanghai) was used as a coupling reagent to modify SiC. 64 65 Sodium hydroxide (Lingfeng Chemical Reagent Co., Ltd., Shanghai) was used without further purification. Dimethylacetamide (DMAc, 66 Lingfeng Chemical Reagent Co., Ltd., Shanghai) was purified by 67 distillation under reduced pressure and used as a solvent. All the 68 69 other reagents were obtained from commercial sources and used 70 as received.

2.2. Preparation of modified silicon carbide whisker (mSiC) 71

72 1 mL of KH550 was accurately weighed and mixed with 30 mL 73 of absolute ethyl alcohol and 3 mL of water in a 100 mL beaker. 74 After adjusting the pH to 9–10 with sodium hydroxide, the mix-75 ture was subjected to ultrasonic treatment for 30 min. Then 1 g of 76 SiC was added into the solution and stirred at 80 °C for 4 h. The mixture was cooled down to room temperature and continuously 77 stirred for 24 h. The solution was then dried at 80 °C for 18 h to re-78 move the solvent. The obtained product was washed with alcohol 79 and water until the pH of the centrifugate was 7, and then dried 80 02 81 at 80 °C to a constant weight.

2.3. Preparation of OPBI/mSiC membranes 82

OPBI and OPBI composite membranes were prepared by the 83 solution-casting method. mSiC was added into DMAc in an ultra-84 sonic bath for 1 h to obtain a homogeneously dispersed solution. 85 Then, approximately 5 wt% OPBI solution was mixed with mSiC 86 87 (DMAc) in different proportions under stirring. All homogeneous 88 solutions were cast onto a clean glass plate at 80 °C for 24 h to obtain smooth membranes. Later, these membranes were immersed 89 in water for 24 h to remove residual solvent, and then dried at 90 80 °C in vacuum overnight. Pure OPBI membrane and OPBI/SiC 91 92 composite membranes were also prepared by the same procedure and served as the control. OPBI/mSiC membranes and PA doped 93 94 membranes were 60–70 μ m and 100–120 μ m thick, respectively.

2.4. Characterization and measurements 95

96 The functional groups in mSiC were analyzed using a Nicolet 6700 Fourier transform infrared (FT-IR) spectrometer in the range 97 of $400-4000 \text{ cm}^{-1}$ with a resolution of 4 cm^{-1} , and the element 98 contents were measured using an energy dispersive spectrometer 99 (EDS). 100

The membranes were immersed in a 14.6 M PA aqueous solu-101 tion at 80 °C for 48 h and then wiped with filter paper repeatedly 102

to obtain PA-doped membranes. Then, these membranes were im-103 mersed into alkaline aqueous solution to remove absorbed PA and 104 washed several times with water. The obtained membranes were 105 then dried overnight at 80 °C in vacuum. The PADL was determined 106 by the following equation: 107

PADL(mol PA per repeating unit) =
$$(W_{PA}-W_{dry})/W_{dry}$$

 $\times M_{RU}/M_{PA}$ (1)

where W_{drv} , and W_{PA} are the weight of dry and PA-doped mem-108 branes, and M_{RU} and M_{PA} are the molar mass of repeating unit of 109 OPBI and PA, respectively. 110

The mechanical properties of PA-doped membranes were deter-111 mined using an MTS E43 universal testing machine in ambient at-112 mosphere at a drawing rate of 10 mm min⁻¹. The tensile fracture 113 surface of membranes was observed by a scanning electron mi-114 croscope (SEM, FE-S4800, Hitachi, Japan) at an acceleration voltage 115 of 15 kV. The thermal stability of PA-doped membranes was mea-116 sured on an HCT-1 thermogravimetric analyzer (TGA, Beijing Hen-117 ven Scientific Instrument Factory, China). Samples were maintained 118 at 120 °C for 30 min to remove absorbed moisture, and then cooled 119 to 80 °C and reheated to 800 °C at a heating rate of 10 °C min⁻¹ un-120 der nitrogen atmosphere. The grafting ratio of mSiC was calculated 121 via weight loss in the TGA curves. 122

Dry membranes were immersed in Fenton's reagent (3 wt% 123 H_2O_2 and 4 ppm Fe²⁺) at 68 °C, and the oxidative stability was 124 evaluated by weight remaining of membranes [30]. All membranes 125 were immersed for 168 h, washed, and then dried at 120 °C to a 126 constant weight. The weight remaining was calculated by: 127

Weight remaining (wt%) = $W_1/W_0 \times 100\%$

where W_0 and W_1 are the weight of initial and final membranes, 128 respectively. 129

The proton conductivity of composite membranes was deter-130 mined by AC impedance spectroscopy over a frequency range of 131 10⁶–10³ Hz at 100–170 °C using a PARSTAT 2273 electrochemical 132 analyzer. All membranes were dried at 120 °C to remove absorbed 133 moisture, and the measurement was carried out under anhydrous 134 conditions. The in-plane proton conductivity (σ , S cm⁻¹) was mea-135 sured and calculated according to the following equation: $\sigma = L/RS$, 136 where L is the distance between the two electrodes, R is the re-137 sistance of the membrane, and S is the cross-sectional area of the 138 membrane. 139

3. Results and discussion

3.1. Characterization of mSiC

The FT-IR spectra of SiC and mSiC are shown in Fig. 1. Two new 142 absorption peaks observed at about 2848 and 2923 cm⁻¹ in Fig. 143 1(b) can be assigned to the $-CH_2$ - stretching vibration of alkane 144 chain; and the peak at 1640 cm⁻¹ becomes stronger and is as-145 signed to the -NH₂ group, indicating the successful reaction of 146 KH550 with SiC. This can be further confirmed by the TGA curves 147 in Fig. 2. The weight of pre-dried SiC remains almost constant, 148 whereas that of mSiC decreases at first and then keeps constant 149 after 600 °C. The weight remaining at 600 °C is 99.0 wt% and 93.2 150 wt% for SiC and mSiC, respectively, indicating that the content of 151 coupling reagent is about 12.1 wt% via the following formula. 152

Grafting ratio (wt%) = $(m_2 - m_1) \times M_2/M_1$

where m_1 and m_2 are the weight remaining of SiC and mSiC at 153 600 °C and M_1 and M_2 are the molecular weight of thermal decom-154 position part (-CH₂CH₂CH₂NH₂) and hydrolyzed coupling reagent 155 (-Si(OH)₂CH₂CH₂CH₂NH₂), respectively. 156

Table 1 shows the EDS results of SiC and mSiC. The atomic ratio 157 of hydrolyzed KH550 should be N/Si =1:1. Thus, the atom percent 158

(2)

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