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Carbon-coated manganese dioxide nanoparticles and their enhanced electrochemical properties for zinc-ion battery applications

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ABSTRACT

In this study, we report the cost-effective and simple synthesis of carbon-coated α -MnO₂ nanoparticles (α -MnO₂@C) for use as cathodes of aqueous zinc-ion batteries (ZIBs) for the first time. α -MnO₂@C was prepared via a gel formation, using maleic acid (C₄H₄O₄) as the carbon source, followed by annealing at low temperature of 270 °C. A uniform carbon network among the α -MnO₂ anoparticles was observed by transmission electron microscopy. When tested in a zinc cell, the α -MnO₂@C exhibited a high initial discharge capacity of 272 mAh/g under 66 mA/g current density compared to 213 mAh/g, at the same current density, displayed by the pristine sample. Further, α -MnO₂@C demonstrated superior cycleability compared to the pristine samples. This study may pave the way for the utilizing carbon-coated MnO₂ electrodes for aqueous ZIB applications and thereby contribute to realizing high performance eco-friendly batteries.

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1 1. Introduction

Since manganese dioxide (MnO₂) is relatively cheap, environ-2 mentally benign, and electrochemically active, it is considered 3 as a promising energy storage electrode material. Different crys-4 5 tallographic polymorphs of MnO₂ namely α , β , γ , δ , and λ can be obtained by different synthetic routes [1]. Among all these 6 7 polymorphs, α -MnO₂ has a distinctive [2 × 2] tunnel structure and it has received much attentions. In our previous report, α -MnO₂ 8 nanorods were investigated as zinc-ion battery (ZIB) cathodes. 9 10 This electrode showed a first discharge capacity of 233 mAh/g at 83 mA/g current density [2]. However, capacity fading under 11 prolonged cycles, which has also been reported by many other 12 13 groups, is a major drawback of this electrode [3,4]. Additionally, the intrinsically poor electrical conductivity of MnO₂ impedes the 14 15 electronic path way and consequently, limits its electrochemical 16 performance [4,5].

Naturally abundant, low-cost, and environmentally benign 17 18 highly conducting materials, such as carbon can be used to overcome the aforementioned limitation of MnO₂. Carbon-coated elec-19 20 trode materials such as transition metal oxides have been widely 21 studied to obtain high performance electrodes [6–9]. Hashem et al. prepared carbon-coated rod-shaped MnO₂ in order to achieve an 22 electrode material with enhanced electrochemical performance for 23 24 lithium-ion battery (LIB) application [10]. Incorporation of carbon

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materials including carbon nanotubes, graphene, and fullerene 25 into MnO₂ increases its surface area and enhances its conductivity 26 because of the formation of a MnO₂/C composite. Hence, these 27 composite materials have been used in high performance super-28 capacitors and LIBs [11,12]. Mesoporous MnO₂/C spheres with a 29 very large surface area $(324 \text{ m}^2 \text{ g}^{-1})$ have been synthesized and 30 investigated as an active electrode by Pan et al. [13]. Kim et al. 31 demonstrated the electrochemical performance of an electrode-32 posited mesoporous MnO₂/C composite electrode for advanced 33 supercapacitors [14]. Moreover, Li et al. reported on carbon coated 34 MnO₂ nanorods obtained via a facile solid state grinding and low 35 temperature calcination as a high performance electrode for super-36 capacitors [6]. Therefore, from the above discussion, it is clear that 37 the carbon coating strategy is an effective method of improving 38 conductivity and increasing the stability of electrode materials. 39

Here, for the first time, we report the enhanced electrochemical 40 performance of carbon-coated α -MnO₂ nanoparticles for ZIBs. We 41 have successfully prepared carbon-coated α -MnO₂ nanoparticles 42 by just stirring α -MnO₂ nanoparticles in an ethanolic solution of 43 maleic acid at a mild temperature until a gel is formed and finally 44 it is annealed at lower temperature under Ar atmosphere. The 45 carbon coatings on the electrode material not only increase the 46 electrical conductivity and discharge capacity but also improve the 47 electrode cycleability and capacity retention by preventing man-48 ganese dissolution from the active material during electrochemical 49 discharge reaction. Higher discharge capacity and improved cy-50 cleability were obtained for the prepared carbon-coated MnO₂ 51 sample. 52

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53 2. Experimental

The pristine α -MnO₂ nanoparticles were synthesized before 54 55 preparing the carbon-coated α -MnO₂ nanoparticles (α -MnO₂@C). First, 150 mmol of Mn(CH₃COO)₂·4H₂O was dissolved in 30 mL of 56 distilled water (denoted: solution A). Then, a 50 mL solution of 57 100 mmol KMnO₄ was added dropwise to solution A. A dark brown 58 precipitate was obtained immediately after the addition of KMnO₄ 59 60 solution followed by continuous stirring for 5 h. The resultant products were then filtered, washed with distilled water followed 61 62 by ethanol, and dried at 80 °C for 12 h. Finally, the as-prepared 63 sample was ground using an agate mortar and annealed at 450 °C for 5 h before characterization. For preparing α -MnO₂@C, maleic 64 65 acid was used as the carbon source. $100 \text{ mg} \alpha$ -MnO₂ nanoparticles were added slowly to a 5 mL ethanolic solution of 25 mg maleic 66 acid and stirred at a mild temperature (50 °C) until gel formation 67 was obtained. Then, the obtained gel was heated at 270 °C for 3 h 68 under Ar atmosphere. 69

The crystalline nature of the powder samples was characterized 70 using X-ray diffraction (XRD, Shimadzu X-ray diffractometer, Chon-71 nam National University, Republic of Korea) with Cu-K α radiation 72 73 $(\lambda = 1.54056 \text{ Å})$ operating at 40 kV and 30 mA over the 2 θ range of 74 10° to 80° in steps of 0.01°. The morphology and elemental distri-75 bution of the final products were characterized by scanning electron microscopy (SEM, Hitachi, Chonnam National University, Re-76 public of Korea), field-emission transmission electron microscopy 77 (FE-TEM, Tecnai-F20) with an accelerating voltage of 200 kV, and 78 79 energy dispersive X-ray (EDX) spectroscopy (EX-200, Hitachi, Chonnam National University, Republic of Korea). For FE-TEM studies, 80 the samples were dispersed in ethanol by ultrasonication before a 81 few drops of the dispersion were put on copper grids and held un-82 83 til the solvent evaporated at room temperature.

84 Both pristine and carbon-coated MnO₂ electrodes in this study were prepared by mixing the active materials, Ketjen black, and 85 teflonated acetylene black (TAB) with a ratio of 7:2:1, and were 86 then pasted onto a stainless-steel mesh. The prepared cathodes 87 were then vacuum-dried at 120 $^\circ C$ for 12 h. The anode was a Zn 88 89 metal foil with a thickness of 0.25 mm and 1 M ZnSO₄ (pH 4.0) aqueous solution was used as the electrolyte. To assemble 2032-90 type coin cells, a glass fiber was sandwiched between the cathode 91 and the Zn foil in the electrolyte. The coin cells assembled in an 92 93 open-air atmosphere were aged overnight before electrochemical measurements. Cyclic voltammetry (CV) was performed at a scan 94 95 rate 0.5 mV/s in the voltage range 1.0-1.8 V and electrochemical 96 impedance spectroscopy (EIS) was performed with an amplitude of 5 mV at frequencies varying from 100 mHz to 10 kHz using a 97 98 Bio Logic Science Instrument (VSP 1075). The electrochemical discharge/charge measurements at room temperature were performed 99 using a BTS 2004H (Nagano Keiki Co. Ltd., Ohta-ku, Tokyo, Japan) 100 analyzer. 101

102 3. Results and discussion

¹⁰³ In this study, α -MnO₂ was prepared by a simple redox reaction ¹⁰⁴ in DI water between manganese acetate and potassium perman-¹⁰⁵ ganate and the reaction can be described as follows:

 $\begin{array}{l} 3Mn(CH_{3}COO)_{2}+2KMnO_{4}+2H_{2}O\rightarrow5MnO_{2}\\ +4CH_{3}COOH+2CH_{3}COOK \end{array}$

A dark brown precipitate was formed immediately after the addition of the $KMnO_4$ solution. This simple redox reaction is widely used to prepare MnO_2 nanoparticles. MnO_2 exhibits poor electrical conductivity, thereby limiting electronic transport and hence its performance for electrochemical applications. The techniques of utilizing highly conductive materials such as carbon to improve the electronic transport and hence electrochemical performance in

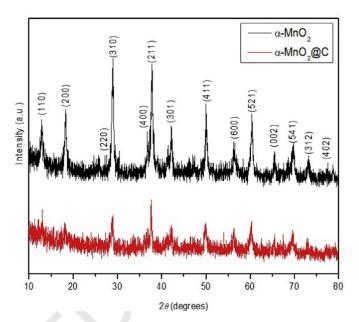


Fig. 1. XRD pattern of the pristine and carbon-coated α -MnO₂ powder.

MnO2 electrodes are well recorded in previous studies. Recently,113Li et al. have prepared carbon-coated MnO2 nanorods by solid-114state grinding and annealing at lower temperature using malic acid115as the carbon source [6]. In the present study, we have slightly116modified that method for preparing carbon-coated MnO2 by using117maleic acid as the carbon source.118

Fig. 1 shows the XRD patterns for both pristine α -MnO₂ and 119 α -MnO₂@C samples. Both of the patterns match well with the pat-120 tern of α -MnO₂ (JSPDS card no. 44-0141). This observation sug-121 gests that carbon coating does not influence the structure of α -122 MnO_2 [10]. The calculated values of the pristine α -MnO₂ unit cell 123 parameters were a = 9.837 Å and c = 2.860 Å. On the other hand, 124 the lattice parameters of the α -MnO₂@C were a = 9.731 Å and 125 c = 2.814 Å. The peak intensities appear relatively low for the α -126 MnO₂@C sample [5,6]. 127

The surface morphologies of the pristine α -MnO₂ and α -128 MnO2@C samples were studied using electron microscopy. The FE-129 SEM images of the pristine α -MnO₂ and α -MnO₂@C are shown 130 in Fig. 2(a) and (b), respectively. Uniform particles with spherical 131 shaped sizes ca. 20 nm for α -MnO₂ and α -MnO₂@C were observed 132 from these images. Similar SEM images were obtained for the pris-133 tine α -MnO₂ and α -MnO₂@C, indicating that no obvious change in 134 morphology occurred after carbon coating [6]. Furthermore, in or-135 der to investigate the carbon coating, the sample was characterized 136 by FE-TEM as well. The FE-TEM images recorded for α -MnO₂@C at 137 different magnifications are presented in Fig. 2(c) and (d). The rel-138 atively dark shade regions appear to indicate the presence of the 139 clustered α -MnO₂@C. The high resolution TEM image presented 140 in Fig. 2(d) displays distinct fringes with interplanar spacing val-141 ues of approximately 0.49 nm, corresponding to the (200) planes 142 of α -MnO₂ (JCPDS card no. 44-0141). HR-TEM image recorded for 143 α -MnO₂@C also reveals that the nanoparticles are coated with a 144 thin carbon layer. 145

EDX mapping for the chemical elements in the sample was also 146 performed and the obtained images in Fig. 3 showing the distribu-147 tion of C, Mn, and O confirm the presence of carbon in the sam-148 ple. Also, it is worth noting that the carbon distribution is uni-149 formly spread throughout the selected area of study. This obser-150 vation appears to suggest that the carbon is uniformly wrapped 151 on the α -MnO₂ particles. Moreover, the EDX analysis estimated 152 the carbon percentage to be 7.9%, the amount being adequate for 153

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