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Few-layered graphene via gas-driven exfoliation for enhanced supercapacitive performance

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ABSTRACT

High-quality graphene flakes have long been desirable for numerous applications including energy storage, printable electronics, and catalysis. In this contribution, we report a green, efficient, facile gas-driven exfoliation process for the preparation of high-quality graphene in large scale. The gas exfoliation process was realized by the interplay between the expansion of interlayer at high temperature and the gasification of liquid nitrogen within the interlayer. Detailed experiments demonstrated that the higher temperature was critical to the formation of fewer layers. The exfoliated graphene was demonstrated to be of high quality. We further investigated the electrochemical behavior of this exfoliated graphene. As a result, this few-layered graphene demonstrated an enhanced capability as a supercapacitor, much higher than its counterpart parent material.

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1. Introduction 1

2 Graphene consisted of carbon atoms arranging in a honeycomb 3 lattice, has been regarded as a harbinger of the both experimen-4 tally and industrially coming of next-generation material, because of its remarkable electronic, optical, thermal and mechanical prop-5 erties [1–4]. Although graphenes, including graphene oxides (Go), 6 reduced graphene oxides (rGo), etc., have shown promising per-7 8 formance in many fields such as catalysis [5-7] electrochemistry [8-10] photochemistry [11,12], even in their disordered forms, one 9 can anticipate that high-quality graphene would demonstrate ex-10 tensive enhancement in performance [13]. 11

The first discovery of graphene was reported in 2004, Geim and 12 13 Novoselov harvested graphene monolayers from original graphite 14 crystals by micromechanical cleavage [14]. Since then, a broad 15 range of fabrication methods have been developed. Generally, the production methods of graphene can be summarized as two types: 16 17 bottom-up process and top-down strategy, i.e., carbonization of or-18 ganic precursors (bottom-up) and reducing layers by chemical, mechanical, physical methods (top-down) [2,15–17]. While numerous 19

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bottom-up methods have been proposed for generating graphene, 20 the top-down process is still the most versatile strategy for fabri-21 cating graphene in mass production [18]. Up to date, several top-22 down methods, i.e. ball milling, high-shear mixing, ultrasound-23 assisted liquid exfoliation, have been reported [19-26]. Neverthe-24 less, these pathways struggle from their inherent disadvantages. 25 For example, ball milling, including dry milling and wet milling, was developed for producing graphene from the parent graphite [21,22,27]. However, a high concentration of amorphous carbon is inevitable during the dry milling process and long milling time is required in wet milling procedure [20]. Meanwhile, in both the ultrasound-assisted liquid exfoliation and high-shear mixing methods, liquid medium has turned to be a hurdle. Several issues, including the dispersion of graphene in liquid and subsequent removal of liquid solvent, are usually associated with the liquid medium. As a result, the processes involving the liquid medium generate very poor exfoliating efficiency (usually lower than 3%) and chemical impurity [19]. Thusly, it is of significance to develop a top-down stage which can produce high-quality graphene without a liquid medium. 39

Recently, our group has developed a gas exfoliation strategy for 40 producing few-layer hexagonal boron nitride from its parent ma-41 terial [28,29]. The operating process employed a gas medium in-42 stead of liquid medium for exfoliation, dramatically improved the 43

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chemical purity of the final products. Additionally, the final prod-44 45 ucts were proved to be in high-quality and high-yield. Inspired by 46 the finding, herein, we employed a gas-driven exfoliation approach 47 to gain few-layer graphene from commercial graphite. The key in the process is treating the pristine graphite under extreme condi-48 tions to destroy the Van der Waals' force between graphene layers. 49 Generally, the graphite was treated at a high temperature, lead-50 ing to the interlayer expansion and enabling exfoliating media (liq-51 52 uid nitrogen, L-N₂) to enter the graphite interlayers. Then the hot graphite was quickly immersed into extreme cold L-N₂, and the L-53 54 N₂ entered into the graphene interlayers. Because of the very hot status of graphite, the L-N2 quickly gasified and consequently de-55 stroyed the interlayer Van der Waals' force to generate few-layer 56 57 graphene. In this contribution, the exfoliation process and mechanism was investigated in detail. We also studied the influence of 58 treating temperature and found that a higher treating tempera-59 ture is in favor of fewer-layered structure. The gas exfoliation strat-60 egy for the preparation of graphene is environmentally benign. We 61 further investigated the electrochemical behavior of the obtained 62 few-layered graphene and demonstrated an enhanced supercapac-63 itive performance at all scan rate. This work provides a new strat-64 egy for producing high-quality graphene in large scale to meet 65 66 the requirements for some electrochemical applications. Also, some other few-layer materials with high quality may be generated by 67 68 the current procedure as well.

69 2. Experimental

70 2.1. Materials

Flake graphite was obtained from Sigma-Aldrich without further treatment. Liquid nitrogen was provided by Suzhou Jinhong Co. Ltd. Ultra-pure water was obtained by a water purifier in our laboratory.

75 2.2. Preparation process

76 Typically, 1 g of pristine graphite was placed in a 25 mL of the crucible. After that, the crucible was transferred to a muffle 77 oven with a temperature of 500 °C, 600 °C and 700 °C, respec-78 tively. After ~ 1 min, the graphite turned to be a red-hot status. 79 Then the graphite was poured into a 4 L of Teflon beaker with 80 \sim 200 mL of L-N₂ inside. Afterwards, shake the beaker until the 81 total gasification of L-N2. So far, one exfoliation process was com-82 pleted. Then, the whole process was repeated for ten times. Then, 83 the products were carefully collected and dispersed in 200 mL of 84 DI water. Subsequently, the dispersed solution was centrifuged at 85 1500 rpm to remove un-exfoliated bulk graphite, and the upper 86 dispersed solution was further centrifuged at 12,000 rpm to ob-87 tain the final product. The final product was finally vacuum dried 88 at 50 °C for overnight. With different heating temperature, the ex-89 90 foliated graphenes were marked as exf-G-500, exf-G-600, and exf-91 G-700, respectively. The yield of graphene from its parent graphite was meausred to be \sim 0.2–03 g, depending on the treatment tem-92 93 perature. Caution that during the preparation process, the high and 94 ultra-low temperature is unavoidable and necessary protection is 95 required.

96 2.3. Characterizations

97 In situ X-ray diffraction (XRD) patterns were recorded on an 98 RINT2000 vertical goniometer equipped with Anton-Paar XRK. The 99 scanning voltage and current were 40 kV and 100 mA, respectively; 100 and the scanning range was $5-75^{\circ}$. A D8 ADVANCE X-ray diffrac-101 tion with Cu $K\alpha$ radiation ($\lambda = 1.541$ Å) was employed for the de-102 termination of XRD patterns from 10° to 80° with a scanning speed of 7° min⁻¹. While the surface morphologies of samples were char-103 acterized by a JSM-7001F field emission scanning electron micro-104 scope (FE-SEM). Raman spectra were performed on a Thermo Sci-105 entific DXR Smart Raman spectrometer, whose excitation wave-106 length was 532 nm. Atomic force microscope (AFM) was tested by 107 an Asylum MFP-3D from Asylum Research Company. Transmission 108 Electron Microscope (TEM) and High-Resolution Transmission Elec-109 tron Microscope (HRTEM) were carried out on a Tecnai G2 F30 S-110 TWIN from FEI Company, USA. 111

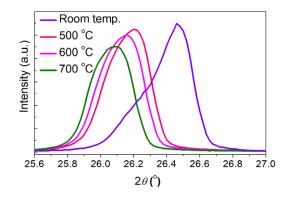
2.4. Electrochemical measurements

The eletrochemical measurements were carried out in 1 M KOH 113 solution with a CHI 760D electrochemical analyzer (ChenHua In-114 struments, Shanghai, China) at room temperature. All the eletro-115 chemical performances were tested with a three-electrode system. 116 Platinum foil and saturated calomel electrode (SCE) were used as 117 the counter and reference electrodes, respectively. The working 118 electrodes were prepared as following: Typically, 5 mg of the sam-119 ples were mixed with 0.5 mg Nafion binder in 1 mL of enthanol to 120 obtain homongeneous slurry. Afterwards, 20 µL of the slurry was 121 doped onto a glassy carbon electrode, whose diameter was 5 mm. 122 Then, the eletrode was dried at 80 °C under vacuum for 24 h. The 123 laoding amount of samples on glassy carbon electrode was mea-124 sured to be 0.5 mg cm^{-2} . 125

3. Results and discussion

As described in the experimental section, the whole process 127 does not introduce any chemical impurities and produce graphene 128 in large scale. Realization of this strategy mainly relies on high 129 temperature enlarged interlayer distance, triggered weakened Van 130 der Waals' force between layers, and gasification of L-N2 for over-131 coming the Van der Waals' force. In situ XRD characterization in 132 Fig. 1 was performed to highlight the impact of the heating pro-133 cess. It is verified that the 2θ degree of (002) peak, representing 134 the inter-layer lattice of graphene, shifted to lower angle gradually 135 from 26.5° to 26.0° with the heating temperature raised from room 136 temperature to 700 °C. Based on Bragger's equation, the interlayer 137 distance enlarged with the increase of heating temperature, facili-138 tating the fluxion of L-N₂ into the interlayer of graphite. Moreover, 139 larger interlayer distance indicates weaker Van der Waals' force be-140 tween interlayers, making the exfoliation easier than its pristine 141 status. 142

The whole exfoliation process was summarized in Fig. 2. Firstly, 143 the extraneous heat gave rise to the enlarged interlayer distance, 144 advocating weaker interlayer Van der Waals' force and a larger interlayer space for L-N₂. Subsequently, with the red-hot graphite 146 immersed into L-N₂, L-N₂ entered into the interlayer of graphite. 147





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