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Solid State Ionics

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Amorphous *x*LiF-FeSO₄ ($1 \le x \le 2$) composites as a cathode material for lithium ion batteries

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 A R T I C L E I N F O
 A B S T R A C T

 Keywords:
 Although the synthesis is not so easy, LiFeSO₄F, a high-voltage iron-based cathode, is attractive by virtue of its low cost and small environmental impact. We prepared amorphous xLiF-FeSO₄ ($1 \le x \le 2$) using the dry ball-milling method

 Dry ball-milling method
 milling method. Moreover, we succeeded to synthesize tavorite-type LiFeSO₄F by sintering amorphous LiF-FeSO₄

 Fluorosulfate compound
 C. Amorphous 1.3LiF-FeSO₄ exhibited the highest reversible capacity, of about 130 mAh g⁻¹, among all the xLiF-FeSO₄ series ($1 \le x \le 2$), with an average voltage of 3.5 V. In addition, we found that the obtained

1. Introduction

Beyond Li-ion batteries are being developed to power electric vehicles and to efficiently utilize renewable energies, such as solar and wind power. In particular, researchers worldwide are intensely pursuing novel electrode materials for further advances in energy density. As potential cathodes for beyond Li-ion batteries, iron-based cathode materials have attracted attention by virtue of their low cost and the abundance of iron resources. Among these materials, LiFePO₄ has a high operating voltage of 3.3 V corresponding to $\text{Fe}^{2+}/\text{Fe}^{3+}$ redox by the inductive effect of PO_4^{3-} polyanions [1]. Lithium iron sulfate Li₂Fe $(SO_4)_2$ [2], which substitutes sulfate SO_4^{2-} with phosphate PO_4^{3-} anions, shows the highest operating voltage of 3.75 V (vs. Li⁺/Li) among iron-based cathode materials, although its theoretical capacity is restricted by the high molecular weight of SO₄. On the other hand, the theoretical capacity can be increased to 150 mAh g^{-1} from 102 mAh g⁻¹ by changing from Li₂Fe(SO₄)₂ to LiFeSO₄F. This LiFeSO₄F has been reported to have two types of crystal structures, tavorite-type and triplite-type LiFeSO₄F, which showed operating voltages of 3.6 V and 3.75 V, respectively [3,4]. However, the reported LiMSO4F (M = Fe, Co, Ni) is not stable at the high temperatures (over 500 °C) used in the normal solid-state method. In addition, the other reported LiFeSO₄F also required a special synthesis route such as ionic liquid, vacuum condition, or $FeSO_4$ hydrate [5–7]. Because these synthesis routes are very costly, it is necessary to develop a simple process for synthesis of LiFeSO₄F. Triplite-type LiFeSO₄F has already been obtained from nonhydrate FeSO₄ and LiF by the dry ball-milling method [8]. However, its rechargeable capacity was 100 mAh g⁻¹, corresponding to < 0.7 electron reaction per mole, and the calculated energy density was only 350 Wh kg⁻¹. In this work, in order to develop a simple and low-cost synthesis method of LiFeSO₄F and to improve the cathode properties, we used the dry ball-milling method to prepare amorphous LiF-FeSO₄ having the same chemical composition as LiFeSO₄F. Moreover, it is already known that the electric conductivity for solid electrolyte can improve by increasing Li concentration in amorphous compound. Therefore, we are interested in determining whether this theory is effective for amorphous cathode materials. So, we also prepared amorphous *x*LiF-FeSO₄ (1 ≤ *x* ≤ 2) with the dry ball-milling method and evaluated its cathode properties against Li metal anodes in Li-salt electrolytes.

2. Experiment

amorphous xLiF-FeSO₄ cathodes have excellent cyclability and rate capability, also.

Amorphous *x*LiF-FeSO₄ composites (x = 1.0, 1.2, 1.3, 1.5, 1.7 and 2.0) were prepared by dry ball-milling. Mixtures of LiF (Wako Pure Chemical Industries) and FeSO₄ with a molar ratio of *x*LiF:FeSO₄ were put in an Ar-filled atmosphere control container with ϕ 3-ZrO₂ balls (ca. 40 g). The mixtures were ball-milled using a planetary mill (Fritsch, Pulverisette7) at 600 rpm under ambient Ar for 6 h. Here, to obtain FeSO₄ as a starting material, FeSO₄·7H₂O (Wako Pure Chemical Industries) was sintered at 300 °C for 12 h under Ar. To obtain a uniform *x*LiF-FeSO₄ ($1 \le x \le 2$) and carbon composite, the obtained

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https://doi.org/10.1016/j.ssi.2018.09.007

Received 8 June 2018; Received in revised form 7 September 2018; Accepted 13 September 2018 0167-2738/ © 2018 Elsevier B.V. All rights reserved.





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Fig. 1. (a) The XRD profiles of the obtained amorphous xLiF-FeSO₄ ($1 \le x \le 2$), (b) EDS mapping for the obtained LiF-FeSO₄ sample.



Fig. 2. (a) Structure of triplite-LiFeSO₄F and tavorite-LiFeSO₄F, (b) DSC profile for the obtained amorphous LiF-FeSO₄, (c) XRD profiles of the samples sintered at 280 °C and 350 °C.

products were subjected to the carbon composite process twice. The obtained amorphous *x*LiF-FeSO₄ was ball-milled with 5 wt% acetylene black (AB, Denki Kagaku) at 600 rpm for 6 h. The product was ball-milled again with 20 wt% AB in Ar. Crystalline LiFeSO₄F was obtained from the amorphous LiF-FeSO₄/C. To determine the sintering temperature, we measured the temperature profiles of amorphous LiF-FeSO₄/C using the Thermo Plus TG-DSC 8230L system (Rigaku). To obtain crystalline LiFeSO₄F, amorphous *x*LiF-FeSO₄ (x = 1.0) was sintered at 280 °C or 350 °C in a sealed SUS container. The obtained powders were characterized using powder X-ray diffraction (XRD, 50 kV and 300 mA, Cu K α , Rigaku TTRIII). The particle size, particle

morphology and EDS (Energy Dispersive X-ray Spectrometer) mapping were observed by using a transmission electron microscope (TEM; JEOL JEM 2100F). The cathode properties of the amorphous xLiF-FeSO₄/C and the crystalline LiFeSO₄F were evaluated with a 2032 coin-type cell using 1 M LiPF₆/EC:DMC = 1:1 in volume (Tomiyama Pure Chemical Industries) and a polypropylene separator (3501, Celgard) against lithium metal (Honjo Metal). The cathode pellets to evaluate the electrochemical properties were fabricated by mixing the xLiF-FeSO₄/C composite powder with a 5 wt% polytetrafluoroethylene (PTFE) Teflon binder (Polyflon PTFE F-104; Daikin Industries, Ltd.) and punched into disks (ca. 30 mg weight and 10 mm diameter). Download English Version:

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