

## Materials science communication

Facile synthesis of hierarchically  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres for lithium-sulfur batteriesYao Wu<sup>a</sup>, Qi Xiao<sup>a,\*</sup>, Suping Huang<sup>b</sup>, Kun Wang<sup>a</sup><sup>a</sup> School of Minerals Processing and Bioengineering, Central South University, Changsha, 410083, China<sup>b</sup> State Key Lab of Powder Metallurgy, Central South University, Changsha, 410083, China

## HIGHLIGHTS

- $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres were synthesized.
- $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell structure can encapsulate the polysulfides.
- There is a synergistic action among C and  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>.
- Sulfur cathodes containing  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres exhibit high initial discharge capacity.
- Sulfur cathodes containing  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres exhibit a superior rate performance.

## ARTICLE INFO

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## ABSTRACT

The practical application of lithium-sulfur batteries (LSBs) remains restricted because of a severe polysulfide-shuttle effect and the insulating nature of sulfur. In this work, a kind of unique  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres comprising carbon coating on the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> yolk-shell microspheres were synthesized using organic adsorbing and in-situ thermal reduction methods, and characterized by X-ray diffraction (XRD), scanning electron microscopy (SEM). It was demonstrated that the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres could encapsulate the polysulfides and increase the sulfur utilization. Accordingly,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres cathode achieved high initial discharge capacity of 1046.9 mAh g<sup>-1</sup>, excellent discharge capacity (765.5 mAh g<sup>-1</sup>) after 70 cycles (the capacity retention was about 73%) and superior rate performance. In addition, the appropriate C content was proved to be an important factor in the influence of electrochemical performances. Hence, our present method paves a new path to improve the performance of Li-S batteries.

## 1. Introduction

Recently, with the development of energy storage devices, rechargeable batteries with high performance have been attracted more and more attention. Among all rechargeable batteries, the lithium-sulfur battery has been widely believed to be one of the most promising devices for next-generation energy storage systems due to its high theoretical energy density (2600 Wh kg<sup>-1</sup>), specific capacity (1675 mAh g<sup>-1</sup>) and the low price of raw materials [1,2]. However, there are still many problems that plague practical application of lithium-sulfur batteries, including the “shuttle effect” resulting from the dissolution of lithium polysulfide produced during the discharge/charge process and the low utilization of sulfur attributing to high resistivity of elemental sulfur and its other products.

In order to deal with these problems, a variety of efforts have been

made toward improving the electrochemical performances of Li-S batteries in recent years. The most popular strategy has been developing various types of carbonous materials to suppress the “shuttle effect” and promote inferior conductivity, including porous carbon [3–6], single-walled or multi-walled carbon nanotubes [7,8], active carbon [9,10], conductive polymers [11–14], graphene-based nanomaterials [15]. However, it should be indicated that the nonpolar conductive carbon materials has no sufficiency to diminish the polysulfide-shuttle effect due to the high-polar nature of polysulfides. The second strategy has been developing polar materials (metal sulfides [16–18] and metal oxides [19,20] et al.) to improve the electrochemical performances by strong absorbing lithium polysulfides during charge-discharge process, but there is a low utilization of sulfur because of their relatively high resistivity. The third effective strategy is to development hybrid composites of carbon and polar materials. For example, Yolk-

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Shelled C@Fe<sub>3</sub>O<sub>4</sub> and Co-VN@C showed excellent performances owing to the physical confinement of the diffused polysulfides [21,22].

Al<sub>2</sub>O<sub>3</sub> is found to trap polysulfides via producing a strong chemical reaction by Choi [23]. This electrodes containing nano-sized Al<sub>2</sub>O<sub>3</sub> exhibits higher discharge capacity than that of the sulfur electrode without nano Al<sub>2</sub>O<sub>3</sub>. Dong et al. [24] reported micron-sized flaky Al<sub>2</sub>O<sub>3</sub> as the adsorbent to reduce the dissolution of polysulfides into electrolytes. Besides these diverse structures, the designation and application of enclosed yolk-shell-structured  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C as host materials have not been reported to date. The yolk-shell structures displayed much improved cycle ability due to their benefits of low density, favourable specific surface area and enclosed structure characteristics.

Here,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres have been obtained using Congo red (CR) adsorption and in-situ thermal reduction methods. Benefiting from the unique yolk-shelled structure,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres has the following advantageous features: (i) This yolk-shell structure can encapsulate the polysulfides. (ii) The large void space of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C not only restricts large volume expansion during the cycling but also improves the sulfur utilization. (iii) There is a synergistic action among C and  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>. Moreover, the obtained  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres electrode showed excellent electrochemical performances and achieved long-cycling stability.

## 2. Experimental section

### 2.1. Preparation of $\gamma$ -Al<sub>2</sub>O<sub>3</sub> yolk-shell microspheres

In a typical preparation,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> yolk-shell microspheres were synthesized by a simple hydrothermal method. 40 mmol of glucose (C<sub>6</sub>H<sub>12</sub>O<sub>6</sub>), 30 mmol of urea (CO(NH<sub>2</sub>)<sub>2</sub>) and 7.5 mmol of aluminum ammonium sulfate (NH<sub>4</sub>Al(SO<sub>4</sub>)<sub>2</sub>·12H<sub>2</sub>O) were added to 140 mL of deionized water under stirring, the mixed solution was then transferred to a Teflon-lined stainless steel autoclave, followed by heated to 180 °C for 6 h in an oven. After cooling to room temperature, a brown product was separated by filtration and washed with deionized water and anhydrous alcohol several times, then it was dried in a vacuum oven at 60 °C for 12 h. Finally, the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> yolk-shell microspheres were obtained by the calcination for precursor at 600 °C for 3 h in air.

### 2.2. Preparation of $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres

The as-synthesized  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> yolk-shell microspheres (0.5 g) were mixed with different concentration of CR solution (100 mg/L, 100 mg/L, 300 mg/L, 500 mg/L) with magnetic stirring for 2 h. Subsequently, the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@CR yolk-shell microspheres were harvested through centrifugation and vacuum dried at 60 °C for 12 h. Finally, the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres (marked as  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C-100,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C-300 and  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C-500, respectively) were annealed at 600 °C for 2 h in N<sub>2</sub> with a heating rate of 2 °C min<sup>-1</sup>.

### 2.3. Preparation of the sulfur composite

Firstly, 0.3 g S, 0.08 g Super P and 0.02 g CoS<sub>2</sub> (the synthetic

process was presented in Supporting Information) were mixed sufficiently before putting in a covered crucible. Then the covered crucible was transferred to a 200 mL Teflon-lined stainless-steel autoclave and maintained at 155 °C for 6 h. The sulfur composite was obtained after cooling to room temperature.

### 2.4. Materials characterization

The phase and morphology structure were characterized by Powder X-ray diffraction (DX-2700 diffractometer with a Cu-K $\alpha$  radiation source), JSM-6490LV (JEOL, Japan) and Nova Nano SEM230 (FEI, America) scanning electron microscope.

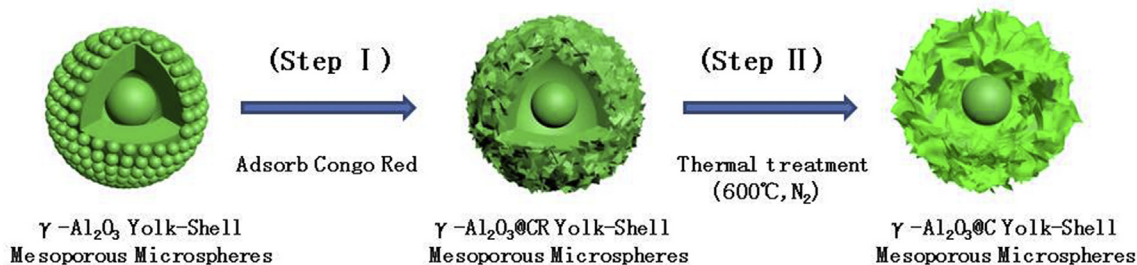
### 2.5. Electrochemical measurement

The Cyclic voltammetry (CV) and electrochemical impedance spectroscopy measurements were measured BTS4000 cell test instrument (Shenzhen Neware, China). Sulfur composite (70 wt %),  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C (−100, −300 and −500, respectively) yolk-shell (10 wt %), Super P (10 wt %) and polyvinylidene fluoride (PVDF) binder (10 wt %) were mixed in N-methyl-2-pyrrolidone to form a uniform slurry. This slurry was then casted onto aluminium foil using doctor blade and dried under a vacuum oven at 60 °C for 12 h to form the Li–S cathode. The electrolyte was made up of 1M LiTFSI, 0.1M lithium nitrate (LiNO<sub>3</sub>) in 1, 2-dimethoxyethane (DME) and 1, 3-dioxolane (DOL) (volume ratio 1:1). The batteries were assembled in an Ar-filled glove box. A loading mass of sulphur was 0.96 mg/cm<sup>2</sup>. CV curves were detected in the voltage range 1.5–3 V. The electrochemical impedance was tested in the frequency range of 0.01 Hz–100 kHz. As comparison, the sulfur cathodes containing  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> yolk-shell mesoporous microsphere (Sulfur composite: Al<sub>2</sub>O<sub>3</sub>: Super P: PVDF = 7:1:1:1) and sulfur cathode containing without Al<sub>2</sub>O<sub>3</sub> (Sulfur composite: Super P: PVDF = 7:2:1) were prepared in the same method and tested under same condition.

## 3. Results and discussion

The formation mechanism of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres is illustrated in Scheme 1. In the Step I, hierarchically  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@CR yolk-shell microspheres were prepared by the adsorption of Congo red on the surface of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> yolk-shell microspheres. In the step II, the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@CR yolk-shell microspheres were calcined at 600 °C in N<sub>2</sub>, resulting in the formation of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell mesoporous microspheres.

The crystal structures of as-prepared materials were investigated by X-ray diffraction (XRD). As shown in Fig. 1a, the XRD of three yolk-shell microspheres samples were identical, and all were indexed to the cubic  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (JCPDS Card 10–0425). The morphology and microstructures were then measured by SEM. Fig. 1b showed a typical SEM image of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> yolk-shell mesoporous microspheres, from which one can see that the obtained yolk-shell architecture with a diameter of 3–4  $\mu$ m and a granular surface. Although the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@CR yolk-shell microspheres (Fig. 1c) and  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell microspheres (Fig. 1d) retained an intact yolk-shell structure and similar size, there was an appearance of



Scheme 1. Schematic illustration of the synthesis of hierarchically  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>@C yolk-shell mesoporous microspheres.

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