



# Trivacancy and Stone-Wales defected silicene for adsorption of small gas molecules

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## ABSTRACT

The adsorption characteristics of small gas molecules ( $\text{CO}_2$ ,  $\text{CO}$ ,  $\text{H}_2\text{O}$ ,  $\text{N}_2$ ,  $\text{NO}$ ,  $\text{NH}_3$ ,  $\text{NO}_2$  and  $\text{O}_2$ ) on trivacancy and Stone-Wales defected silicene were investigated by density functional theory calculations. The results reveal that  $\text{CO}$ ,  $\text{H}_2\text{O}$  and  $\text{N}_2$  are absorbed on trivacancy silicene in a physical way via van der Waals forces, whereas  $\text{CO}_2$ ,  $\text{NO}$ ,  $\text{NH}_3$ ,  $\text{NO}_2$  and  $\text{O}_2$  are chemisorbed on trivacancy silicene via strong covalent (Si-bndN or Si-bndO) bonds. For Stone-Wales defected silicene,  $\text{N}_2$ ,  $\text{H}_2\text{O}$  and  $\text{CO}_2$  are physisorbed whereas  $\text{CO}$ ,  $\text{O}_2$ ,  $\text{NO}$ ,  $\text{NH}_3$ , and  $\text{NO}_2$  are chemisorbed. The chemisorption of gas molecules on trivacancy and Stone-Wales defected silicene widens the band gap. In particular, the  $\text{NO}_2$  chemisorption on trivacancy or Stone-Wales defective silicene shows significant hole doping. Additionally,  $\text{O}_2$  is found to be readily decomposed into two O atoms on trivacancy silicene due to the small energy barrier and large exothermic reaction heat. Our work offers a possible method to regulate the electronic properties of silicene for applications at the nanoscale.

## 1. Introduction

Two-dimensional materials such as graphene were found to have some special properties. For instance, with  $\text{sp}^2$  hybridization of carbon atoms, graphene has huge specific surface area ( $2630 \text{ m}^2/\text{g}$ ) [1,2]. Past investigators found that modifying the surface of the nanosheet with doping or defects is an important way to tune its electronic properties [3]. Since the discovery of graphene, some other two-dimensional materials such as silicene, phosphorene, transition metal carbide and nitride-based MXenes have been investigated [4]. As carbon and silicon are both in group-IV of the periodic table, silicene is expected to behave like graphene [5]. It is a highly ordered silicon structure, arranged within a honeycomb lattice, known to have  $\text{sp}^2$ - $\text{sp}^3$  mixed hybridizations, and can be prepared on Ag, Ir and  $\text{ZrB}_2$  substrates [6–8]. Theoretical studies showed that the defects can improve the electronic behavior of silicene [9,10], tailor thermal conductivity at room temperature [11], as well as improving magnetic response [12–14]. On the other hand, the adsorption behaviors of atoms or molecules on silicene have been theoretically explored to realize charge carrier doping [15], bandgap engineering [16,17], silicene-based catalysis [18], hydrogen separation [16], and use in field effect transistors [19,20] and gas sensing [21–24].

However, as far as we know, there are few systematic theoretical studies on the adsorption of small gas molecules on the surface of

defective silicene. In this study, density functional theory (DFT) calculations were carried out to compare the electronic properties of trivacancy and Stone-Wales defected silicene upon the adsorption of small gas molecules ( $\text{CO}_2$ ,  $\text{CO}$ ,  $\text{H}_2\text{O}$ ,  $\text{N}_2$ ,  $\text{NO}$ ,  $\text{NH}_3$ ,  $\text{NO}_2$  and  $\text{O}_2$ ). One of the benefits of the vacancy defects is that it can enrich the adsorption configuration of the small gas molecules on the surface of silicene monolayer [23]. In addition, Stone-Wales defect, which exists generally in carbon nanotubes, graphene and similar materials, involves the change of connectivity of two  $\pi$ -bonded atoms with their rotation by  $90^\circ$  relative to the midpoint of their bond. In these materials, the repositioning is anticipated to have an imperative influence on the chemical, electrical, and mechanical properties [25]. These factors motivated us to choose trivacancy and Stone-Wales defected silicene sheets as the subject of our study. For convenience, we refer trivacancy silicene as T-silicene and Stone-wales defected silicene as S-silicene hereafter. Our theoretical investigation on adsorption structures, energies and band structures would provide prior knowledge on the adsorptions of different gases on defected silicene before carrying out experiments.

## 2. Methodology

DFT calculations were performed using the Dmol<sup>3</sup> program package in Material Studio of Accelrys Inc. The geometries and total energies were based on Perdew-Burke-Ernzerhof (PBE) functional calculated

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with generalized-gradient approximation (GGA) [26–28]. The k-point of the Brillion zone is set as  $6 \times 6 \times 1$  for geometry optimization. The self-consistent field (SCF), which is used to control the electronic minimization, is set with a tolerance of  $10^{-6}$  a.u. Here, long-range dispersion correction of Grimme [26] is integrated to treat the adsorption systems. The van der Waals correction is introduced based on the DFT-D method of Grimme. In addition, climbing-image nudged elastic band (NEB) method in Quantum Espresso package [26] was implemented to investigate the dissociation of  $O_2$  on T-silicene. In Quantum Espresso, GGA with PBE functional was also chose to describe the exchange-correlation interaction. Van der Waals (vdW) correction was using the DFT-D method. The kinetic energy cutoff for wave functions and charge density were 30 Ry and 120 Ry given by standard solid state pseudopotentials [29]. The convergence threshold of  $10^{-5}$  Ry/Borr for total energy and  $10^{-4}$  Ry for total force were applied.

In our calculation, the periodic boundary condition is arranged with the supercell setting. A model of  $5 \times 5$  pristine silicene (P-silicene) containing 50 silicon atoms is built. A vacuum height of 20 Å is added above the silicene sheet, for avoiding the interaction between two adjacent silicene sheets in z-axis (perpendicular to the silicene sheet). The adsorption energies ( $E_a$ ) of the adsorption systems are calculated by

$$E_a = E_{\text{gas-silicene}} - (E_{\text{gas}} + E_{\text{silicene}}) \quad (1)$$

where  $E_{\text{gas-silicene}}$  donates the total energy of the gas molecular adsorbing on intrinsic or defective silicene sheet;  $E_{\text{gas}}$  and  $E_{\text{silicene}}$  donate the energies of gas molecule and pristine or silicene sheet with trivacancy and Stone-Wales defect, respectively. An adsorption system with large negative value of  $E_a$  is regarded as stable.

### 3. Results and discussion

#### 3.1. Band structures of T-silicene and S-silicene

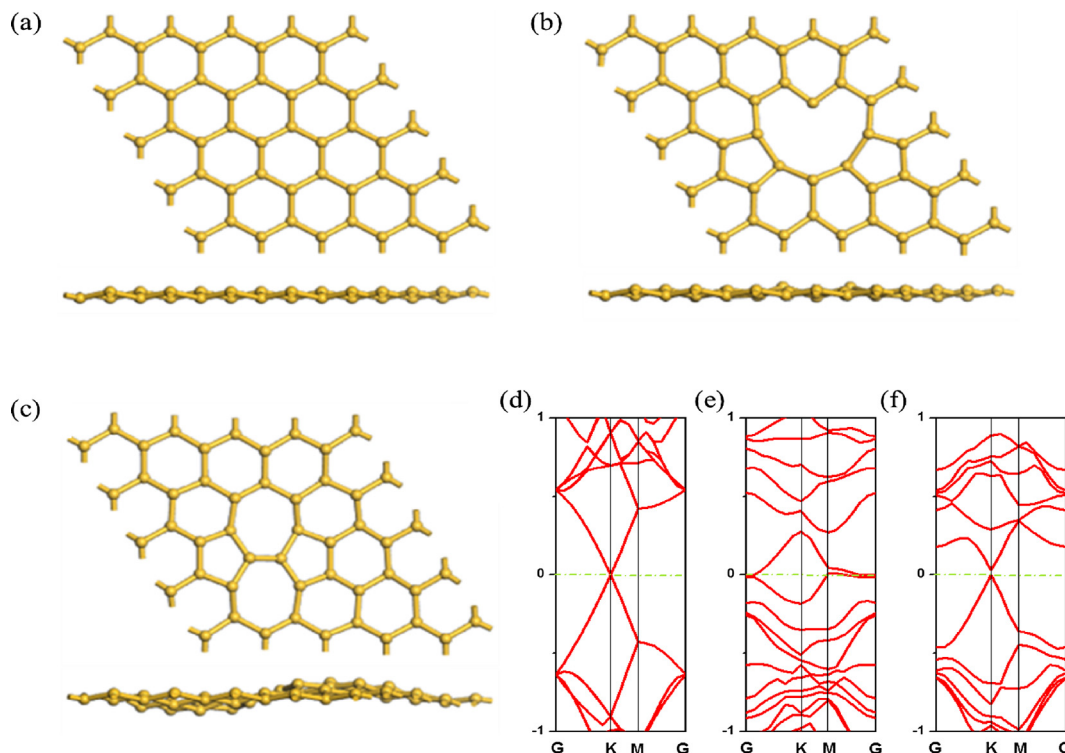
The P-silicene in a  $5 \times 5$  supercell after optimization is shown in Fig. 1a. It was found to have lattice constant of 3.860 Å and layer

thickness of 0.470 Å, and agreed well with previous theoretical results [6]. The T-silicene, created by removing three Si atoms in P-silicene, has the optimized structure shown in Fig. 1b. The stability of such vacancy defect with up to six missing atoms had already been studied in a previous work based on first-principle calculations [23]. The formation energy of our T-silicene is 5.88 eV, which agrees well with a previous study [23]. For S-silicene, the four adjacent six-membered Si rings of a pyrene-like region are changed into two five-membered and two seven-membered rings, as shown in Fig. 1c. The formation energy of S-silicene was found to be 2.21 eV, which is similar with former study [21].

Fig. 1d shows that there is a distinct Dirac point at the high symmetry point of P-silicene, reflecting its semi-metal performance. As shown in Fig. 1e, the Dirac point disappeared in T-silicene and exhibited an indirect bandgap. The valence band was moved to above the Fermi level due to the reduced electrons of the valence band after removing the three Si atoms. This phenomenon is consistent with previous study such that the Dirac points were easily changed position with importing defects or dopants in graphene and silicene [24]. The bandgap of T-silicene has a very small value of 0.002 eV. The band structure of S-silicene is quite different from that of P-silicene. As shown in Fig. 1f, although the bandgap is direct, the energy bands of S-silicene appear to be condensed, with bandgap of 0.026 eV.

#### 3.2. Energetics of the adsorption systems

Before investigating the adsorption effect of  $CO$ ,  $CO_2$ ,  $N_2$ ,  $H_2O$ ,  $NO$ ,  $NH_3$ ,  $NO_2$  and  $O_2$  on T- and S-silicene, we calculated the adsorption energy, charge transfer, shortest distance between the gas molecule and P-silicene, and the change of band gaps after adsorption of small gas molecules. Similar to previous studies [30,31], P-silicene was found to exhibit remarkably high activity toward  $O_2$  and  $NO_2$  with the adsorption energies being greater than 1.00 eV. Besides,  $NO$  and  $NH_3$  can be adsorbed on silicene with a mild adsorption energy (0.34 and 0.59 eV), whereas  $N_2$ ,  $H_2O$  and  $CO_2$  are physisorbed on P-silicene with weak van



**Fig. 1.** The structure of (a) pristine silicene (the side and top view, respectively); (b) trivacancy silicene; (c) Stone-Wales defective silicene; the band structure (with Fermi level set to zero) of (d) pristine silicene; (e) trivacancy silicene; (f) Stone-Wales defective silicene.

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