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Design of high resistivity light-rare-earth-based $\text{PrFe}_{1.93}$ magnetostrictive alloys: Si doping and high-pressure annealing

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Abstract

Light-rare-earth-based $\text{Pr}(\text{Fe}_{1-x}\text{Si}_x)_{1.93}$ alloys ($0 \leq x \leq 0.1$) with pure cubic Laves phase were synthesized by high-pressure annealing method. Si substitution effects on magnetic, electrical and magnetostrictive properties of light-rare-earth-based $\text{Pr}(\text{Fe}_{1-x}\text{Si}_x)_{1.93}$ alloys were systematically investigated. It was found that the lattice parameter a and Curie temperature T_C of the cubic Laves phase in the alloys increase with the increasing Si content up to $x = 0.05$, which might be ascribed to the preferential occupation of Si in the Laves phase interstitial sites. The magnetization at the maximum available field of 15 kOe, σ_{15k} , decreases monotonically with the increasing x . A significant increase of 67% in electrical resistivity was observed in $\text{Pr}(\text{Fe}_{0.9}\text{Si}_{0.1})_{1.93}$ alloy at room temperature. The magnetostriction at the field of 3 kOe of $\text{Pr}(\text{Fe}_{0.95}\text{Si}_{0.05})_{1.93}$ is about 542 ppm, which is even larger than the saturation magnetostriction of heavy-rare-earth-based $\text{Tb}_{0.2}\text{Dy}_{0.22}\text{Ho}_{0.58}\text{Fe}_2$ single crystal ($\lambda_s=530$ ppm). The attractive price, lower eddy current loss, together with high magnetostrictive response suggest that the $\text{Pr}(\text{Fe}_{0.95}\text{Si}_{0.05})_{1.93}$ alloy might be a good candidate material for the potential magnetostrictive applications.

Keywords: Cubic Laves phase, PrFe_2 , High-pressure annealing, Magnetostriction

1. Introduction

Giant magnetostrictive materials (GMMs), as modern smart materials, are playing increasingly important roles in many applications, such as ultra-sensitive sensors, high energy density actuators, acoustic transducers, etc[1, 2, 3, 4]. In order to exploit the giant magnetostriction at low field, it is necessary to minimize the magnetocrystalline anisotropy by combining two RFe_2 (R=rare earth) with opposite anisotropy signs.[1] The pseudobinary compounds $\text{Tb}_{0.27}\text{Dy}_{0.73}\text{Fe}_2$ and $\text{Tb}_{0.15}\text{Ho}_{0.85}\text{Fe}_2$ were firstly proposed as candidate materials for the magnetostrictive application[5, 6]. Based on that, further interest has been paid to the element-doped and multicomponent pseudobinary systems, for the purpose of minimizing the magnetocrystalline anisotropy and improving their magnetic, magnetostrictive and electrical properties[7, 8, 9, 10, 11,

12]. Recently, Wutting *et al.* have experimentally found that the easy magnetic direction (EMD) boundary of $\text{Tb}_{1-x}\text{Dy}_x\text{Fe}_2$ (Terfenol-D) between $\text{Ms}||[111]$ and $\text{Ms}||[100]$ indeed corresponds to the structure rhombohedral (R) and tetragonal (T) boundary, and the anisotropy compensation area could be physically analogous to ferroelectric **morphotropic phase boundary (MPB)**. [13] After that, many experimental and theoretical works on ferromagnetic MPB have being done.[14, 15, 16] However, the raw materials of these compounds mainly consist of expensive heavy-rare-earths Tb, Dy, or Ho. Hence, there is a need for searching new advanced magnetostrictive materials based on inexpensive light-rare-earths. According to the single ion model,[1] the spontaneous magnetostriction of PrFe_2 is as large as 5600 ppm at 0 K, which is even larger than that of TbFe_2 (4400 ppm) and DyFe_2 (4200 ppm). Many efforts were made for increasing the Pr content in magnetostrictive materials under ambient atmosphere. However, when the Pr content was over 30 at. %, many unanticipated PrFe_3 and $\text{Pr}_2\text{Fe}_{17}$ phases appear, which

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