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Water induced phase segregation in hydrocarbon proton exchange membranes

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ABSTRACT

Proton exchange membranes (PEMs) are a key material for proton exchange membrane fuel cells (PEM-FCs). Non-fluorinated hydrocarbon PEMs are low-cost alternatives to Nafion, but limited by the low proton conductivity, because of the weak phase segregation structure and narrow ion-transport channels. Various efforts have been taken to improve the performance of hydrocarbon PEMs, but mostly with complex methodologies. Here we demonstrate a simple, yet very efficient method to create phase segregation structure inside a typical hydrocarbon PEM, sulfonated poly(ether ether ketone) (SPEEK). By simply adding appropriate amounts of water into the DMF solvent, the resulting SPEEK membrane exhibits widened ion-transport channels, with the phase size of 2.7 nm, as indicated by both molecular dynamic (MD) simulations and transmission electron microscope (TEM) observations, and the proton conductivity is thus improved by 200%. These findings not only further our fundamental understanding of hydrocarbon PEMs, but are also valuable to the development of low-cost and practical fuel cell technologies.

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Proton exchange membrane fuel cells (PEMFCs) are one of the **Q2**₂ most promising energy conversion devices with zero emissions and 3 high efficiency [1–3]. After decades of development, PEMFCs are approaching commercialization, but the high manufacture cost of 4 5 key components is still a barrier to widespread applications [1]. One of the key components is the proton exchange membranes 6 (PEMs) [1,3-6]. Up to now, Nafion is a benchmark of PEMs owing 7 to its excellent chemical stabilities and high proton conductivity 8 9 [7]. However, the manufacture of perfluorocarbon polymers makes it very difficult to reduce its cost [5]. 10

To address this issue, much effort has been devoted to de-11 veloping non-fluorinated PEMs with stable hydrocarbon back-12 bones, such as sulfonated poly(ether ether ketone) (SPEEK), 13 sulfonated poly(arylene ether sulfone) (SPAES), and sulfonated 14 poly(phthalazinone ether sulfone ketone) (SPPESK) [8]. Although 15 16 cost has been reduced, the proton conductivity of these PEMs is relatively low, mostly because the phase segregation structure (ion 17 conduction channels) cannot be efficiently formed inside the poly-18 19 mer [9].

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Several strategies have been taken to build appropriate phase segregation structures for sulfonated hydrocarbon PEMs. One strat-21 egy is to design multi-block copolymers or flexible side chains 22 [10-12]. For example, McGrath et al. designed a 6FK-BPSH multi-23 block copolymer possessing a higher proton conductivity than 24 Nafion 112 over a wide range of relative humidity (RH) and tem-25 perature [10], the ionic groups were located closer to each other 26 in the multi-block copolymer than in random copolymer, facili-27 tating the phase segregation and resulting in larger domains and 28 greater differences in topography. Also, Pang et al. introduced a 29 flexible tetra-sulfonated side chain to the poly(aryl ether ketone) 30 membrane (TS-PAEK-25) [12], resulting in a quite clear phase-31 segregated morphology. The small-angle X-ray scattering (SAXS) 32 result showed that the morphology of the membrane has a phase 33 size of 8.17 nm, which was larger than that of Nafion (3.53 nm). 34 This unique phase-segregated structure resulted in an excellent 35 proton conductivity of 151 mS/cm at 100 °C. Another strategy is 36 to use additives, which seems to be a simpler method in com-37 parison to the synthesis of complex structures. Some additives, 38 such as PTFE, SiO₂, and heteropolyacid, have been reported to be 39 beneficial to the proton conductivity, durability and mechanical 40 stability [13]. For example, Dong reported an SPEEK/PANTs (poly 41 vinylphosphonic acid functionalized silica nanotube) composite 42 membrane which showed a high proton conductivity of 103 mS/cm 43 at low temperature (30 °C) [14]. The nanotubes (1240 nm in length) 44 distributed evenly within the SPEEK matrix and constructed 45

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Fig. 1. Atomistic MD simulation results for SPEEK membranes containing DMF (a) or H₂O/DMF (b) with a volume ration of 1:1.

46 continuous pathway that facilitates proton transportation along the 47 nanostructure. Although introducing additives to PEMs is consid-48 ered as a facile way to increase the phase segregation and pro-49 ton conductivity, there is always a compatibility issue between the 50 polymers and the additives, and additives may also aggregate in 51 the membrane and result in inhomogeneous membranes [13,14].

Unlike using the solid additives aforementioned, in this study, 52 we are interested in the influence of water to the phase segre-53 gation of sulfonated hydrocarbon PEMs. Water is abundant, envi-54 ronmental friendly, and easy to remove from membranes without 55 56 causing any inhomogeneous issue. We focused on the influence of water to the phase segregation of SPEEK, a promising proton-57 exchange membrane with stable backbone but week phase segre-58 59 gation structure [9,15].

Atomistic MD simulations were first applied to study the mor-60 61 phology of a SPEEK membrane containing DMF or H₂O/DMF mixture solvents. To determine the ratio between SPEEK and solvent 62 molecules in a simulation box, the SPEEK membrane was first 63 simulated under a fully wet state, namely, with a hydration level 64 $\lambda = 10$ (where λ denotes the number of water molecules per H₃O⁺) 65 [16]. The volume of all ion channels formed in such a simulated 66 wet membrane was taken as the space to be fully filled with sol-67 vent molecules in the following simulations. 68

The simulations were then started with the box containing SPEEK chains and solvent molecules (DMF or H_2O/DMF mixture) with above-determined volume ratio. After 30 ns of equilibration, phase segregation morphologies were generated. As demonstrated in Fig. 1, the resulting ionic channels in the DMF-containing SPEEK membrane seems to be narrow, and the DMF molecules were squeezed to line up in a wire state in the channels. By contrast, the ionic channels formed in the H_2O/DMF -containing SPEEK membrane are wide, inside which water molecules aggregate together with the sulfonate groups and hydronium ions. These results indicate that, after solvent evaporation, the H_2O/DMF -containing SPEEK membrane may solidify with a structure of enhanced phase segregation.

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Although the atomistic MD simulations are informative, the box 82 is too small to identify any long-ranged structural pattern possibly 83 generated by phase segregation. Coarse grained (CG) simulations 84 were thus employed to extend the space and time scales. Martini 85 CG force field [17] was used for simulating $3 \times 3 \times 3$ systems. After 86 equilibration, the CG systems were transformed back to atomistic 87 configurations and further equilibrated. The simulation results are 88 present in Fig. 2. Charged particles are not shown so as to have 89 a clear view of the ion channels. Stronger phase segregation with 90 wider ion channels in the membrane can be clearly observed in 91 SPEEK containing H₂O/DMF mixture (Fig. 2(b)) than that with DMF 92 (Fig. 2(a)), which is consistent to the atomistic MD simulation re-93 sults 94

To obtain structural information of the ion channels, DMF 95 molecules were removed from both configurations, and water 96 molecules were added to the channels to make the final hydration 97 level $\lambda = 10$. The refilled systems were then equilibrated. Structure 98 factors (SFs) of the hydrophilic species (H_2O , H_3O^+ , and $-SO_3^-$) 99 were then calculated based on their radial distribution functions 100 (RDFs) [18,19]. As shown in Fig. 2(c), the SF peak for the SPEEK 101 membrane formed with H₂O/DMF appears at $q = 2.3 \text{ nm}^{-1}$, with a 102 greater intensity than the SF peak for the membrane formed with 103



Fig. 2. CG simulation results for SPEEK membranes containing DMF (a) or H_2O -DMF (b) in nanometer scale. The hydrophobic parts are presented in grey. Structure factors (c) of hydrophilic species (H_2O , H_3O^+ , $-SO_3^-$) was calculated to quantify the phase segregation structures.

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