ARTICLE IN PRESS JID: JECHEM [m5G;June 27, 2018;7:38]

Journal of Energy [Chemistry](https://doi.org/10.1016/j.jechem.2018.06.008) xxx (2018) xxx–xxx

Contents lists available at [ScienceDirect](http://www.ScienceDirect.com)

Journal of Energy Chemistry

http://www.journals.elsevier.com/ journal-of-energy-chemistry/

journal homepage: www.elsevier.com/locate/jechem

Correlating Fe source with Fe-N-C active site construction: Guidance for rational design of high-performance ORR catalyst^{$\dot{\mathbf{x}}$}

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a r t i c l e i n f o

Article history: Received 10 April 2018 Revised 11 June 2018 Accepted 16 June 2018 Available online xxx

Keywords: Carbon-nitrogen-coordinated iron $(FeN₄)$ Oxygen reduction reaction Iron source Molecular size Hydrolysis

A B S T R A C T

Pyrolyzed Fe-N*X*/C materials derived from Fe-doped ZIF-8 are recently emerged as promising alternatives to noble metal platinum-based catalysts towards oxygen reduction reaction (ORR) and elucidating the dependacne of Fe source on the active site structure and final ORR performance is highly desirbale for further development of these materials. Here, we designed and synthesized a series of Fe-N-C catalysts using ZIF-8 and various iron salts (Fe(acac)₃, FeCl₃, Fe(NO₃)₃) as precusors. We found that the iron precursors, mainly the molecular size, hydrolysis extent, do play a major role in determining the final morphology of Fe, namely forming the Fe-N_{*x*} coordination or Fe₃C nanoparticles, as well as the site density, therefore, significantly affecting the ORR activity. Among the three iron sources, $Fe(acac)_3$ is most advantageous to the preferential formation of single-atom Fe-N*x* active sites and the derived catalyst demonstrated best ORR performance.

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1 **1. Introduction**

Fuel cell is regarded as one of the most promising clean en-2 ergy conversion devices due to its higher energy efficiency and lower emissions than internal combustion engines [\[1–4\].](#page--1-0) However, the high cost, which is mainly originated from the noble metal platinum-based cathode catalysts, seriously limits the practical ap- plication of PEM fuel cells [\[5\]](#page--1-0). To overcome this obstacle, devel- oping high-performance nonprecious metal catalysts (NPMCs) for oxygen reduction reaction (ORR) has been considered as the most effective solution and captured intensive attention in recent years [\[6–9\].](#page--1-0) A variety of NPMCs have been reported catalytically active to ORR, among which, carbon-supported transition metal-nitrogen co-

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ordinations $(M-Nx/C, M = Fe, Co)$ have been generally recognized as 13 the most promising candidates, ascribable to their superior activity 14 the most promising candidates, ascribable to their superior activity and stability in acidic electrolyte $[10-12]$. As the active sites M- 15 N_x are more inclined to formed in certain specific pore structures 16 of the carbon substrate [\[13,14\],](#page--1-0) and various microporous carbon 17 materials are developed to promote the formation of the desired 18 active sites [15-17], such as Ketjenblack [\[18\],](#page--1-0) BP 2000 carbons 19 [\[19\]](#page--1-0) and some microporous metal-organic frameworks [\[13,20,21\].](#page--1-0) 20 Especially, zeolite imidazole framework-8 (ZIF-8) is demonstrated 21 as an ideal host for the synthesis of M-N*x*/C due to its exception- 22 ally abundant micropores [\[14,22\],](#page--1-0) inherent presence of coordinated 23 M-N species and tunable structure at molecular level [\[23,24\].](#page--1-0) 24 Notably, the catalytic performance of the ZIF-8 derived Fe-N-C 25

catalysts differs greatly in the reported literatures [\[1,18,23,25–28\],](#page--1-0) 26 which may be caused by different synthetic parameters; however, 27 such effect is rarely studied. Elucidating the origins of the per- 28 formance difference is significant for further designing the highly 29 efficient catalysts. As is known to all, the metal content [\[18,29\],](#page--1-0) 30 the precursor concentration [\[18\],](#page--1-0) and the pyrolytic temperature 31 [\[26\]](#page--1-0) will all affect the final chemical composition and structure of 32 the resulted material, thus leading to the difference in their cat- 33 alytic property. Lai et al. demonstrate that the increase in iron con- 34 tent results in decrease of ORR activity, due to fact that inactive 35 iron species dominate at high iron content [\[25\].](#page--1-0) Wu's group in- 36 vestigates the dependence of particle size of Fe-doped-ZIF-8 and 37

<https://doi.org/10.1016/j.jechem.2018.06.008>

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The work is supported by the National Natural Science [Foundation](http://dx.doi.org/10.13039/501100001809) of China (21633008, 21433003, U1601211, 21733004), National Science and Technology Major Project (2016YFB0101202), Jilin Province Science and Technology Development Program (20150101066JC, 20160622037JC, 20170203003SF, 20170520150JH), Hundred Talents Program of Chinese Academy of Sciences and the Recruitment Program of Foreign Experts (WQ20122200077).

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* The imperfectly carbon-coated Fe₃C particles would be form due to adding too much FeCl₃ in precursor then be etched after pickling. It can be prvoed by TEM as shown in Fig. S5(c, d).

 pyrolytic temperature on the performance of the catalysts via con- trolling the precursors' concentration [\[26\].](#page--1-0) Previous study showed that iron–ligand coordination influences the fuel-cell performance [\[1,29,30\],](#page--1-0) however, the effect of iron precursor, which is closely re- lated with coordination nature and the final catalytic activity, is yet to be studied to date.

 Here, we designed and synthesized a series of Fe-N-C cata-45 lysts derived from ZIF-8 and various iron salts (Fe(acac)₃, FeCl₃, 46 Fe($NO₃$)₃). The structural and performance differences between the different samples were studied in-depth by combination of varied physical characterization techniques coupled with electrochemical measurements. From the study, it is found that although ZIF-8 is featured with high capability in synthesizing highly dispersed Fe- N*x* type of catalysts, the iron precursors do play a major role in determining the final forms of Fe, namely forming the Fe-N*x* coor-53 dination or $Fe₃C$ nanoparticles. Specifically, two properties of the Fe precursors, i.e., the molecular size and the hydrolysis feature of the iron source, are found crucial. Meanwhile, the ORR catalytic be- havior dependency on the Fe precursor sources and content were studied in detail.

58 **2. Experimental**

59 *2.1. Material synthesis*

 Typically, ZIF-8-FeX (FeX indicates a variety of iron sources) is 61 made by rapidly adding a 40 mL 0.1 M $\text{Zn}(\text{NO}_3)$ ₃ solution into the solution of 80 mL 0.2 M dimethylimidazole in methanol a specific iron source pre-added (list in Table 1) and stirring 24 h in room temperature. The obtained product was separated by centrifuga- tion and washed with methanol and finally dried at 60°C under vacuum for overnight, which was marked as ZIF-Fe-X-Y. The power of ZIF-Fe-X was then transferred into a ceramic boat and placed in a tube furnace. The sample was heated to 950°C with a heat-69 ing rate of 5°C·min⁻¹, kept at 950°C for 1 h under 10% H₂/Ar mix 70 gas. naturally cooled to room temperature, and then preleached in gas, naturally cooled to room temperature, and then preleached in 71 1 M HNO₃ to remove unstable species. The resultant was denoted as Czif-Fe-X-Y.

73 *2.2. Electrochemical measurement*

 All electrochemical measurements were conducted in a con- ventional three-electrode cell at room temperature (∼25°C) using the 750E Bipotentiostat (CH Instruments). Non-noble metal catalyst ink was prepared by ultrasonically dispersing 5 mg cat- alyst in a suspension containing 25 μL Nafion (5 wt%) solution and 475 μL ethanol; the catalyst film coated electrode was ob- tained by dispersing the catalyst ink on a glassy carbon rotating ring-disk electrode followed by drying in air. The catalyst load-82 ing on RRDE was 0.6 mg cm⁻² for Non-noble metal catalysts. And the comparison platinum was adopted as 20% Pt on Vulcan car-84 bon with an electrode Pt loading of 20 μ g·cm⁻² The ink was

Scheme 1. Schematic illustration of the preparation of the Czif- $Fe (acac)₃/FeCl₃/Fe (NO₃)₃$.

prepared by ultrasonically dispersing 5 mg catalyst in a suspen- 85 sion containing 500 μL isopropanol, 5 μL Nafion (5 wt%) solution 86 and 1.995 mL H_2O . And the catalyst film coated electrode was ob- 87 tained by dispersing the catalyst ink on a glassy carbon rotating 88 ring-disk electrode followed by drying in air with 500 rpm rotat- 89 ing. The ORR stability was investigated by continuous potential cy-
90 cling in oxygen-saturated 0.1 M $HClO₄$ solution between 0.6 V and 91 1.2 V with the scan rate at 0.2 V·s⁻¹. And after 5000/10,000 cy-
cles the ORR steady-state polarization measurements were concles, the ORR steady-state polarization measurements were conducted in O₂-saturated 0.1 M HClO₄ solution with scanning rates 94
of 5 mV-s⁻¹ and rotation rate at 1600 rpm. The ORR catalytic acof 5 mV·s⁻¹ and rotation rate at 1600 rpm. The ORR catalytic ac-
tivity can be further investigated based on the Kouteckye-Levich tivity can be further investigated based on the Kouteckye–Levich equations: 97

$$
\frac{1}{J} = \frac{1}{J_K} + \frac{1}{J_L} = \frac{1}{J_K} + \frac{1}{B\omega^{1/2}}
$$

B = 0.62nFC₀(D₀)^{2/3}v^{- $\frac{1}{6}$}

Where *J* is the measured current density, J_K is the kinetic cur- 99 rent density, J_L is the diffusion-limited current density, ω is the 100 electrode rotation rate, *F* is the Faraday constant (96,485 C/mol), 101 *C*₀ is the bulk concentration of O₂ (1.3 × 10⁻³ mol/L), *D*₀ is the dif- 102 fusion coefficient of O₂ (1.9 × 10⁻⁵ cm²/s) in acidic solution and v 103 fusion coefficient of O₂ (1.9×10^{-5} cm²/s) in acidic solution and *v* is the kinetic viscosity of the electrolyte $(1.0 \times 10^{-2} \text{ cm}^2/\text{s})$. 104

2.3. Materials characterization 105

The products were characterized using the below methods: 106 transmission electron microscopy (TEM) on a JEM-2100 transmis- 107 sion electron microscope (JEOL, Japan) operating at 120 kV; X-ray 108 diffraction (XRD) on a TD-3500 powder diffractometer (Tongda, 109 China) operating at 30 kV and 20 mA, using Cu *K*α radiation ¹¹⁰ sources; Nitrogen adsorption on ASIQCUF60U-5 (Quantachrome in- 111 struments, USA) at 77 K, scanning electron microscopy (SEM) on a 112 Merlin field emission SEM (Carl Zeiss); X-ray photoelectron spec- 113 troscopy (XPS) on a VG ESCALAB MK2 X-ray photoelectron spec- 114 trometer (VG corporation, UK), using an Al *K*α X-ray source. Fe ¹¹⁵

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