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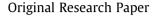
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A smart processing of silicon oxynitride ceramic powders with variable morphology controlled by hard template assistance

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ABSTRACT

Spherical and rod-like Si_2N_2O ceramic powders were successfully prepared using a hard template strategy to meet different application background. The mesoporous silica spheres and rod-like silica (SBA-15) particles were selected as hard templates and also served as reactants. Highly active mesoporous silica is reacted through a combination of sucrose-nano-casting and carbothermal reduction nitridation (CRN) to successfully synthesize spherical and rod-like powders. Temperature and holding time of calcination are found to be significant factors for the formation of pure and well-crystallized powders. Insufficient temperature and holding time cause an incomplete reaction between silica, carbon and nitrogen, and also lead to the formation of amorphous phase. In the processing, sucrose was used as carbon source to incorporate evenly into mesopores of templates to assist CRN reaction. Big surface areas and high porosity of mesopores in SiO₂ provide an additional driving force to decrease the formation temperature of Si_2N_2O phase to 1280 °C. As-synthesized powders inherit same morphology as their parental mesoporous templates, which indicate that two mesoporous templates play an important role in guiding the growth of ceramic particles. In the research, we develop a smart processing for synthesis of oxynitride powders with variable morphology controlled by template.

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1. Introduction

Si₂N₂O is a significant compound in silicon–oxygen–nitrogen system [1], as an advanced engineering material which can keep low thermal conductivity, excellent oxidation resistance in air, high flexural strength, and low dielectric constant and loss [2–8]. Having these excellent properties, Si₂N₂O has been a promising candidate in varied applications. For example, the powders with rod-like morphology or fiber will benefit the reinforcement of fracture toughness in engineering material matrix [9,10]. On the contrary, the powders with low aspect ratio will be more suitable to use in areas requiring relative low dielectric constant and loss [11]. Herein, synthesizing morphology-controllable Si₂N₂O powders crucially affects their further development and extended applications. As for particle shape control of ceramic powders, there are essentially three ways to approach: (1) growth directed syntheses typical of precipitation processes [11–15]; (2) template directed syntheses, wherein the growth is directed by epitaxy via a preexisting structure upon which nucleation and growth take place [16–18]; (3) parameters-adjusted wet chemical methods. [19,20]. However, among the three methods, the template directed way is relatively easy and simple to control, while the finding of effective templates is full of challenges.

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Since ordered mesoporous SiO_2 was successfully synthesized [21,22], the preparation and application of various mesoporous materials have attracted great attention of researchers, including SiO_2 -series [23–25], other oxides (Al_2O_3 , V_2O_5 , etc.) [26–28], and even non-oxides (carbon, ZnS, CdS, etc.) [29–31]. With hot focus on mesoporous materials of SiO_2 (such as MCM-type, SBA-type, and KIT-type), owing to their good stability and diversity of selectable structural-guide agents, hard template directed methods using mesopore channels of SiO_2 have shown the effectiveness in directing particle growth. To date, considerable works have devoted to control macrostructure/microstructure of different products using hard template approach. Kim and Cho [32] reported

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the preparation of mesoporous SnO_2 lithium battery anode materials using KIT-6 and SBA-15 templates (two kinds of mesoporous SiO_2), in which the shapes and pore sizes of SnO_2 changed with different templates. Wang et al. [33] employing two kinds of mesoporous silica (SBA-15 and spheres) synthesized morphologycontrollable Co_3O_4 and CeO_2 . The Mesoporous MFI zeolite using CMK carbon templates was reported recently [34]. In a word, the hard template method with morphology flexibility from a copy of template structure can readily realize the precise morphological control. Based on advantages mentioned above, the "migrating" of hard template method into preparation of ceramic powder will potentially solve the difficulties in precisely morphology controlling and limited kinds of powder shapes in present methods.

Very importantly, carbon can be readily introduced into the nanosized channels or skeletons of mesoporous SiO₂. Previous studies have also demonstrated that in situ composited carbon in mesoporous SiO₂ and large specific surface area of mesoporous SiO₂ are greatly helpful to decrease the synthesis temperature of pure α -Si₃N₄ and β -SiAlON powders through carbothermal reduction and nitridation method (CRN), utilizing SBA-15 powders as starting active SiO₂ raw materials [35,36].

Predictably, using SiO₂ hard template for synthesis of morphological controlled Si₂N₂O powder would give a novel technical method to further extend the applications of Si₂N₂O, in which SiO₂ serves as not only a structural guide agent but also Si source. In the present work, by utilizing two different kinds of mesoporous silica hard template (SBA-15 and spheres), rod-like and spherical Si₂N₂O powders were successfully synthesized at lower temperature of 1280 °C.

2. Experimental

2.1. Materials preparation

AR grade sucrose (as carbon source), sulfuric acid, tetraethyl orthosilicate (TEOS, Si source), ammonia, cetyltrimethylammonium bromide (CTAB, structural guide agent), and C_2H_5OH were purchased from Sinopharm Chemical Reagent Co., Ltd. (Shanghai, China), and used without further purification. Additive Y_2O_3 (99.9 wt%) was purchased from Shanghai Yuelong Rare Earth New Materials Co. Ltd (Shanghai, China). The rod-like mesoporous silica SBA-15, one of hard templates, was purchased from Fudan University of China and mesoporous silica sphere (MSS), another hard template, was prepared in our laboratory. Before usage as templates, SBA-15 and MSS were fired at 550 °C for 6 h to remove the remaining impurities.

To make a comparison with rod-like SBA-15 hard-template prepared in rigid bolaform surfactants by Zhao et al. [37]. MSSs were synthesized in our laboratory using a similar method reported by Teng [38], aiming at preparation of spherical Si_2N_2O . Generally, ethanol-to-water volume ratio was selected as 0.56 with an aqueous ammonia solution (25 wt% NH₃ in water). Then, TEOS was added drop-like into the above solution with stirring time of 12 h.

As for the synthesis of Si_2N_2O powders, SBA-15 and the assynthesized MSS particles were used as silica source and hard templates to control the morphology of final products. Sucrose was selected as carbon source, with a C:SiO₂ molar ratio of 4:1. A typical procedure of processing is as follows: 1.8 g sucrose and 0.2 g H₂SO₄ were dissolved in deionized water and 1.0 g SiO₂ (SBA-15 or MSS) was added subsequently, followed with stirring and drying. Afterwards, Y₂O₃ was added in the amount of 3.0 mol% (relative to the mesoporous silica source) and the solution was stirred for 30 min. After carbonized at 800 °C for 4 h in nitrogen flow of 0.3 L/min, the mixture powders form a SiO₂–C composite with carbon introduced into the nanosized channels or skeletons of mesoporous SiO₂ (Fig. 1a and b). This composite consequently underwent further nitridation in a graphite furnace at 1250– 1300 °C for 5–7 h with a 0.8 L/min nitrogen flow. The residual carbon was removed by heating at 600 °C for 6 h in air environment. It has to be emphasized herein that Y_2O_3 assistant is required to be added before carbonization to obtain a stable and highly-pure Si₂N₂O phase. The role of Y_2O_3 additive has been explained in a separate literature [39]. Generally, the addition of Y_2O_3 guarantees the CRN reaction occurred below 1300 °C due to the formation of an eutectic intermediate phase as Y_2SiO_5 , which was the reaction product of Y_2O_3 and mesoporous SiO₂.

2.2. Characterization

Nitrogen adsorption–desorption isotherms were measured on a Micromeritics ASAP2010 surface area and pore size analyzer at liquid nitrogen temperature (77 K). Prior to measurements, the samples were dehydrated at 373 K and then outgassed at 473 K in vacuum for 4 h. The Brunauer–Emmett–Teller (BET) method was utilized to calculate the specific surface areas (S_{BET}). The pore volume (V_{BJH}) and the mean pore size (D_{BJH}) were derived from the adsorption branches of the isotherms using the Barrett–Joyner–H alanda (BJH) method. X-Ray powder diffraction patterns were collected on a Rigaku D/MAX-c β instrument using Cu K α_1 ($\lambda = 0.15406$ nm) radiation at 40 kV and 60 mA. Morphology of the particles was observed with field emission scanning electron microscope (FSEM, JSM-6700F JEOL, Japan, 10.0 kV) equipped with EDS analysis.

3. Results and discussions

3.1. Texture of hard templates

To identify the difference between rod-like SBA-15 particles and mesoporous silica spheres (MSS), the textural parameters of two templates were analyzed using nitrogen adsorption–desorption isotherms and results are listed in Table 1. Both templates possess relatively high BET surface areas and relatively high pore volume. While MSS has a higher surface about 994 m²/g and SBA-15 possess a bigger pore diameter about 6.6 nm. The particle morphology of two templates is shown in Fig. 1c and d.

3.2. Phase formation of Si₂N₂O

To compare the different phase formation behavior of Si₂N₂O by two hard template guiding, mesoporous SBA-15 and MSS were used either as silica sources or as templates. And XRD analysis was carried out to record the phase formation of Si₂N₂O resulted from the two mesoporous SiO₂ templates. However, based on our experimental results, only SBA-15 derived Si₂N₂O phase formation is discussed in detail herein, because of a very similar phase formation and reaction mechanism shown from the two mesoporous silica precursors as starting materials.

3.2.1. Calcination temperature effect

Fig. 2 provides XRD patterns of final powders Si_2N_2O with SBA-15 and MSS as starting material of SiO_2 , calcined at 1250 °C, 1260 °C, 1280 °C, and 1300 °C, respectively, for 7 h. As the temperature of CRN increases, SBA-15 is gradually transformed into Si_2N_2O (JCPDS Card No. 47-1627). When the temperature is below 1280 °C, the unreacted amorphous SBA-15 clearly exists, indicated by a big "broad" diffraction band. When the calcination temperature or CRN temperature is above 1280 °C, considerable a-Si₃N₄ phase appears as a detectable impurity in the matrix of Si_2N_2O . As a result, the optimal temperature is fixed at 1280 °C. A phase

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