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Letter 2

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Solution-processed single-crystalline organic transistors 6 4 7 on patterned ultrathin gate insulators

8 Q1 Junto Tsurumi ^{a,b}, Atefeh Yousefi Amin ^{c,d}, Toshihiro Okamoto ^{b,e}, Chikahiko Mitsui ^b,
9 Kazuo Takimiya ^f, Hiroyuki Matsui ^{b,*}, Marcus Halik ^{d,*}, Jun Takeya ^{b,*}

10 ^a Department of Applied Physics, Osaka University, 2-1 Yamadaoka, Suita 565-0871, Osaka, Japan

11 ^b Department of Advanced Materials Science, Graduate School of Frontier Sciences, The University of Tokyo, 5-1-5 Kashiwanoha, Kashiwa 277-8561, Chiba, Japan

12 ^c Plastic Logic GmbH, An der Bartlake 5, 01109 Dresden, Germany

13 ^d Organic Materials and Devices, Institute of Polymer Materials, Department of Materials Science, Friedrich-Alexander-University of Erlangen-Nürnberg,

14 Martensstraße 7, 91058 Erlangen, Germany

15 e PRESTO, JST, 4-1-8 Honcho, Kawaguchi, Saitama 332-0012, Japan

16 ^f Emergent Materials Department, Advanced Science Institute, RIKEN, Wako, Saitama 351-0198, Japan

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ABSTRACT

Single-crystalline organic transistors of 3,11-didecyl-dinaphtho[2,3-d:2',3'-d']benzo 29 30 [1,2-b:4,5-b']dithiophene (C₁₀-DNBDT-NW) and 2,9-didecyl-dinaphtho[2,3-b:2',3'-f] thieno[3,2-b]thiophene (C10-DNTT) were fabricated by solution processes on top of the 31 patterned hybrid ultrathin gate dielectrics consisting of 3.6 nm-thick aluminum oxide 32 and self-assembled monolayers (SAMs). Due to the excellent crystallinity of the channel 33 films, bottom-gate and top-contact field-effect transistors exhibited the average field-34 effect mobility of 3.7 cm²/V s and 4.3 cm²/V s for C_{10} -DNBDT-NW and C_{10} -DNTT, respec-35 36 tively. These are the first successful devices of solution-processed single-crystalline 37 transistors on ultrathin gate dielectrics with the mobility above $1 \text{ cm}^2/\text{V}$ s, opening the way to develop low-power-consumption and high-performance printed circuits. 38 39

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42 1. Introduction

The development of organic thin-film transistors (OTFTs) 43 44 has attracted great interest for next-generation flexible de-45 vices such as electric papers and conformable sensor arrays [1]. In particular, the reports about high-mobility OTFTs 46 beyond $10 \text{ cm}^2/\text{V} \text{ s}$ [2–4] raised the expectation for the 47 48 application of the organic semiconductors to high-speed 49 logic circuits such as radio-frequency identification tags, flexible display drivers and wearable computers [5]. A 50 straightforward approach to realizing high-performance 51 52 OTFTs is to use single-crystalline organic semiconductor 53 channels, since the grain boundaries are known to reduce

O2 * Corresponding authors. Tel.: +81 4 7136 3763; fax: +81 4 7136 3790 (H. Matsui).

E-mail address: h-matsui@k.u-tokyo.ac.jp (H. Matsui).

http://dx.doi.org/10.1016/j.orgel.2014.02.028 1566-1199/© 2014 Published by Elsevier B.V. carrier mobility significantly as well as device stability and 54 reproducibility. Several groups have already succeeded to 55 fabricate quite large crystalline domains of organic semi-56 conductors by solution processes, and all these OTFTs exhib-57 ited considerably high mobilities [3,6,7]. However, the 58 application of these methods has been limited so far to 59 the preliminary devices where the crystals were grown on 60 the smooth and homogeneous surface of SiO₂/Si wafers. In 61 order to realize logic circuits and active matrix for displays, 62 the single-crystalline films need to be formed on top of 63 substrates with patterned gate electrodes while the inho-64 mogeneous surface energy, topography and/or morphology 65 of the substrates may cause random nucleation of crystals. 66 Thus it is a challenging issue to form single-crystalline films 67 on top of the patterned substrates by suppressing the 68 undesirable nucleation of crystals. 69

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Edge casting is one of the methods to grow large crys-70 talline domains of organic semiconductor films from a 71 72 small amount of solution [6]. The method uses a solu-73 tion-holding piece on substrates in order to control the 74 drying direction of the solution. Because the crystal nucle-75 ation occurs only at one side of the solution droplet, inch-76 size crystal domains along the crystal growth direction can 77 be obtained [8]. One unique feature of edge cast is that the 78 crystallization occurs at the liquid/air interface slightly above the contact line; Thus, it is expected that the nucle-79 ation of crystals should not be affected as much as other 80 solution methods. It is also important to combine this 81 method with the ultrathin gate insulators consisting of a 82 83 few nm of aluminum oxide and self-assembly monolayers (SAMs) [9-12], since the reduction of operation voltage 84 85 down to a few volts should be essential for realizing mobile devices driven by batteries or wireless power sources [13]. 86

87 In this paper, we demonstrate the solution process of 88 single-crystalline OTFTs on top of a variety of ultrathin gate dielectrics. High performances were successfully obtained 89 with two kinds of organic semiconductors, 3,11-didecyl-90 91 dinaphtho[2,3-d:2',3'-d']benzo[1,2-b:4,5-b']dithiophene (C 92 10-DNBDT-NW, Fig. 1(a)) [8] and 2,9-didecyl-dinaphtho [2,3-*b*:2',3'-*f*]thieno[3,2-*b*]thiophene (C₁₀-DNTT, Fig. 1(b)) 93 94 [14], and with four kinds of gate dielectrics by edge cast method. The highest mobility of 5.2 cm²/V s was achieved 95 at the small gate voltage of 2 V for the combination of C_{10} -96 DNTT and AlO_x without SAM. The variation in device prop-97 98 erties is also discussed in terms of the film crystallinity and the surface property of gate insulators. 99

100 2. Experimental

101 Aluminum gate electrode was deposited on silicon/ 102 thermal silicon oxide (500 nm) substrate by thermal evap-103 oration under 10^{-7} mbar through a shadow mask [4]. 104 Approximately 3.6 nm of aluminum oxide was formed by

O2 plasma to fabricate hybrid dielectric. The substrates 105 were then immersed in the 2-propanol solutions of SAMs 106 about 24 h: 0.2 mM 16-phosphonohexadecanoic acid 107 (PHDA, Fig. 1(c)), 0.05 mM 11-hvdroxyundecylphosphonic 108 acid (HO–C₁₁–PA, Fig. 1(d)), 0.1 mM phenylphosphonic 109 acid (PhPA, Fig. 1(e)), 0.05 mM tetradecylphosphonic acid 110 (C₁₄-PA, Fig. 1(f)), and 0.05 mM 12,12,13,13,14,14,15, 111 15,16,16,17,17,18,18,18H-pentadecafluoro-octadecyl phos 112 phonic acid (F₁₅C₁₈–PA, Fig. 1(g)) [11]. The bare aluminum 113 oxide gate insulator was also used for comparison. The 114 capacitances were measured at 0.88 µF/cm² for PHDA. 115 $0.75 \,\mu\text{F/cm}^2$ for HO-C₁₁-PA, $1.1 \,\mu\text{F/cm}^2$ for PhPA and 116 $1.27 \,\mu\text{F/cm}^2$ for bare AlO_x, which were measured with 117 the Au/insulator/Al structure of 50 \times 50 μm in size. 118

We used C₁₀-DNBDT-NW and C₁₀-DNTT as channel 119 materials. The single crystals of the organic semiconduc-120 tors were grown on the hybrid dielectrics by the edge-cast 121 method [15] as shown in Fig. 2(a). C₁₀-DNBDT-NW and 122 C10-DNTT were dissolved in 1,2-dimethoxybenzene at 123 0.07 wt% and in tetralin at 0.05 wt%, respectively. Then 124 the solution was dropped near the edge of a glass piece 125 which was placed on the substrate to hold the solution 126 [16]. As the solvent evaporated, the edge of the droplet 127 shifted toward the glass piece so that the direction of 128 the crystal growth is perpendicular to the edge of the 129 glass piece. The processes were carried out on a hot plate 130 kept at 125 °C for C₁₀-DNBDT-NW and at 100 °C for 131 C_{10} -DNTT. After the crystal growth, the films were an-132 nealed in vacuum oven at 100 °C under the pressure of 133 10 mbar in order to remove the remaining solvent. The 134 thickness of the semiconductor films was estimated at 135 10-30 nm by atomic force microscopy (AFM). Finally, 136 30 nm-thick gold electrodes for source and drain were 137 thermally evaporated in vacuum through shadow masks. 138 The channel length was designed to be 20–300 µm, and 139 the channel width 500 µm. The device performance was 140 measured in air by semiconductor device analyzer 141 (B1500A, Agilent). 142



Fig. 1. Molecular structures of (a) C₁₀-DNTT, (b) C₁₀-DNBDT-NW, (c) PHDA, (d) HO-C₁₁-PA, (e) PhPA, (f) C₁₄-PA, and (g) F₁₅C₁₈-PA.

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