

# Phase formation in the systems $\text{Li}_2\text{MoO}_4\text{--K}_2\text{MoO}_4\text{--Ln}_2(\text{MoO}_4)_3$ ( $\text{Ln} = \text{La}, \text{Nd}, \text{Dy}, \text{Er}$ ) and properties of triple molybdates $\text{LiKLn}_2(\text{MoO}_4)_4$

O.M. Basovich<sup>a</sup>, E.G. Khaikina<sup>a,\*</sup>, S.F. Solodovnikov<sup>b</sup>, G.D. Tsyrenova<sup>a</sup>

<sup>a</sup>*Baikal Institute of Nature Management, Siberian Branch, Russian Academy of Sciences, Sakh'yanova Street 6, Ulan-Ude, 670047 Buryat Republic, Russia*

<sup>b</sup>*Nikolaev Institute of Inorganic Chemistry, Siberian Branch, Russian Academy of Sciences, Akad. Lavrent'ev Ave. 3, Novosibirsk 630090, Russia*

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## Abstract

Subsolidus phase relations in the systems  $\text{Li}_2\text{MoO}_4\text{--K}_2\text{MoO}_4\text{--Ln}_2(\text{MoO}_4)_3$  ( $\text{Ln} = \text{La}, \text{Nd}, \text{Dy}, \text{Er}$ ) were determined. Formation of  $\text{LiKLn}_2(\text{MoO}_4)_4$  was confirmed in the systems with  $\text{Ln} = \text{Nd}, \text{Dy}, \text{Er}$  at the  $\text{LiLn}(\text{MoO}_4)_2\text{--KLn}(\text{MoO}_4)_2$  joins. No intermediate phases of other compositions were found. No triple molybdates exist in the system  $\text{Li}_2\text{MoO}_4\text{--K}_2\text{MoO}_4\text{--La}_2(\text{MoO}_4)_3$ . The join  $\text{LiLa}(\text{MoO}_4)_2\text{--KLa}(\text{MoO}_4)_2$  is characterized by formation of solid solutions.

Triple molybdates  $\text{LiKLn}_2(\text{MoO}_4)_4$  for  $\text{Ln} = \text{Nd--Lu}, \text{Y}$  were synthesized by solid state reactions (single phases with ytterbium and lutetium were not prepared). Crystal and thermal data for these molybdates were determined. Compounds  $\text{LiKLn}_2(\text{MoO}_4)_4$  form isostructural series and crystallized in the monoclinic system with the unit cell parameters  $a = 5.315\text{--}5.145 \text{ \AA}$ ,  $b = 12.857\text{--}12.437 \text{ \AA}$ ,  $c = 19.470\text{--}19.349 \text{ \AA}$ ,  $\beta = 92.26\text{--}92.98^\circ$ . When heated, the compounds decompose in solid state to give corresponding double molybdates. The dome-shaped curve of the decomposition temperatures of  $\text{LiMLn}_2(\text{MoO}_4)_4$  has the maximum in the  $\text{Gd--Tb--Dy}$  region.

While studying the system  $\text{Li}_2\text{MoO}_4\text{--K}_2\text{MoO}_4\text{--Dy}_2(\text{MoO}_4)_3$  we revealed a new low-temperature modification of  $\text{KDy}(\text{MoO}_4)_2$  with the triclinic structure of  $\alpha\text{-KEu}(\text{MoO}_4)_2$ <sup>1</sup> ( $a = 11.177(2) \text{ \AA}$ ,  $b = 5.249(1) \text{ \AA}$ ,  $c = 6.859(1) \text{ \AA}$ ,  $\alpha = 112.33(2)^\circ$ ,  $\beta = 111.48(1)^\circ$ ,  $\gamma = 91.30(2)^\circ$ , space group  $P\bar{1}$ ,  $Z = 2$ ).

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## 1. Introduction

Existence of the extensive family of the isostructural triple molybdates  $\text{LiMR}_2(\text{MoO}_4)_4$  ( $M = \text{K}, \text{Tl}, \text{Rb}$ ;  $R = \text{Bi}, \text{Ln}$ ) was early reported in [1–4]. Their stability fields shift toward larger  $R^{3+}$  cations along with increasing  $M^+$  radius and include Bi, Nd–Lu, Y for the potassium-containing series; Bi, Ce–Eu for the

thallium-containing row, and Bi, La–Eu for the rubidium-containing one. The most extensive group of the  $\text{LiMR}_2(\text{MoO}_4)_4$  family is the potassium-containing compounds (13 of 26 representatives). However, whereas information on phase formation in the systems  $\text{Li}_2\text{MoO}_4\text{--}M_2\text{MoO}_4\text{--}R_2(\text{MoO}_4)_3$  ( $M = \text{Tl}, \text{Rb}$ ) is rather completed [4–6], data for  $M = \text{K}$  are limited by the system containing bismuth [4].

To fill this gap we undertook the investigation of phase relations in a number of the systems  $\text{Li}_2\text{MoO}_4\text{--K}_2\text{MoO}_4\text{--Ln}_2(\text{MoO}_4)_3$  that allowed us to observe a pattern of their changes depending on the nature of a rare-earth element.

\*Corresponding author. Fax: +73012434753.

E-mail address: [ekha@binm.bsc.buryatia.ru](mailto:ekha@binm.bsc.buryatia.ru) (E.G. Khaikina).

<sup>1</sup>Hereinafter,  $\alpha$  refers to a low-temperature modification.

Numerous publications are devoted to phase relations in the boundary binary systems of such triple systems. The phase diagram of the system  $\text{Li}_2\text{MoO}_4\text{--K}_2\text{MoO}_4$  was constructed in [7,8]. According to these data the system is characterized by formation of the only intermediate phase  $\text{LiKMoO}_4$  crystallizing in four polymorphous modifications [9]. Phase equilibria in the systems  $M_2\text{MoO}_4\text{--Ln}_2(\text{MoO}_4)_3$  ( $M = \text{Li, K}$ ) are rather complex and change significantly depending on the rare-earth element and the structure of its normal molybdate.

In the systems  $\text{Li}_2\text{MoO}_4\text{--Ln}_2(\text{MoO}_4)_3$  ( $\text{Ln} = \text{La--Tb}$ ) between  $\text{Ln}_2(\text{MoO}_4)_3$  and  $\text{LiLn}(\text{MoO}_4)_2$  with distorted and undistorted scheelite structures, respectively, intermediate phases were found which have an approximate composition  $\text{LiLn}_5(\text{MoO}_4)_8$  with monoclinic scheelite superstructures and wide homogeneity regions. Homogeneity regions of the compounds with an approximate 1:1 composition are also significant. Solid solutions on the base of 1:1 and 1:5 compounds are decomposed peritectically. Subsolidus phase transitions of these compounds are not revealed [10], except of  $\text{LiLa}(\text{MoO}_4)_2$  undergoing an irreversible polymorphous transformation [10,11]. Only  $\beta\text{-LiLa}(\text{MoO}_4)_2$  possessing the tetragonal scheelite-type structure forms solid solutions [12]. The authors of the latter work suppose that  $\alpha\text{-LiLa}(\text{MoO}_4)_2$  has no appreciable homogeneity region. No intermediate phases were revealed in the lithium containing systems with  $\text{Ln} = \text{Dy--Lu, Y}$  [10] in the range of 50–100 mol%  $\text{Ln}_2(\text{MoO}_4)_3$ . Formation of incongruently melting compounds  $\text{LiLn}(\text{MoO}_4)_2$  and  $\text{Li}_7\text{Ln}_3(\text{MoO}_4)_8$  with neither noticeable homogeneity regions, nor polymorphism was established. In the system  $\text{Li}_2\text{MoO}_4\text{--Dy}_2(\text{MoO}_4)_3$ , the monotectic reaction  $L_2 \leftrightarrow L_1 + \beta\text{-Dy}_2(\text{MoO}_4)_3$  was detected within the range 40–50 mol%  $\text{Li}_2\text{MoO}_4$ .

The systems  $\text{K}_2\text{MoO}_4\text{--Ln}_2(\text{MoO}_4)_3$  are characterized by formation of a greater number of double molybdates such as  $\text{K}_5\text{Ln}(\text{MoO}_4)_4$  and  $\text{KLn}(\text{MoO}_4)_2$  [13–15]. Compounds  $\text{K}_5\text{Ln}(\text{MoO}_4)_4$  have no distinct homogeneity regions. While there is a significant structural similarity between  $\text{KLn}(\text{MoO}_4)_2$  and  $\text{Ln}_2(\text{MoO}_4)_3$  based on the scheelite-type structure, 1:1 compounds possess rather extensive homogeneity regions. In this case, as well as in the analogous lithium–lanthanide systems, formation of scheelite-type solid solutions on the base of the phases with an approximate 1:5 composition was determined. When studying the systems with neodymium and europium [13,14] double molybdates  $\text{KLn}_5(\text{MoO}_4)_8$  were obtained. The authors of [13] suggest that  $\text{KLn}_5(\text{MoO}_4)_8$  should exist for the other rare-earth elements forming simple and double molybdates of a 1:1 composition with scheelite-like structures. Nevertheless, despite of scheelite-like structures of  $\text{KLa}(\text{MoO}_4)_2$  and  $\text{La}_2(\text{MoO}_4)_3$ , the existence of a similar phase in the potassium–lanthanum system is still

uncertain. As it was shown in [15], where  $T\text{--}x$  diagram of the system  $\text{K}_2\text{MoO}_4\text{--La}_2(\text{MoO}_4)_3$  was constructed using X-ray powder diffraction and thermal analysis, the obtained results suggested a possible formation of both the intermediate 1:5 phase and the extended solid solution on the base of  $\alpha\text{-KLa}(\text{MoO}_4)_2$ . According to [15], only preparation and XRD study of single crystals containing 50 and 83 mol%  $\text{La}_2(\text{MoO}_4)_3$ , as well as comparing the types of their superstructures allow us to make the unequivocal conclusion about existence of the 1:5 phase and the character of the phase diagram of this system. In [16,17], the crystals of  $\alpha\text{-}$  and  $\beta\text{-KLa}(\text{MoO}_4)_2$  were obtained with hydrothermal crystallization, but their complete structure study was not conducted. The authors of [17] determined that the higher stability limit of the  $\alpha\text{-}$ modification lies within 560 and 565 °C and indexed the powder pattern of this modification in a body-centered monoclinic subcell with the parameters  $a = 5.437(1)\text{ \AA}$ ,  $b = 12.205(2)\text{ \AA}$ ,  $c = 5.417(1)\text{ \AA}$ ,  $\beta = 90.05(1)^\circ$ . However, they failed to index a group of the first superstructure reflections. No information on the synthesis of single crystals of  $\text{KLa}_5(\text{MoO}_4)_8$  and its crystal structure is available in literature.

This paper presents results of our studies of subsolidus phase formation in the systems  $\text{Li}_2\text{MoO}_4\text{--K}_2\text{MoO}_4\text{--Ln}_2(\text{MoO}_4)_3$  ( $\text{Ln} = \text{La, Nd, Dy, Er}$ ) and physicochemical properties of triple molybdates  $\text{LiKLn}_2(\text{MoO}_4)_4$  for  $\text{Ln} = \text{Nd--Lu, Y}$ .

## 2. Experimental

As starting materials molybdenum trioxide, lithium and potassium carbonates (all reagent grade) and rare-earth oxides with content of the major substance more than 99.9% were taken. Molybdates  $M_2\text{MoO}_4$  ( $M = \text{Li, K}$ ) and  $\text{Ln}_2(\text{MoO}_4)_3$  were prepared by solid-state reactions. To avoid a loss of  $\text{MoO}_3$  due to its evaporation, annealing was started from 500 °C, stepwise increasing temperature up to 650 °C (to prepare  $M_2\text{MoO}_4$ ) or 750 °C (to obtain rare-earth molybdates). Total duration of solid-state synthesis was 70 h. For better reactivity, the reaction mixtures were ground with ethanol in agate mortars in each 15 h of annealing.

The investigations of interaction in the ternary salt systems were carried out by the “intersecting joins” method [18,19]. The method is based on determination of phase compositions of the interception points of all joins connecting the composition points of the components, binary and ternary compounds. Using X-ray powder diffraction we revealed quasibinary sections to construct triangulated phase diagrams of the systems  $\text{Li}_2\text{MoO}_4\text{--K}_2\text{MoO}_4\text{--Ln}_2(\text{MoO}_4)_3$ .

X-ray diffraction analysis (XRD) was performed on a diffractometer DRON-UM1 ( $\text{CuK}\alpha$ -radiation). In the most important cases such as determining homogeneity

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