

# Microstructures and mechanical behavior of Inconel 718 fabricated by selective laser melting

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## Abstract

In this study Inconel 718 cylinders were fabricated by selective laser melting in either argon or nitrogen gas from a pre-alloyed powder. As-fabricated cylinders oriented in the build direction ( $z$ -axis) and perpendicular to the build direction ( $x$ -axis) exhibited columnar grains and arrays of  $\gamma''$  (body-centered tetragonal)  $\text{Ni}_3\text{Nb}$  oblate ellipsoidal precipitates oriented in a strong  $[200]$  texture determined by combined optical metallography, transmission electron microscopy, and X-ray diffraction analysis. Fabricated and hot isostatic pressed (HIP) components exhibited a more pronounced  $[200]$  columnar  $\gamma''$  phase precipitate architecture parallel to the laser beam or build direction (spaced at  $\sim 0.8\ \mu\text{m}$ ), and a partially recrystallized fcc grain structure. Fabricated and annealed ( $1160\ ^\circ\text{C}$  for 4 h) components were  $\sim 50\%$  recrystallized and the recrystallized regions contained spheroidal  $\gamma'$  precipitates distributed in a dense field of fine  $\gamma''$  precipitates. The  $\gamma''$  precipitates were always observed to be coincident with  $\{100\}$  planes of the  $\gamma$ -fcc NiCr matrix. Some  $\delta$  phase precipitates in the unrecrystallized/recrystallized interfaces and recrystallized grain boundaries were also observed in the annealed samples. The microindentation (Vickers) hardness was 3.9 GPa for the as-fabricated materials, 5.7 GPa for the HIP material, and 4.6 GPa for the annealed material. Corresponding tensile properties were comparable with wrought Inconel 718 alloy.

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## 1. Introduction

Nickel-based superalloys with a broad range of alloy compositions have found a wide range of industrial and aerospace applications over the past four decades in cast, wrought and powder metallurgy forms, especially after the development of pre-alloyed powders by rapid solidification technologies [1,2]. Inconel 718, a Ni–Fe–Cr (or Ni–Cr solid solution hardened) austenite ( $\gamma$ ) has found particular applications in gas turbine blades, combustors,

turbocharger rotors, a variety of corrosion containments and structural applications up to  $\sim 700\ ^\circ\text{C}$ . In its wrought form Inconel 718 normally exhibits three intermetallic precipitation phases:  $\gamma'$ , having the composition  $\text{Ni}_3(\text{Al}, \text{Ti}, \text{Nb})$  and a cubic (ordered face-centered)  $\text{Ll}_2$  crystal structure;  $\gamma''$ , having the composition  $\text{Ni}_3\text{Nb}$  and a body-centered tetragonal (bct) ( $\text{DO}_{22}$ ) crystal structure; an orthorhombic ( $\text{DO}_{19}$ )  $\delta$  phase having the composition  $\text{Ni}_3\text{Nb}$ . The nominal base composition is 51Ni–22Fe–19Cr–5Nb–3Mo–1Co (plus small additions of Ti ( $\sim 1\%$ ) and Al ( $\sim 0.5\%$ )). Studying alloys with a base composition Ni–25Fe–16Cr, Kirman and Warrington [3] showed that both the  $\gamma'$  and  $\gamma''$  phases occur when 1.8Ti and 3.5Nb were added, but when the Ti content was increased to 3.5Ti only the  $\gamma'$  phase was obtained, while increasing the Nb content to 6Nb allowed

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only the  $\gamma''$  phase to form. The  $\gamma'$  precipitates are cuboidal or spherical, while the  $\gamma''$  precipitates are lenticular, disc-shaped (oblate spheroid) particles with the following orientation relationships [4,5]:

$$(001)\gamma'' \parallel \{001\}\gamma; [100]\gamma'' \parallel \langle 100 \rangle \gamma$$

In a recent study of Inconel 718 produced by electron beam melting (EBM) an additive (layer) manufacturing process utilizing pre-alloyed, atomized powder Strondl et al. [6] observed columnar, [200] textured  $\gamma$  grains having low angle boundaries containing coarse (50–100 nm)  $\gamma''$  precipitates. Coarse (<2  $\mu\text{m}$ ) (Ti, Nb)(C, N, B) type precipitates with a B1 structure were also observed to be aligned along the growth (or layer building) direction. The  $(001)\gamma \parallel (001)\gamma''$  orientation relationship was also confirmed.

In the present study pre-alloyed, rapidly solidified (atomized) Inconel 718 powder served as a precursor for the production of additive manufactured components using selective laser melting (SLM). In contrast to EBM, a vacuum environment process, the SLM process utilizes an inert atmosphere (argon or nitrogen gas). Consequently, different process parameters produce cooling variations which can affect the residual microstructure and mechanical behavior [7]. This study represents a comprehensive microstructural and mechanical behavior characterization using optical metallography (OM), scanning electron microscopy (SEM) and transmission electron microscopy (TEM) employing energy-dispersive X-ray spectrometry (EDS), and X-ray diffraction (XRD) analysis. The mechanical properties (hardness and tensile strength, including fracture surface analysis by SEM) were also measured for as-fabricated and heat-treated components.

## 2. Experimental methods

The precursor, pre-alloyed atomized Inconel 718 powder, had the composition 53.5Ni–19Cr–18.3Fe–5Nb–3Mo–1Ti–0.43Al. Fig. 1a shows an SEM view illustrating the powder size and morphology, with an average particle size of 17  $\mu\text{m}$ . An OM view of a mounted, polished and etched particle section of the microdendritic structure is shown in Fig. 1b.

Fig. 2 shows a schematic view representing the EOS M270 SLM system which utilized a 200 W Yb:YAG fiber laser (Fig. 2-1). Cylindrical components measuring 1.6 cm diameter  $\times$  8.75 cm length were fabricated in both a vertical and horizontal build orientation, with the cylinder axis parallel to or perpendicular to the beam direction, respectively. The 100  $\mu\text{m}$  diameter laser beam was scanned at 800 or 1200  $\text{mm s}^{-1}$  in either argon or nitrogen gas environments surrounding the building components. The build platform (Fig. 2-5) was preheated to 80  $^{\circ}\text{C}$  and maintained at that temperature. Re-coater formed and melted layers (Fig. 2-6) were alternately scanned along the  $x$ - or  $y$ -axis (at 90 $^{\circ}$ ). The as-fabricated cylindrical specimens were

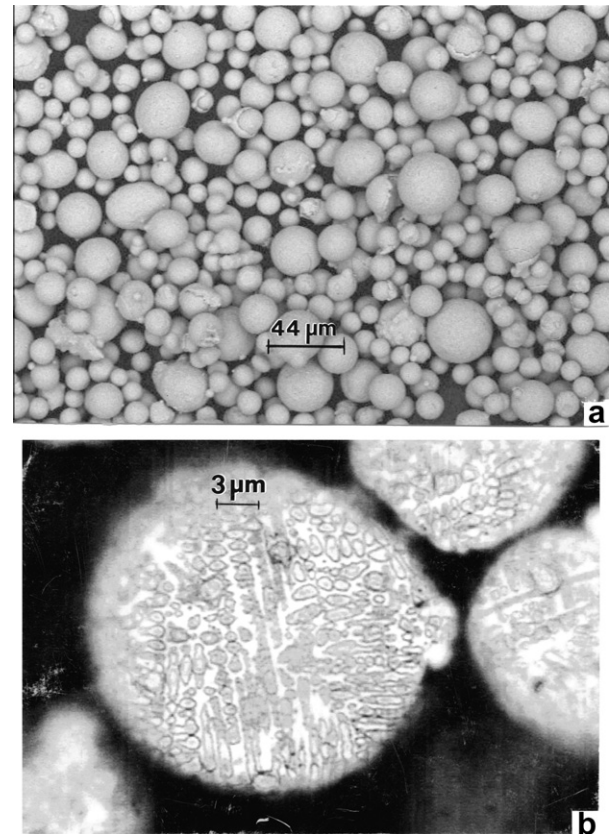


Fig. 1. The pre-alloyed (Inconel 718) precursor powder. (a) Low magnification SEM image showing the range of powder sizes and spherical morphology. (b) Optical metallographic image of a powder particle etched cross-section showing the internal microdendritic microstructure.

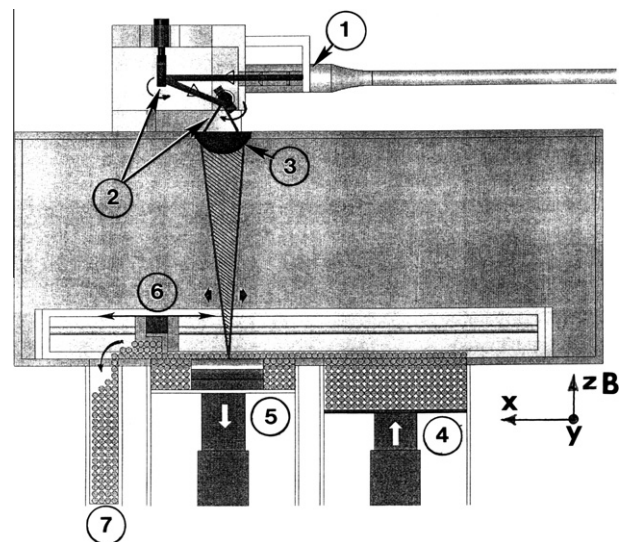


Fig. 2. Components indicated are: (1) laser input; (2) mirror system; (3) beam focus lens; (4) powder feed system; (5) build platform; (6) re-coater system for powder layer formation; (7) powder recovery/recycle bin.

annealed at 982  $^{\circ}\text{C}$  for 0.5 h under vacuum and then hot isostatic pressed (HIP) at 1163  $^{\circ}\text{C}$  at 0.1 GPa pressure for

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