

Investigation of Eu^{2+} luminescence in barium tetrphosphate $\text{Ba}_3\text{P}_4\text{O}_{13}$ polycrystalline ceramics

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Abstract

In this paper, $\text{Ba}_3\text{P}_4\text{O}_{13}:\text{Eu}^{2+}$ phosphor was synthesized by a solid-state reaction. The photoluminescence (PL) emission spectrum and luminescence decay kinetics confirm that the doped Eu^{2+} ions can occupy two different Ba^{2+} sites. The PL excitation spectrum shows a broad band matching well with the emission of near-UV chip. $\text{Ba}_3\text{P}_4\text{O}_{13}:\text{Eu}^{2+}$ is a promising phosphor for near-UV chip excited white LEDs. The doped Eu^{3+} ions can also be reduced to Eu^{2+} ions in air atmosphere at high temperature. Charge compensation mechanism is applied to explain this kind of abnormal reduction.

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1. Introduction

Recently, more and more attention is focused on phosphor converted white light-emitting-diodes (White LEDs) because of their advantages, such as high efficiency, long lifetime, energy saving and environment-friendly [1–3]. Developing the excellent phosphor materials matching well with near-UV or blue chips is the research focus in the field of solid-state-light (SSL) industry [4–6].

Up to now, Eu^{2+} ion has been widely investigated as an efficient activator in phosphors for display and lighting due to its intense broad excitation and strong emission band [7]. The excellent luminescent properties can be attributed to the allowed inter-configurational $4f \rightarrow 5d$ transitions with high oscillator strength [8,9]. In general, the $5d \rightarrow 4f$ emissions of Eu^{2+} depend strongly on the host composition, crystal structure and lattice symmetry [5,10,11]. So the colors of the Eu^{2+} emission can vary from long-wavelength ultraviolet to red. In addition, its emission peak can easily shift and its full

width at half maximum will broaden if electron–phonon interaction is stronger [12].

Phosphates are a large family of compounds. The luminescent properties of rare earth doped materials have been investigated widely due to their application in the field of display and lighting [13]. For $\text{Ba}_3\text{P}_4\text{O}_{13}$ crystal, the structure has been reported by Gatehouse in 1991 [14]. The luminescent properties of $\text{Ba}_3\text{P}_4\text{O}_{13}:\text{Eu}^{2+}$ phosphor has only been investigated simply by Lagos in 1968 [15]. In this paper, the PL emission spectrum at lower temperature and luminescence decay kinetics have been utilized to confirm that the doped Eu^{2+} ions can occupy two different Ba^{2+} sites. At the same time, it is found that the Eu^{3+} can be reduced to Eu^{2+} in $\text{Ba}_3\text{P}_4\text{O}_{13}$ host in air at high temperature. Charge compensation mechanism is applied to explain this kind of abnormal reduction.

2. Experimental

$\text{Ba}_3\text{P}_4\text{O}_{13}:\text{Eu}^{2+}$ phosphor was synthesized by a solid-state reaction. The stoichiometric amounts of materials BaCO_3 (Aldrich, 99.9%), $(\text{NH}_4)_2\text{HPO}_4$ (Aldrich, 99.9%) and Eu_2O_3 (Aldrich, 99.99%) were well mixed. The materials were pre-fired at 600 °C for 2 h in air to pyrolyze phosphate; then

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the raw materials were re-ground and calcined at 750°C for 8 h in air atmosphere.

The crystalline structure of the final product was examined by using a Philips XPert/MPD diffraction system with CuK α ($\lambda=1.5405$ Å) radiation. The photoluminescence (PL) excitation and emission spectra were measured with a Fluorescence Spectrophotometer (PTI) with a 150 W Xe lamp as an excitation source. The PL emission spectrum at 18 K was obtained by 355 nm pulsed Nd/YAG (yttrium aluminum garnet) laser (Spectron Laser Systems SL802G). The sample was placed at cold finger in a He gas recycled cryostat. The decay curves were recorded by the 500 MHz digital oscilloscope (LeCroy 9350A).

3. Results and discussion

3.1. Identification of crystalline phase of the sample

Ba₃P₄O₁₃ compound crystallizes in triclinic space group $P1^-$ with lattice constants of $a=5.691$ (5), $b=7.238$ (7) and $c=8.006$ (5) Å; $\alpha=83.65$ (5), $\beta=75.95$ (8) and $\gamma=70.49$ (7)° [14]. Fig. 1 shows X-ray diffraction pattern of Ba_{2.96}Eu_{0.04}P₄O₁₃ sample. The XRD pattern of Ba₃P₄O₁₃ crystal (JCPDS, no. 79-1530) is also shown for comparison. The diffraction peaks are in good agreement with data given in JCPDS 79-1530, which indicates that doped Eu²⁺ ions do not change the crystal structure of Ba₃P₄O₁₃. Because the electric charges of Ba²⁺ and Eu²⁺ are identical and the radii of them are similar ($r_{Ba^{2+}}=0.142$ nm, $r_{Eu^{2+}}=0.125$ nm CN=8) [16], we argue that the doped Eu²⁺ ions prefer to occupy Ba²⁺ sites.

3.2. PL properties of Ba₃P₄O₁₃:Eu²⁺ prepared in air

It is well known that Eu²⁺ usually gives broadband emission due to d \rightarrow f transition, and the spectral position of the emission

band depends strongly on the crystal structure of host [13]. The coordination numbers for the barium ions in Ba₃P₄O₁₃ crystal have been taken to be eight for Ba(1) and seven for Ba(2) (the crystal structure of Ba₃P₄O₁₃ emphasizing the coordination environment of Ba(1) and Ba(2) cations is shown in the inset of Fig. 1). The mean distance of Ba(1)–O bond is 2.82 (13) Å and that of Ba(2)–O bond is 2.76 (7) Å [14]. As discussed above, the doped Eu²⁺ ions will substitute Ba²⁺ ions sites in Ba₃P₄O₁₃. Two different Ba sites [Ba(1) and Ba(2)] can be partially occupied by Eu²⁺ ions. According to the above structure analysis, the mean distance of Ba(1)–O is longer than that of Ba(2)–O, so the Eu²⁺ ions substituting Ba(1) sites experience weaker crystal-field strength which is inversely proportional to R^5 (R : chemical bond length between a cation with d orbital electrons and the coordinating anion) [17].

Room temperature excitation and emission spectra of Ba_{2.96}Eu_{0.04}P₄O₁₃ are depicted in Fig. 2. The excitation spectra show broadband in the near-UV region. Although the two excitation spectra show some difference when monitored by different emission wavelengths, they should be ascribed to the 4f⁷ \rightarrow 4f⁶5d transition of Eu²⁺ ions. The 5d levels of Eu²⁺ that are not shielded from the outside environment are split under different ligand field strengths and the number of split levels is determined by the local symmetry around Eu²⁺ ions [18]. For BaCl₂:Eu²⁺ phosphor, Eu²⁺ ions occupy the Ca²⁺ site having O_h symmetry. So the 5d orbitals of Eu²⁺ are split into two levels, t_{2g} and e_g [19]. In the case of CaSr_{1-x}Eu_xSi₅N₈, the excitation spectrum is broadband and cannot be exactly resolved because Eu²⁺ could occupy two different Sr sites [20]. In the present case, the relative complicated environment of the Ba²⁺ also results in a broad excitation band, which is similar to CaSr_{1-x}Eu_xSi₅N₈ sample. Because the PL excitation spectra show strong absorption in the range of near-UV, Ba₃P₄O₁₃:Eu²⁺ could be a potential phosphor for white LEDs.

The broad emission band of Ba_{2.96}Eu_{0.04}P₄O₁₃ sample is assigned to the allowed 4f⁶5d \rightarrow 4f⁷ transition of Eu²⁺. It is found that the no Eu³⁺ emission (f–f transition) is observed within the sensitivity of our Fluorescence Spectrophotometer,

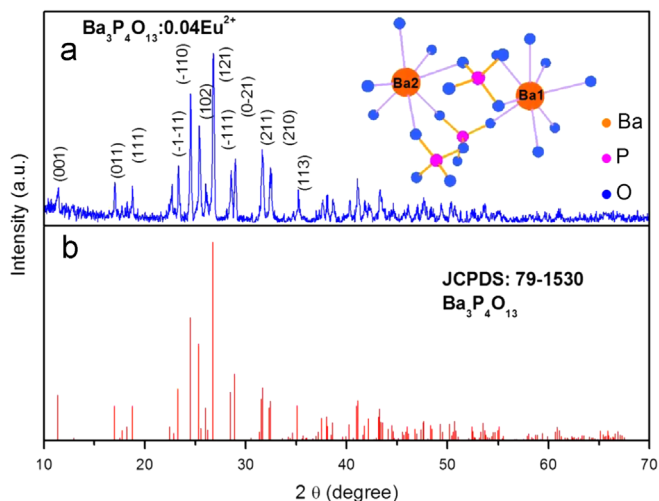


Fig. 1. XRD pattern of Ba_{2.96}Eu_{0.04}P₄O₁₃ phosphor, the XRD pattern of Ba₃P₄O₁₃ crystal (JCPDS, no. 79-1530) is also shown for comparison (the inset shows the crystal structure of Ba₃P₄O₁₃ emphasizing the coordination environment of Ba1 and Ba2 cations).

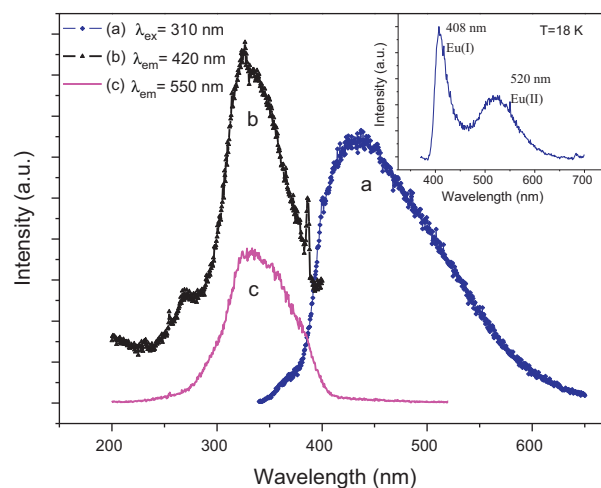


Fig. 2. Photoluminescence excitation ($\lambda_{em}=420, 550$ nm) and emission ($\lambda_{ex}=310$ nm) spectra of Ba_{2.96}Eu_{0.04}P₄O₁₃.

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