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# Structural characterization of barium titanate–cobalt ferrite composite powders

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#### Abstract

Barium titanate–cobalt ferrite composite powders prepared by sol–gel method and by mixing of precursors were characterized by Raman spectroscopy, scanning electron microscopy and Rietveld refinement of XRD data. Ba vacancies and  $BaTiO_3$ –CoFe<sub>2</sub>O<sub>4</sub> interfacial stress were associated to lowering of tetragonal distortion in the ferroelectric structure. Raman spectroscopy suggested the encapsulation of ferrite inside clusters of particles in the sol–gel composite.

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### 1. Introduction

Magnetoelectric (ME) materials are promising candidates for technological applications since the multiferroic coupling allows the interconnection between magnetic and electric fields. New memory devices electrically written and magnetically read have been proposed based on ME materials [1].

ME structures can be prepared as single-phase materials or composites combining magnetostrictive and piezoelectric components. In the past few years, ME composites have been subject of many investigations [2–5] since they can yield a giant ME response at room temperature while the single-phase materials exhibit weak magneto-electric coupling at low temperature. The ME coupling in the composite materials depends strongly on their microstructure. Many inherent preparation problems as atomic diffusion and undesirable chemical reactions between phases [6] can affect the interface, modifying its chemical and structural properties and thus lowering the coupling.

BaTiO<sub>3</sub>-CoFe<sub>2</sub>O<sub>4</sub> composites are interesting magnetoelectric materials due to the individual properties of their components. CoFe<sub>2</sub>O<sub>4</sub> has a spinel structure with large coercivity and magnetostriction [7]. BaTiO<sub>3</sub> exhibits polymorphs depending on the temperature. The tetragonal polymorph (t-BaTiO<sub>3</sub>) is the most interesting due to its ferroelectric and piezoelectric properties. The cubic polymorph (c-BaTiO<sub>3</sub>) is paraelectric. Although the tetragonal-tocubic transition occurs at the Curie temperature ( $T_c \approx 130$  °C in single-crystal), it is well known that c-BaTiO<sub>3</sub> can be found at room temperature in very fine particles, below the critical size of 10-35 nm [8-12]. Studies on BaTiO<sub>3</sub> ferroelectric behavior relate the cubic structure of ultrafine particles to the lack of long-range cooperative interactions, the presence of internal strains, depolarization fields and structural defects [13,14].

X-ray diffraction (XRD) is a limited method to distinguish between c- and t-BaTiO<sub>3</sub> in ultrafine particles due to the extensive broadening of Bragg reflexions. Raman vibrational spectroscopy can provide information on local lattice disorder, defects and has proved to be more sensitive to detect small distortions of the tetragonal unit cell than XRD.

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This work aimed at evaluating the structural properties of  $BaTiO_3$  and  $CoFe_2O_4$  in single-phase and in composite powders prepared by sol-gel process focusing mainly on the tetragonal distortion of  $BaTiO_3$ .

#### 2. Synthesis

The syntheses of BaTiO<sub>3</sub>, CoFe<sub>2</sub>O<sub>4</sub> and BaTiO<sub>3</sub>–CoFe<sub>2</sub>O<sub>4</sub> composite powders were described elsewhere [15]. Briefly, BaTiO<sub>3</sub> powder was prepared by sol–gel method and calcined at 900 °C (sample B9) and 1100 °C (sample B11). CoFe<sub>2</sub>O<sub>4</sub> powder was prepared by coprecipitation followed by freezedrying and calcination at 900 °C (sample F9) and 1100 °C (sample F11). The composites BaTiO<sub>3</sub>–CoFe<sub>2</sub>O<sub>4</sub> 14:11 wt% were prepared by: (a) mixing and calcination of precursor powders of BaTiO<sub>3</sub> and CoFe<sub>2</sub>O<sub>4</sub> at 900 °C (sample M9) and 1100 °C (sample M11); (b) dispersing the CoFe<sub>2</sub>O<sub>4</sub> precursor powder into BaTiO<sub>3</sub> wet gel, drying of the resulting brown gel and calcination of the dried gel at 900 °C (sample C9) and 1100 °C (sample C11).

#### 3. Characterization

Raman spectra were collected using a T64000 Horiba–Jobin Yvon triple–monochromator spectrometer equipped with a confocal microscope and a liquid N<sub>2</sub> cooled charge coupled device (CCD). The scattering was excited using an argon ion laser at 514.5 nm. The spot size of the laser was 1  $\mu$ m<sup>2</sup> using a 100× objective and the laser power was kept at 1.0 mW in order to avoid sample heating.

XRD data were recorded using a Siemens D5000 diffractometer with Cu K $\alpha$  radiation and graphite monochromator,  $2\theta$  step of 0.05°. The samples were measured under spinning at 60 rpm to prevent preferred orientation. The major crystalline phases were assigned to spinel CoFe<sub>2</sub>O<sub>4</sub> (JCPDS-22-1086, 1997), c-BaTiO<sub>3</sub> (JCPDS-01-0104, 1997) or t-BaTiO<sub>3</sub> (JCPDS-05-0626); BaFe<sub>12</sub>O<sub>19</sub> (JCPDS-34-0129, 1997) was identified as minor crystalline phase.

Rietveld refinement with FullProf program was performed to evaluate particle size, lattice parameters and microstrain. Due to the low solubility of Ba in comparison with Ti, the stoichiometric Ba/Ti ratio was also evaluated. The Thompson– Cox–Hastings pseudo-Voight function with axial divergence asymmetry was chosen as profile function. The statistics  $R_{exp}$ ,  $R_{wp}$  and GOF were used to evaluate fitness [16].

Scanning electron microscopy (SEM) was carried out using a secondary electron detector (FEI, Quanta 200 FEG). The particulate material was previously dispersed in isopropanol and deposited on a Si plate.

#### 4. Results and discussion

#### 4.1. Raman spectroscopy

The Raman spectra of samples B9, B11, shown in Fig. 1a, were characteristic of t-BaTiO<sub>3</sub> as evidenced by the vibrational modes at around 257, 306, 520,  $720 \text{ cm}^{-1}$  (Table 1). Both

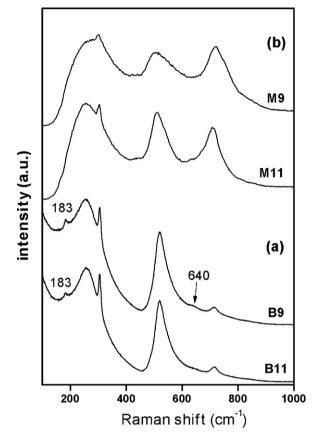


Fig. 1. Raman spectra of samples (a) B9, B11 and (b) M9, M11 (white region).

spectra exhibited a small peak at  $183 \text{ cm}^{-1}$  that could be ascribed to lattice defects or small amounts of orthorhombic phase [17,18]. Sample B9 showed an interesting feature at  $640 \text{ cm}^{-1}$  which might be assigned to hexagonal BaTiO<sub>3</sub> [19,20]. As this feature decreased with increasing thermal treatment temperature, it was considered to be a satellite peak due to planar defects in the tetragonal structure [21].

In the mixture samples M9, M11, the laser beam could clearly discriminate between white and dark regions assigned to t-BaTiO<sub>3</sub> and CoFe<sub>2</sub>O<sub>4</sub> particles, respectively. Typical Raman spectra of white regions, shown in Fig. 1b, presented t-BaTiO<sub>3</sub> characteristic modes. The bands were broader than those found in samples B9, B11, suggesting decreased structural coherence [11]. The higher intensity of the band at  $\approx$ 720 cm<sup>-1</sup> relative to the band at  $\approx$ 510 cm<sup>-1</sup> was also found by Busca et al. [22] and assigned to particles with sub-stoichiometric Ba/Ti ratio due to Ba segregation as BaCO<sub>3</sub>. BaTiO<sub>3</sub> vibrational modes were downshifted in comparison with the corresponding modes in samples B9, B11, as shown in Table 1.

Table 1 Raman vibrational modes in  $BaTiO_3$  samples and white region of the composite samples (M).

Sample	Raman shift (cm <sup>-1</sup> )			
B9, B11	256, 259	305, 306	521, 519	718, 722
M9, M11	274, 255	301, 304	506, 510	722, 711
BaTiO <sub>3</sub> <sup>a</sup>	242, 266	303, 311	514, 530	713, 720

<sup>a</sup> Polycrystalline as reported in Refs. [9,11,12,17,18,21,32-34].

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