



# Influence of molybdenum ions on spectroscopic and dielectric properties of $\text{ZnF}_2\text{--Bi}_2\text{O}_3\text{--P}_2\text{O}_5$ glass ceramics

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## ABSTRACT

Glasses doped with different concentrations of  $\text{MoO}_3$  (ranging from 0 to 10 mol %) in the system  $\text{ZnF}_2\text{--Bi}_2\text{O}_3\text{--P}_2\text{O}_5$  were crystallized and characterized by X-ray diffraction (XRD), scanning electron microscopy (SEM) and differential thermal analysis (DTA) techniques. The presence of  $\text{BiPO}_4$ ,  $\alpha\text{-Zn}_3(\text{PO}_4)_2$ ,  $\text{Bi}_2\text{MoO}_6$ ,  $\gamma\text{-Bi}_2\text{MoO}_6$ ,  $\text{Bi}_2\text{Mo}_2\text{O}_9$ ,  $\text{MoOPO}_4$ ,  $\alpha\text{-ZnMoO}_4$  microcrystalline phases were exposed by XRD and SEM studies. The DTA indicated that the surface crystallization prevails over the bulk crystallization as the concentration of the crystallizing agent is increased. Spectroscopic properties (FTIR, Raman, optical absorption and ESR) at room temperature and dielectric properties (dielectric constant, loss  $\tan\delta$ , ac conductivity) over a range of frequency and temperature are studied. The IR and Raman spectra exhibit bands due to  $\text{MoO}_4$  and  $\text{MoO}_6$  units in structural groups in addition to various conventional phosphate groups. The existence of  $\text{Mo}^{5+}$  state in addition to  $\text{Mo}^{6+}$  state in these glass ceramic samples has been identified by the optical absorption, ESR and magnetic susceptibility studies. The variations observed in all these properties with the concentration of the crystallizing agent have been analyzed in the light of different oxidation states and environment of molybdenum ions in the glass ceramic network.

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## 1. Introduction

Glass-ceramic materials are expected to have good mechanical, electrical and thermal properties, high chemical durability and very low coefficient of thermal expansion. By monitoring crystallization temperature and using appropriate nucleating agents the characteristics of these materials can be adjusted so as to exhibit excellent physical properties [1,2].

Among various glass ceramics, zinc bismuth phosphate glass ceramics are being widely used as fast ion conducting materials, laser hosts, energy storage devices and in solid state batteries [3–5]. Amid different crystallizing agents,  $\text{MoO}_3$  is expected to be a more effective mineralizer, particularly in the glass systems like bismuth phosphate that exhibits non-linear optical properties. In fact, in several glass matrices, molybdenum oxide has been successfully used as a crystallizing catalyst [6,7]. The molybdenum ions exist in two valence states viz.,  $\text{Mo}^{5+}$  and  $\text{Mo}^{6+}$  and participate in the glass network with octahedral ( $\text{MoO}_6$ ) and tetrahedral ( $\text{MoO}_4$ ) structural units [8,9]. Additionally, the presence of molybdenum ions in bismuth phosphate glasses makes them to be useful for potential applications in high-density memories, light modulation, large area display devices like smart windows and other electro-chromic devices [10–12].

The objective of the present investigation is to prepare  $\text{ZnF}_2\text{--Bi}_2\text{O}_3\text{--P}_2\text{O}_5$  glass ceramics with  $\text{MoO}_3$  as nucleating agent and to

study the effect of compositional change of  $\text{MoO}_3$  on the crystallization behavior and microstructural aspects of glass ceramics products. The prepared samples were characterized by XRD, SEM and DTA techniques and their spectroscopic (IR, Raman, optical absorption and ESR), magnetic and dielectric properties have been investigated and the results were analyzed in the light of variations in the concentration of different crystal phases embedded in the samples.

## 2. Experimental

In the present investigation, a particular composition  $20\text{ZnF}_2\text{--}(20-x)\text{Bi}_2\text{O}_3\text{--}60\text{P}_2\text{O}_5\text{:}x\text{MoO}_3$  ( $x$  – ranging from 0 to 10 mol%) was chosen. The detailed compositions (all in mol%) are as follows.

|  |
|--|
| $M_0$ : $20\text{ZnF}_2\text{--}20\text{Bi}_2\text{O}_3\text{--}60\text{P}_2\text{O}_5$                          |
| $M_2$ : $20\text{ZnF}_2\text{--}18\text{Bi}_2\text{O}_3\text{--}60\text{P}_2\text{O}_5\text{:}2\text{MoO}_3$     |
| $M_4$ : $20\text{ZnF}_2\text{--}16\text{Bi}_2\text{O}_3\text{--}60\text{P}_2\text{O}_5\text{:}4\text{MoO}_3$     |
| $M_6$ : $20\text{ZnF}_2\text{--}14\text{Bi}_2\text{O}_3\text{--}60\text{P}_2\text{O}_5\text{:}6\text{MoO}_3$     |
| $M_8$ : $20\text{ZnF}_2\text{--}12\text{Bi}_2\text{O}_3\text{--}60\text{P}_2\text{O}_5\text{:}8\text{MoO}_3$     |
| $M_{10}$ : $20\text{ZnF}_2\text{--}10\text{Bi}_2\text{O}_3\text{--}60\text{P}_2\text{O}_5\text{:}10\text{MoO}_3$ |

Analytical grade reagents of grade  $\text{ZnF}_2$  (Sigma Aldrich 99.99%),  $\text{Bi}_2\text{O}_3$ ,  $\text{P}_2\text{O}_5$ ,  $\text{MoO}_3$  (LOBA, AR grade) reagents in the batches of 15 g were thoroughly mixed in an agate mortar and melted for 20 minutes in a thick walled platinum crucible at  $970 \pm 10^\circ\text{C}$  in PID controlled furnace until a bubble free liquid is formed. The melt was then poured on a brass mould at room temperature and subsequently annealed at  $175^\circ\text{C}$ . The dopant free specimen ( $M_0$ ) appeared to be a transparent

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brown color, later the color of the glass specimens (from  $M_2$  to  $M_{10}$ ) was observed to change gradually from light greenish blue to thick blue. All the glass specimens with various concentrations of  $\text{MoO}_3$  were heat treated in a furnace at their crystalline temperature about  $450^\circ\text{C}$  identified from the DTA of the glass samples for 6 h. Automatic controlling furnace was used to keep the temperature at a required level. After the heat treatment in a furnace at specified temperature, the samples were quenched in air to room temperature.

The crystalline phases in the glass ceramic samples were identified from the X-ray diffraction patterns of powdered glass samples recorded with Philips Xpert system using the step scan method with  $\text{Cu-K}\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ) operated at 40 kV, 25 mA. The surface morphology of prepared samples was carried out using Carl Zeiss, OXFORD INSTRUMENTS, Inca Penta FET x3 Scanning Electron Microscope to observe the crystallinity. Differential thermal analysis (DTA) of the samples was carried out using METTLER STAR SW 8.10 instrument (to an accuracy of  $\pm 0.1^\circ\text{C}$ ) with a heating rate of  $10^\circ\text{C}/\text{min}$  in the temperature range of  $30$ – $1000^\circ\text{C}$ . A programmable VIBRA HT density measurement kit was used to determine the densities and molar volumes of bulk samples automatically (with readability  $0.0001 \text{ g}/\text{cm}^3$ ) by means of Archimedes' principle with O-Xylene (99.99% pure) as buoyant liquid. The FTIR spectra were recorded by dispersing the glass powders in KBr with a Perkin-ELMER FTIR spectrometer in the range of  $400$ – $1300 \text{ cm}^{-1}$ . Micro Raman spectra were recorded using a Horiba Jobin-Yuon-UV800 Lab Ram HR spectrometer with a 17 mW internal He-Ne laser source of excitation wavelength of  $514 \text{ nm}$  with a spectral resolution of about  $\pm 0.33 \text{ cm}^{-1}$ . The EPR spectra were recorded on JEOL-FE-IX ESR spectrometer operating at X-band frequency ( $9.125 \text{ GHz}$ ) with a field modulation frequency of  $100 \text{ kHz}$ . The magnetic field was scanned from  $0$  to  $500 \text{ mT}$  and the micro wave power used was  $10 \text{ mW}$ . Glass sample in powdered form of  $100 \text{ mg}$  is taken in a quartz tube for EPR measurements. Magnetic susceptibility measurements were also carried out on these samples by Guoy's method using fine powders.

The optical absorption spectra of the samples (thickness  $1 \text{ mm}$ ) were recorded in the wavelength range of  $200$ – $1200 \text{ nm}$  up to a resolution of  $0.1 \text{ nm}$  using JASCO V-670 UV-Vis-NIR spectrophotometer. For dielectric studies, the glasses were grounded and polished to get  $1 \text{ cm} \times 1 \text{ cm} \times 0.2 \text{ cm}$  dimensions. A thin coating of silver paint was applied on either side of the samples to serve as electrode for electrical measurements. The dielectric measurements were taken on LF-impedance analyzer (Hewlett-Packard model 4192A) in the frequency range of  $10^3$ – $10^6 \text{ Hz}$  and in the temperature range of  $30$ – $300^\circ\text{C}$ . The accuracy in the measurement of dielectric constant  $\epsilon'$  is  $\pm 0.01$  and that in dielectric loss ( $\tan\delta$ ) is  $\pm 0.001$ . The dielectric breakdown strength of all these glasses was determined at room temperature in air medium using a high a.c. voltage breakdown tester (ITL Model BOV-7, Hyderabad) operated with an input voltage of  $230 \text{ V}$  at a frequency of  $50 \text{ Hz}$ .

### 3. Results

The density measurements indicate the gradual decrease in their value with increase in concentration of  $\text{MoO}_3$ . From the measured values of the density and average molecular weight  $\bar{M}$  of the samples, various other physical parameters such as molybdenum ion concentration  $N_i$ , mean molybdenum ion separation  $R_i$ , polaron radius  $R_p$  of  $\text{ZnF}_2\text{-Bi}_2\text{O}_3\text{-P}_2\text{O}_5\text{:MoO}_3$  glass-ceramics are evaluated and presented in Table 1.

#### 3.1. XRD

Fig. 1 shows the X-ray diffraction patterns (XRD) of the  $\text{MoO}_3$  free samples before and after crystallization and one of the glass ceramic samples doped with  $2.0 \text{ mol\%}$  of  $\text{MoO}_3$ . The XRD analysis of the pre-heated glass samples indicates the amorphous in nature. After

**Table 1**

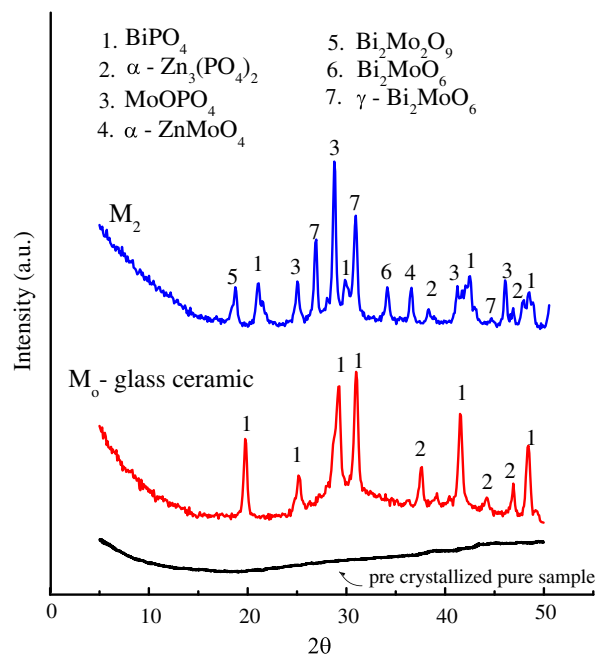
Various physical properties of  $\text{ZnF}_2\text{-Bi}_2\text{O}_3\text{-P}_2\text{O}_5$  glass ceramics doped with different concentrations of  $\text{MoO}_3$ .

| Sample   | $M_0$  | $M_2$  | $M_4$  | $M_6$  | $M_8$  | $M_{10}$ |
|--|--------|--------|--------|--------|--------|----------|
| Concentration of $\text{MoO}_3$ (mol%)   | 0      | 2      | 4      | 6      | 8      | 10       |
| Average molecular weight, $\bar{M}$ (g/mol)  | 199.03 | 192.59 | 186.15 | 179.71 | 173.27 | 167.19   |
| Density, $\rho$ ( $\text{g}/\text{cm}^3$ ) ( $\pm 0.0001$ )                                      | 4.2891 | 4.1532 | 3.9891 | 3.8389 | 3.6808 | 3.5216   |
| Molybdenum ion concentration, $N_i$ ( $\times 10^{20} \text{ ions}/\text{cm}^3$ ) ( $\pm 0.01$ ) | –      | 2.60   | 5.11   | 7.74   | 10.23  | 12.69    |
| Inter ionic separation, $R_i$ ( $\text{\AA}$ ) ( $\pm 0.01$ )                                    | –      | 15.67  | 12.51  | 10.89  | 9.92   | 9.24     |
| Polaron radius, $R_p$ ( $\text{\AA}$ ) ( $\pm 0.01$ )  | –      | 6.32   | 5.04   | 4.39   | 4.00   | 3.72     |

the samples are crystallized at  $450^\circ\text{C}$ , the concerned diffraction patterns exhibit peaks due to  $\text{BiPO}_4$  (JCPDS No. 89-0287),  $\alpha\text{-Zn}_3(\text{PO}_4)_2$  (JCPDS No. 76-0518),  $\text{MoOPO}_4$  (JCPDS No. 73-2333),  $\text{Bi}_2\text{MoO}_6$  (JCPDS No. 84-0787),  $\gamma\text{-Bi}_2\text{MoO}_6$  (JCPDS No. 82-2067),  $\text{Bi}_2\text{Mo}_2\text{O}_9$  (JCPDS No. 84-0829), and  $\alpha\text{-ZnMoO}_4$  (JCPDS No. 25-1023) [13]. These are kinetically and thermodynamically feasible crystalline products which are identified in all the present investigated samples. The patterns indicate that bismuth ions exist in  $\text{Bi}^{3+}$  oxidation state. Similarly, the crystalline phases of  $\text{Mo}^{5+}$  ions were also detected in addition to  $\text{Mo}^{6+}$  ion phases. We observed a diffraction peak with a maximum intensity due to  $\text{MoOPO}_4$  crystal phase at about  $2\theta = 28.77^\circ$ . The presence of such phase suggests that more fraction of molybdenum ions do exist in  $\text{Mo}^{5+}$  state. As the concentration of  $\text{MoO}_3$  increased, the considerable hike in the intensity of the peak is observed.

#### 3.2. SEM

Fig. 2 refers some of the SEM pictures of the samples  $M_0$ ,  $M_2$  and  $M_{10}$  and they show an increasing crystallinity with increasing concentration of the crystallizing agent  $\text{MoO}_3$ . The remaining glass phase is being acted as interconnecting zones among the crystallized areas of all the samples, avoiding voids and cracks. The electron



**Fig. 1.** XRD patterns of  $\text{ZnF}_2\text{-Bi}_2\text{O}_3\text{-P}_2\text{O}_5$  glass ceramics doped with  $0$  and  $2.0 \text{ mol\%}$  of  $\text{MoO}_3$  showing different possible crystalline phases. XRD pattern of pre-crystallized dopant free sample is also included.

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