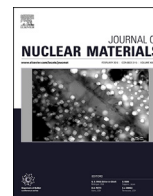




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## Non-linear effects of alumina concentration on Product Consistency Test response of waste glasses

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## ABSTRACT

The effect of alumina mole fraction on the 7-day Product Consistency Test (PCT) responses,  $\ln[NL_\alpha]$ , of simulated nuclear waste glasses was examined. It was found that the effect is highly non-linear. At low mole fractions of  $\text{Al}_2\text{O}_3$  ( $x_{\text{Al}_2\text{O}_3} \leq 0.035$ ) the effect of  $\text{Al}_2\text{O}_3$  additions is highly negative ( $\frac{d \ln NL_\alpha}{dx_{\text{Al}_2\text{O}_3}} = -72$ ). At intermediate concentrations ( $0.035 < x_{\text{Al}_2\text{O}_3} \leq 0.19$ ) the effect of  $\text{Al}_2\text{O}_3$  additions is moderately negative ( $\frac{d \ln NL_\alpha}{dx_{\text{Al}_2\text{O}_3}} = -11$ ). At high concentrations ( $0.19 < x_{\text{Al}_2\text{O}_3}$ ) the effect of  $\text{Al}_2\text{O}_3$  additions is highly positive ( $\frac{d \ln NL_\alpha}{dx_{\text{Al}_2\text{O}_3}} = +21$ ). This variable impact of  $\text{Al}_2\text{O}_3$  on  $\ln[NL_\alpha]$  is speculated to be caused by the rates at which the glass corrosion process changes through various reaction regimes in static conditions at  $90^\circ\text{C}$ . A model for prediction of  $\ln[NL_\alpha]$  as a function of glass composition is presented. This model represents the data from 2669 glass compositions spanning a broad nuclear waste glass composition region and it is useful up to much higher  $\text{Al}_2\text{O}_3$  concentrations than previous models (from  $\sim 10$  to  $25 \text{ mol}\%$   $\text{Al}_2\text{O}_3$ ). The model was validated using data subset validation methods and shown to predict validation data in the same composition region with roughly equal certainty as the model fit data. The potential causes for non-linear effects of  $\text{Al}_2\text{O}_3$  on  $\ln[NL_\alpha]$  were discussed in context of the progression of glass corrosion rates. Time resolved static corrosion data is needed to better understand these non-linear effects.

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## 1. Introduction

## 1.1. Background and motivation

Approximately  $210,000 \text{ m}^3$  of high-level radioactive waste (HLW) is stored in underground tanks at the Hanford site near Richland WA. This waste resulted from the processing of irradiated nuclear fuels and targets for the separations of heavy elements by a number of processes [1]. The Hanford Tank Waste Treatment and Immobilization Plant (WTP) is being designed and built to manage these tank wastes. The HLW will be retrieved, separated into low-activity waste (LAW) and HLW fractions, and vitrified to form alkali-alumino-borosilicate glass waste forms for disposal. The resulting LAW glass will be disposed on the Hanford site and the HLW glass will be stored temporarily and then ultimately disposed in a federal HLW repository. The waste acceptance criteria for disposal of both HLW and LAW include a limit on the response to the Product Consistency Test (PCT) Method A – a static dissolution

test performed for 7 d at  $90^\circ\text{C}$  in deionized water with a targeted glass surface area to solution volume ratio of  $2000 \text{ m}^{-1}$  [2]. For HLW glass, the normalized loss ( $NL_\alpha$ ) for  $\alpha = \text{B}$ ,  $\text{Na}$ , and  $\text{Li}$  must be below those for the Defense Waste Processing Facility Environmental Assessment Glass ( $NL_{\text{B}} = 8.35 \text{ g m}^{-2}$ ,  $NL_{\text{Na}} = 6.67$ , and  $NL_{\text{Li}} = 4.78$ ; respectively) [3–5]. For LAW glass, the  $NL_{\text{B}}$ ,  $NL_{\text{Na}}$ , and  $NL_{\text{Si}}$  must each be below  $2 \text{ g m}^{-2}$  [6]. These PCT constraints are met using models that predict the  $NL_\alpha$  of glass to be produced based on the glass composition. Current Hanford models predict the PCT responses of glasses planned for the early stage of WTP operation (called “baseline” Hanford glasses) [7,8]. These models correlate the natural logarithm ( $\ln$ ) of  $NL_\alpha$  with linear combinations of glass components plus a limited number of second-order polynomial terms:

$$\ln[NL_\alpha] = \sum_{i=1}^q a_{\alpha,i} x_i + \text{selected} \left\{ \sum_{i=1}^q a_{\alpha,ii} x_i^2 + \sum_{i=1}^{q-1} \sum_{j=i+1}^q a_{\alpha,ij} x_i x_j \right\} \quad (1)$$

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where,  $a_{\alpha,i}$ ,  $a_{\alpha,ii}$ , and  $a_{\alpha,ij}$  are the coefficients for component  $i$ , component  $i$  squared, and component  $i$  crossed with component  $j$ , respectively; and  $x_i$  and  $x_j$  are the  $i$ th and  $j$ th component concentration in glass. The concentrations can be either mass fraction or mole fraction; in the case of the baseline models concentrations are in mass fractions, while models developed for this study use concentrations in mole fractions.

The breadth of the expected Hanford glass compositions (called “advanced” Hanford glasses) is not covered by the current baseline models. The HLW baseline glasses were designed to immobilize the first few Hanford tanks that are relatively high in either iron or zirconium and thorium and low in most other waste components. These HLW baseline glasses were also designed for relatively low waste loading as their purpose was to start up the new facility. By contrast, the advanced glasses are under development to immobilize the full range of tank wastes with very broad composition ranges and focused on high waste loadings to improve mission efficiency [9–14]. Of particular interest is the difference in concentration ranges for  $\text{Al}_2\text{O}_3$ , which was restricted to 13 mass% (~8.89 mole%) in baseline HLW glasses and projected to be up to 30 mass% (22.2 mole%) in advanced Hanford HLW glasses. Higher  $\text{Al}_2\text{O}_3$  wastes include neutralized cladding removal waste and REDOX solvent extraction wastes that were not included in the baseline waste tanks.

It has previously been shown that  $\text{Al}_2\text{O}_3$  has a non-linear effect on  $\ln[NL_\alpha]$  [15–18]. Fig. 1 shows the effect of changing  $\text{Al}_2\text{O}_3$  concentration on PCT response. For each of the studies shown  $\text{Al}_2\text{O}_3$  concentration is increased while maintaining the ratios of other glass components constant. As alumina content increases, the  $\ln[NL_\alpha]$  decreases dramatically until a concentration point near 3.5 mole% (roughly 5 mass%), after which  $\text{Al}_2\text{O}_3$  additions only modestly decrease  $\ln[NL_\alpha]$ . It is interesting to note that the EM-07 glasses with  $\text{Al}_2\text{O}_3$  concentrations ranging from 4 to 14 mole% show slightly “U-shaped” behavior [19]. Two hypotheses were offered to explain this non-linear effect. Feng et al. suggested that small additions of  $\text{Al}_2\text{O}_3$  increase the total fraction of glass formers

until a threshold is achieved above which significant effects are not realized [18]. Vienna et al. suggested that small additions strongly reduced non-bridging oxygen concentrations, improving short-term durability, while larger additions had a lesser effect [20]. Neither of these speculations were supported by data or further developed.

Vienna et al. developed PCT response models with a broader database of glasses that extended many of the component concentration ranges [21]. Notably, the maximum  $\text{Al}_2\text{O}_3$  concentration was increased to 20 mass% (compared to the  $\leq 13$  mass% in the baseline WTP models). This model includes non-linear terms for  $\text{Al}_2\text{O}_3$  concentration. This model was applied to advanced glasses, including those with relatively high  $\text{Al}_2\text{O}_3$  concentrations. Fig. 2 shows the residual  $\ln[NL_B]$  (measured minus predicted  $\ln[NL_B]$ ) for advanced glasses using the Vienna 2009 model [21]. The model under-predicts  $\ln[NL_B]$  progressively more as the concentration of  $\text{Al}_2\text{O}_3$  increases. This suggests that as  $\text{Al}_2\text{O}_3$  concentration increases above roughly 19 mass% increases PCT responses.

## 1.2. Previous studies evaluating $\text{Al}_2\text{O}_3$ impacts on glass durability

The impacts of  $\text{Al}_2\text{O}_3$  concentration on other glass durability related measurements have been reported with the results summarized here [23–29].

- Kim et al. [17] reported the impact of  $\text{Al}_2\text{O}_3$  concentration on the Materials Characterization Center Test 1 (MCC-1); a test performed at  $90^\circ\text{C}$  and  $10\text{ m}^{-1}$ , in deionized water for 28 d. The results showed a “U-shaped” response for glasses with 6 or 9 mass%  $\text{B}_2\text{O}_3$  and a monotonically negative response for soda-lime silicate glasses or sodium-alumino-borosilicate glasses with 12 mass%  $\text{B}_2\text{O}_3$ . They attribute the negative effect to removal of non-bridging oxygen (NBO) concentrations caused by the alkali ion charge compensating added  $\text{Al}_2\text{O}_3$  in the four-coordinated state ( $^4\text{Al}$ ). The increase in MCC-1 responses at  $\text{Al}_2\text{O}_3$  concentrations above 12 mass% in glass was not understood, nor was the reason that it occurred only for the glasses

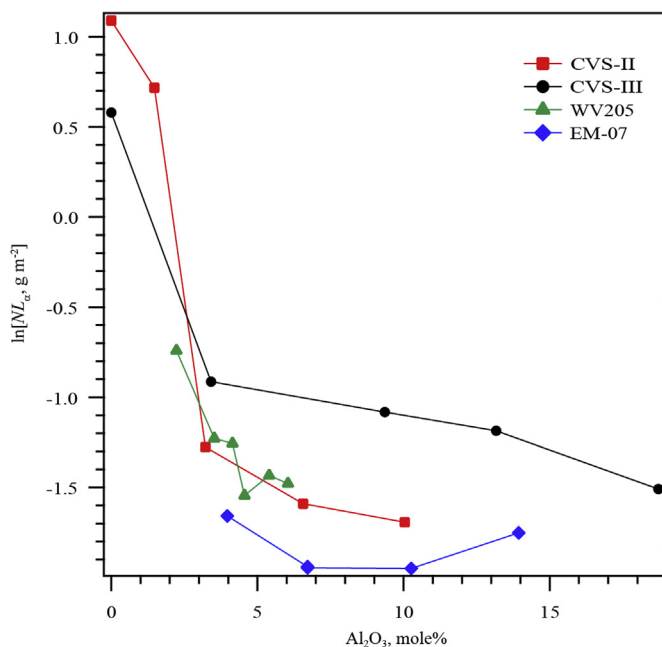


Fig. 1. Effect of  $\text{Al}_2\text{O}_3$  mole% change on  $\ln[NL_\alpha, \text{g m}^{-2}]$  based on data from CVS-II study [ $\alpha = \text{B}$ ] [15], CVS-III study [ $\alpha = \text{Na}$ ] [16], WV205 study [ $\alpha = \text{B}$ ] [18], and EM-07 study [ $\alpha = \text{B}$ ] [19].

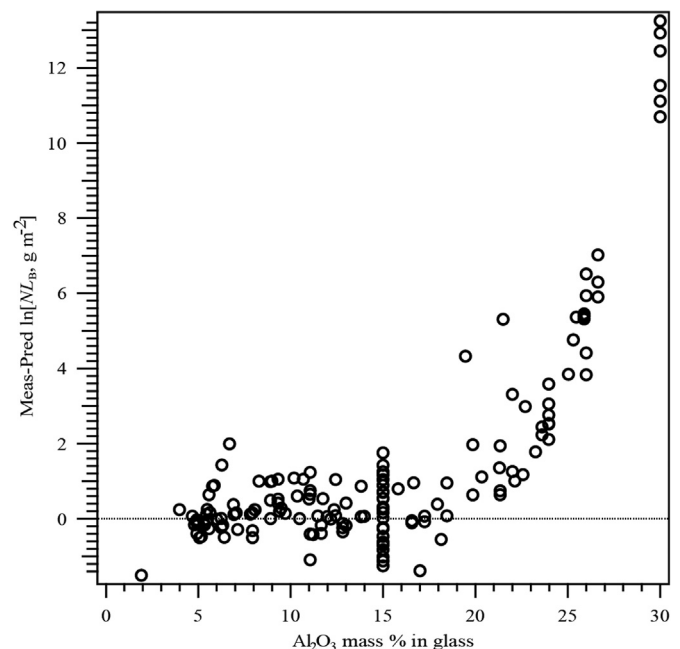


Fig. 2. Impact of  $\text{Al}_2\text{O}_3$  concentration on PCT prediction residuals (after Vienna et al. [22]).

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