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Full length article

Numerical and experimental analysis of the effect of volumetric energy absorption in powder layer on thermal-fluidic transport in selective laser melting of Ti6Al4V

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HIGHLIGHTS

- Volumetric heat source has been used in modeling of SLM of Ti6Al4V.
- Simulation of moving single scan and multi-scan.
- Porosity is estimated and compared with the experiment.
- Porosity forms due to improper melting in the powder bed.
- The solidification parameters are calculated to estimate solidified grain structure.

ARTICLE INFO

Keywords: Selective laser melting Volumetric heat absorption Experimental validation Single and multi-track build Porosity Solidification interface parameters

ABSTRACT

A volumetric heat source is used in numerical modeling of selective laser melting (SLM) of Ti6Al4V powder. Single track and multi-track SLM simulations are performed by varying the two key process parameters-laser power and scan speed. The model is validated with the published experimental results for melt pool shape, size and temperature. The predictions are in good agreement with the experiments at low to medium energy density. The validated model is used for investigating the thermo-fluidic transport during SLM of Ti6Al4V and examining the dependence of the melt pool characteristics on the process parameters. As-solidified porosity is calculated numerically for the multi-track simulations and its formation is delineated with the transport phenomena. The predicted porosity compares reasonably well with the experimental values. Solidification parameters, such as temperature gradients and cooling rate are calculated at the instantaneous location of the solidification front and analyzed. This analysis suggests the formation of fully columnar grains of different sizes along the width and depth of the melt pool. Overall, the model provides a good description of thermo-fluidic transport in SLM of Ti6Al4V powder and the resulting temperature field, melt pool characteristics, as-solidified porosity and the expected grain structure. Based on the current analysis, an optimum processing window of 50–70 J mm⁻³ energy density is suggested for SLM of Ti6Al4V powder.

1. Introduction

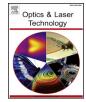
Selective laser melting (SLM) can offer the metal manufacturing industries the much-needed ability to produce parts without being concerned of design complicacies. It has the potential to help in reducing the longer lead times involved with the traditional manufacturing of intricate parts or products such as scaffolds, turbine blades etc. SLM is a laser assisted powder layer fusion based additive manufacturing process. It uses a focused laser beam to melt the metal powder layer to form one layer or cross section of the part being manufactured. The process is repeated till the complete part is built layer by layer. The SLM produced parts have shown unique microstructures and mechanical properties that are unlike the conventional manufacturing processes [1]. Along with these, the very high energy and material efficiency of SLM makes it highly cost-effective for single and small batch production. These aspects have helped this process gain considerable interest from industries like aerospace, automobile, tooling, biomedical and jewelry. However, there exist several issues that impede the large scale commercial utilization of the SLM process, such as (i) Inadequate understanding of the various observed defects, such as porosity,

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Nomenclature Pe			Peclet number
		Φ	powder layer porosity
α	absorptivity	q	heat source
$\alpha_{\rm m}$	mass fraction	$q_{\rm loss}$	heat loss flux
a, b	coefficients of equation	Ŕ	solidification front velocity
$\beta_{\rm T}$	coefficient of thermal expansion	$R_{ m L}$	laser beam spot radius
$c_{\rm p}$	specific heat capacity at constant pressure	Re	Reynolds number
$c_{\rm pS}$	specific heat capacity of solid at constant pressure	ρ	density
$c_{\rm pL}$	specific heat capacity of liquid at constant pressure	$\rho_{\mathbf{P}}$	density of powder layer
δ	powder layer thickness	ρs	density of solid
$D_{ m L}$	diffusion coefficient	ρ_L	density of liquid
ε	emissivity	S	absorption depth (Optical penetration depth)
Ε	energy density	σ	Stefan-Boltzmann constant
\overrightarrow{F}	source term in the momentum conservation equation	t	time
$f_{\rm L}$	liquid fraction	Т	temperature
$\begin{array}{c} f_{\rm L} \\ \overrightarrow{g} \end{array}$	acceleration due to gravity	$T_{\rm S}$	solidus temperature of Ti6Al4V
γ	surface tension coefficient	$T_{ m L}$	liquidus temperature of Ti6Al4V
G	temperature gradient magnitude	T_{∞}	ambient temperature
GR	cooling rate	$T_{\rm ref}$	reference temperature for Boussinesq approximation
G/R	interface stability factor	T_{β}	beta transition temperature of Ti6Al4V
h	convective heat transfer coefficient	$T_{\rm max}$	maximum (peak) temperature
k	thermal conductivity	\vec{u}	molten metal flow velocity field
$k_{ m P}$	thermal conductivity of powder layer	$ \overrightarrow{u} _{\max}$	maximum (peak) magnitude of molten metal velocity field
$k_{ m L}$	thermal conductivity of liquid	и	x-component of molten metal flow velocity
$k_{\rm S}$	thermal conductivity of solid	ν	y-component of molten metal flow velocity
λ	scan spacing	vol.	melt pool volume
L	latent heat of melting	$V_{ m L}$	laser beam scan speed
μ	dynamic viscosity	x	x- coordinate
n	exponent	ξ	Mushy zone constant
ω	Mushy zone constant	у	y-coordinate
р	pressure	Z	z-coordinate
$P_{\rm L}$	laser power	z_0	maximum height of domain in z-direction

delamination, cracking and thermal distortion; (ii) Lack of sufficient know-how of SLM manufacturability of different materials; (iii) Very high cost of the SLM equipment and raw materials. In addition, the SLM parts are seldom used as-built and require considerable post processing such as heat treatment etc. further obstructing the large scale commercial utilization. This has resulted in a significant amount of resources being invested in SLM research in recent times, both in industries and academia.

A number of interconnected thermal and physical phenomena occur

during SLM (Fig. 1), which control the solidified microstructure and mechanical properties of the end product. This includes laser-matter interaction, rapid heating and solidification, conduction-convectionradiation heat transfer and Marangoni and buoyancy driven flow of molten metal [2]. The process parameters and material properties play a significant role in controlling these phenomena. Therefore, for a particular material, the desired microstructure and mechanical properties in the end product can be obtained by using the optimum process parameters. The SLM process has a number of process parameters-laser

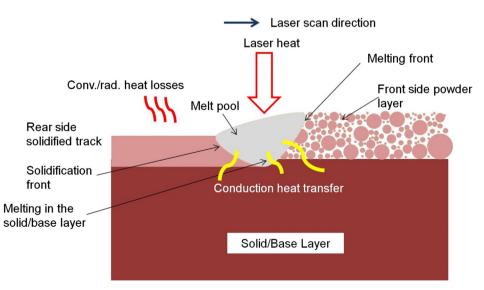


Fig. 1. Schematic of thermal and physical phenomena occurring during SLM.

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