

Spectroscopic properties and energy transfer of Nd³⁺/Ho³⁺-doped Ga₂O₃-GeO₂ glass by codoping Yb³⁺ ion

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Abstract: This study presented the luminescence properties of Nd³⁺/Yb³⁺/Ho³⁺ dopant ions inside a host based on Ga₂O₃-GeO₂-Li₂O (GGL) glass. The measured differential scanning calorimetry result showed that GGL glass exhibited excellent stability against devitrification with $\Delta T=135$ °C. Obvious 543 and 657 nm emissions were observed in Nd³⁺/Ho³⁺-codoped sample. The incorporation of Yb³⁺ into Nd³⁺/Ho³⁺-codoped glass system had resulted in enhanced upconversion emission intensity under the excitation of 808 nm and/or 980 nm laser diode (LD). The possible mechanisms and related discussions on this phenomenon were presented. It was noted that the presence of Yb³⁺ yielded an enhancement about 7 and 11 times in the 543 and 657 nm emission intensities respectively under 808 nm excitation due to the energy transfer from Nd³⁺ to Ho³⁺ via Yb³⁺ ion. Here Yb³⁺ played a major role as a bridging ion. While enhanced 543 and 657 nm emission intensities under the excitation of 980 nm LD originated from the sensitization effect of Yb³⁺. Our results showed that Nd³⁺/Ho³⁺/Yb³⁺ triply doped GGL glass might be a promising candidate for the development of visible-laser materials.

Keywords: upconversion luminescence; Nd³⁺/Yb³⁺/Ho³⁺; Ga₂O₃-GeO₂ glass; energy transfer; rare earths

In recent years, high-power laser diode (LD) has attracted a great deal of interest because of their technological importance in developing compact up-conversion (UC) visible-laser systems as pumping sources^[1,2]. Although several approaches have been pursued, the frequency UC laser by developing rare-earth (RE)-doped glass materials has important advantages over the others^[3–8]. Using trivalent RE ions, such as Er³⁺, Tm³⁺ and Ho³⁺ luminescence centers have been the focus of the research on UC luminescence materials^[1,7–12]. In particular, the Ho³⁺ ion is an extensively studied RE ion due to its numerous radiative emissions in the visible and IR, which provides broad potential applications toward NIR/Vis solid-state lasers^[9,13]. However, Ho³⁺ cannot be excited directly by the commercial 980 nm and/or 808 nm LD^[14,15]. Codoping with other RE ions with strong absorption bands at a wavelength of ~800 nm, such as Nd³⁺, could efficiently pump the Ho³⁺ via the ⁴I_{9/2}→⁴F_{5/2}, ²H_{9/2} transition of Nd³⁺ ions^[1,14,15]. On the other hand, UC luminescence could be obtained from the levels of RE³⁺ ions which can not be excited and/or enhanced without incorporation of Yb³⁺ ions due to intense absorption band of Yb³⁺ located at 1 μm corresponding to the emission band of a commercially available high-power InGaAs LD^[3].

Over the past years glass materials have aroused wide

interest based on their obvious advantages such as relatively easy synthesis compared with crystals, which urges us to study it in a wide range of composition and to obtain large samples^[9]. Holmium has been widely studied in various fluoride glasses^[13,16]. However, by considering the poor chemical durability of fluoride glasses, it is considered to be more important to search for novel glass systems to be used as host materials. The glasses based on Ga₂O₃-GeO₂ system as a promising group of heavy metal oxide glasses have been paid great attention for optoelectronics applications because of their excellent optical properties, such as the good transparency in the mid-infrared, and the possibility to incorporate a large amount of RE ions, etc.^[17–20]

Herein, the main objective of this work is to carry out a detailed study on photoluminescence properties of the Nd³⁺/Yb³⁺/Ho³⁺-doped Ga₂O₃-GeO₂-Li₂O (GGL) glass upon excitation of 808 nm and/or 980 nm to examine their suitability as potential UC emission laser. Effects of Yb³⁺ concentration on UC luminescence characteristics and mechanisms have also been investigated and discussed under the excitation of 808 and 980 nm. Simultaneously the fluorescence spectra in the range of 1000–2050 nm have been investigated under the 808 nm excitation.

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1 Experimental

GGL glasses with the molar composition of (99-*x*)(15Ga₂O₃-75GeO₂-10Li₂O)-0.5Nd₂O₃-0.5Ho₂O₃-*x*Yb₂O₃ (*x*=0, 0.1, 0.3, 0.5, 0.7, named as GGL1, GGL2, GGL3, GGL4 and GGL5) were prepared using a conventional melting-quenching method from powders with analytical-reagent chemicals of Li₂CO₃, Ga₂O₃ (99.99%), GeO₂ (99.999%) and R₂O₃ (R=Nd, Yb, Ho) (99.99%) in platinum crucible for 20 min at around 1480 °C, followed by a quenching in air on a stainless-steel plate. After annealing, all the samples were polished and cut into the size of 20 mm×20 mm×2.0 mm. Meanwhile the GGL glasses based on 15Ga₂O₃-75GeO₂-10Li₂O and 99(15Ga₂O₃-75GeO₂-10Li₂O)-0.5Nd₂O₃-0.5Yb₂O₃ were prepared.

Thermal analyses of samples were carried out using a Netzsch STA 449/C differential scanning calorimetry (DSC) at a heating rate of 10 °C/min from room temperature to 850 °C. Densities were measured according to the Archimedes' principle using distilled water as the medium. Refractive index was obtained by prism coupler. Measured density and refractive index of host glass are 3.995 g/cm³ and 1.6699 respectively. The absorption spectra of GGL1-GGL5 samples and transmission spectrum of host glass were measured with a PERKIN-ELMER Lambda 900 UV-Visible-NIR spectrophotometer with the resolution of 1 nm respectively. The fluorescence spectra in the range of 1000–2050 nm were obtained through a computer-controlled Triax 320 spectrofluorimeter with the resolution of 1 nm under excitation of an 808 nm LD, and the signal was detected with a PbSe detector (1000–5000 nm). The UC luminescence measurements were obtained with a computer-controlled Triax 320 spectrofluorimeter (Jobin-Yvon Corp.) upon continuous wave excitation of 808 nm and 980 nm LD (Coherent Corp). Emitted light was focused onto the monochromator and monitored at the exit slit by R928 photomultiplier tube (500–750 nm). All the optical measurements were performed at room temperature.

2 Results and discussion

2.1 Thermal property

Fig. 1 illustrates the DSC curve of GGL host glass. The measured glass transition temperature (*T_g*), the onset crystallization temperature (*T_x*), and the difference (ΔT) between *T_x* and *T_g* are 652, 517, and 135 °C, respectively, as shown in Fig. 1. The difference ΔT between the *T_g* and *T_x* could be a rough indicator of glass stability against devitrification. Fiber drawing contains a reheating process and should avoid any crystallization which could lead to the scattering loss and decrease the optical properties of fiber^[21]. Herein it is necessary to obtain a large ΔT value as much as possible. The ΔT value achieved to

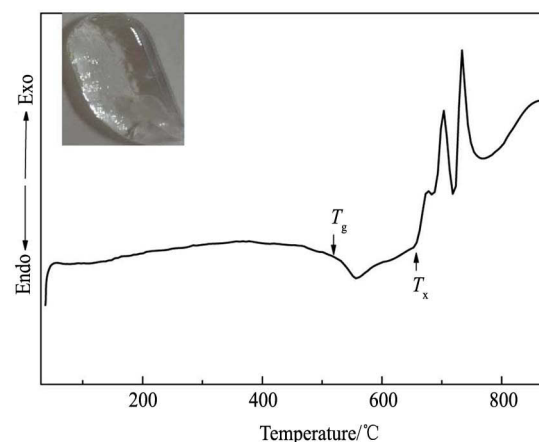


Fig. 1 DSC curve of GGL host glass (Inset shows the picture of transparent host glass)

135 °C for GGL host glass, indicating excellent glass stability against devitrification for fiber drawing in comparison with that of the fluorozirconate glass^[21]. Transparent GGL host glass has been prepared as exhibited in the inset of Fig. 1.

2.2 Absorption spectra

Fig. 2 shows the absorption spectra in the range of 250–2000 nm for Nd³⁺/Ho³⁺/Yb³⁺-doped samples when Yb³⁺ concentration changes from 0 to 0.7 mol.%. Each assignment corresponds to the excited level of RE³⁺ ions. Thirteen absorption bands of RE³⁺ are observed in Fig. 2, centered at the respective wavelengths of 1948, 1151, 970, 871, 802, 740, 680, 643, 580, 530, 452, 415 and 355 nm. The peak positions of RE³⁺-doped samples do not change with the incorporation of Yb³⁺. The samples illustrate the short cutoff band less than 350 nm. Fig. 2 inset shows the transmission spectrum of host glass.

2.3 Analysis of UC and fluorescence spectra under the excitation of 808 nm

Fig. 3 shows typical room-temperature visible emission for the Nd³⁺/Ho³⁺/Yb³⁺-doped samples pumped at 808 nm LD. The insets (a) and (b) of Fig. 3 depict the

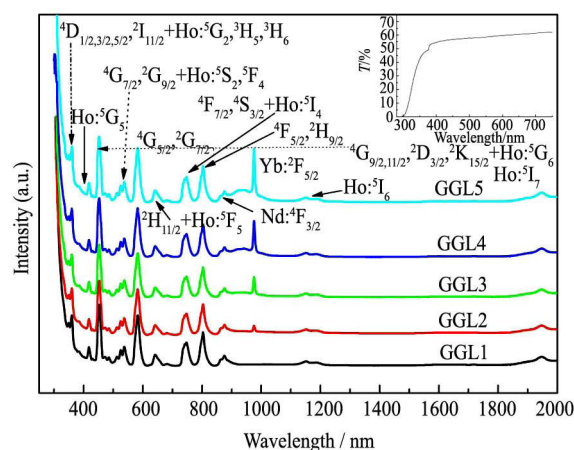


Fig. 2 Absorption spectra of Nd³⁺/Yb³⁺/Ho³⁺-doped GGL glasses (Inset shows the transmission spectrum of host glass)

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