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# Effect of polyaniline on hydrogen absorption–desorption properties and discharge capacity of AB<sub>3</sub> alloy

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#### Abstract

A series of AB<sub>3</sub>/PANI composites were prepared by adding polyaniline (PANI) into  $La_{0.7}Mg_{0.25}Ti_{0.05}Ni_{2.975}Co_{0.525}$  (AB<sub>3</sub>) hydrogen storage alloy, which was prepared by magnetic levitation melting, and the composites were investigated by XRD and SEM. The effects of PANI concentration on the hydrogen absorption–desorption properties and discharge capacities of AB<sub>3</sub> alloy were systematically studied by P–C–T isotherms and LAND battery test system, respectively. The results indicated that the addition of PANI did not change the hydrogen absorption capacity of AB<sub>3</sub> alloy distinctly, while the hydrogen desorption plateau pressure of AB<sub>3</sub> alloy decreased firstly, and then increased with the increase in the PANI concentration, the minimum plateau pressures of 0.022, 0.1, 0.321 and 0.55 MPa were obtained with PANI concentration of 2 wt% at 25, 50, 80 and 100 °C, respectively. It was theoretically verified by the changes in enthalpy and entropy of AB<sub>3</sub>/PANI hydrides dehydrogenation which were calculated by Van't Hoff equation. In the present paper, the phenomenon that PANI improved the hydrogen absorption kinetics of AB<sub>3</sub> alloy was found; the influence of PANI concentration on hydrogen absorption kinetics of AB<sub>3</sub> alloy was more apparent at higher temperature. The activation energies of dissolved hydrogen in AB<sub>3</sub>/PANI hydrides were calculated from hydrogen absorption kinetics and the Arrhenius equation. LAND experiments demonstrated that, the AB<sub>3</sub>/PANI electrodes composites possessed higher cycling discharge capacity retention rates than AB<sub>3</sub> electrode alloy.

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### 1. Introduction

As a hydrogen absorption media for metal alloys, polymer attracted much attention in the last several years [1]. Uemura and coworkers [2] reported the effect of polystyrene on the hydrogen absorption properties of LaNi<sub>5</sub> alloy. They found that the hydrogen absorption capacity of the alloy was not influenced by the encapsulating treatment. Tinge [3] also obtained the same conclusion by studying the hydrogen absorption and desorption kinetics of the polyethylene/LmNi<sub>4</sub> (Lm = lanthanide mixture).

Recently, a study on the mechanism of hydrogen storage in polymer was reported by Schmidt [4], who considered that polymer could improve the hydrogen movement in metal alloy. And then, the experimental results of Isobe and Yamauchi et al. [5,6] showed that addition of certain polymer could increase the hydrogen absorption capacity of Pt or Pd nanoparticles. PANI, as a normal polymer, does not have the hydrogen absorption kinetics of AB<sub>2</sub> alloy [8]. So it would be interesting to systematically study the effect of PANI on hydrogen absorption–desorption properties of AB<sub>3</sub> alloy, a typical hydrogen storage alloy. In the present work, PANI was added

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into  $AB_3$  alloy to improve unfavorable properties such as hydrogen absorption kinetics of  $AB_3$  alloy. The effect of PANI concentration on the hydrogen absorption–desorption and kinetic properties of  $AB_3$  alloy was investigated by a P-C-T apparatus. The discharge capacities of  $AB_3/PANI$  electrodes composites were also studied in this paper.

### 2. Experimental

### 2.1. Sample preparation

PANI was prepared by  $(NH_4)_2S_2O_8$  oxidation of aniline in 2 M HCl aqueous solution [9]. Before compounding with AB<sub>3</sub> alloy, it was dried at 100 °C under vacuum for 4 h to wipe off water.

The  $AB_3$  alloy was prepared by magnetic levitation melting under argon atmosphere. The alloy button was crushed into fine powders below 200 meshes.

### 2.2. Preparation and characterization of AB<sub>3</sub>/PANI composites

The AB<sub>3</sub> alloy below 200 meshes was activated by 4 MPa hydrogen at room temperature for three times [10], then the activated AB<sub>3</sub> alloys were compounded with 0, 1, 2, 4 wt% (to AB<sub>3</sub> alloy) PANI, respectively, by hand milling for 30 min. The AB<sub>3</sub>/PANI composites were characterized by XRD (Rigaku D/max-2500, Cu K $\alpha$ , 40 kV, 250 mA) and scanning electron microscopy (JSM-6360LV).

### 2.3. Hydrogen pressure-composition isotherm and kinetics studies

Hydrogen absorption–desorption behaviors of AB<sub>3</sub>/PANI composites were measured using a Severts-type apparatus up to a hydrogen pressure of 4 MPa similar to our previous work [11]. Repeated measurements of absorption–desorption sequence were carried out in the temperature range of 25–100 °C. The absorption kinetics data were automatically collected by recording the change in pressure as a function of time at constant temperature.

### 2.4. Charge-discharge capacity

The charge–discharge measurements were performed on a LAND automatic battery testing system as described elsewhere [12,13]. The tested electrodes were fabricated by mixing 0.1 g AB<sub>3</sub>/PANI composites with 0.3 g electrolytic Ni powder. The above mixture containing AB<sub>3</sub>/PANI composite and Ni powder was pressed into a pellet under a pressure of 30 MPa. Both sides of the pellet were coated with two foam nickel sheets, then pressed at 10 MPa and tightly spot-welded. A nickel lead wire was attached to this pressed foam nickel sheet by spot welding. The NiOOH/Ni(OH)<sub>2</sub> electrode and Hg/HgO electrode were used as counter and reference electrode, respectively. The electrolyte was 6 M KOH aqueous solution. The electrodes were charged for 3 h at a current density of 300 mA g<sup>-1</sup>, rested



Fig. 1. XRD patterns of the AB<sub>3</sub>/PANI composites.

5 min and then discharged to -0.6 V versus Hg/HgO electrode at a current density of  $100 \text{ mA g}^{-1}$ . The measurements were carried out at room temperature of  $25 \,^{\circ}\text{C}$ .

### 3. Results and discussion

### 3.1. Crystal structure

Fig. 1 shows the X-ray diffraction patterns for the AB<sub>3</sub> alloy, PANI and AB<sub>3</sub>/2 wt% PANI composite. For PANI, a broad Bragg peak was observed at approximately  $25^{\circ}(2\theta)$ , which confirmed that the prepared PANI was amorphous. As to samples of AB3 alloy and AB3/2 wt% PANI composite, two main phases, i.e. the (La, Mg)Ni<sub>3</sub> phase with PuNi<sub>3</sub>-type rhombohedral structure and LaNi<sub>5</sub> phase with CaCu<sub>5</sub>-type hexagonal structure were observed in their XRD patterns [14,15]. However, in the XRD pattern of AB<sub>3</sub>/2 wt% PANI composite, the PANI phase was not observed for its low intensity. SEM images of (a)  $AB_3$  (< 200 meshes, without other treatment), (b) AB<sub>3</sub>/0 wt% PANI composite (with activation and hand milling), (c) AB<sub>3</sub>/2 wt% PANI composite are shown in Fig. 2. It was found that the average particle size of these materials reduced with the treatment of activation and hand milling, the average particle size of AB<sub>3</sub>/2 wt% PANI composite was the smallest, about 25 µm.

### 3.2. P-C-T isotherms

Hydrogen absorption-desorption isotherms and the maximum hydrogen storage capacities of AB<sub>3</sub>/PANI composites at different temperatures (25, 50, 80 and 100 °C) are shown in Fig. 3 and Table 1, respectively. It can be seen that at 25, 50, 80 and 100 °C, the maximum hydrogen absorption capacities of AB<sub>3</sub> alloy were the highest, and they decreased with the increase in PANI concentration. This result was in good agreement with that observed in Ref. [8], because PANI does not Download English Version:

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