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# High hydrogen content super-lightweight intermetallics from the Li–Mg–Si system

Volodymyr Pavlyuk<sup>*a,b*</sup>, Grygoriy Dmytriv<sup>*a*</sup>, Ihor Chumak<sup>*c*</sup>, Oliver Gutfleisch<sup>*d,e*</sup>, Inge Lindemann<sup>*d*</sup>, Helmut Ehrenberg<sup>*c,d,e,\**</sup>

<sup>a</sup> Ivan Franko Lviv National University, Department of Inorganic Chemistry, Kyryla i Mefodia 6, 79005 Lviv, Ukraine <sup>b</sup> Częstochowa Jan Długosz University, Institute of Chemistry, Environmental Protection and Biotechnology, al. Armii Krajowej 13/15, 42200 Częstochowa, Poland

<sup>c</sup>Karlsruhe Institute of Technology (KIT), Institute for Applied Materials (IAM), Hermann-von-Helmholtz-Platz 1,

D-76344 Eggenstein-Leopoldshafen, Germany

<sup>d</sup> IFW Dresden, PO Box 270116, 01171 Dresden, Germany

<sup>e</sup> Technische Universität Darmstadt, Institut für Materialwissenschaft, Petersenstr. 23, 64287 Darmstadt, Germany

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#### ABSTRACT

The existence of Li-rich super-lightweight intermetallics in the Li-Mg–Si ternary system has attracted attention for high capacity hydrogen storage materials. The hydrogenation properties of the alloys were studied by thermogravimetric analysis, differential scanning calorimetry in H<sub>2</sub> atmosphere and X-ray diffraction. The Li-rich alloy absorbs the highest amount of hydrogen (8.8% w/w for Li<sub>70</sub>Mg<sub>10</sub>Si<sub>20</sub>), while the Mg-rich alloy (Li<sub>30</sub>Mg<sub>40</sub>Si<sub>30</sub>) absorbs 6.0% w/w H<sub>2</sub> and shows the first experimental evidence for LiMgH<sub>3</sub> formation with LiNbO<sub>3</sub>-type structure during hydrogenation.

The extension of homogeneity range of existing phases from Li–Mg–Si system was found and the crystal structures of four ternary phases,  $Li_{2+x}Mg_{1-x}Si$  (x = 0.268),  $Li_{65-x+y}Mg_{30+x-y}Si_{33}$  (x = 9.15, y = 1.23),  $Li_{7-x+y}Mg_{5+x-y}Si_4$  (x = 1.14, y = 0.61) and  $Li_{x+y}Mg_{2-x}Si$  (x = 0.51, y = 0.39), were solved and more precise refined using X-ray single crystal diffraction data. Electronic structure calculations reveal an increased occupation of electronic states at the Fermi level for  $Li_{12+x}Mg_{3-x}Si_4$  in comparison to  $Li_{2+x}Mg_{1-x}Si$ . The results of crystal orbital Hamilton population (COHP) and integrated crystal orbital Hamilton population (iCOHP) calculations for both structures indicate the strongest interactions between Mg–Si (-iCOHP = 1.999 eV) and the weakest between Li–Li (-iCOHP = 0.049 eV). Copyright © 2013, Hydrogen Energy Publications, LLC. Published by Elsevier Ltd. All rights reserved.

#### 1. Introduction

The main requirements for modern hydrogen storage materials in automotive applications are a high gravimetric density (above 6.5% w/w gravimetric capacity of hydrogen), absorption/desorption of hydrogen at moderate temperatures and pressures, low-cost and environmental safety [1]. Conventional metal hydrides, such as  $LaNi_5$  and its substitutional derivatives, titanium and zirconium alloys which are commonly used for hydrogen storage systems have storage

<sup>\*</sup> Corresponding author. Karlsruhe Institute of Technology (KIT), Institute for Applied Materials (IAM), Hermann-von-Helmholtz-Platz 1, D-76344 Eggenstein-Leopoldshafen, Germany.

E-mail address: helmut.ehrenberg@kit.edu (H. Ehrenberg).

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capacities less than 2% w/w H<sub>2</sub> and cannot satisfy this need. New materials developed on the basis of light metals such as Li and Mg are promising candidates as potential light-weight hydrogen storage material. The binary system of lithiummagnesium does not form intermetallic phases, but ternary phases Li-Mg-Si and the isothermal section of the Li-Mg-Si phase diagram at 200 °C are known [2,3]. Later, the experimentally supported calculated phase equilibria in the entire composition and temperature range of the Li-Mg-Si system are presented, including the liquidus surface and invariant reactions [4]. Accordingly, five intermetallic compounds are known: Li2MgSi, LiMg2Si, Li12Mg3Si4, Li8MgSi6 and Li5MgSi4. The crystal structure data of cubic  $\mathrm{Li}_{12}Mg_3Si_4$  and monoclinic Li<sub>8</sub>MgSi<sub>6</sub> ternary phases are reported [5-7]. Later, the two compounds Li<sub>8</sub>MgSi<sub>6</sub> (53.3 at.% Li, 6.7 at.% Mg, 40.0 at.% Si) and Li<sub>5</sub>MgSi<sub>4</sub> (50.0 at.% Li, 10.0 at.% Mg, 40.0 at.% Si) were identified as the same phase with an extended homogeneity range [8]. Two possible structural models with  $a \approx 6.4$  Å (I) and  $a \approx 12.8$  Å (II), based on the space groups F-43m, P-43m and Fm-3m were proposed [2,9-11] for Li<sub>2</sub>MgSi. These phases are described by the general formula  $Li_{2x}Mg_{2-x}Si$ , and model I is realized at 0 < x < 0.547 in Fm-3m, at 0.55 < x < 0.65 in P-43m, and the model II exists at 0.65 < x < 0.8 [3]. The initial structure data for the  $Li_{2+x}Mg_{1-x}Si_{1-x/4}$  hexagonal phase with Na<sub>3</sub>As-type structure (a = 4.48 Å, c = 8.00 Å) were described by Nesper and Curda [12]. Unfortunately, atomic parameters are not given.

The properties of  $Li_2MgSi$  as a hydrogen storage material were investigated and a reversible hydrogen uptake and release of about ~2.8% w/w at T ~ 300 °C was measured [10].

This work reports on syntheses, phase diagram, crystal structures, electron density distributions and hydrogenation properties of some alloys from the ternary Li–Mg–Si system.

#### 2. Material and methods

#### 2.1. Synthesis and phase analysis

The investigated samples were prepared from the following reactants: lithium rod (99.9%, Alfa Aesar, Karlsruhe, Germany), magnesium and silicon (99.999 at.%). Appropriate amounts were mixed according to the aimed stoichiometry of the product and filled into tantalum crucibles in the glove box under argon atmosphere. These crucibles have been sealed by arc-welding in dry argon atmosphere. The reaction between the metals was carried out in an induction furnace at 1100 °C. After 15 min, the samples were rapidly cooled down to room temperature by removing the crucibles from the furnace into ambient conditions. The reaction product was powdered in an agate mortar, filled into a capillary of 0.3 mm diameter and sealed for X-ray powder diffraction (XRD) on a STOE STADI P diffractometer (Co-Ka1 radiation) in Debye-Scherrer mode with a linear position sensitive detector with 6° aperture.

#### 2.2. Structure analysis and refinement

The crystal structures of  $Li_{2+x}Mg_{1-x}Si$  (x = 0.268;  $\tau_3$ ),  $Li_{x+y}Mg_{2-x}Si$  (x = 0.51, y = 0.39;  $\tau_7$ ),  $Li_{65-x+y}Mg_{30+x-y}Si_{33}$ (x = 9.15, y = 1.23;  $\tau_5$ ) and  $Li_{7-x+y}Mg_{5+x-y}Si_4$  (x = 1.14, y = 0.61) were determined by single crystal X-ray diffraction and confirmed by powder diffraction. The single crystals were selected under dried paraffin, fixed and sealed in a glass capillary. Intensity data for Li<sub>2+x</sub>Mg<sub>1-x</sub>Si and Li<sub>x+y</sub>Mg<sub>2-x</sub>Si were collected using a BRUKER Kappa Apex II diffractometer, while for  $Li_{65-x+y}Mg_{30+x-y}Si_{33}$  and  $Li_{7-x+y}Mg_{5+x-y}Si_4$  intensity data were collected using a SuperNova (Agilent Technologies) diffractometer, equipped with a CCD detector. The structures have been solved by direct methods and were refined using the SHELX-97 program package [13]. An initial parameter set automatic interpretations of obtained from was direct methods using SHELXS-97, and this structure model was further refined until convergence was reached using SHELXL-97.

The crystal structure refinement from X-ray powder diffraction was performed by the Rietveld method using the FullProf software package [14].

Further details of the crystal structure investigations can be obtained from the Fachinformationszentrum Karlsruhe, 76344 Eggenstein-Leopoldshafen, Germany, (Fax: +(49)7247-808-666; e-mail: crysdata@fiz-karlsruhe.de, http://www. fizkarlsruhe.de/request\_for\_deposited\_data.html) on quoting the appropriate CSD numbers 425657, 425658, 425659, 425660 and 425661.

#### 2.3. Hydrogenation measurements

Thermogravimetric analysis (TGA) was performed with the IGA (Hiden Isochema) in 5 bar static  $H_2$  pressure heating with 5 K/min up to 500 °C. The reaction products were sealed in 0.7 mm capillaries and analysed by XRD on a STOE STADI P diffractometer (Mo-K $\alpha_1$  radiation). Differential scanning calorimetry (DSC) was carried out using a Netzsch DSC 204 HP heating in different  $H_2$  pressures with 5 K/min up to 590 °C.

#### 3. Calculation

#### 3.1. Electronic structure calculations

The linear muffin-tinorbital (LMTO) method [15-17] in its tight-binding (TB) representation [18] is a fast linearized form of the Korringa-Kohn-Rostoker (KKR) method [19,20] for electronic structure calculations and was applied on the Li-Mg-Si system. Basis sets composed of short-ranged atom-centered TB-LMTOs without empty spheres were used for the calculations using the TB-LMTO-ASA4.7 program with a scalar-relativistic Hamiltonian and atomicspheres approximations [21]. Electronic energies were calculated via density-functional theory (DFT) based on the local-density approximation (LDA) for the exchangecorrelation functional as parametrized by von Barth and Hedin [22]. Diagonalization and integration in reciprocal space were performed with the help of an improved tetrahedron method [23]. In order to evaluate various orbital interactions, density of states (DOS), the crystal orbital Hamilton population (COHP) curves [24], and the integrated COHP values (iCOHPs) were also calculated. The mapping of the electrons in the real space was obtained using the

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