



## Short communication

## Lithium storage mechanism in superior high capacity copper nitrate hydrate anode material



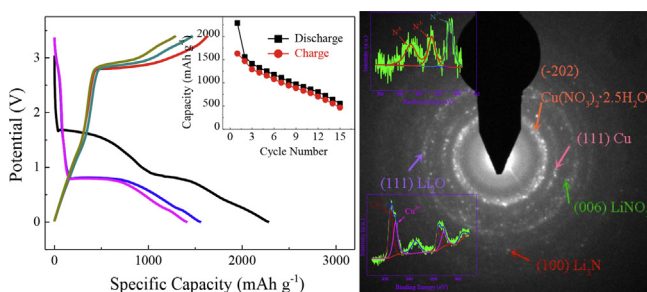
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## HIGHLIGHTS

- Lithium storage process in  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  is studied by various *ex-situ* techniques.
- $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  reveals quasi-reversible conversion mechanism for lithium storage.
- $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  shows a lithium storage capacity of  $2285.0 \text{ mAh g}^{-1}$ .

## GRAPHICAL ABSTRACT



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## ABSTRACT

Copper nitrate hydrate ( $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$ ) exhibits superior high lithium storage capacity ( $2285 \text{ mAh g}^{-1}$ ) as anode material for lithium-ion batteries. The structural transformation and lithium storage mechanism of  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  are thoroughly studied by various advanced analytical techniques. It is found that the lithium storage process of  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  is associated with a quasi-reversible electrochemical conversion reaction. During the discharge process, the electrochemical reaction of  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  with lithium results in the formation of  $\text{Cu}$ ,  $\text{LiNO}_3$ ,  $\text{Li}_3\text{N}$ ,  $\text{Li}_2\text{O}$  and  $\text{H}_2\text{O}$ . In the reverse charge process,  $\text{Cu}(\text{NO}_3)_2$  can be generated by a conversion reaction. As a result,  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  shows the reversible charge capacities of  $1632.1 \text{ mAh g}^{-1}$  in  $0.0\text{--}3.4 \text{ V}$  and  $689.1 \text{ mAh g}^{-1}$  in  $1.0\text{--}3.4 \text{ V}$ , respectively.

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## 1. Introduction

Recently, energy storage batteries for transportation and communication have gradually become the main power sources owing to global energy issues. Among these rechargeable energy storage batteries, lithium-ion batteries are dominant in the electric

vehicles and portable electronics market for their advanced characteristics, such as high energy density and long cycling life [1–4]. However, anode materials have been focused on the carbonaceous materials, such as graphite [5,6], carbon nanotube [7,8], since the lithium-ion batteries were developed in 1991. Nowadays, these carbonaceous materials cannot satisfy the demands of the market. Therefore, intensive worldwide attempts have been done to develop novel high capacity materials to take place of carbonaceous materials in the past two decades.

According to the previous studies, Cu-based metal oxides and nitrides, such as  $\text{CuO}$  [9–11] and  $\text{Cu}_3\text{N}$  [12], attract intensive

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attentions from researchers all over the world. CuO with high theoretical capacity ( $670 \text{ mAh g}^{-1}$ ), various structural patterns and inexpensive price has activated lots of attentions in recent years.  $\text{Cu}_3\text{N}$ , which exhibits high reversible capacity of  $1280 \text{ mAh g}^{-1}$ , good cycle life and excellent rate capability, is also examined as a candidate anode material for rechargeable lithium-ion batteries. Whether CuO,  $\text{Cu}_3\text{N}$ ,  $\text{CuF}_2$  or  $\text{CuCl}_2$ , their lithium storage mechanisms are based the reversible conversion reactions between  $\text{Li}_x\text{M}/\text{Cu}$  and  $\text{Li}/\text{Cu}_y\text{M}_z$  ( $M = \text{O}, \text{N}, \text{Cl}, \text{F}$ ) [9–14].

In most recent, copper nitrate hydrate ( $\text{Cu}(\text{NO}_3)_2 \cdot x\text{H}_2\text{O}$ ) with high specific capacity has been investigated by our group as a novel anode material for lithium-ion batteries [15]. It is found that  $\text{Cu}(\text{NO}_3)_2 \cdot x\text{H}_2\text{O}$  can deliver an initial discharge capacity higher than  $2200 \text{ mAh g}^{-1}$ . This superior high lithium storage capacity is much higher than all the ever reported transition metal oxides and nitrides, which generally show the initial discharge capacities of  $1000\text{--}1500 \text{ mAh g}^{-1}$  as anode materials. After 30 cycles, a large reversible charge capacity of  $597.6 \text{ mAh g}^{-1}$  can be maintained for  $\text{Cu}(\text{NO}_3)_2 \cdot x\text{H}_2\text{O}$ . As a result,  $\text{Cu}(\text{NO}_3)_2 \cdot x\text{H}_2\text{O}$  shows outstanding potential as high capacity anode material for lithium-ion batteries. However, the structural transformation and lithium storage mechanism of high-capacity  $\text{Cu}(\text{NO}_3)_2 \cdot x\text{H}_2\text{O}$  material during charge–discharge cycles were not investigated. To inspire the researchers to develop novel high capacity anode materials, it is necessary to discover the characteristics of superior high lithium storage capability.

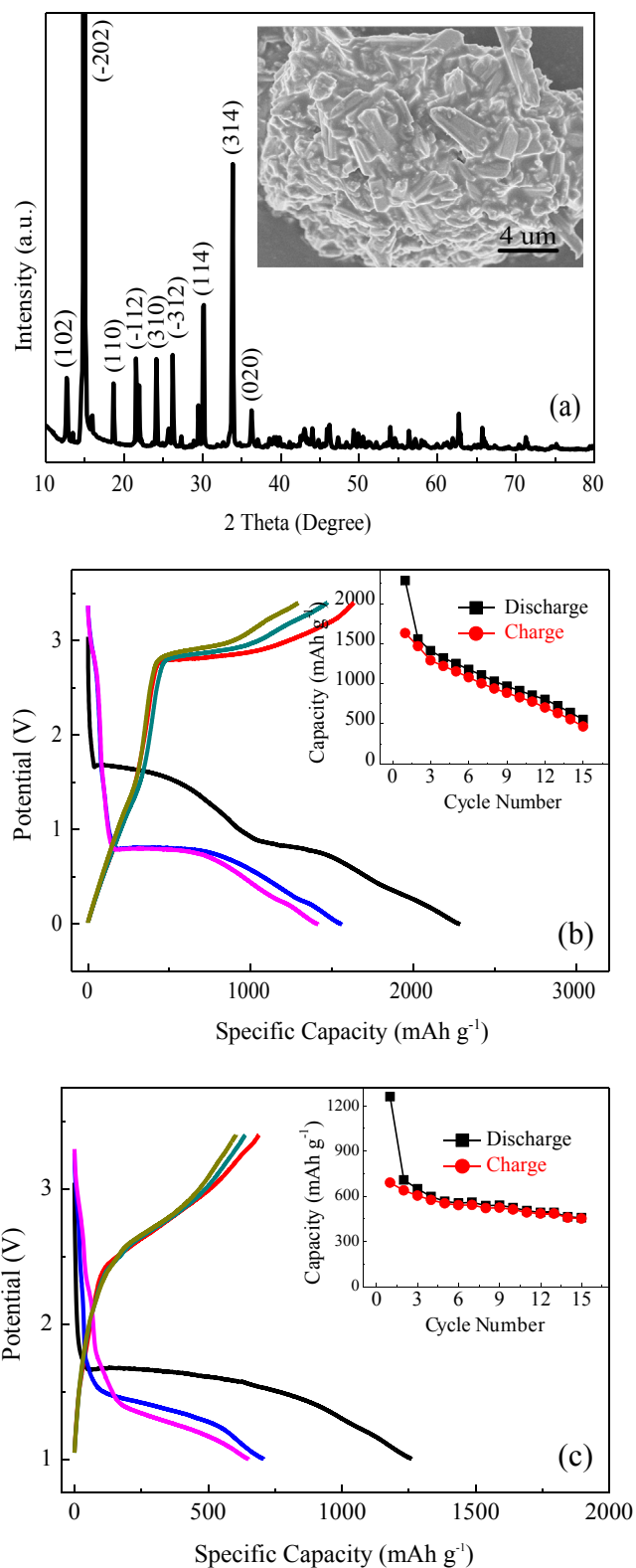
In this paper, the structural transformation and lithium storage mechanism of high-capacity  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  materials are thoroughly studied by *ex-situ* X-ray photoelectron spectroscopy (XPS), *ex-situ* high-resolution transmission electron microscopy (HRTEM), *ex-situ* selected-area electron diffraction (SAED), *ex-situ* Fourier transform infrared spectroscopy (FTIR) techniques. A quasi-reversible conversion reaction mechanism between  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5 \text{H}_2\text{O}$  with Li is discussed and proposed for the first time in this work.

## 2. Experimental

For being as active material,  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  powder (analytical grade, Aladdin Chemistry) was used as received and dried at  $80^\circ\text{C}$  under vacuum before electrode preparation. The slurry for working electrode was composed of  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  powder as active material, carbon black as conductive additive, and polyvinylidene fluoride as a binder with a weight composition of 4:1:1 in N-methyl pyrrolidinone solvent. Then the slurry was coated on a Cu-foil current collector and dried at  $80^\circ\text{C}$  under vacuum for 12 h. Discs with a diameter of 15 mm were cut and used as working electrodes.

The simulated  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}/\text{Li}$  cells were assembled by using  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  disc as working electrode, metal lithium foil as counter electrode, Whatman glass fiber as separator and  $1 \text{ mol L}^{-1}$   $\text{LiPF}_6$  dissolved in ethylene carbonate–dimethyl carbonate (1:1 in volume) as electrolyte. All the  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}/\text{Li}$  cells were assembled in an argon-filled glove box at room temperature.

The charge–discharge cycles were measured by a constant-current density ( $50 \text{ mA g}^{-1}$ ) on multi-channel Land battery test system. X-ray diffraction (XRD) pattern of  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  was collected by a Bruker D8 Focus powder X-ray diffraction instrument with Cu  $K\alpha$  radiation. SEM image was achieved by a Hitachi S3400 scanning electron microscopy. XPS investigation was measured by a focused and monochromatized Al  $K\alpha$  radiation with a Kratos Axis Ultra spectrometer. The structural evolution of  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  during charge–discharge cycles was directly imaged by a JEOL JEM-



**Fig. 1.** (a) XRD pattern and corresponding SEM image of  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  powders, (b) charge–discharge curves and corresponding cyclic performance of  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  in 0.0–3.4 V and (c) charge–discharge curves and corresponding cyclic performance of  $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$  in 1.0–3.4 V.

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