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# Li<sub>2</sub>CuVO<sub>4</sub>: A high capacity positive electrode material for Li-ion batteries



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# HIGHLIGHTS

# • We have been able to grow Li<sub>2</sub>CuVO<sub>4</sub> single crystals.

- We solved the crystal structure using single crystal data.
- Li<sub>2</sub>CuVO<sub>4</sub> is a good ionic conductor at temperatures above 100 °C.
- Interestingly, 2 lithium atoms can be reversibly extracted/inserted from/in Li<sub>2</sub>CuVO<sub>4</sub>.
- Li<sub>2</sub>CuVO<sub>4</sub> shows high rate capability with a capacity of 175 mA h g<sup>-1</sup> at 1C rate.

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# ABSTRACT

The new compound Li<sub>2</sub>CuVO<sub>4</sub> was synthesized by a solid state reaction route, and its crystal structure was determined from single crystal X-ray diffraction data. Li<sub>2</sub>CuVO<sub>4</sub> was characterized by galvanometric cycling, cycle voltammetry, and electrochemical impedance spectroscopy. The structure of Li<sub>2</sub>CuVO<sub>4</sub> is isotypic to *Pmn*2<sub>1</sub>-Li<sub>3</sub>VO<sub>4</sub>. It can be described as a disordered wurtzite structure with rows of Li1/Cu1 atoms alternating with rows of (Li2/Cu2)-V-(Li2/Cu2) atoms along [100]. All cations are tetrahedrally coordinated. The lithium and copper atoms are statistically disordered over two crystallographic sites. The electrochemical cycling between 2.0 and 4.7 V indicates that almost two lithium atoms could be extracted and re-intercalated. This delivers a maximum discharge capacity of 257 mA h g<sup>-1</sup> at a C/50 rate (theoretical capacity = 139 mA h g<sup>-1</sup> for one lithium). Li<sub>2</sub>CuVO<sub>4</sub> shows also high rate capability with a capacity of 175 mA h g<sup>-1</sup> at 1C rate. This demonstrates that Cu-based compounds can be very interesting as electrodes for Li-ion batteries if Cu-dissolution is avoided.

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# 1. Introduction

Polyanionic compounds have attracted much attention during the last decades because they have a potential for use as positive electrode in lithium or sodium-ion batteries. Several families of compounds [ $AMBO_3$ ,  $AMPO_4$ ,  $A_2MSiO_4$ ,  $AMSO_4X$ , and  $A_2MPO_4X$  (A: Li, Na; M: Mn-Ni; X: F, OH)] have particularly been intensively investigated [1–11 and ref. therein]. The LiMPO<sub>4</sub> are perhaps the most interesting for practical applications. Indeed, the olivine





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LiFePO<sub>4</sub> (theoretical capacity = 170 mA h  $g^{-1}$ ) is a particularly promising example of positive electrode material, due to its advantages of being environmentally benign, inexpensive, and safe.

Compared to phosphates with the olivine-type structure, the homologous vanadates  $LiMVO_4$  (M: Mn, Co, Ni, Cu) crystallize with different crystal structures at atmospheric pressure. LiMnVO<sub>4</sub> is isostructural with Na<sub>2</sub>CrO<sub>4</sub> whereas LiCoVO<sub>4</sub>, LiNiVO<sub>4</sub>, and LiCuVO<sub>4</sub> crystallize with the spinel-type structure [12–15]. As in the spinel-type structure is ccp. The main structural difference is the distribution of the cations on the available sites. This is at the origin of the superior electrochemical performances of the spinel materials. Nevertheless, among these vanadates, only LiNiVO<sub>4</sub> thin films were successfully used as positive electrode in lithium-ion batteries [15].

The LiMVO<sub>4</sub> vanadates have been also tested as negative electrode materials [16–20]. The intercalation of *x* Li into LiCuVO<sub>4</sub> led to the formation of multiple phases (Li<sub>2.5</sub>Cu<sub>0.5</sub>VO<sub>4</sub> + Cu for *x* = 1.5, Li<sub>3</sub>VO<sub>4</sub> + Cu for 1.5 < *x* < 2, and Li<sub>3+y</sub>VO<sub>4</sub> + Cu for 2 < *x* < 5) [20]. Since in our previous studies related to LiMnVO<sub>4</sub>, Li<sup>+</sup> in tetrahedral sites was successfully replaced by Cu<sup>+</sup> [21], we believe that the formation of these intermediate phases (Li<sub>2.5</sub>Cu<sub>0.5</sub>VO<sub>4</sub> and Li<sub>3</sub>VO<sub>4</sub>) is an indication of the existence of a solid solution Li<sub>3-x</sub>Cu<sub>x</sub>VO<sub>4</sub>. Particularly, the composition *x* = 1 would be of great interest as positive electrode material for lithium-ion batteries since it would allow the deintercalation of two lithium atoms during the oxidation of Cu<sup>+</sup> to Cu<sup>3+</sup> delivering a theoretical capacity of 278 mA h g<sup>-1</sup>. To confirm this hypothesis, we prepared the composition Li<sub>2</sub>CuVO<sub>4</sub> by a solid state reaction route.

In the present work, we first describe the synthesis and the crystal structure of Li<sub>2</sub>CuVO<sub>4</sub> determined by single-crystal X-ray diffraction. We then describe the electrochemical properties of Li<sub>2</sub>CuVO<sub>4</sub> determined by electrochemical impedance spectroscopy, galvanometric cycling, and cycle voltammetry measurements.

### 2. Experimental section

## 2.1. Synthesis

Powder sample of Li<sub>2</sub>CuVO<sub>4</sub> was first prepared by a direct solid state reaction from stoichiometric mixtures of Li<sub>2</sub>O, CuO, and VO<sub>2</sub> (2CuO + 2VO<sub>2</sub> => Cu<sub>2</sub>O + V<sub>2</sub>O<sub>5</sub>). The mixture was ground in an agate mortar, pelletized and heated at 600 °C for 12 h in a platinum crucible under argon or vacuum. The progress of the reaction was followed by powder XRD and a mixture of Li<sub>2</sub>CuVO<sub>4</sub> and LiCuVO<sub>4</sub> was observed. Li<sub>2</sub>CuVO<sub>4</sub> was then prepared from stoichiometric mixtures of Li<sub>2</sub>Co<sub>3</sub>, Cu<sub>2</sub>O, and V<sub>2</sub>O<sub>5</sub>. The mixture was heated at 600 °C for 12 h in a platinum crucible under argon. This led to Li<sub>2</sub>CuVO<sub>4</sub> as a major phase beside a little amount of unreacted Cu<sub>2</sub>O. The resulting powder was then ground and fired at 600 °C for 12 h several times, however traces of Cu<sub>2</sub>O were still observed. Since the DTA analysis (Fig. S1) indicated a partial decomposition around 500 °C, pure Li<sub>2</sub>CuVO<sub>4</sub> could be obtained only by quenching the sample (fired at 600 °C) in liquid nitrogen.

It is well known that the particle size has an important impact on the electrochemical performances therefore, two samples were prepared. One was ground in mortar for 0.5 h (G-Li<sub>2</sub>CuVO<sub>4</sub>) and the other was ball milled for 12 h in ethanol (M-Li<sub>2</sub>CuVO<sub>4</sub>).

### 2.2. Electron microprobe analysis

Semiquantitative energy dispersive X-ray spectrometry (EDX) analyses of the powder and different single crystals including the one investigated on the diffractometer were carried out with a JSM-500LV (JEOL) scanning electron microscope (Fig. S2). The experimentally observed Cu/V molar ratios were close to one, as expected

for Li<sub>2</sub>CuVO<sub>4</sub>. The images of the hand mixed and ball milled powders were taken using a JSM-6700FV (JEOL) field emission scanning electron microscope (Fig. S3).

# 2.3. Powder X-Ray diffraction measurements

To ensure the purity of G-Li<sub>2</sub>CuVO<sub>4</sub> powder, high precision powder XRD measurements were performed. The data were collected at room temperature over the  $2\theta$  angle range of  $13^{\circ} \le 2\theta \le 93^{\circ}$  with a step size of 0.015° using a RINT-TTR diffractometer (Rigaku) operating with Cu K $\alpha$  radiations. Full pattern matching refinement was performed with the Jana2006 program package (Fig. 1) [22]. The background was estimated by a Legendre function, and the peak shapes were described by a pseudo-Voigt function. The refinement of peak asymmetry was performed using four Berar-Baldinozzi parameters. Evaluation of these data revealed the refined cell parameters a = 6.34323(6), b = 5.46621(6), c = 5.09767(5) Å, V = 176.754(3) Å<sup>3</sup>, in good agreement with the single crystal data listed in Table 1. Similar results were obtained for M-Li<sub>2</sub>CuVO<sub>4</sub> powder.

## 2.4. Single crystal X-ray diffraction measurements

Single crystals of Li<sub>2</sub>CuVO<sub>4</sub> suitable for XRD were selected on the basis of the size and the sharpness of the diffraction spots. The data collections were carried out on a Smart Apex diffractometer (Bruker AXS) using Mo K $\alpha$  radiation. Data processing and all refinements were performed with the Jana2006 program package. A multi-scan-type absorption correction was applied using SADABS program [23]. For data collection details see Table 1.

# 2.5. Electrochemical cycling

Positive electrodes were made from mixtures of Li<sub>2</sub>CuVO<sub>4</sub> powder, acetylene black (AB) and Polyvinylidene fluoride (PVDF) in a weight ratio of 62:30:8. The resulting electrode film was pressed with a twin roller, cut into a round plate ( $\Phi = 14$  mm) and dried at 130 °C for 12 h under vacuum. Li<sub>2</sub>CuVO<sub>4</sub>/LiPF<sub>6</sub>+EC + DMC/Li cointype cells (CR2032) were assembled in an argon-filled glove box,



Fig. 1. Final observed, calculated and difference plots for powder XRD (Cu K $\alpha$  radiation) refinement of G-Li<sub>2</sub>CuVO<sub>4</sub>.

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