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Enhanced performance in hole transport material free perovskite solar cells via morphology control of PbI₂ film by solvent treatment



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HIGHLIGHTS

- A simple solvent treatment process is introduced.
- Morphology of PbI₂ film is modified by the solvent treatment.
- More complete conversion of PbI₂ to CH₃NH₃PbI₃ is observed.
- Enhanced performance and reproducibility is obtained.

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ABSTRACT

The morphology of PbI₂ film plays a critical role in determining the quality of the resultant CH₃NH₃PbI₃ film and power conversion efficiency of CH₃NH₃PbI₃ perovskite solar cell. Here, we propose a solvent treatment method in the two-step sequential deposition process to control the morphology of PbI₂ film, which leads to enhanced power conversion efficiency. Hole transport material free perovskite solar cell is chosen as a paradigm to demonstrate this idea. Solvent (isopropanol, chlorobenzene, or ethanol) treated PbI₂ films exhibit dendrite-like or flake-like morphologies, which facilitate more complete conversion of PbI₂ to CH₃NH₃PbI₃ perovskite in ambient atmosphere with a relative high humidity. Therefore, enhanced performance is obtained with the solvent treated PbI₂ films. Average power conversion efficiency has been improved from 9.42% in the traditional two-step sequential deposition to 11.22% in solar cells derived from ethanol treated PbI₂ films.

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1. Introduction

Organic-inorganic hybrid perovskites have been demonstrated to be an outstanding class of light harvest materials [1–8]. In a few years, power conversion efficiencies (PCEs) of perovskite solar cells have taken a huge leap forward, from 3.9% in 2009 [9] to a recently reported 20.1% [6]. The perovskites light absorb layer can be easily prepared by cost-effective solution processing method, i.e., onestep spin-coating [10–12] or two-step sequential deposition [13–15]. In one-step spin-coating process, a mixed PbX₂ (X = Cl, Br, or I) and CH₃NH₃X solution is spin-coated onto the substrate and annealed. Both evaporation of solvent and crystallization of perovskite take place simultaneously during the spin-coating and subsequently annealing process, thus it is difficult to control the growth and final morphology of perovskite films [12,16]. Usually low quality perovskite films with poor surface coverage and pinholes are obtained. To prepare high quality perovskite films with controlled morphologies, the two-step sequential deposition method is proposed. In a typical two-step sequential deposition process, PbI₂ film is first deposited onto substrate by spin-coating, the obtained PbI₂ film is then dipped into CH₃NH₃I isopropanol solution. The CH₃NH₃I molecular will diffuse and intercalate into PbI₂ crystal and convert it into CH₃NH₃PbI₃ perovskite. However, the intercalation process will expand the PbI₂ lattice and form a compact CH₃NH₃PbI₃ surface layer. This compact layer will block the further diffusion of CH₃NH₃I into the inner PbI₂, leading to an incomplete conversion of PbI₂ to CH₃NH₃PbI₃ [17–19]. Too much residual PbI2 decreases the PCEs of the perovskite solar cells, and

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the uncontrolled PbI₂ amount also greatly deteriorates the reproducibility of perovskite solar cells. It has been demonstrated that morphologies of PbI₂ films play a critical role in determining the conversion of PbI₂ to CH₃NH₃PbI₃ in two-step sequential deposition process, which influences the morphologies of the resultant perovskite films and their power conversion efficiencies [20,21]. Various methods have been developed to engineer the morphologies of PbI₂ films, including dissolving PbI₂ into different solvent (*N*,*N*-dimethylformamide, or dimethyl sulfoxide, or a mixture of them) [22], using mixed PbI₂ and PbCl₂ precursor solution [23], careful thermal annealing treatment [5,24], and adding a small amount of CH₃NH₃I into PbI₂ solution [17,18]. All these methods enable a better conversion of PbI₂ to CH₃NH₃PbI₃ perovskite, and better control of the final morphologies of the CH₃NH₃PbI₃ films, thus enhanced PCEs are observed.

In this manuscript, we have proposed a solvent treatment method to modify the morphologies of PbI₂ films in the two-step sequential deposition process, in order to ensure more complete conversion of PbI₂ to CH₃NH₃PbI₃ perovskite and better control of the morphologies of CH₃NH₃PbI₃ films. The spin-coated wet PbI₂ films are immediately treated with different solvent, i.e., isopropanol, chlorobenzene and ethanol, before the subsequently thermal drying process. Due to the low solubility of PbI₂ in these solvent, PbI₂ would quickly precipitate and crystallize, leading to a dendrite-like or flake-like morphologies with many nano-sized voids. The existence of voids in the PbI₂ films facilitates the diffusion of CH₃NH₃I, and the nano-sized PbI₂ crystal is easily converted into CH₃NH₃PbI₃ perovskite. Thus high quality CH₃NH₃PbI₃ films with controllable morphology and low amount of residual PbI₂ are obtained. Hole transport material free perovskite solar cells fabricated on these CH₃NH₃PbI₃ films prepared from solvent treated PbI₂ films exhibit superior PCEs. Average PCE has been improved from 9.42% for the traditional two-step sequential deposition to 11.22% in solar cells derived from ethanol treated PbI₂ films.

2. Experiments

2.1. Preparation of CH₃NH₃PbI₃ perovskite film

1.0 M PbI₂ and 0.2 M PbCl₂ was dissolved into N,N-dimethylformamide (DMF) at 70 °C under stirring overnight. Fig. 1a presents the schematic two-step sequential deposition process to prepare the CH₃NH₃PbI₃ films without solvent treatment. At first, PbI₂ precursor solution was spin-coated onto the mesoporous TiO₂ film at 6000 rpm for 20 s. The obtained PbI₂ films were directly dried at 100 °C for 5 min. Then to convert PbI₂ film into perovskite, CH₃NH₃I isopropanol solution (7 mg mL⁻¹) was loaded onto the PbI₂ film for 2 min, and spun at 2000 rpm for 20 s. Fig. 1b presents the two-step sequential deposition process with solvent treatment. Immediately after the first-step spin-coating of the PbI₂ films, a few drops of solvent, i.e., isopropanol (IPA), chlorobenzene (CB), or ethanol (ET), were dropped onto the wet PbI₂ film, soaked for 10 s and spun at 2000 rpm, the solvent treated PbI₂ film was dried at 100 °C for 5 min. The second-step spin-coating process is the same as the procedure in Fig 1a. The as-prepared CH₃NH₃PbI₃ films were further dried at 100 °C for 15 min.

2.2. Fabrication of perovskite solar cells

Solar cells were fabricated under ambient atmosphere with a relative high humidity, usually around 50%. FTO glass was ultrasonic cleaned with ethanol and acetone. A 0.15 M solution of titanium diisopropoxide bis(acetylacetonate) in 1-butanol was spincoated onto the FTO glass at 3000 rpm for 30 s, and dried at 150 °C for 30 min to form the compact TiO₂ layer. A homemade P25 paste was prepared by ball milling 2.845 g commercial P25 TiO₂ particles, 1.423 g ethyl cellulose, 11.538 g terpineol and 63.222 g ethanol together for 24 h and was spin-coated onto the compact TiO₂ layer at 3000 rpm for 30 s, and then annealed at 500 °C for 30 min to form the mesoporous TiO₂ layer. The TiO₂ films were further treated with 0.4 M TiCl₄ aqueous solution at 70 °C for 30 min, then sintered again at 500 °C for 30 min. The CH₃NH₃PbI₃ perovskite film was prepared using two-step sequential deposition method as discussed in the previous context. Finally, a carbon paste was doctor-bladed onto the perovskite film and dried at 100 °C to serve as a counter electrode.

2.3. Characterization

X-ray diffraction (XRD) measurements were carried out on a Bruker D8 focus X-ray diffractometer with Cu K α radiation. Absorption of Pbl₂ films and CH₃NH₃Pbl₃ perovskite films were characterized by UV–vis spectrometer (Lambda 650S, PerkinElmer). SEM images were obtained using scanning electron microscopy (SEM, Sirion FEG, USA). J-V characteristics of the perovskite solar cells were measured on a CHI660C electrochemical

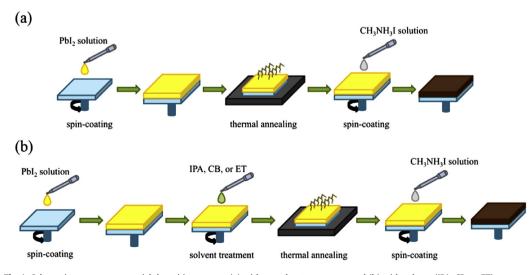


Fig. 1. Schematic two-step sequential deposition process (a) without solvent treatment, and (b) with solvent (IPA, CB, or ET) treatment.

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