

Contents lists available at ScienceDirect

Journal of Solid State Chemistry



journal homepage: www.elsevier.com/locate/jssc

Solid solubility of $Yb_2Si_2O_7$ in $\beta\text{-},\,\gamma\text{-}$ and $\delta\text{-}Y_2Si_2O_7$

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ARTICLE INFO

Article history: Received 9 March 2011 Received in revised form 26 May 2011 Accepted 26 May 2011 Available online 2 June 2011

Keywords: Rare earth silicates Y₂Si₂O₇ Yb₂Si₂O₇ Solid solubility Phase diagram

ABSTRACT

This paper examines the structural changes with temperature and composition in the Yb₂Si₂O₇-Y₂Si₂O₇ system; members of this system are expected to form in the intergranular region of Si₃N₄ and SiC structural ceramics when sintered with the aid of Yb₂O₃ and Y₂O₃ mixtures. A set of different compositions have been synthesised using the sol–gel method to obtain a xerogel, which has been calcined at temperatures between 1300 and 1650 °C during different times. Isotherms at 1300 and 1600 °C have been analysed in detail to evaluate the solid solubility of Yb₂Si₂O₇ in β-Y₂Si₂O₇ and γ-Y₂Si₂O₇. Although Yb₂Si₂O₇ shows a unique stable polymorph (β), Yb³⁺ is able to replace Y³⁺ in γ-Y₂Si₂O₇ and δ-Y₂Si₂O₇ at high temperatures and low Yb contents. IR results confirm the total solid solubility in the system and suggest a constant SiOSi angle of 180° in the Si₂O₇ unit across the system. The temperature–composition diagram of the system, obtained from powder XRD data, is dominated by the β-*RE*₂Si₂O₇ polymorph, with γ-*RE*₂Si₂O₇ and δ-*RE*₂Si₂O₇ showing reduced stability fields. The diagram is in accordance with Felsche's diagram if average ionic radii are assumed for the members of the solid solution at any temperature, as long as the β-γ phase boundary is slightly shifted towards higher radii.

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1. Introduction

Rare Earth disilicates (*RE*₂Si₂O₇) exhibit different polymorphic forms depending on the RE ionic radius, temperature and pressure [1,2]. Y₂Si₂O₇ shows, in particular, up to five polymorphs with increasing temperature at room pressure (y, α , β , γ and δ , also called, respectively, y, B, C, D and E); while Yb₂Si₂O₇ exhibits a unique polymorph (β , also called C) up to the melting point of the compound. These silicates are important materials in the sinterization of the structural advanced ceramic Si₃N₄. It has been shown that when RE oxides (RE=lanthanides and yttrium), single or mixed, are added to the powder of pure Si₃N₄ as sintering aids, a glassy disilicate phase $(RE_2Si_2O_7)$ forms in the intergranular regions, which, upon crystallisation, improves the high-temperature mechanical properties of the material [3,4]. Knowledge of the crystalline structures adopted by the $RE_2Si_2O_7$ intergranular phase at different temperatures and RE contents is therefore of great value in understanding the behaviour of these materials.

The aim of this research is to analyse the solid solubility of $Yb_2Si_2O_7$ in β -, γ - and δ - $Y_2Si_2O_7$ and to describe a temperature– composition diagram showing the polymorphism of the system. Study of the solid solubility of $Yb_2Si_2O_7$ in *y*- and α - $Y_2Si_2O_7$ is not reported due to the slow kinetics at the stability temperatures of these phases (< 1200 °C) [5].

2. Experimental section

2.1. Synthesis of the xerogel

The following reactants have been used as starting materials: Yb(NO₃)₃ · 5H₂O (99.99% Aldrich Chemical Co.), Y(NO₃)₃ · 4H₂O (99.99% Aldrich Chemical Co.), Si(OC₂H₅)₄ (TEOS, 99% Aldrich Chemical Co.) and absolute ethanol (AnalaR Normapur). To verify the amount of water per formula unit present in the nitrates, thermogravimetric analyses (TGA) were carried out using a SDT Q600 (TA instruments). $Yb_{2-x}Y_xSi_2O_7$ members with nominal x=0.00, 0.40, 0.80, 1.00, 1.20, 1.40, 1.60, 1.80 and 2.00 were synthesised. A TEOS solution in ethanol (1:3 in volume) was added over the appropriate amounts of Yb(NO₃)₃·5H₂O and $Y(NO_3)_3 \cdot 4H_2O$ previously dissolved in 5 ml of absolute ethanol. The mixture was stirred at 40 °C for 5–7 h and the transparent gels obtained were dried at 60 °C for 24 h in air. Nitrates were eliminated by calcination at 500 °C for 2 h at a heating rate of 1 °C min⁻¹. Importantly, only a slight excess of TEOS in the initial mixture yielded satisfactorily the $Yb_{2-x}Y_xSi_2O_7$ phase, otherwise oxiorthosilicate phase was obtained in a large proportion.

2.2. Calcination experiments

The xerogel of each composition was ground in an agate mortar and subsequently divided into different portions, which

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^{0022-4596/\$ -} see front matter \circledcirc 2011 Elsevier Inc. All rights reserved. doi:10.1016/j.jssc.2011.05.034

were calcined in air at different temperatures between 1300 °C and 1650 °C and different periods of time (Table 1), using a heating rate of 5 °C min⁻¹ and a platinum crucible. Finally, the samples were slowly cooled down to room temperature.

2.3. Characterisation

The global composition of the samples was examined by X-Ray fluorescence (Panalytical, AXIOS model). Table 2 shows the agreement between the nominal and real compositions.

An analytical transmission electron microscope (ATEM, Philips CM20FEG) operating at 200 kV and equipped with an energydispersive X-ray spectrometer (EDX, NORAN Ge detector) was used to observe the morphology of the samples as well as to examine chemical composition. The EDA software was used to index the electron diffraction patterns [6]. EDX analyses on $Yb_{2-x}Y_xSi_2O_7$ single crystals were carried out to determine yttrium and ytterbium contents. The quantification included an absorption correction and the calibration of Cliff-Lorimer factors for ytterbium and yttrium with respect to silicon, determined from pure Yb₂Si₂O₇ and Y₂Si₂O₇, respectively. At least 10 rare earth disilicate single crystals were analysed in each sample to calculate the mean compositions (results are displayed in third column of Table 2). A counting time from 100 to 300 s has been used in order to accumulate enough X-ray counts. The samples were dispersed in ethanol by sonication and dropped on a conventional carbon-coated copper grid.

X-Ray diffraction patterns (PANalytical XPert Pro Diffractometer, CuK α and X-Celerator detector) were recorded over the angular range $10^{\circ} < 2\theta < 120^{\circ} 2\theta$ with step width of 0.02° and 10 s counting time. The patterns were analysed using the Rietveld method with the TOPAS software (TOPAS version 4.2, Bruker AXS

Table 1

Polymorphs obtained after calcination of the $Yb_{(2-x)}Y_xSi_2O_7$ compositions at different temperatures and times.

T (°C)	<i>t</i> (h)	0.00	0.40	0.80	1.00	1.20	1.40	1.60	1.80	1.94	2.00
1300	24	β	β	β	β	β		β			β
1400	12										β
	36										$\gamma + \beta$
	72						β	$\beta + \gamma$	$\beta + \gamma$		
1500	24							$\beta + \gamma$	$\beta + \gamma$		$\gamma + (\beta)$
1550	24						$\beta + \gamma$	$\beta + \gamma$			
	48						γ	γ			
1600	24		β	β	β	β	$\beta + \gamma$				
	36			-		-					γ
	48							$\gamma + \beta$			
	63.2							γ.			
1650	12					β	$\gamma + \beta$	γ	γ	$\gamma + \delta$	$\gamma + \delta$
	24					$\beta + \gamma$	γ.	γ	γ		$\gamma + \delta$
							•	•	•		•

Table 2 Compositional data obtained with different techniques for $Yb_{2-x}Y_xSi_2O_7$ samples.

imes Nominal	imes FRX	imes Rietveld	\times EDX/TEM	
		β	γ	
0.00 0.40 0.80 1.00 1.20 1.40 1.60 1.80	0.00 0.35 0.81 1.00 1.21 Non-recorded 1.59 Non-recorded	0.00 (0.00) 0.36 (0.01) 0.80 (0.01) 1.00 (0.01) 1.26 (0.01) - 1.64 (0.01)	- - - 1.40 (0.01) 1.58 (0.03) 1.81 (0.01)	0.00 0.40 0.85 0.96 Non-recorded Non-recorded 1.61 Non-recorded
2.00	2.00	2.00 (0.00)	1.01 (0.01)	2.00

2009). Refined parameters were background coefficients, zero correction, scale, unit cell parameters, site occupation factors for the rare earth sites, isotropic atomic displacement parameters, atomic positions and microstructure parameters.

Finally, Infrared Fourier Transform spectra (Nicolet 510 FTIR instrument equipped with DTGS detectors) were recorded over the $1500-400 \text{ cm}^{-1}$ range with a resolution of 2 cm^{-1} . The measurements were made using dried KBr pellets, which were prepared by mixing and pressing 1.5 mg of sample with 100 mg KBr.

3. Results and discussions

3.1. Solid solubility of Yb_2Si_2O_7 in β -Y_2Si_2O_7: Study of the 1300 °C isotherm

3.1.1. X-ray diffraction study

Representative portions of the XRD patterns of different $Yb_{2-x}Y_xSi_2O_7$ members calcined at 1300 °C for 24 h are shown in Fig. 1. The diagram of the *x*=0.0 sample matches the standard pattern of pure β -Yb₂Si₂O₇ (PDF 00-025-1345), as expected. Increasing Y content produces very similar patterns, with slight variations in peak positions and intensity; see, for example, the shift of the peaks under the dashed lines, the intensity decrease in the



Fig. 1. Selected portions of the XRD patterns of $Yb_{2-x}Y_xSi_2O_7$ samples annealed at 1300 °C for 24 h. Asterisks: X2-*RE*₂SiO₅; c: cristobalite. Dashed lines are guides to the eye to show shift of reflections with composition.

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