

Preparation, crystal structure and chemical bonding analysis of the new binary compounds $\text{Rh}_4\text{Ga}_{21}$ and $\text{Rh}_3\text{Ga}_{16}$

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Abstract

The two new binary compounds $\text{Rh}_4\text{Ga}_{21}$ (space group $Cmca$ ($Cmce$), $a = 40.135(6) \text{ \AA}$, $b = 6.470(2) \text{ \AA}$, $c = 6.473(1) \text{ \AA}$, Pearson symbol $oC136$) and $\text{Rh}_3\text{Ga}_{16}$ (space group $Ccca$ ($Ccce$), $a = 30.424(7) \text{ \AA}$, $b = 6.476(2) \text{ \AA}$, $c = 6.468(2) \text{ \AA}$, Pearson symbol $oC76$) were synthesised and their crystal structures were solved from single-crystal X-ray diffraction data. From a topological point of view, both these two crystal structures and the crystal structure of PdGa_5 can be described either as inhomogeneous intergrowth structures containing three different kinds of segments, or as built up by layers of capped square antiprisms condensed via their capping atoms. Bonding analysis with bonding indicators revealed that the crystal structures of $\text{Rh}_4\text{Ga}_{21}$ and $\text{Rh}_3\text{Ga}_{16}$ have to be considered as framework polyanions formed by covalently bonded gallium atoms with embedded rhodium cations.

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1. Introduction

Four binary phases are known in the Rh–Ga system; RhGa (CsCl type, Pearson symbol $cP2$) [1], $\text{Rh}_{10}\text{Ga}_{17}$ (prototype, $tP108$) [2], RhGa_3 (IrIn_3 type [3], $tP16$) [4] and Rh_2Ga_9 (prototype, $mP22$) [5,6]. No phase diagram has hitherto been published. In course of a systematic investigation of Ga- and Al-rich compounds of Ir, several new phases were found [6,7]. This encouraged an investigation of the chemically related Rh–Ga system, of which the first results are the subject of the present work.

2. Experimental

Single crystals of $\text{Rh}_4\text{Ga}_{21}$ were prepared in a Ga-rich self-flux containing 0.0538 g Rh (99.9%, Chempur) and 0.952 g Ga (99.999%, Chempur), hence having the nominal composition $\text{Rh}_{3.7}\text{Ga}_{96.3}$. The elements were sealed inside an evacuated quartz tube already prepared with a quartz wool filter on a support of small pieces of crushed quartz

glass according to the high-temperature centrifugation aided filtration technique (HTCAF) [8,9]. The sample was heat treated at 300°C for 18 days before the melt was removed from the solid using a centrifuge operating at $2000g$, without significantly decreasing the temperature of the sample.

Single crystals of $\text{Rh}_3\text{Ga}_{16}$ were prepared analogously from 0.0239 g Rh and 1.003 g Ga (nominal composition $\text{Rh}_{1.6}\text{Ga}_{98.4}$), but cooled down from 900°C at the rate of 5°C/h down to 150°C , where the sample was kept for 18 days before the melt was removed.

X-ray powder diffraction patterns of the powdered single crystals were collected with a Huber Imaging Plate Guinier Camera 670 using $\text{CuK}\alpha_1$ radiation ($\lambda = 1.54059 \text{ \AA}$) for $\text{Rh}_3\text{Ga}_{16}$ and $\text{CoK}\alpha_1$ radiation ($\lambda = 1.78897 \text{ \AA}$) for $\text{Rh}_4\text{Ga}_{21}$. Silicon ($a = 5.43119(1) \text{ \AA}$) and LaB_6 powder ($a = 4.15692(1) \text{ \AA}$), respectively, were added as internal standard to the samples. The lattice parameters were refined from the powder data with the program CELLREF [10].

Single crystals with well-exhibited faces were isolated and parts of them were used for the X-ray diffraction experiment. Single-crystal X-ray diffraction data were

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collected on a Rigaku Mercury CCD diffraction system employing MoK α radiation ($\lambda = 0.71069$ Å). The measured data sets were corrected for absorption using multi-scans in the Laue-class *mmm*. Relevant details for the data collection and handling are summarized in Table 1.

The crystal structures of Rh₄Ga₂₁ and Rh₃Ga₁₆ were solved by direct methods with the Shelxs97 program [11]. The refinement was performed using the program Jana2000 [12]. The refined parameters were scale factor, atomic coordinates, anisotropic displacement parameters, extinction coefficient and a twinning factor for Rh₃Ga₁₆.

The TB–LMTO–ASA program package [13] with exchange correlation potential (LDA) according to Barth and Hedin [14] was used for quantum chemical calculations. Because of the clear similarity between the crystal structures of Rh₃Ga₁₆ and Rh₄Ga₂₁, the calculations were made only for the latter. The radial scalar-relativistic Dirac equation was solved to obtain the partial waves. A calculation within the atomic sphere approximation (ASA) includes corrections for the neglect of partial waves

of higher order and interstitial regions [15], hence no addition of empty spheres was found to be necessary. The radii of the atomic spheres used in the calculations were: $r(\text{Rh1}) = 1.609$ Å, $r(\text{Rh2}) = 1.630$ Å, $r(\text{Ga1}) = 1.543$ Å, $r(\text{Ga2}) = 1.537$ Å, $r(\text{Ga3}) = 1.579$ Å, $r(\text{Ga4}) = 1.581$ Å, $r(\text{Ga5}) = 1.605$ Å, $r(\text{Ga6}) = 1.597$ Å, $r(\text{Ga7}) = 1.571$ Å. A basis set containing the Rh(5s, 5p, 4d) and Ga(4s, 4p) orbitals was employed for the self-consistent calculations with the Rh(4f) and Ga(4d) functions being downfolded. The electron localization function (ELF, η) was evaluated according to [16] within the TB–LMTO–ASA program package with an ELF module already implemented. To gain a deeper insight in the chemical bonding, the topology of ELF was analysed with the program Basin [17]. The electron density was integrated in basins which were bound by zero flux surfaces in the ELF gradient. This method, analogous to the procedure proposed by Bader [18] for charge calculations from electron densities resulted in electron counts for each basin, which revealed basic information for the description of the bonding situation.

Table 1
Crystallographic information and data handling for Rh₄Ga₂₁ and Rh₃Ga₁₆

Crystal data	Rh ₄ Ga ₂₁		Rh ₃ Ga ₁₆
Chemical formula			
Crystal system		Orthorhombic	
Space group	<i>Cmca</i> (<i>Cmce</i>)		<i>Ccca</i> (<i>Ccce</i>)
Z	4		4
<i>a</i> (Å) ^a	40.135(6)		30.424(7)
<i>b</i> (Å) ^a	6.470(2)		6.476(2)
<i>c</i> (Å) ^a	6.473(1)		6.468(2)
<i>V</i> (Å ³) ^a	1681		1274
Density calc. (g cm ^{−3})	7.41		7.42
Crystal form		Irregular	
Crystal size (mm ³)	0.050 × 0.040 × 0.020		0.040 × 0.030 × 0.015
Colour		Grey metallic	
Absorption coefficient (mm ^{−1})	36.8		37.0
Data collection			
Diffraction system		Rigaku AFC7 Mercury CCD	
Radiation, λ (Å)		MoK α , 0.71069	
No. of measured reflections	6820		2858
Range of <i>hkl</i>	−43 ≤ <i>h</i> ≤ 57 −9 ≤ <i>k</i> ≤ 9 −8 ≤ <i>l</i> ≤ 8		−41 ≤ <i>h</i> ≤ 41 −8 ≤ <i>k</i> ≤ 7 −6 ≤ <i>l</i> ≤ 8
Absorption correction		Multi-scan	
<i>T</i> _{min} / <i>T</i> _{max}	0.523		0.541
<i>R</i> _{int}	0.055		0.045
Refinement			
Refinement on		<i>F</i>	
No. of independent reflections	1286		759
No. of independent observed reflections	967		565
Observation criterion		<i>I</i> > 3 σ (<i>I</i>)	
No. of reflections used in refinement	1286		759
No. of parameters refined	62		47
Weighting scheme		Unit	
<i>R</i> (<i>F</i>) _{all} , <i>wR</i> (<i>F</i>) _{all} , <i>R</i> (<i>F</i>) _{obs}	0.053, 0.046, 0.036		0.076, 0.061, 0.060
<i>GOF</i> _{all}	1.0		5.0
$\Delta\rho_{\text{max}}$, $\Delta\rho_{\text{min}}$, (e Å ^{−3})	2.0, −1.8		6.2, −1.9
Extinction model	Gaussian isotropic		—
Extinction coefficient	0.051(2)		—

^aCell parameters from powder diffraction data with internal standard.

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