

# Why pressure induces an abrupt structural rearrangement in PdTe<sub>2</sub> but not in PtTe<sub>2</sub>

C. Soulard<sup>a</sup>, P.E. Petit<sup>a</sup>, P. Deniard<sup>a</sup>, M. Evain<sup>a</sup>, S. Jobic<sup>a,\*</sup>,  
M.-H. Whangbo<sup>b</sup>, A.-C. Dhaussy<sup>c,1</sup>

<sup>a</sup>Laboratoire de Chimie des Solides (UMR 6502), Institut des Matériaux Jean Rouxel, 2 rue de la Houssinière, BP 32229, 44322 Nantes cedex 3, France

<sup>b</sup>Department of Chemistry, North Carolina State University, Raleigh, North Carolina 27695-8204, USA

<sup>c</sup>ESRF, 6 rue Jules Horowitz, BP 220, 38043 Grenoble Cedex, France

Received 28 January 2005; received in revised form 29 March 2005; accepted 7 April 2005

Available online 4 May 2005

## Abstract

High-pressure X-ray diffraction measurements were carried out for polymeric CdI<sub>2</sub>-type compounds *M*Te<sub>2</sub> (*M* = Pt, Pd) to investigate if they undergo a structural phase transition under pressure as does IrTe<sub>2</sub>. Up to 27 GPa at room temperature PtTe<sub>2</sub> does not undergo any structural phase transition. In contrast, however, an abrupt change in the inter-atomic distances occurs in PdTe<sub>2</sub> above 15.7 GPa at room temperature, and above 5 GPa at 300 °C, but the volume vs. pressure curve exhibits no discontinuity. To account for the differences between the isostructural compounds PtTe<sub>2</sub>, PdTe<sub>2</sub> and IrTe<sub>2</sub>, their electronic structures and bonding were analyzed on the basis of first principles electronic band structure calculations.

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**Keywords:** PdTe<sub>2</sub>; PtTe<sub>2</sub>; X-ray diffraction

## 1. Introduction

The CdI<sub>2</sub>-type transition metal dichalcogenides *MQ*<sub>2</sub> (*M* = transition metal, *Q* = chalcogen) consist of *MQ*<sub>2</sub> layers made up of edge-sharing *MQ*<sub>6</sub> octahedra, where adjacent *MQ*<sub>2</sub> layers are held together by van der Waals interactions. IrTe<sub>2</sub> adopts the polymeric CdI<sub>2</sub>-type structure, i.e., a derivative of the CdI<sub>2</sub>-type structure [1,2], in which adjacent *MQ*<sub>2</sub> layers have short *Q*...*Q* contacts thereby leading to a low *c/a* ratio, i.e., 1.38 instead of 1.67 expected on the basis of a classical hexagonal close stacking of the anions. From the viewpoint of the ionic electron counting leading to the charge balance *M*<sup>4+</sup>(*Q*<sup>2-</sup>)<sub>2</sub>, the occurrence of short

interlayer *Q*...*Q* contacts is explained by considering the electron transfer from the *p*-block bands of *Q* to the *d*-block bands of *M* [3]. When the polymeric CdI<sub>2</sub>-type IrTe<sub>2</sub> [2] (hereafter named h-IrTe<sub>2</sub>) is subjected to pressure up to 32 GPa, it undergoes two structural phase transitions [4-6]. The hexagonal phase h-IrTe<sub>2</sub> is converted to a monoclinic-type structure (m-IrTe<sub>2</sub>) at 5 GPa and at room temperature, and then to a pyrite structure (c-IrTe<sub>2</sub>) at 32 GPa under laser heating.

PtTe<sub>2</sub> and PdTe<sub>2</sub> are similar to IrTe<sub>2</sub> in several structural and electronic properties under ambient conditions. They both adopt the polymeric CdI<sub>2</sub>-type structure [7,8] with a low *c/a* ratio (i.e., 1.30 in PtTe<sub>2</sub> and 1.27 in PdTe<sub>2</sub>). The three compounds IrTe<sub>2</sub>, PtTe<sub>2</sub> and PdTe<sub>2</sub> have similar cell volumes per formula unit (i.e., 72.23(1) Å<sup>3</sup> for IrTe<sub>2</sub>, 72.43(6) Å<sup>3</sup> for PdTe<sub>2</sub> and 73.35(4) Å<sup>3</sup> for PtTe<sub>2</sub>), and their short Te...Te contacts are in the range of 3.44–3.56 Å. In addition, IrTe<sub>2</sub>, PtTe<sub>2</sub> and PdTe<sub>2</sub> should be similar in electronic structure because the electronegativities of Ir, Pt and Pd are

\*Corresponding author. Tel.: +33 2 40 37 39 22;  
fax: +33 2 40 37 39 95.

E-mail address: [stephane.jobic@cnrs-imn.fr](mailto:stephane.jobic@cnrs-imn.fr) (S. Jobic).

<sup>1</sup>Present address: Laboratoire CRISMAT, UMR 6508, 6 boulevard Maréchal Juin, 14050 CAEN Cedex.

nearly the same (i.e., 2.20 for Ir and Pd, and 2.28 for Pt in Pauling scale), although  $M\text{Te}_2$  ( $M = \text{Pt}, \text{Pd}$ ) has one more valence electron per formula unit than does  $\text{IrTe}_2$ . Thus, one might suggest that the high-pressure behavior of  $M\text{Te}_2$  ( $M = \text{Pt}, \text{Pd}$ ) would be similar to that of  $\text{IrTe}_2$ . To test this speculation, we performed high-pressure X-ray diffraction experiments for  $M\text{Te}_2$  ( $M = \text{Pt}, \text{Pd}$ ) in both angle- and energy-dispersive configurations.  $M\text{Te}_2$  ( $M = \text{Pt}, \text{Pd}$ ) was not found to undergo a first-order structural phase transition up to 27 GPa. Only a sudden change in the pressure dependence of the interatomic distances was observed in  $\text{PdTe}_2$  at about 19 and 6 GPa at room temperature and at 300 °C, respectively. To understand why  $\text{IrTe}_2$ ,  $\text{PdTe}_2$  and  $\text{PtTe}_2$  behave differently under pressure, we compared their electronic structures on the basis of first principles electronic band structure calculations.

## 2. Synthesis and structure determination under pressure

### 2.1. Synthesis

A stoichiometric mixture of elemental powders (Pd sponge, 99.9% Aldrich; Pt powder, 325 mesh, 99.9%, Alfa Aesar; Te granules, 30 mesh, 99.99%, Aldrich) was ground and loaded into a quartz tube (~9 cm in length, 10 mm in diameter). The tubes were evacuated to  $10^{-2}$  Torr, sealed and placed into a furnace whose temperature was raised to 850 °C at the rate of 5 °C per hour and kept there for 360 h, following the procedure of Lieth et al. [9] The powder sample was subsequently cooled to room temperature at the rate of 5 °C/h. X-ray powder diffraction data were collected on an INEL CPS 120 diffractometer (Debye–Scherrer geometry). The structure refinements were carried out using the JANA2000 software package [10]. The cell parameters obtained in the present work are in good agreement with those reported in the literature [7] (in italics and within parenthesis), i.e.,  $a = 4.027(1)$  (*4.026*) Å and  $c = 5.223(2)$  (*5.221*) Å for  $\text{PtTe}_2$ , and  $a = 4.037(2)$  (*4.037*) Å and  $c = 5.132(2)$  (*5.126*) Å for  $\text{PdTe}_2$ . The corresponding volumes are 73.35(4) (*73.28*) Å<sup>3</sup> for  $\text{PtTe}_2$  and 72.43(6) (*72.33*) Å<sup>3</sup> for  $\text{PdTe}_2$ .

### 2.2. Structure determination

Preliminary high-pressure energy-dispersive X-ray diffraction (EDX) measurements at 20 and 300 °C were carried out at the LURE (France). Experiments reported herein were performed in an angle-dispersive (ADX) mode on the ID30 beamline (ESRF, France) at room temperature (RT) for  $\text{PtTe}_2$ , and at RT and at 300 °C for  $\text{PdTe}_2$ . The experimental procedure has already been described elsewhere [11].  $M\text{Te}_2$  powder samples were ground to a fine powder in an agate

mortar and loaded in the pressure chamber of a membrane-type diamond anvil cell (DAC). The pressure-transmitting medium added to provide quasi-hydrostatic pressure conditions was  $\text{N}_2$  for the experiments at RT, and LiF for those at 300 °C. The ruby fluorescence method [12] was used to measure the in situ pressure.

The diffraction patterns were recorded up to 27 GPa for  $2\theta$  angles ranging from 2° to 20° (Figs. 1 and 2). For the 300 °C experiments, the  $\text{PdTe}_2$  powder placed in the DAC was resistively heated. The storage ring was operated at 6 GeV and 7–15 mA (single-bunch mode). A channel-cut monochromator was used to select the wavelength of 0.3738 Å. The energy calibration was carried out using the *K*-edge of an iodine foil. The beam was focused via two multilayer mirrors settled in the Kirkpatrick-Baetz (KB) configuration, and then collimated with slits, thus defining a spot of 20 μm in

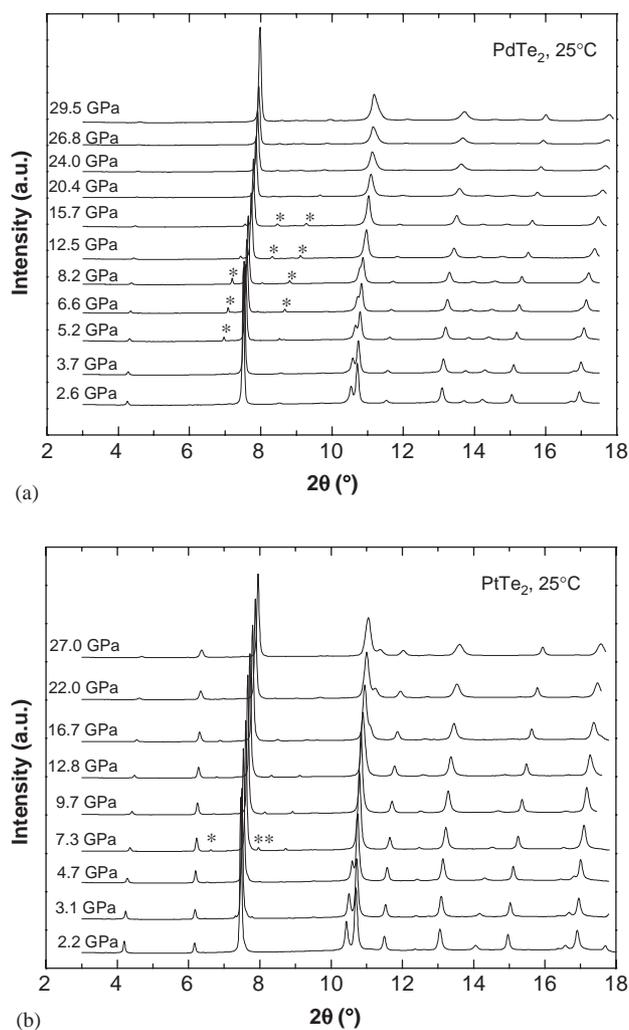


Fig. 1. Angle dispersive diffraction patterns of (a)  $\text{PdTe}_2$  and (b)  $\text{PtTe}_2$  under pressure during the decompression cycle at room temperature. The asterisks refer to the  $\beta\text{-N}_2$  high-pressure phase.

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