



# Alumina–tantalum composite for femoral head applications in total hip arthroplasty

T.S. Huang<sup>a</sup>, M.N. Rahaman<sup>a,\*</sup>, B.S. Bal<sup>b</sup>

<sup>a</sup> Department of Materials Science and Engineering, Missouri University of Science and Technology, Rolla, MO 65409, USA

<sup>b</sup> Department of Orthopaedic Surgery, University of Missouri–Columbia, MO 65212, USA

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## ABSTRACT

Dense composite laminates of alumina ( $\text{Al}_2\text{O}_3$ ) and tantalum (Ta) were fabricated by hot pressing and tested *in vitro* for potential use as a femoral head material in total hip arthroplasty (THA).  $\text{Al}_2\text{O}_3$ –Ta composite laminates hot pressed at 1450 °C and 1650 °C had flexural strengths of  $940 \pm 180$  MPa and  $1090 \pm 340$  MPa, respectively, which were far larger than the values of  $420 \pm 140$  MPa and  $400 \pm 130$  MPa for  $\text{Al}_2\text{O}_3$  hot pressed at 1450 °C and 1650 °C, respectively. The interfacial shear strength, determined by a double-notched specimen test, was  $310 \pm 80$  MPa for the composite laminate hot pressed at 1650 °C, indicating strong interfacial bonding between  $\text{Al}_2\text{O}_3$  and Ta. Scanning electron microscopy (SEM), energy dispersive X-ray (EDS) analysis, and X-ray mapping of polished sections of the hot-pressed laminates showed the presence of an interfacial region formed presumably by diffusion of O (at 1450 °C) or O and Al (1650 °C) from  $\text{Al}_2\text{O}_3$  into Ta. Composite femoral heads of  $\text{Al}_2\text{O}_3$  and Ta could combine the low wear of an  $\text{Al}_2\text{O}_3$  articulating surface with the safety of a ductile metal femoral head.

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## 1. Introduction

Total hip arthroplasty (THA) articulations consist of a femoral head (ball) rotating inside a hemispherical acetabular shell (socket). Typical bearing materials are cobalt–chromium alloy (CoCr) for the femoral head, and ultra-high molecular weight polyethylene (UHMWPE), for the socket. After 10–15 years *in vivo*, polyethylene wear particles lead to localized inflammation in the hip joint, periprosthetic bone resorption, and ultimately, aseptic loosening of the prosthetic components. Repeat hip surgery is morbid, risky, and costly. It accounts for about 25% of the nearly 250,000 THA operations performed annually in the U.S. and Europe [1].

Increased wear resistance of THA bearings can favorably impact implant durability. Crosslinking of polyethylene leads to better wear properties in polyethylene sockets [2–4]. Substituting  $\text{Al}_2\text{O}_3$  femoral heads for CoCr also decreases wear [5,6].  $\text{Al}_2\text{O}_3$  heads can also be used with sockets of the same material. The resulting  $\text{Al}_2\text{O}_3$ – $\text{Al}_2\text{O}_3$  (hard-on-hard) articulations have the lowest wear of any THA bearing combination [7–9]. At intermediate follow-up, THA with  $\text{Al}_2\text{O}_3$ – $\text{Al}_2\text{O}_3$  articulations is associated with less femoral bone loss than THA with metal-on-polyethylene bearings [10]. Ceramic bearings contribute to the longevity of THA by dramatically reducing bearing wear, and, hence, the likelihood of revision (repeat) THA surgery [11,12].

Despite its wear advantage,  $\text{Al}_2\text{O}_3$ , like other ceramics, is brittle. Catastrophic failure of  $\text{Al}_2\text{O}_3$  femoral heads *in vivo*, while rare, is a

serious complication. The incidence of catastrophic failure *in vivo* is 1 in 5000–10,000 [13], and this risk has persisted during the last decade or so despite many improvements in  $\text{Al}_2\text{O}_3$  starting materials, design, manufacturing, and quality control [14]. Another concern with  $\text{Al}_2\text{O}_3$ – $\text{Al}_2\text{O}_3$  hip articulations is audible noise or squeaking [15,16], which has an overall occurrence in the range 0.5–10% [16]. Squeaking has been the subject of considerable investigation and discussion in the last 5 years, but the mechanism of squeaking is not clear [15,16].

The design of modular femoral heads in THA is outlined schematically in Fig. 1a. The articulating surface, which must be polished and smooth to minimize friction, is loaded in compression. Femoral heads have a modular taper bore, in order to allow attachment to a matched Morse taper on the metal femoral stem. Taper depth allows intra-operative control of leg lengths. Once installed during surgery, the modular taper-bore junction is mechanically stable, but micro-motion can occur which can lead to fretting wear and corrosion [17,18]. When a compressive stress is applied to the femoral head during physiological activity, it gives rise to a tensile (or hoop) stress component in the taper bore. Ceramic taper bores typically have much larger and more numerous surface flaws from drilling than the smooth articulating surface. Catastrophic failure of ceramic bearings *in vivo* commonly results from slow crack growth, under the static and repetitive loading experienced in the body, until fracture occurs.

Catastrophic failure of  $\text{Al}_2\text{O}_3$  femoral heads *in vivo* has been addressed by the development of alternative ceramic materials, such as  $\text{Al}_2\text{O}_3$  matrix composites and silicon nitride ( $\text{Si}_3\text{N}_4$ ), which have higher strength and fracture toughness than  $\text{Al}_2\text{O}_3$ , in addition to low wear [19]. However, the fracture toughness values of these alternative ceramic materials, typically lower than  $\sim 10 \text{ MPa}\cdot\text{m}^{1/2}$ , are still far below the values for CoCr alloy ( $50$ – $100 \text{ MPa}\cdot\text{m}^{1/2}$ ).

\* Corresponding author. Tel.: +1 573 341 4406; fax: +1 573 341 6934.

E-mail address: [rahaman@mst.edu](mailto:rahaman@mst.edu) (M.N. Rahaman).

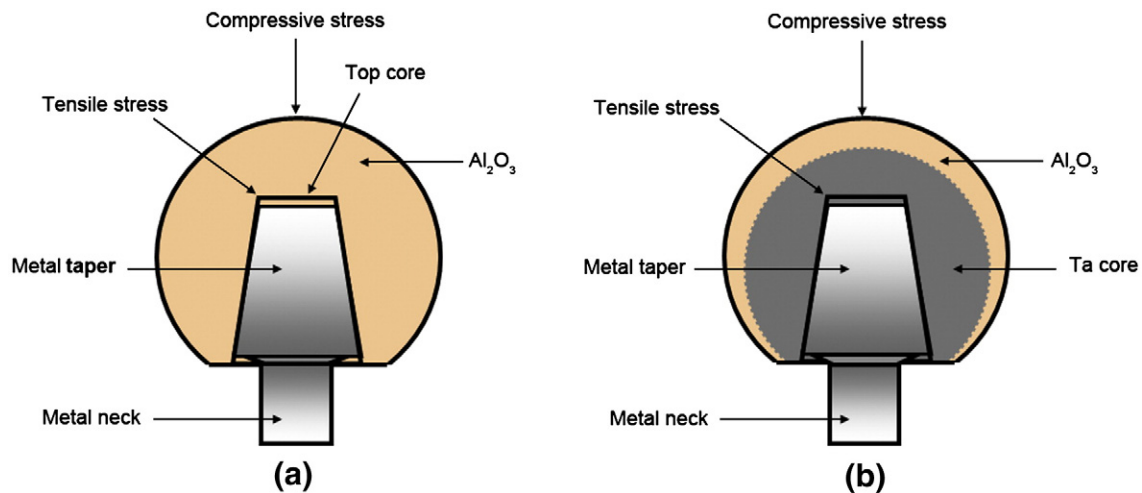


Fig. 1. Schematic diagrams of (a)  $\text{Al}_2\text{O}_3$  femoral head on a metal taper, and (b) proposed design of  $\text{Al}_2\text{O}_3$ -Ta composite femoral head consisting of an  $\text{Al}_2\text{O}_3$  articulating surface and a ductile Ta core.

Surface treatments such as ion implantation, thermal diffusion, or the deposition of hard ceramic coatings have been widely investigated for improving the surface hardness and, hence, the wear and corrosion resistance of metal implants [19,20]. However, many of these treatments have not proved to be successful in clinical applications, because of problems such as delamination of the coating, corrosion, poor interfacial bonding, and porosity [21,22].

Modification of the surface of a metal to form a ceramic has been successfully used in the manufacture of THA femoral heads (as well as bearings for total knee arthroplasty, TKA) from a zirconium alloy (Zr-2.5% niobium) that is oxidized by thermal diffusion to create a 5–10  $\mu\text{m}$  oxidized zirconium layer [23]. The  $\text{ZrO}_2$ -based ceramic surface forms the articulating surface of the femoral head, whereas the metal substrate provides the strength and ductility to resist fracture. Oxidized zirconium has had a successful clinical experience when used as an articulating bearing against UHMWPE, providing favorable reduction in wear when compared to CoCr [24,25]. However, oxidized zirconium is not intended to be used for hard-on-hard bearing applications, whereas  $\text{Al}_2\text{O}_3$  can be used as an articulating bearing against UHMWPE, as well as for  $\text{Al}_2\text{O}_3$ - $\text{Al}_2\text{O}_3$  (hard-on-hard) bearing applications.

The objective of the present work was to explore the feasibility of developing an  $\text{Al}_2\text{O}_3$ -metal composite system which would address concerns about catastrophic failure of  $\text{Al}_2\text{O}_3$  femoral heads *in vivo*. The composite is intended to combine the low wear rate of  $\text{Al}_2\text{O}_3$  with the safety of a ductile metal femoral head (Fig. 1b). This design differs from that of oxidized zirconium in that the ceramic articulating layer has a far higher thickness (2–3 mm). In addition to the low wear of dense  $\text{Al}_2\text{O}_3$ , the thick articulating layer can also contribute substantially to the strength of the composite femoral head. The composite femoral head is intended to provide an alternative to  $\text{Al}_2\text{O}_3$  articulating against UHMWPE, as well as  $\text{Al}_2\text{O}_3$ - $\text{Al}_2\text{O}_3$  (hard-on-hard) bearings. The work reported here explored the fabrication of  $\text{Al}_2\text{O}_3$ -Ta test specimens, and the evaluation of the microstructure and mechanical properties of the composite specimens.

$\text{Al}_2\text{O}_3$  has been used in orthopaedic applications for decades [19]. Tantalum was selected as the metal because of its biocompatibility, corrosion resistance, and engineering properties. Tantalum is being increasingly used for a variety of orthopaedic applications, and porous Ta has shown the ability to support cell and tissue ingrowth [26]. Because of the existence of a stable passivating oxide layer on its surface [27], Ta has low electrochemical potential and high corrosion resistance. Tantalum (density = 16.6 g/cm<sup>3</sup>; melting point = 2996 °C) has a tensile strength of 420 MPa, and an elastic modulus of 186 GPa in tension at 20 °C [28]. The average coefficient of thermal expansion of Ta in the range 20 °C–1000 °C is  $6.73 \times 10^{-6} \text{ }^\circ\text{C}^{-1}$ , compared to a value

of  $8.44 \times 10^{-6} \text{ }^\circ\text{C}^{-1}$  for  $\text{Al}_2\text{O}_3$ , so thermal mismatch stresses and, presumably, the risk of delamination at the interface between the two materials should be small.

## 2. Experimental procedure

### 2.1. Preparation of $\text{Al}_2\text{O}_3$ -Ta composite laminates

The starting materials were high-purity  $\text{Al}_2\text{O}_3$  powder (A16SG; average particle size  $\leq 0.4 \mu\text{m}$ ; Almatix, Leetsdale, PA) and Ta powder (–325 mesh; purity = 99.8%; Atlantic Equipment Engineers, Bergenfield, NJ). Composite laminates of  $\text{Al}_2\text{O}_3$ -Ta with the shape of disks (35 mm in diameter  $\times$  3 mm) were prepared by hot pressing in a graphite die, the contact surfaces of which were coated with boron nitride to limit reaction between the composite and the die. The thickness of the  $\text{Al}_2\text{O}_3$  and Ta layers in the laminate was approximately the same.

The required mass of Ta powder was poured into the graphite die, and the system was vibrated to settle the powder into a layer with a nearly uniform thickness. The process was repeated for the  $\text{Al}_2\text{O}_3$  layer on top of the Ta layer. Hot pressing was performed for 30 min in flowing Argon gas ( $\sim 60 \text{ cm}^3/\text{min}$ ) at 1450 °C or 1650 °C (heating and cooling rates = 10 °C/min) under a pressure of 35 MPa. The pressure was applied when the temperature reached 1200 °C. On cooling after the hot pressing, the pressure was released at 1100 °C. For comparison,  $\text{Al}_2\text{O}_3$  disks were hot pressed under the same conditions.

### 2.2. Mechanical testing

The flexural strength of the  $\text{Al}_2\text{O}_3$ -Ta laminates was determined in four-point loading and compared with the value determined for  $\text{Al}_2\text{O}_3$ . Mechanical testing in flexure is commonly used to measure the strength of (brittle) ceramics, and the main goal of the flexural strength testing was to determine the strength of the  $\text{Al}_2\text{O}_3$ -Ta composite laminate relative to that of  $\text{Al}_2\text{O}_3$ . The interfacial shear strength between  $\text{Al}_2\text{O}_3$  and Ta in the hot-pressed composite laminates was measured using a double-notched coupon test [29,30]. This method was chosen because it has been shown to be more reliable than the short-beam shear test [30,31]. Furthermore, the specimen preparation is simple, and no extensive set-up or fixture is required for the test.

#### 2.2.1. Measurement of flexural strength

Hot-pressed disks of  $\text{Al}_2\text{O}_3$ -Ta composite and  $\text{Al}_2\text{O}_3$  were sectioned and diamond-machined to give rectangular beams (25.0  $\times$  2.0  $\times$  1.5 mm) corresponding to the dimensions of ASTM C1161-02c 'Standard A' bars. The thicknesses of the  $\text{Al}_2\text{O}_3$  and Ta layers in the composite bars were

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