



## Original Research Paper

# One-step hydrothermal synthesis of few-layered and edge-abundant MoS<sub>2</sub>/C nanocomposites with enhanced electrocatalytic performance for hydrogen evolution reaction



Lin Ma<sup>a,\*</sup>, Xiaoping Zhou<sup>a</sup>, Xuyao Xu<sup>a</sup>, Limei Xu<sup>a</sup>, Lingling Zhang<sup>a</sup>, Weixiang Chen<sup>b</sup>

<sup>a</sup>School of Chemistry and Chemical Engineering, Institute of Physical Chemistry, Development Center for New Materials Engineering & Technology in Universities of Guangdong, Lingnan Normal University, Zhanjiang 524048, PR China

<sup>b</sup>Department of Chemistry, Zhejiang University, Hangzhou 310027, PR China

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## ABSTRACT

MoS<sub>2</sub>/C nanocomposite has been successfully synthesized by a one-step hydrothermal method with citrate ammonium as carbon source. The obtained samples are well characterized by X-ray diffraction (XRD), scanning electron microscopy (SEM), energy dispersive X-ray spectroscopy (EDX), high-resolution transmission electron microscopy (HRTEM), Raman spectroscopy, thermo gravimetric analyzer (TG) and X-ray photoelectron spectroscopy (XPS). Moreover, they are also evaluated as electrochemical catalysts in the hydrogen evolution reaction (HER). The results indicate that numerous ultrathin MoS<sub>2</sub> nanosheets which are characteristic of few layers (1–3 layers) and abundant exposed active edges and defects are well-dispersed in the amorphous carbon network to form MoS<sub>2</sub>/C composite. As a consequence, the desirable microstructures endow the resultant MoS<sub>2</sub>/C composite with significantly enhanced catalytic activity for HER such as more positive onset potential, larger current density as well as smaller Tafel slope of 64 mV dec<sup>-1</sup> in comparison with the bare MoS<sub>2</sub>.

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## 1. Introduction

Depletion of fossil fuels and increased environmental concerns have prompted an urgent search for clean and sustainable alternative energy sources [1]. The electrochemical splitting of water via proton reduction in the hydrogen evolution reaction (HER) provides a feasible route toward this without dependence on consumable resources. Electrocatalyst always plays an indispensable role in reducing the overpotential of electrodes in HER, consequently promoting the reaction yield and thus the energy storage efficacy [2,3]. To date, the best known catalysts for HER are precious metals, such as platinum, ruthenium, and iridium, but the scarcity and high cost of these materials prohibit their wide-scale deployment [4]. It is desirable but rather challenging to develop a large-scale, environmentally friendly synthesis of inexpensive alternatives with high electrocatalytic activity to replace Pt-based catalysts.

Molybdenum sulfide (MoS<sub>2</sub>) has a layered structure consisting of S–Mo–S arrangements connected to each sheet by van der Waals forces, which is a long-known catalyst for

hydrodesulfurization [5,6]. Recent studies have shown that nanostructured MoS<sub>2</sub> is a promising electrocatalyst for HER, while the catalytic activity of its bulk counterpart is poor [7,8]. Theoretical calculations combined with experiments have revealed that the edge sites of MoS<sub>2</sub>, especially the sulfur edge-terminated sites are identified as active sites for HER, on which the theoretical free energy of adsorption of hydrogen is similar to that of Pt, while the basal surfaces are catalytically inert [9,10]. Moreover, it has been demonstrated that the exchange current density of MoS<sub>2</sub> nanocrystals is proportional to the length of edge sites [9,10]. For this reason, great efforts have been unremittingly made to enlarge the edge sites of MoS<sub>2</sub> and various MoS<sub>2</sub> nanostructures with preferentially exposed edge sites such as doublegyroid MoS<sub>2</sub> [11], vertically aligned MoS<sub>2</sub> layers [12,13], porous films [14,15], ultrathin MoS<sub>2</sub> nanoplates [16,17], nanospheres [18,19], nanoflowers [20–23], highly divided or distorted MoS<sub>2</sub> [24,25], optimized and tailored MoS<sub>2</sub> nanosheets [26–30] as well as amorphous molybdenum sulfide and clusters [31,32] have been successfully prepared. However, the edge sites are highly energetic and unstable compared with basal plane. Therefore, the design and synthesis of nano-sized edge-terminated MoS<sub>2</sub> are quite difficult and still

\* Corresponding author. Tel.: +86 759 3174025; fax: +86 759 3183510.

E-mail address: [mal@lingnan.edu.cn](mailto:mal@lingnan.edu.cn) (L. Ma).

remain a critical issue for application of MoS<sub>2</sub> as a catalyst for HER [33,34].

Another pivotal inherent problem persecuting MoS<sub>2</sub> in HER is originated from its poor conductivity, especially the extremely low conductivity between two adjacent van der Waals bonded S–Mo–S layers, which greatly compromises the catalytic performance [35]. It is confirmed that the conductivity along two vertically stacked S–Mo–S interlayers is about 2 orders of magnitude lower than that of the intralayers [36]. In addition, the stacking of MoS<sub>2</sub> layers would severely decrease the amount of exposed active sites [37]. Fortunately, layered features along with the weak interlayer interaction has provided the potential capability of exfoliation of bulk MoS<sub>2</sub> into graphene-like single-layered (SL-) or few-layered (FL-) nanosheets through mechanical or chemical techniques [38]. Apparently, reduction of the layers is favorable to decreasing the interlayered resistance and improving the hopping efficiency of electrons between MoS<sub>2</sub> layers [35]. To the present, graphene-like MoS<sub>2</sub> nanosheets has gradually become a very competitive alternative electrocatalyst in HER [17,39–41]. Another approach toward improving the electrical contact is to couple MoS<sub>2</sub> with other conductive materials such as carbon nanotubes, pyrolyzed carbon as well as graphene to fabricate engineered composites [12,17,20,42–48]. It has been demonstrated that incorporation of these conductive materials can effectively promote electron transport from the active sites to the electrodes, accelerate the kinetics of electrode reaction and thus enhance the HER catalytic activity [12,17,20].

Herein, we present a one-step hydrothermal approach to synthesize MoS<sub>2</sub>/C nanocomposite with citrate ammonium as carbon source and appraise them as an electrocatalyst for HER activity. The samples are characterized by XRD, SEM, EDX, TEM/HRTEM, Raman, TG and XPS techniques. The experimental results indicate that the as-prepared MoS<sub>2</sub>/C composite is composed of numerous few-layered and edge-abundant ultrathin MoS<sub>2</sub> nanosheets dispersed in amorphous carbon. As a result, the as-prepared MoS<sub>2</sub>/C nanocomposite demonstrates prominently enhanced electrocatalytic performance for hydrogen evolution reaction compared with the bare MoS<sub>2</sub>.

## 2. Experimental Method

### 2.1. One-step hydrothermal synthesis of MoS<sub>2</sub>/C nanocomposite

In a typical synthetic procedure, 1.5 mmol of ammonium citrate were dissolved into 50 mL deionized water to form a solution. Then 1.5 mmol of ammonium tetrathiomolybdate (NH<sub>4</sub>)<sub>2</sub>MoS<sub>4</sub> were dissolved to the above solution with constant stirring. After that, 10 ml of hydrazine hydrate solution was dropwise added into the above solution under vigorous stirring. Afterwards, the solution was transferred into the 100 mL Teflon-lined stainless steel autoclave, which was heated and maintained at 240 °C for 24 h. After cooling naturally, the black solid product was collected by centrifugation and washed several times with deionized water and ethanol, and dried in the vacuum oven at 80 °C for 12 h. In control experiment, the bare MoS<sub>2</sub> sample was also prepared by a similar synthetic route without ammonium citrate in hydrothermal solution.

### 2.2. Characterizations

XRD patterns were recorded with a Thermo X'TRA X-ray diffractometer with Cu K $\alpha$  radiation ( $\lambda = 0.154056$  nm). The morphologies of the samples were observed by using a SIRION-100 field emission SEM (FESEM). The elemental composition of the samples was analyzed by energy dispersive X-ray spectroscopy (EDX,

GENENIS-4000) and thermogravimetric analysis (TG), which was carried out with a NETZSCH STA 409 PC apparatus at a heating rate of 10 °C min<sup>-1</sup> in flowing air. TEM and HRTEM characterizations were performed on a JEOL JFL-2010 TEM operating at 200 kV, for which each sample was prepared by dispersing it in acetone and drop-casting onto a 200 mesh copper grid coated with holey carbon. Raman spectra were recorded with 514 nm Ar ion laser at 6 mW for 50 s employing a Jobin Yvon LabRam HR spectrometer. X-ray photoelectron spectra (XPS) were processed on a Perkin-Elmer PHI5000c XPS, using C1s (B.E. 284.6 eV) as a reference.

### 2.3. Electrochemical measurements

Electrochemical measurements were performed in a three-electrode system at an electrochemical station (CHI660D). Typically, 4 mg of catalyst and 80  $\mu$ L Nafion solution (5 wt%) were dispersed in 1 mL water–ethanol solution with volume ratio of 4:1 by sonicating for 1 h to form a homogeneous ink. Then 5  $\mu$ L of the dispersion (containing 20  $\mu$ g of catalyst) was loaded onto a glassy carbon electrode with 5 mm diameter (loading ca. 0.102 mg cm<sup>-2</sup>). Linear sweep voltammetry with scan rate of 5 mV s<sup>-1</sup> was conducted in 0.5 M H<sub>2</sub>SO<sub>4</sub> (purged with pure N<sub>2</sub>) using saturated calomel as the reference electrode, a Pt wire as the counter electrode, and the glassy carbon electrode with various samples as the working electrode. All the potentials were calibrated to a reversible hydrogen electrode (RHE). Number of active sites was estimated according to the methods reported previously [16]. AC impedance measurements were carried out in the same configuration at  $\eta = 200$  mV from 10<sup>6</sup> to 0.01 Hz with an AC voltage of 5 mV.

## 3. Results and discussion

### 3.1. Microstructure and morphology of MoS<sub>2</sub>/C composite

The crystalline structure and crystallinity of the as-prepared materials are inspected by X-ray diffraction (XRD) and the results are plotted in Fig. 1. Fig. 1 shows that the pattern of the bare MoS<sub>2</sub> agrees well with a hexagonal structure (JCPDS 37-1492) which belongs to the space group *P6<sub>3</sub>/mmc*. The peaks at  $2\theta = 13.9^\circ$ ,  $33.7^\circ$ ,  $57.6^\circ$  and  $69.2^\circ$  are attributed to the (002), (100), (110) and (201) planes, respectively. The appearance of the (002) peak signifies a stacked layered structure though the weak diffraction peaks are indicative of the low crystallinity of the bare MoS<sub>2</sub>. In sharp contrast to that of the bare MoS<sub>2</sub>, the (002) plane peak of the MoS<sub>2</sub>/C composite become extremely weak and obscure, indicating that the stacking of MoS<sub>2</sub> layers

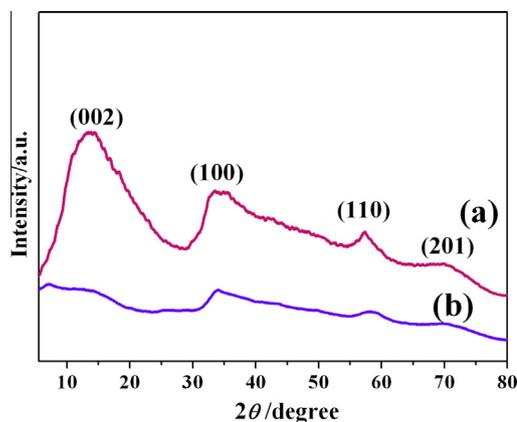


Fig. 1. XRD patterns of the as-prepared (a) bare MoS<sub>2</sub> and (b) MoS<sub>2</sub>/C composite.

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