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**CERAMICS**INTERNATIONAL

Ceramics International 39 (2013) S491-S495

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# Microwave dielectric properties of bismuth zinc niobate thin films deposited on alumina by pulsed laser deposition

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Available online 24 October 2012

#### Abstract

Bi<sub>2</sub>Zn<sub>2/3</sub>Nb<sub>4/3</sub>O<sub>7</sub> thin films were prepared on Al<sub>2</sub>O<sub>3</sub> substrates by pulsed laser deposition. The phase compositions and microstructures were characterized by X-ray diffraction and atomic force microscopy. The as-deposited films were all amorphous in nature. All films were crystallized after the post annealing at the temperature range of 700–900 °C for 30 min in air. The texture characteristics change with annealing temperature. A split post dielectric resonator method was used to measure the microwave dielectric performance at the resonant frequencies of 10, 15 and 19 GHz. For the films annealed at 900 °C, the preferential orientation is similar to the monoclinic BZN bulk. The microwave dielectric constants at 10, 15 and 19 GHz are 69.4, 58.9 and 47.9, respectively, which are closer to these of the monoclinic BZN bulk.

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Keywords: A. Films; C. Dielectric properties; E. Capacitors; Bi<sub>2</sub>Zn<sub>2/3</sub>Nb<sub>4/3</sub>O<sub>7</sub>

#### 1. Introduction

In recent years, thin films of Bi<sub>2</sub>O<sub>3</sub>–ZnO–Nb<sub>2</sub>O<sub>5</sub> (BZN) has drawn a lot of attention for their excellent dielectric properties such as moderate high dielectric constant, low dielectric loss, lower crystallization temperature and dielectric tunability [1–3]. There are two main phases in the BZN system: a cubic pyrochlore based on Bi<sub>1.5</sub>ZnNb<sub>1.5</sub>O<sub>7</sub> (c-BZN) composition and a monoclinic zirconolite based on Bi<sub>2</sub>Zn<sub>2/3</sub>Nb<sub>4/3</sub>O<sub>7</sub> (m-BZN) composition [4]. In previous works, it was found that Bi<sub>2</sub>Zn<sub>2/3</sub>Nb<sub>4/3</sub>O<sub>7</sub> (m-BZN) bulk material has low sintering temperature, relatively moderate dielectric constant (~80 at 10 kHz), low dielectric loss ( $5 \times 10^{-4}$  at 10 kHz) and low temperature coefficient of dielectric permittivity (150 ppm/K) [4,5]. Several deposition methods have been used to prepare BZN thin films, such as metal organic decomposition, RF magnetron sputtering, metal organic chemical vapour deposition, and pulsed laser deposition (PLD) [6-8]. Compared to other deposition methods, PLD has merits in excellent control of stoichiometric composition of oxide films with complex components, especially for BZN thin films due to high volatility of Zn and Bi [9]. The dielectric properties of m-BZN thin films have been studied by several researchers [10,11]. In this paper, PLD deposition was used to prepare the m-BZN thin films. The post annealing crystallization and microwave properties of m-BZN thin films have been investigated.

#### 2. Experimental details

 $Bi_2Zn_{2/3}Nb_{4/3}O_7$  (m-BZN) ceramic targets were synthesized by a solid state reaction processing. Raw materials of  $Bi_2O_3$ , ZnO and  $Nb_2O_5$  of high purity were weighed in stoichiometric proportions, ball milled for 24 h, then sintered for 3 h at 960 °C. The m-BZN films were prepared by PLD using a KrF excimer laser. The thin films were deposited directly on alumina substrates, without use of a metallic electrode. The PLD chamber was evacuated to a pressure of  $4 \times 10^{-4}$  Pa. The laser was operated at a laser repetition frequency of 3 Hz, with an energy density of 250 mJ/pulse. The substrate temperature was set at 600 °C and the chamber oxygen pressure during deposition was set at 10 Pa. The deposition was carried out for 60 min. After the deposition, the films were postannealed in a rapid thermal annealing furnace at the

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temperature range of 700–900 °C for 30 min in air. The thicknesses of these films were measured by a stylus profiler (Dektak 6 M, Veeco) and are 540–580 nm.

The phase composition and crystallization of the BZN thin films were characterized by X-ray diffraction (XRD, D/Max-2400, Rigaku). Surface morphology was examined by an atomic force microscope (AFM, Nanoscope, Veeco). Electron probe micro-analysis (JXA-8100, JEOL) was used to determine the composition of the films. Microwave dielectric properties of these films were measured by a split-post dielectric resonator (SPDR) technique. The dielectric resonator was connected to a microwave network analyzer (HP8720ES, Agilent) and the resonant frequencies were 10, 15 and 19 GHz, respectively.

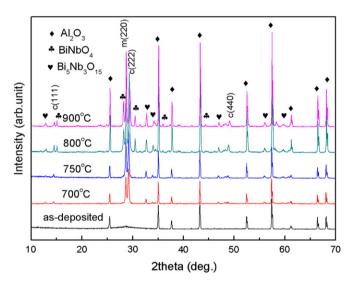


Fig. 1. X-ray diffraction patterns of  $Bi_2Zn_{2/3}Nb_{4/3}O_7$  (m-BZN) thin films as a function of annealing temperature (c for cubic pyrochlore; m for monoclinic zirconolite-like pyrochlore).

#### 3. Results and discussion

Fig. 1 shows the XRD patterns of both as-deposited and annealed m-BZN thin films. There are no other peaks detected in the as-deposited films except those from the substrates, which indicates that the as-deposited films are amorphous in nature. This is different from the asdeposited films deposited on Pt/TiO<sub>2</sub>/SiO<sub>2</sub>/Si substrate with the same deposition conditions, which shows a crystalline structure. It suggests that the substrates have strong influence on the crystallization of m-BZN thin films. However, after post-annealing in air, the m-BZN thin films are crystallized. For the films annealed at 700 °C and 750 °C for 30 min, the mixing peaks of cubic pyrochlore BZN (222) and monoclinic zirconolite BZN (220) could be detected, which suggests that a cubic and a monoclinic zirconolite co-existed phase existed in BZN films. Meanwhile, the secondary phases of BiNbO<sub>4</sub> and Bi<sub>5</sub>Nb<sub>3</sub>O<sub>15</sub> appear when annealed over 800 °C, due to the significant loss of Zn volatility at high temperature (as discussed next). Meanwhile, the full width half maximum (FWHM) of m(220) peak gradually decreases and the FWHM of c(2 2 2) peak also decreases with annealing temperature, as shown in Fig. 2. It is well known that the FWHM values are related with the film crystallization. Hence, high annealing temperature can improve the film crystallization. However, high annealing temperature also introduces impurity secondary phases, like BiNbO<sub>4</sub> and Bi<sub>5</sub>Nb<sub>3</sub>O<sub>15</sub>. It is noted that as the annealing temperature increases, the m(2 2 0) peak intensity increases and the c(222) peak intensity does not change much. Thin films annealed at 800 °C and 900 °C show stronger  $m(2\ 2\ 0)$  peak than the  $c(2\ 2\ 2)$  peak. This indicates that the monoclinic phase is dominant in the films when annealed at high annealing temperature. It indicates that the annealing temperature plays an important role in the crystallization of m-BZN thin films.

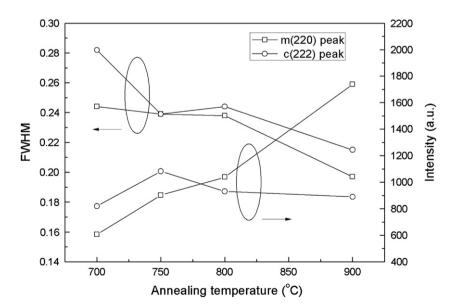


Fig. 2. FWHM and intensity values of c(2 2 2) and m(2 2 0) diffraction peaks of m-BZN thin films at various annealing temperatures.

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