

Low temperature neutron diffraction studies on $\text{Bi}_4\text{Ti}_3\text{O}_{12}$

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Abstract

The detailed crystal structure of $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ obtained by the Rietveld refinement of the powder neutron diffraction data in the temperature range of 300–15 K is being reported. At ambient temperature $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ has an orthorhombic structure (Space Group: B2cb) with unit cell parameters: $a = 5.4432(5) \text{ \AA}$, $b = 5.4099(5) \text{ \AA}$, $c = 32.821(2) \text{ \AA}$, and $V = 966.5(1) \text{ \AA}^3$. The low temperature neutron diffraction studies revealed the retention of the orthorhombic structure without any significant change in the atoms arrangement. A marginal decrease in the unit cell parameters is observed with the lowering of the temperature.

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1. Introduction

Crystalline materials of Aurivillius [1] type structure have been studied owing to their interesting ferroelectrics [2] and fast ion conduction Abraham et al. [3] properties. The Aurivillius type layered compounds can be represented by the general formula $[\text{M}_2\text{O}_2]^{+2}[\text{A}_{m-1}\text{B}_m\text{O}_{3m+1}]^{2-}$, where M is generally Bi^{3+} . The crystal structure of such compounds is built by the intergrowth of fluorite-type layers and perovskite type layers [1,4]. The perovskite-type layer is formed with A and B cations and the thickness of this slab is governed by the integer m . Thus, the $m = 1, 2, 3$, etc. indicate the one, two and three layer perovskite layers. The selection of the A and B cation is mainly based on the nature and ionic radii requirement of the perovskite lattice. The perovskite slab is formed with the corner sharing of the BO_6 octahedra and the interstices are occupied by the A type cations. The fluorite-type layer is formed by the Bi and O atoms, and Bi have square pyramidal polyhedra with lone pair of the Bi^{3+} as an apex. The presently studied compound bismuth titanate ($\text{Bi}_4\text{Ti}_3\text{O}_{12}$) is a three layers Aurivillius

compound, i.e., $m = 3$, and the composition can be written as $(\text{Bi}_2\text{O}_2)(\text{Bi}_2\text{Ti}_3\text{O}_{10})$ [5].

Bismuth titanate ($\text{Bi}_4\text{Ti}_3\text{O}_{12}$) bears a special interest in the Aurivillius family of compounds due to its ferroelectric properties, which has switching application with low leakage and fatigue and high remnant polarization [6,7]. Due to these properties this material is a promising candidate for nonvolatile memory and dynamic memory of computers. This compound shows ferroelectric to paraelectric phase transition at about 675 °C [5,8]. There are several structural studies at room temperature and higher temperature available in the literature explaining the crystal structure of both ferro- and paraelectric phase [5,8,9,10]. The early powder XRD studies on $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ indicated an orthorhombic lattice with space group Fmmm [1]. Later the detailed crystal structure of ferroelectric phase of $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ showed the similar orthorhombic lattice but the space group is changed to B2cb [8,9]. Further, the crystal structure was re-determined by [10,11] with a monoclinic (Pc) lattice. However, the monoclinic distortion in the lattice is reported to be too small to differentiate these two lattice types. Kim and Jeon [12] studied this compound by combined neutron and X-ray diffraction of the polycrystalline sample and supported the monoclinic lattice. Besides, there are few reports dealing with structure below ambient temperature [13–15]. A possible existence of low temperature phase was earlier indicated from dielectric anomalies [13]. Recently, the change

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in crystal structure at low temperature has been observed by variable temperature Raman spectroscopy [15]. But, no detailed natures of crystallographic transformations are available in literature.

All the available structural studies on $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ are carried out at either ambient or high temperatures. To the best of our knowledge, no diffraction data are available below ambient temperature. In the present study our intention was to look for the low temperature crystal structure.

2. Experimental

The titled compound was prepared by heating a homogeneous mixture of appropriate amounts of pre-dried Bi_2O_3 and TiO_2 at 750°C for 12 h in a platinum boat. The product obtained was reground and pelletized and the pellet was further heated at 850°C for another 12 h. The product obtained was characterized by powder XRD for its phase purity. The powder XRD data was collected using a Philips PW1710 model diffractometer in the two theta range $10\text{--}70^\circ$, with Ni filtered $\text{Cu K}\alpha$ radiation. The observed diffraction data agree well with reported powder data JCPDS 35-795. Further the crystal structure was verified by the Rietveld refinement of the powder neutron diffraction data. For the neutron diffraction studies, the sample was packed in a vanadium can (8 mm diameter and 5 cm height) and diffraction data were recorded using 1.249 \AA wavelength, with an array of five linear PSD based powder diffractometer at 100 MW Dhruva Research Reactor at BARC, Mumbai. Low temperature diffraction data were collected using an APD make closed cycle helium refrigerator. The diffraction data were analyzed by Rietveld refinements of the observed data using the Fullprof-2K software package [16].

3. Results and discussion

The Rietveld refinement of the observed powder neutron diffraction data was initiated with the starting model based on the orthorhombic (B2cb) data reported by [9]. This space group shows an excellent fit to our diffraction data. The profile was

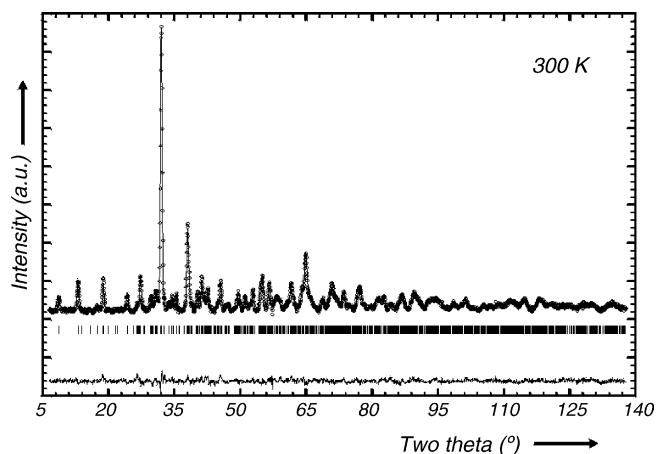


Fig. 1. Experimental (○) and calculated (continuous line) neutron diffraction pattern of $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ at 300 K. The difference profile is given at the bottom. The Bragg positions are indicated by the vertical marker below the observed pattern.

fitted with Pseudo-Voigt profile function. The profile refinement was started with scale and background parameters followed by the unit cell parameters. The typical half-width parameters, mixing parameter and preferred orientation parameter were also refined. After getting a proper match in the profile model, the positional parameters and overall thermal parameters were refined. The goodness of the refinements was observed by the residuals (R -values). The typical observed and calculated neutron diffraction patterns at 300 K are shown in Fig. 1. The refined unit cell parameters and the residuals of refinement are given in Table 1. The observed refined positional coordinates do not show any significant difference from the earlier reported values [9]. The low temperature diffraction data were similarly analyzed using the observed ambient temperature structural model. The observed and calculated neutron diffraction patterns at the lowest studied temperature (15 K) are identical to Fig. 1. The refined unit cell parameters at different temperatures are collected in Table 1. A three-dimensional representation of the $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ unit cell is shown in Fig. 2.

As it was mentioned in the introduction section, the crystal structure of the $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ is made by stacking of layers of

Table 1
Typical crystallographic and refinement parameters of $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ at various temperatures

	300 K	150 K	75 K	15 K
Crystal system	Orthorhombic	Orthorhombic	Orthorhombic	Orthorhombic
Space group	B2cb	B2cb	B2cb	B2cb
a (Å)	5.4432(5)	5.4393(6)	5.4363(4)	5.4354(4)
b (Å)	5.4099(5)	5.4035(6)	5.4012(4)	5.4004(4)
c (Å)	32.821(2)	32.772(3)	32.756(2)	32.743(2)
V (Å ³)	965.51(1)	963.2(2)	961.8(1)	961.1(1)
Neutron wavelength (Å)	1.249	1.249	1.249	1.249
2θ Range (°)	6.61–138.00	6.61–138.00	6.61–138.00	6.61–138.00
Number of variables parameters	46	46	46	46
R_p %	5.21	7.78	5.62	5.85
R_{wp} %	6.66	9.90	7.17	7.44
R_c %	3.94	3.89	3.91	3.93
R_{Bragg} %	5.24	6.55	5.65	5.94
χ^2	2.85	6.39	3.37	3.58

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