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Characterization of Eu³⁺-doped fluorophosphate glasses for red emission

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Abstract

Fluorescence spectra and decay curves of the 5D_0 level for different concentrations of Eu^{3+} (4f 6) ions in K–Ba–Al fluorophosphate glasses have been measured at room temperature and are analyzed. The Judd–Ofelt intensity parameters Ω_2 and Ω_4 have been determined from the intensity ratios of emission peaks corresponding to $^5D_0 \rightarrow ^7F_J$ (J=2 and 4) to $^5D_0 \rightarrow ^7F_1$ transitions for 1.0 mol% glass. The intensity parameters thus obtained are in turn used to calculate the radiative properties of the fluorescent levels of Eu^{3+} ions. Second and fourth rank crystal-field parameters have been evaluated by assuming a C_{2V} site symmetry for the local environment of Eu^{3+} ions to estimate the crystal-field strength experienced by Eu^{3+} ions in the present host. The decay profiles of the $^5D_0 \rightarrow ^7F_2$ transition of Eu^{3+} ions in the present glasses are found to be single exponential for all the studied Eu^{3+} ion concentrations. A marginal increase in lifetime of the 5D_0 level has been noticed with Eu^{3+} ion concentration up to 2.0 mol% and then the lifetime marginally decreases for higher Eu^{3+} ion concentrations.

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1. Introduction

Alkali aluminometaphosphate glasses are the promising hosts among oxide glasses for lanthanide (Ln) ions due to their relatively higher refractive indices, lower dispersion, larger stimulated emission cross-sections, higher gain, better chemical durability and lower thermo-optical qualities when compared with many other hosts such as silicates [1]. They are very easy to prepare in various compositions in bulk form and they preserve useful properties upon the introduction of a significant amount of Ln ions. Addition of fluoride contents will further improve the radiative prop-

erties by reducing the number of hydroxyl (OH⁻) groups in the glass network [2].

Among Ln ions, trivalent europium (Eu^{3+}) ion is the best candidate to be used as a probe to investigate the local structure around Ln ions in condensed matter. This is because, the low lying energy level scheme of Eu^{3+} ions is simple and the transition between 7F_0 and 5D_0 non-degenerate levels is convenient to apply the fluorescence line narrowing technique to analyze changes from site to site through energy level scheme analysis [3]. Also Eu^{3+} -doped materials are frequently used in photonic applications as red phosphors for field emission technology due to the narrow and monochromatic emitting nature of the $^5D_0 \rightarrow ^7F_2$ transition of Eu^{3+} ions at 610 nm. The present work reports the fluorescence and local field properties of different concentrations of Eu^{3+} ions in K–Ba–Al fluorophosphate glass.

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2. Experimental

Eu³⁺-doped fluorophosphate glasses with composition (in mol%) $(56 - x/2)P_2O_5 + 17.0K_2O + (15 - x/2)BaO + 8.0Al_2O_3 + 4.0AlF_3 + xEu_2O_3$ (referred as PKBAFEu) for x = 0.01, 0.1, 1.0, 2.0, 4.0 and 6.0 mol%, were prepared by the conventional melt quenching technique using high purity chemicals of KH₂PO₄, Ba(PO₃)₂, Al(PO₃)₃, AlF₃ and Eu₂O₃ as starting materials. The batch composition was melted around 1075 °C for 1 h. The samples were then kept for annealing at 350 °C for 10 h and then slowly allowed to cool to room temperature (RT). Refractive index, density, optical absorption, emission and fluorescence decay measurements were carried out following the similar methods mentioned in Refs. [4,5].

3. Results

3.1. Emission spectra – determination of Judd–Ofelt parameters

The optical absorption and emission spectra for the 1.0 mol% Eu³⁺-doped PKBAFEu fluorophosphate glass (referred as PKBAFEu10) were recorded at RT and are shown in Figs. 1 and 2, respectively. The bands in the absorption spectra correspond to the transitions originating from the 7F_0 and 7F_1 levels and the peaks in the emission spectra correspond to the well-known $^5D_0 \rightarrow ^7F_J$ (J=0–6) transitions. The Judd–Ofelt (JO) intensity parameters, Ω_{λ} ($\lambda=2$, 4 and 6) have been calculated using the ratio of intensities of $^5D_0 \rightarrow ^7F_J$ (J=2,4 and 6) transitions to that of the $^5D_0 \rightarrow ^7F_1$ transition [5–9].

3.2. Radiative properties

The calculated JO parameters have been used to determine the radiative properties by following the procedure

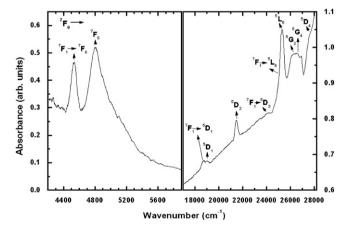


Fig. 1. Optical absorption spectrum of PKBAFEu10 glass at room temperature (all the transitions are originated from the ground 7F_0 level, unless otherwise specified).

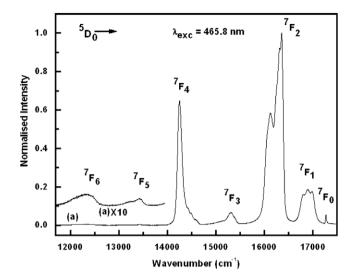


Fig. 2. Emission spectrum of the PKBAFEu10 glass at room temperature (weak intense $^5D_0 \rightarrow ^7F_{5,6}$ peaks are also shown with tenfold magnification).

outlined in the earlier reported works [7–10]. The Ω_{λ} , $(\times 10^{-20} \, \text{cm}^2)$, experimental (τ_{exp}) and calculated (τ_{cal}) lifetimes (ms), quantum efficiencies (η), non-radiative transition probabilities (W_{nr}, s^{-1}) and intensity ratios (R(J/1) = $(^{5}D_{0} \rightarrow {^{7}F_{J}})/(^{5}D_{0} \rightarrow {^{7}F_{1}}), J = 2 \text{ and 4})$ have been determined for the PKBAFEu10 glass and are presented in Table 1. Similar results obtained for other Eu³⁺:glasses such as $58.5P_2O_5 + 17K_2O + 14.5BaO + 9Al_2O_3 + 1Eu_2O_3$ (PKBAEu10) [5], $55.5P_2O_5 + 17K_2O + 11.5BaO + 6BaF_2 +$ $9Al_2O_3 + 1Eu_2O_3$ (PKBFAEu10) [5], Eu^{3+} : Ba(PO₃)₂ (PBEu) [11], $Al(NO_3)_3 + SiO_2 + Eu_2O_3$ (ASEu) [12] and $57ZrF_4 +$ $36BaF_2 + 3LaF_3 + 3AlF_3 + 1EuF_3$ (Eu³⁺:ZBLA) [13]; and $SiO_2 + Al_2O_3 + Eu_2O_3$ xerogels (SAEu) [14] are also presented in Table 1 for comparison. The emission peak wavelengths (λ_p , nm), radiative transition probabilities (A, s⁻¹), experimental and calculated branching ratios (β_R), effective bandwidths ($\Delta \lambda_{eff}$, nm) and peak stimulated emission crosssections $(\sigma(\lambda_p), \times 10^{-21} \text{ cm}^2)$ have been determined for different transitions originating from the luminescent ⁵D₀ level of Eu³⁺ ions in PKBAFEu10 glass and are shown in Table 2.

3.3. Crystal-field analysis

A careful observation of ${}^5\mathrm{D}_0 \to {}^7\mathrm{F}_J (J=1 \text{ and } 2)$ peaks in the emission spectrum shown in Fig. 2 yields Stark splittings of these peaks into 3 and 5 components, respectively. The second and fourth rank crystal-field (CF) parameters B_{kq} and the CF strength parameter (S) have been calculated by assuming the $C_{2\mathrm{V}}$ site symmetry around Eu³⁺ ions in PKBAFEu10 glass [15–18]. The CF parameters B_{kq} and their ratios B_{22}/B_{20} and B_{44}/B_{40} for the present PKBAFEu10 glass are presented in Table 3 along with similar parameters for the PKBAEu10 and PKBFAEu10 glasses [5] for comparison.

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