



# Red-emitting LaOF:Eu<sup>3+</sup> phosphors: Synthesis, structure and their Judd–Ofelt analysis for LED applications

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## ARTICLE INFO

### Article history:

Received 23 May 2015

Received in revised form 24 October 2015

Accepted 17 November 2015

Available online 30 November 2015

### Keywords:

A. Inorganic compounds

B. Chemical synthesis

C. X-ray diffraction

C. Infrared spectroscopy

D. Phosphors

## ABSTRACT

In the present study, we have synthesized a series of La<sub>1-x</sub>Eu<sub>x</sub>OF (0.01 ≤ x ≤ 0.09) phosphors by the conventional solid-state reaction route at relatively low temperature (500 °C) and shorter duration of 2 h. The compounds were crystallized in the rhombohedral structure with the space group R-3m (No. 166). Upon UV excitation (254 nm), the photoluminescence spectra exhibit characteristic luminescence <sup>5</sup>D<sub>0</sub> → <sup>7</sup>F<sub>J</sub> (J = 1, 2, 3, and 4) intra-4f shell Eu<sup>3+</sup> ion transitions. An intense red emission peak at 610 nm was observed due to electric dipole (<sup>5</sup>D<sub>0</sub> → <sup>7</sup>F<sub>2</sub>) transition. Judd–Ofelt theory was employed to evaluate various radiative parameters such as radiative emission rates, lifetime, branching and asymmetry ratios. CIE color coordinates confirmed the red emission of the phosphors. The luminescent results reveal that LaOF:Eu<sup>3+</sup> phosphor can be used as potential candidate for developing red component in white LED applications.

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## 1. Introduction

In the recent years, rare earth ion-doped inorganic materials have attracted considerable interest because of its excellent luminescent properties [1]. It has been considered that the host matrix plays a significant role on the luminescent properties of the rare-earth ion [2]. Among various materials, lanthanum oxyfluoride (LaOF) has been drawn much attention due to their high chemical stability, low phonon energy and lasing characteristics [3,4]. The lanthanum oxyfluoride compounds have been used for a long time as hosts for various activators, such as Yb<sup>3+</sup>, Er<sup>3+</sup>, Eu<sup>3+</sup>, Ce<sup>3+</sup>, Pr<sup>3+</sup> [5–9]. It is generally accepted that trivalent europium ions (Eu<sup>3+</sup>) in phosphor materials have been widely used as an activator, which exhibit red luminescence under UV excitation due to <sup>5</sup>D<sub>0</sub> → <sup>7</sup>F<sub>2</sub> transition [10]. Eu<sup>3+</sup> provides a favorable situation for substitution in La<sup>3+</sup> sites with suitable isostructural replacement. Trivalent europium ions exhibit narrow band emissions, large Stokes shifts and long lifetimes [11].

Various synthesis methods have been reported to prepare LaOF materials such as hydrothermal, sol–gel, chemical vapor deposition, stearic acid and combustion methods [12–16]. However, most

of these methods cannot be applied economically on large scale due to their complicated synthesis route, high reaction temperature, long reaction time, less yield and difficult to control stoichiometric compositions.

In this paper, we report the synthesis of La<sub>1-x</sub>Eu<sub>x</sub>OF (0.01 ≤ x ≤ 0.09) phosphors by the facile solid-state method at 500 °C for 2 h. The crystallographic structural parameters, functional groups, morphology were analyzed by powder X-ray diffraction (PXRD), fourier transform infrared (FTIR) spectroscopy and field emission scanning electron microscopy (FESEM), respectively. Electron paramagnetic resonance (EPR) and UV–vis absorption spectroscopy were also studied. The influence of Eu<sup>3+</sup> concentration on the photoluminescence properties of LaOF phosphors was reported in detail. Further, the Judd–Ofelt intensity parameters, radiative emission rates, lifetime, branching and asymmetry ratios were calculated from the luminescence data.

## 2. Experimental

### 2.1. Materials

Analytical reagent grade La<sub>2</sub>O<sub>3</sub>, Eu<sub>2</sub>O<sub>3</sub> and NH<sub>4</sub>F were procured from Sigma–Aldrich and used as precursors for the synthesis of LaOF:Eu<sup>3+</sup> phosphors. Before weighing, La<sub>2</sub>O<sub>3</sub> and Eu<sub>2</sub>O<sub>3</sub> were

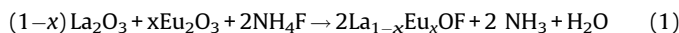
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calcined at 800 °C for 6 h to remove moisture from the starting materials.

## 2.2. Synthesis of $\text{LaOF:Eu}^{3+}$ phosphors

A series of  $\text{La}_{1-x}\text{Eu}_x\text{OF}$  ( $0.01 \leq x \leq 0.09$ ) phosphors were synthesized using the conventional solid-state reaction route. The required amount of  $\text{La}_2\text{O}_3$ ,  $\text{Eu}_2\text{O}_3$  and  $\text{NH}_4\text{F}$  (20% excess) were ground thoroughly in an agate mortar with pestle. The mixed powder was transferred into porcelain crucible, calcined at 500 °C for 2 h (ambient atmosphere) and cooled down to room temperature. The chemical reaction for the synthesis of  $\text{La}_{1-x}\text{Eu}_x\text{OF}$  phosphors are represented as follows:



## 2.3. Characterization

Powder X-ray diffraction (PXRD) of the phosphors were performed on PANalytical X'Pert Pro Powder diffractometer using  $\text{Cu K}\alpha$  radiation ( $\lambda = 1.5418 \text{ \AA}$ ) with a nickel filter. Rietveld refinement data were collected at a scan rate of  $1^\circ/\text{min}$  with a  $0.02^\circ$  step size for  $2\theta$  from  $10^\circ$  to  $80^\circ$ . FullProf suite–2000 programme were used to estimate the structural parameters using Rietveld refinement method. The surface morphology was examined using FEI Quanta 200 scanning electron microscope. FTIR spectra were recorded using Perkin Elmer Frontier Spectrometer in the wave number range of  $300\text{--}4000 \text{ cm}^{-1}$ . The data was collected at room temperature using pellet method with transmission mode and KBr as a reference. The 13 mm diameter thin pellet was made by mixing the sample with dried KBr powder. UV–vis absorption spectra have been recorded for solid powders on PerkinElmer

Lambda 750 spectrophotometer with deuterium and tungsten halogen light sources and double holographic grating monochromators using  $\text{BaSO}_4$  as a background or standard reference. Electron paramagnetic resonance (EPR) were performed using Bruker X-band ESR spectrometer and DPPH (2,2-diphenyl-1-picrylhydrazyl) was used as a field marker. The photoluminescence (PL) studies have been carried out using a Jobin Yvon spectrofluorometer (Fluorolog-3, Horiba) equipped with a 450 W xenon lamp as the excitation source. All the measurements were performed at room temperature.

## 3. Results and discussion

Fig. 1 shows the powder X-ray diffraction patterns of  $\text{La}_{1-x}\text{Eu}_x\text{OF}$  ( $0.01 \leq x \leq 0.09$ ) phosphors. All diffraction peaks could be indexed with the standard rhombohedral phase of LaOF (JCPDS No. 06-0281) with space group R-3m (No. 166), that indicates the synthesized compounds were in single crystalline phase. On increasing the substitution of  $\text{Eu}^{3+}$  ion to  $\text{La}^{3+}$  ion in LaOF matrix, we observed a small impurity of  $\text{Eu}_2\text{O}_3$ . The structural parameters were obtained from the Rietveld refinement method using PXRD data. The patterns were typically refined for lattice parameters, scale factor, backgrounds, pseudo-Voigt profile function ( $u$ ,  $v$  and  $w$ ), atomic coordinates and isothermal temperature factors ( $B_{\text{iso}}$ ). Fig. 2 shows the observed, calculated and the difference PXRD patterns of  $\text{La}_{1-x}\text{Eu}_x\text{OF}$  ((a)  $x = 0.01$ , (b)  $x = 0.05$ , and (c)  $x = 0.09$ ). The difference between XRD pattern profiles experimentally observed and calculated data display near to zero in the intensity scale as illustrated by a line ( $Y_{\text{obs}} - Y_{\text{calc}}$ ). The refined structural parameters for  $\text{La}_{1-x}\text{Eu}_x\text{OF}$  ( $x = 0.01$ ,  $0.05$ , and  $0.09$ ) phosphor are summarized in Table 1. With the substitution of 8-coordinated  $\text{Eu}^{3+}$  ion ( $r_{\text{Eu}^{3+}} = 1.066 \text{ \AA}$ ) to  $\text{La}^{3+}$  ( $r_{\text{La}^{3+}} = 1.160 \text{ \AA}$ ) ion in LaOF matrix, we did not observe any appreciable change in the structural parameters. The crystal structure of rhombohedral LaOF was modeled through VESTA software using lattice parameters and atom positions obtained from the Rietveld refinement data (Fig. 2(d)). The  $\text{La}^{3+}$  ions or  $\text{Eu}^{3+}$  ions are coordinated by four oxide and four fluoride anions. All ions occupy the six-fold 6c Wyckoff positions and the symmetry for  $\text{La}^{3+}$  ions is  $\text{C}_{3v}$ .

Fig. 3 shows the FESEM micrographs of (a)  $\text{La}_{0.99}\text{Eu}_{0.01}\text{OF}$  and (b)  $\text{La}_{0.93}\text{Eu}_{0.07}\text{OF}$  phosphors. The images revealed the plate-like morphology which is agglomerated due to the solid-state reaction route. The FTIR spectra of  $\text{La}_{1-x}\text{Eu}_x\text{OF}$  phosphors were measured in the wave number region of  $300\text{--}4000 \text{ cm}^{-1}$  as shown in Fig. 4. The spectra show two characteristic absorption bands around  $500$  and  $370 \text{ cm}^{-1}$  for all the compositions. These bands can be attributed to La–O vibrations [17]. The low intensity band at  $1430 \text{ cm}^{-1}$  was assigned to the vibrational mode of adsorbed  $\text{CO}_3^{2-}$  group from the atmosphere. The peak around  $3436 \text{ cm}^{-1}$  can be ascribed to the bending vibration of surface adsorbed water molecule.

Fig. 5 shows the UV–vis absorption spectra of  $\text{La}_{1-x}\text{Eu}_x\text{OF}$  phosphors in the range of  $250\text{--}800 \text{ nm}$ . The absorption peak and edge of  $\text{LaOF:Eu}^{3+}$  are located around  $260$  and  $330 \text{ nm}$ , respectively. Further, we observed the small sharp peaks in  $400\text{--}600 \text{ nm}$  range which was due to the absorption of  $\text{Eu}^{3+}$  ion. The absorption peaks at  $416$ ,  $465$ , and  $536 \text{ nm}$  were assigned to  ${}^7\text{F}_0 \rightarrow {}^5\text{D}_3$ ,  ${}^7\text{F}_0 \rightarrow {}^5\text{D}_2$ , and  ${}^7\text{F}_1 \rightarrow {}^5\text{D}_1$ , respectively (inset of Fig. 5). The optical energy band gap ( $E_g$ ) was estimated from these absorption spectra using Wood and Tauc relation [18]. The energy band gap for all compounds was  $\sim 4.77 \text{ eV}$  which is close to the reported value [19,20].

The EPR spectra of  $\text{La}_{1-x}\text{Eu}_x\text{OF}$  ( $x = 0.01$  and  $0.09$ ) phosphors were measured at room temperature and is shown in Fig. 6. It was observed that the EPR signals consist of broad spectra in the range of  $0\text{--}7000 \text{ Gauss}$ . The  $g$  value calculated for  $x = 0.01$  and  $0.09$  was found to be  $1.78$  and  $1.80$ , respectively which suggests that there is no variation in EPR spectra with the higher  $\text{Eu}^{3+}$  ion content.

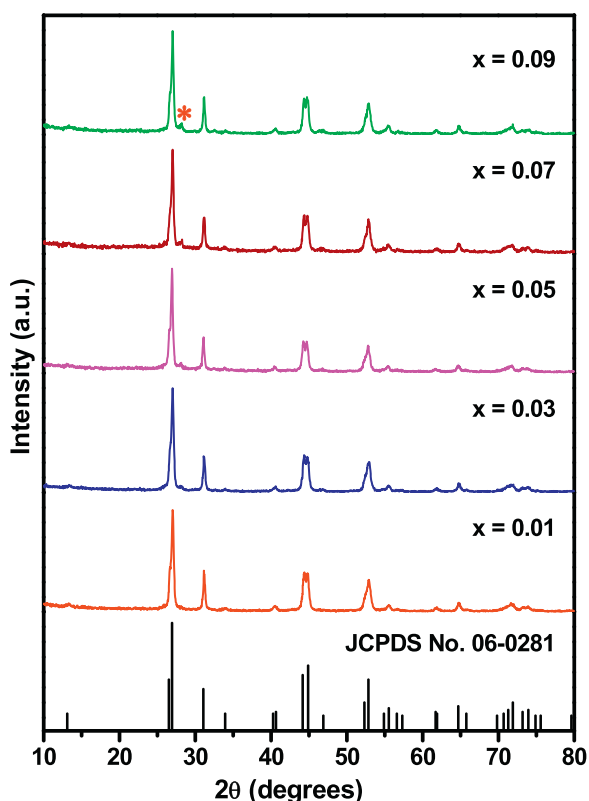


Fig. 1. Powder XRD patterns of  $\text{La}_{1-x}\text{Eu}_x\text{OF}$  ( $0.01 \leq x \leq 0.09$ ) phosphors (The asterisk represents the small  $\text{Eu}_2\text{O}_3$  impurity).

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