



Short communication

Hydrothermal synthesis of superparamagnetic Fe₃O₄ nanoparticles with ionic liquids as stabilizerXiao-Di Liu^{*}, Hao Chen, Shan-Shan Liu, Li-Qun Ye, Yin-Ping Li

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ARTICLE INFO

Article history:

Received 25 June 2014

Received in revised form 17 October 2014

Accepted 8 November 2014

Available online 11 November 2014

Keywords:

A. Magnetic materials

B. Chemical synthesis

B. Magnetic properties

ABSTRACT

Superparamagnetic Fe₃O₄ nanoparticles have been successfully synthesized under hydrothermal condition with the assistant of ionic liquid 1-hexadecyl-3-methylimidazolium chloride ([C₁₆mim]Cl). The structure and morphology of the sample have been investigated by X-ray diffraction (XRD), X-ray photoelectron spectra (XPS), transmission electron microscopy (TEM), and high-resolution TEM (HRTEM), and the results indicate that the as-synthesized inverse spinel Fe₃O₄ nanoparticles have an average diameter of about 10 nm and exhibit relatively good dispersity. More importantly, it is found that [C₁₆mim]Cl acts as stabilizer for the Fe₃O₄ nanoparticles by adsorbing on the particles surfaces to prevent the agglomeration. In addition, the obtained superparamagnetic Fe₃O₄ nanoparticles have a saturation magnetization of 67.69 emu/g at 300 K.

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1. Introduction

Fe₃O₄ nanoparticles (NPs), as one kind of the most important magnetic materials, have been widely applied in biomedicine, magnetic storage devices, ferrofluids, and separation processes due to their excellent magnetic properties [1,2]. The magnetic property of Fe₃O₄ NPs strongly depends on their sizes, and Fe₃O₄ NPs with diameters between 3 and 15 nm possess superparamagnetic property [3]. Most recently, superparamagnetic magnetic materials have attracted much attention in the biomedical field such as drug delivery and magnetic resonance imaging (MRI) [4]. To date, several synthesis processes, including hydrothermal/solvothermal technique, sol–gel route, and co-precipitation method, have been developed to synthesize Fe₃O₄ NPs [2,5–7]. In the synthesis, Fe₃O₄ NPs tend to agglomerate for their large surface energy and magnetization, which would seriously affect the applications; therefore, several stabilizers such as surfactants and polymer matrixes have been used to improve the dispersity of Fe₃O₄ NPs [4,8,9]. However, the final products would possess relatively low saturation magnetizations for the nonmagnetic layers [10]. Thus, it is still a challenge to explore novel routes for the fabrication of Fe₃O₄ NPs with both good dispersity and excellent magnetic property.

Ionic liquids (ILs) are organic salts with low melting points and consist of predominantly ionic species [11]. Most recently, ILs

possessing 1-alkyl-3-methylimidazolium cation ([C_nmim]⁺) have gained much attention for their favorable physico-chemical properties, and they have served as reactants, solvents, and templates in the fabrication of nanomaterials [12–14]. ILs can be manufactured to be hydrophobic or hydrophilic by designing their structures; furthermore, the low surface tensions of ILs are beneficial to improve the stability of the as-formed nanomaterials. In these regards, some imidazolium-based ILs have been chosen as stabilizers to prepare dispersible nanomaterials [15–17]. For example, Dupont et al. have synthesized stable transition-metal NPs (2–3 nm in size) by reduction of transition-metal compounds dissolved in [Bmim]PF₆. Therefore, it is reasonable to conceive that ILs with special cations and anions can be employed as stabilizers to improve the monodispersity of Fe₃O₄ NPs.

Herein, we have developed a novel and effective hydrothermal method for the fabrication of superparamagnetic Fe₃O₄ NPs. IL 1-hexadecyl-3-methylimidazolium chloride ([C₁₆mim]Cl) have been used as stabilizer in the synthetic system. The effect of [C₁₆mim]Cl on the morphology of product have been investigated in details.

2. Material and methods

2.1. Materials

All chemicals were analytical-grade and used without further purification. [C₁₆mim]Cl was prepared according to the literature procedures [18].

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2.2. Synthesis of Fe_3O_4 NPs

In a typical synthesis, $\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$ (0.15 mmol) and $\text{C}_6\text{H}_5\text{Na}_3\text{O}_7 \cdot 2\text{H}_2\text{O}$ (0.30 mmol) were dissolved into deionized water (7.5 mL) with magnetic stirring until a clear solution was obtained. Then, $[\text{C}_{16}\text{mim}]\text{Cl}$ (0.4 g) and $\text{N}_2\text{H}_4 \cdot \text{H}_2\text{O}$ (2.5 mL) were added to the solution. After it was stirred for 5 min, the solution was transferred and sealed in a teflon-lined stainless steel autoclave with a capacity of about 20 mL. The autoclave was sealed and maintained at 150°C for 12 h. The products were separated by centrifugation, washed by deionized water and anhydrous ethanol, and then dried in a vacuum at 60°C for 5 h. Finally black powder was obtained.

2.3. Instruments and characterizations

The phase compositions of the product were characterized by X-ray diffraction (XRD) on a Rigaku D/max 2500V/PC X-ray diffractometer ($\text{Cu K}\alpha$, $\lambda = 1.54056 \text{ \AA}$) with a 2θ range between 20° and 80° . The generator voltage was 40 kV, the current was 40 mA and the step velocity was $8^\circ/\text{min}$. The X-ray photoelectron spectra (XPS) data was characterized using PHI-1600 ESCA X-ray photoelectron spectroscopy with $\text{Mg K}\alpha$ as radiation. The sizes and structures of the products were performed by a Hitachi H-7650 transmission electron microscopy, which were operated at an accelerating voltage of 100 kV and 200 kV, respectively. The magnetic studies were performed on a Quantum Design MPMS-XL-7 SQUID susceptometer and a vibrating sample magnetometer (VSM, LDJ-9600, USA).

3. Results and discussions

3.1. Characterization results of the Fe_3O_4 NPs

Fig. 1a depicts the XRD pattern of Fe_3O_4 NPs. All of the diffraction peaks are indexed as face-centered cubic Fe_3O_4 with the lattice parameters of $a = 8.381 \text{ \AA}$, which matches well with the reported values (JCPDS No. 75-0033). No peaks corresponding to impurities are detected, showing the high purity of the sample. The mean crystallite size of Fe_3O_4 NPs is calculated to be about 10.5 nm based on Scherrer formula. Fe_3O_4 has an inverse spinel structure similar to that of $\gamma\text{-Fe}_2\text{O}_3$; therefore, XRD result cannot provide enough evidence to confirm the formation of Fe_3O_4 [19]. Thus, we employ XPS, which is sensitive to the iron valence state, to distinguish these two different phases [20]. As shown in the XPS spectra of Fe2p in Fe_3O_4 (Fig. 1b), the peaks at 724.1 and 710.5 eV,

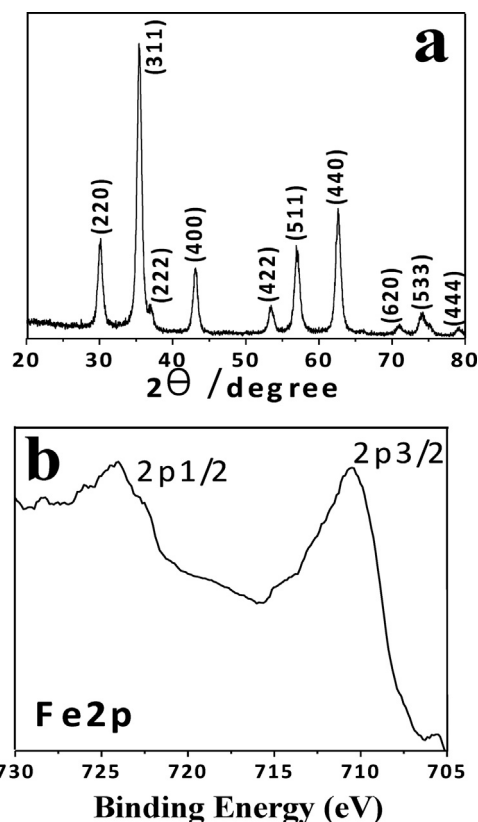


Fig. 1. (a) XRD pattern of the obtained Fe_3O_4 NPs; (b) Fe2p peaks in the XPS spectrum of the Fe_3O_4 NPs.

assigned to Fe2p1/2 and Fe2p3/2, are consistent with the values reported for Fe_3O_4 in the literature; furthermore, no satellite peak around 719.0 eV is identified, indicating the absence of $\gamma\text{-Fe}_2\text{O}_3$ in the sample [21].

TEM image (Fig. 2a) gives a general view of the as-synthesized Fe_3O_4 NPs. It can be clearly seen that the sample is quasi-round NPs and has a mean diameter of about 10 nm, consisting with the XRD result. In the HRTEM image taken at a typical particle (Fig. 2b), the Fe_3O_4 nanoparticle has a d -spacing of about 0.255 nm, corresponding to the lattice spacing of the (311) plane in cubic Fe_3O_4 ; moreover, its corresponding fast Fourier transform (FFT) pattern

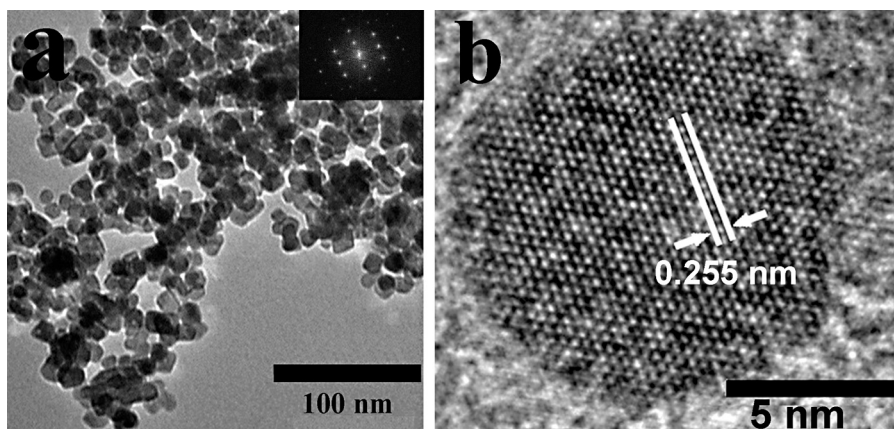


Fig. 2. (a) TEM image of the as-synthesized Fe_3O_4 NPs; (b) HRTEM image of a typical Fe_3O_4 NPs with the electron beam incident along the $[001]$ direction, and the inset of (a) is its FFT pattern.

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