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Synthesis of homogeneous $(Na_{1-x}K_x)NbO_3$ nanorods using hydrothermal and post-heat treatment processes

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HIGHLIGHTS

- NKN nanorods have been produced using hydrothermal method for the first time.
- ► The growth direction of the NKN nanorod is [001].
- ► (Na_{8x}K_{8-8x})Nb₆O₁₉ nH₂O produced at 160 °C and transformed into NKN nanorods at 500 °C.
- ► The NKN nanorods can be used for the piezoelectric energy harvester.

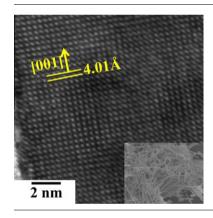
A R T I C L E I N F O

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G R A P H I C A L A B S T R A C T



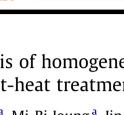
ABSTRACT

To synthesize $(Na_{1-x}K_x)NbO_3$ nanorods, the Nb₂O₅-added (1 - y)NaOH-yKOH specimens with $0.0 \le y \le 0.9$ were heated at 160 °C and subsequently annealed at 500 °C. Homogeneous Na-rich $(Na_{1-x}K_x)NbO_3$ nanorods were only synthesized for the specimen with y = 0.15. However, NaNbO₃ nanorods were formed for specimens with y < 0.15, and K-rich $(Na_{1-x}K_x)NbO_3$ nanoplates were obtained for specimens with y > 0.2. The $(Na_{8x}K_{8-8x})Nb_6O_{19}\cdot nH_2O$ transient phase was formed in the specimen with y = 0.15 heated at 160 °C for 8.0–12.0 h, and this phase transformed into the Na-rich $(Na_{1-x}K_x)NbO_3$ nanorods after annealing at 500 °C. Therefore, the formation of a homogeneous $(Na_{8x}K_{8-8x})Nb_6O_{19}\cdot nH_2O$ phase at a low temperature is very important for the synthesis of the $(Na_{1-x}K_x)NbO_3$ nanorods. The $(Na_{8x}K_{8-8x})Nb_6O_{19}\cdot nH_2O$ phase was considerably influenced by the heating temperature, and the processing time. The Na-rich $(Na_{1-x}K_x)NbO_3$ nanorods have rectangular shape with various sizes, and the growth direction of these nanorods is [001].

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1. Introduction

Wasted mechanical energy in the environment can be converted into electrical energy using piezoelectric nanogenerators. Recently, there has been increasing interest in piezoelectric nanogenerators, because they can be used to power wireless sensors and implantable biosensors without the need for a battery [1–6]. The output energy density (*u*) produced by a piezoelectric nanogenerator is given as $u = 1/2(d_{ij}^2/e_{ij})(F/A)^2$, where d_{ij} is the piezoelectric strain constant, ε_{ij} is the dielectric constant, *F* is the applied force, and *A* is the area [7]. Therefore, piezoelectric materials for the nanogenerator should have a large d_{ij} value to produce a large output energy. ZnO nanowires have been investigated for applications as piezoelectric nanogenerators, but the d_{ij} value of ZnO nanowires is



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relatively small, 14.3–26.7 pm/V [8]. Pb($Zr_{1-x}Ti_x$)O₃ (PZT) nanowires have been also investigated because of their large d_{ij} value of 400 pm/V [9], but they contain more than 60.0 wt% of PbO, which leads to environmental problems [10–12]. Therefore, it is necessary to develop Pb-free piezoelectric nanowires that have good piezoelectric properties.

Toward this goal, Pb-free KNbO₃ nanowires have been investigated. They have been synthesized using the hydrothermal method [13–19] and also via templated crystallization of a precursor sol [20,21]. However, the processing time for KNbO₃ nanowires synthesized using the hydrothermal method is very long (6-7 days), and the templated crystallization of a precursor sol uses two-step processes that involve high processing temperatures. Many investigations have also been conducted on NaNbO3 nanowires because of their piezoelectric properties and photocatalytic properties [22-24]. However, the NaNbO₃ oxides are known to be antiferroelectric [25]. The phase diagram of NaNbO₃ and KNbO₃ phases indicates that they form a complete solid solution of $(Na_{1-x}K_x)NbO_3$ for all composition ranges [26]. In particular, the (Na_{0.5}K_{0.5})NbO₃ oxides have been extensively investigated because of their large d_{33} value of approximately 120 pm/V and high Curie temperatures [10,27]. Moreover, the (Na_{0.5}K_{0.5})NbO₃-Li(Nb, Ta, Sb)O₃ oxides produced using a reactive-templated grain growth method have been reported to exhibit a large d_{33} value of 300 pm/V that is similar to that of the PZT-based piezoelectric oxides [11,28,29]. Therefore, $(Na_{1-x}K_x)NbO_3$ nanowires are expected to have a large d_{33} value and thus may be promising piezoelectric nanogenerators. (Na_{0.5}K_{0.5})NbO₃ nanopowders have been investigated to a considerable extent for applications in bulk piezoelectric devices [30]. However, $(Na_{1-x}K_x)NbO_3$ -based nanowires and nanorods have not been investigated. Therefore, in this work, the synthesis of $(Na_{1-x}K_x)NbO_3$ nanorods has been investigated systematically, and $(Na_{1-x}K_x)NbO_3$ nanorods have been produced for the first time. Furthermore, the variations in the morphology and compositions of the phases with respect to the process conditions have been also studied.

2. Experimental procedures

Analytical-grade solid potassium hydroxide (KOH, >99%, High Purity Chemicals, Osaka, Japan), sodium hydroxide (NaOH, >99%, High Purity Chemicals, Osaka, Japan), and niobium oxide (Nb₂O₅, 99.9%, High Purity Chemicals, Osaka, Japan) were used as the starting materials. A typical synthesis of $(Na_{1-x}K_x)NbO_3$ nanorods is as follows. Nb₂O₅ (0.002 mol) was added to a solution of (1 - y)NaOH–yKOH (14.0 mol) with $0.0 \le y \le 0.9$ in distilled water (40 mL). Then, the mixture was stirred for 2 h, poured into a Teflon vessel, and placed in an autoclave. The autoclave was heated to a temperature range of 140–180 °C for 8–20 h, and the sample was subsequently annealed at 500 °C for 6 h. The product was filtered, washed with distilled water, and dried at 80 °C for 12 h. The structural properties of the specimens were examined using X-ray diffraction (XRD; Rigaku D/max-RC, Tokyo, Japan). The shape and growth direction of the $(Na_{1-x}K_x)NbO_3$ nanorods were determined using scanning electron microscopy (SEM; Hitachi S-4300, Osaka, Japan) and field-emission transmission electron microscopy (FE-TEM; Tecnai F20, FEI, Netherlands).

3. Results and discussion

Fig. 1a–f shows the XRD patterns of the Nb₂O₅-added (1 - y) NaOH–yKOH specimens with $0.0 \le y \le 0.9$ heated at 160 °C for 8 h. For the specimen with y = 0.0, peaks for the Na₂Nb₂O₆·H₂O (SOMS) [31], Na₈Nb₆O₁₉·13H₂O (JCPDS No. 14-0370), and NaNbO₃ (JCPDS No. 82-0606) phases, indicated by stars, filled circles, and filled squares, respectively, were observed, as shown in Fig. 1a.

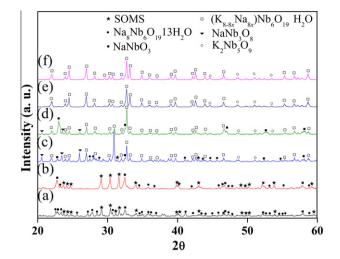


Fig. 1. XRD patterns of the Nb₂O₅-added (1 - y)NaOH–yKOH specimens with $0.0 \le y \le 0.9$ heated at 160 °C for 8 h: (a) y = 0.0, (b) y = 0.1, (c) y = 0.15, (d) y = 0.2, (e) y = 0.35, and (f) y = 0.9.

According to previous work, Na₈Nb₆O₁₉·13H₂O and SOMS are transition phases that form during the formation of the NaNbO₃ phase. Moreover, it has been reported that the Na₈Nb₆O₁₉·13H₂O phase forms first and then transforms into SOMS and NaNbO3 phases. Finally, the SOMS phase is converted into the NaNbO₃ phase at elevated temperatures [32]. Three phases were also observed for the specimen with v = 0.1, but the amounts of the SOMS and NaNbO₂ phases increased while that of the Na₈Nb₆O₁₉·13H₂O phase decreased, as shown in Fig. 1b. For the specimen with y = 0.15, the amounts of the NaNbO₃, SOMS, and Na₈Nb₆O₁₉·13H₂O phases considerably decreased. However, new peaks for the (Na₄K₄)Nb₆O₁₉. 9H₂O (JCPDS No. 14-0360) phase, indicated by open squares, appeared, and it became a major phase in this specimen. In addition, the NaNb₃O₈ (JCPDS No. 20-1143) and K₂Nb₅O₉ (JCPDS No. 47-0081) phases were also newly produced in this specimen. Previously, peaks for the $(Na_4K_4)Nb_6O_{19}\cdot 9H_2O$ phase, which formed during the formation of $(Na_{1-x}K_x)NbO_3$ powders, were indexed as the $(Na_{8x}K_{8-8x})Nb_6O_{19} \cdot nH_2O$ [33] phase based on the molar concentrations of K⁺ and Na⁺. Furthermore, this phase was reported to be a transient phase that transforms into $(Na_{1-x}K_x)NbO_3$ nanopowders at 220-240 °C when sufficient processing time is allowed [33,34]. For the specimen with y = 0.2, the SOMS and Na₈Nb₆O₁₉·13H₂O phases disappeared, but the amount of NaNbO₃ phase considerably increased, as shown in Fig. 1d. When the y value exceeded 0.2, the Na-related phases, such as the NaNbO3 and NaNb₃O₈ phases, disappeared, but the $(Na_{8x}K_{8-8x})Nb_6O_{19} \cdot nH_2O$ phase and a small amount of K₂Nb₅O₉ phase were found, as shown in Fig. 1e and f.

Fig. 2a-f shows the XRD patterns of the Nb₂O₅-added (1 - y)NaOH-yKOH specimens with $0.0 \le y \le 0.9$ specimens annealed at 500 °C for 6 h after being heated at 160 °C for 8 h. A homogeneous NaNbO3 phase with an orthorhombic structure was observed from the specimens with $y \leq 0.1$, as shown in Fig. 2a and b. It is believed that the SOMS and Na₈Nb₆O₁₉·13H₂O transient phases transformed into the orthorhombic NaNbO₃ phase during the annealing at 500 °C, and this transformation has been generally observed during the formation of NaNbO₃ nanowires [32]. For the specimen with y = 0.15, the homogeneous Na-rich $(Na_{1-x}K_x)NbO_3$ phase, which is discussed later, was observed, as shown in Fig. 2c. However, both the K-rich $(Na_{1-x}K_x)NbO_3$ and NaNbO₃ phases, which are shown later, were formed for the specimen with y = 0.2 (see Fig. 2d). Finally, when y exceeded 0.2, the homogeneous K-rich $(Na_{1-x}K_x)NbO_3$ phase developed, as shown in Fig. 2e and f. Therefore, the NaNbO₃ phase formed in the speciDownload English Version:

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