

Synthesis and electrical properties of CeO₂-added (Na_{0.5}Bi_{0.5})_{0.94}Ba_{0.06}TiO₃ ceramics

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Abstract

(Na_{0.5}Bi_{0.5})_{0.94}Ba_{0.06}TiO₃-xwt.%CeO₂ ($x = 0, 0.2, 0.4, 0.6, 0.8, 1.0$) powders were synthesized by a citrate method, and the piezoelectric and ferroelectric properties of the ceramics were investigated. The results indicate that the citrate method is an advantageous route in producing (Na_{0.5}Bi_{0.5})_{0.94}Ba_{0.06}TiO₃-xwt.%CeO₂ ($x = 0, 0.2, 0.4, 0.6, 0.8, 1.0$) ceramics. Homogeneous and fine powders with a pure perovskite structure were derived from precursor solutions with a mole ration of citric acid to total metal cation content (C/M) of 1.3 by calcining at 650 °C for 1 h. The sample containing 0.6 wt.%CeO₂ made by the citrate method shows superior piezoelectric properties and a strong ferroelectricity: high piezoelectric constant ($d_{33} = 157$ pC/N), high electromechanical coupling factor ($k_p = 31.2\%$), enhanced dielectric constant ($\epsilon_{33}^T/\epsilon_0 = 819$) and relatively low dissipation factor ($\tan \delta = 2.72\%$) at 1 kHz, corresponding to a relatively large remanent polarization of $P_r = 38.0$ $\mu\text{C}/\text{cm}^2$ and a relatively low coercive field of $E_c = 37.4$ kV/cm.

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1. Introduction

In recent years, lead-free piezoelectric ceramics has attracted considerable attention as an important piezoelectric material because of its outstanding advantages in no lead pollution. Sodium bismuth titanate, (Na_{0.5}Bi_{0.5})TiO₃ (NBT), is a kind of perovskite-type ferroelectric with complex A-site cations, showing a relatively large remanent polarization ($P_r = 38$ $\mu\text{C}/\text{cm}^2$) at room temperature and a relatively high Curie temperature ($T = 320$ °C) [1]. Due to its strong ferroelectricity at room temperature, NBT has been considered to be a promising candidate material for lead-free piezoelectric ceramics. However, it is difficult to pole due to the high coercive field ($E_c = 73$ kV/cm) and the high conductivity of NBT, making it unsuccessful in obtaining the desired piezoelectric properties. To solve these problems, various NBT-based solid solutions have been developed [2–9]. Among these solid solutions, the (Na_{0.5}Bi_{0.5})_{1-x}Ba_xTiO₃ material has attracted considerable attention owing to an existence of a rhombohedral (F_α)-tetragonal (F_β) morphotropic phase boundary (MPB). Takenaka et al. reported that the (Na_{0.5}Bi_{0.5})_{0.94}TiO₃-

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0.06BaTiO₃ (NBT-BT6) composition near the MPB have a relatively low dielectric constant and a relatively high electromechanical coupling factor [2]. The further enhancement on the piezoelectric and ferroelectric properties of NBT-BT6 ceramics is reported by Wang and Li [10,11]. However, there have been limited reports about rare-earth element doping to NBT-BT6 composition for the further enhancement on the piezoelectric and ferroelectric properties. It has been proved that the synthesis process has an obvious influence on NBT and NBT-based solid solutions in terms of chemical stoichiometry, lattice distortion and electrical properties [12–14]. However, there have been limited investigations into this subject. Therefore, it is necessary to investigate the synthesis and preparation of NBT-based ceramics. In view of this, we reported the synthesis of (Na_{0.5}Bi_{0.5})_{0.94}Ba_{0.06}TiO₃-xwt.%CeO₂ ($x = 0, 0.2, 0.4, 0.6, 0.8, 1.0$) (NBT-BT6- x %CeO₂) powders by a citrate method. In addition, CeO₂ was added into NBT-BT6 ceramics as an additive based on the formula NBT-BT6- x %CeO₂ and its effects on the piezoelectric and ferroelectric properties of the ceramics were investigated.

2. Experimental procedures

The citrate method was used to prepare NBT-BT6- x %CeO₂ powders with the nominal composition of (Na_{0.5}Bi_{0.5})_{0.94}Ba_{0.06}TiO₃-xwt.%CeO₂ ($x = 0, 0.2, 0.4, 0.6, 0.8, 1.0$). Reagent grade NaNO₃, Bi(NO₃)₃·5H₂O, Ba(NO₃)₂, Ce(NO₃)₃·6H₂O, tetrabutyl titanate and citric acid were used as starting materials. Citric acid was first dissolved into deionized water in a beaker. After adjusting the pH value of the solution to 7–9 by dripping an appropriate amount of ammonia solution, tetrabutyl titanate was added under a stirring at 60 °C. After stirring at 60 °C for 1 h, a yellowish two-layer liquid was obtained, comprising a transparent aqueous solution at the lower layer and an oil-like liquid at the upper layer. The aqueous solution was separated from the mixing liquid. Various nitrates were then added into the solution one after another, respectively, followed by a stirring at 80 °C for 1 h to generate a precursor solution with a pH value of about 6. The precursor solution was dehydrated in an oven at 100 °C to form a sol. Subsequent heating at a higher temperature of 160 °C yielded a gel. The gel was pulverized and then calcined at 550–700 °C for 1 h in air. The calcined powders were granulated with PVA as a binder. The granulated powders were pressed into discs in diameter of 19 mm and thickness of 1 mm. The compacted discs were sintered at 1150 °C for 2 h in air.

Thermogravimetry (TG) analysis and differential scanning calorimetry (DSC) analysis of the gel were performed using a netzsch STA449C simultaneous thermal analyzer in air. The crystal structure of calcined powders and ceramic specimens was examined by a Rigaku D/MAX-RB X-ray diffractometer using Cu K α radiation. A scanning electron microscope (SEM, Jeol JSM-5610LV) was used to investigate the morphology of calined powders. Silver paste was fired on both faces of the discs at 650 °C as electrodes. The specimens for measurement of piezoelectric properties were poled in silicon oil at 60 °C under 3 kV/mm for 15 min. After poling, the piezoelectric constant d_{33} was measured using a quasistatic d_{33} meter based on the Berlincourt method. The resonance measurements were performed using a HP4294 impedance analyzer. The electromechanical coupling factor k_p was calculated from the resonance and the anti-resonance frequencies according to Onoe's formulas [15]. Dielectric properties were determined using a HP4192A impedance analyzer at 1 kHz. The coercive field E_c and remanent polarization P_r were determined from P – E hysteresis loops obtained by a Radiant precision workstation at room temperature. The measurement frequency of P – E loops is 20 Hz.

3. Results and discussion

During the synthesis process, it was found that the mole ratio of citric acid to the total metal cation content (C/M) is a main contributing factor to the formation of the sol and gel. It was ascertained that a C/M in the range of 1.1–1.5 produced homogeneous, transparent sol and gel. Fig. 1 shows the TG-DSC curves of NBT-BT6-0.6%CeO₂ gel derived from the precursor solution with C/M = 1.3. The faint endothermic peak at 84.4 °C was caused by the evaporation of residual citric acid and water. The successive endothermic peak at 180.3 °C is attributed to the melting of the citric acid. The exothermic peak at 280.6 °C is assigned to the thermal decomposition of excessive citric acid. The exothermic peak at 477.5 °C is ascribed to the thermal decomposition of the citrate complex. The front four peaks correspond to a total weight loss of 57.14%. The strong exothermic peak at 565.8 °C is attributed to the combustion of remaining organic components. No further DSC peak or weight loss can be seen, indicating the completion of the thermal decomposition of the gel before 600 °C. Fig. 2 shows the X-ray diffraction (XRD) patterns

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