

# A novel nitridogallate fluoride $\text{LiBa}_5\text{GaN}_3\text{F}_5$ – Synthesis, crystal structure, and band gap determination

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## ARTICLE INFO

### Article history:

Received 18 March 2010

Received in revised form

17 May 2010

Accepted 18 May 2010

Available online 26 May 2010

### Keywords:

Nitridogallate

Crystal structure

Electronic structure

Optical measurements

Sodium flux

## ABSTRACT

$\text{LiBa}_5\text{GaN}_3\text{F}_5$  was obtained as red crystals by reaction of Ba, Ga,  $\text{NaN}_3$  and  $\text{EuF}_3$  in a Na/Li flux at 760 °C in weld-shut tantalum crucibles. The crystal structure (*Pnma* (no. 62),  $a = 15.456(3)$ ,  $b = 5.707(1)$ ,  $c = 12.259(3)$  Å,  $Z = 4$ ) was solved on the basis of single-crystal X-ray diffraction data. In the solid there are trigonal planar  $[\text{GaN}_3]^{6-}$  ions and zigzag chains of vertex sharing  $\text{LiF}_6$  octahedrons surrounded by  $\text{Ba}^{2+}$  ions. Optical measurements and calculations of the electronic structure revealed a band gap of  $\leq 1.9$  eV. According to the calculations, the observed transition occurs from a nitrogen state into a hybrid Ba/N state.

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## 1. Introduction

Ternary nitrides of Ga in combination with alkaline earth metals (e.g., Sr, Ba) are known since a couple of years [1–4], but only a small number of quaternary examples have been reported recently [5–7]. Most of these compounds have been synthesized in sodium melts which due to addition of alkaline earth metals exhibit an increased solubility of nitrogen [8]. Syntheses were performed in closed niobium crucibles, utilizing additional azides as nitrogen source.  $\text{Sr}_3\text{GaN}_3$ ,  $\text{Sr}_6\text{Ga}_5$  [2],  $\text{Sr}_4\text{GaN}_3\text{O}$  and  $\text{Sr}_4\text{GaN}_3(\text{CN}_2)$  [5] contain non condensed (“isolated”)  $[\text{GaN}_3]^{6-}$  while  $\text{Ba}_3\text{Ga}_2\text{N}_4$  and  $\text{Sr}_3\text{Ga}_2\text{N}_4$  are made up of trans edge sharing  $\text{GaN}_4$  tetrahedrons building infinite chains [1,4]. Two- or three-dimensional networks of vertex sharing  $\text{GaN}_4$  tetrahedrons have been found in  $\text{Ca}_3\text{Ga}_2\text{N}_4$ ,  $\text{Sr}_3\text{Ga}_3\text{N}_5$  [4] and  $\text{LiSrGa}_2\text{N}_2$  [6], respectively. Thus, structural motifs similar to nitridosilicates (e.g.,  $\text{BaSi}_7\text{N}_{10}$ ;  $\text{Eu}_2\text{SiN}_3$ ) [9,10] occur in the nitridogallates mentioned above. Recently, we have reported about synthetic approaches to control the dimensionality of nitridosilicates employing lithium melts [11]. In comparison with nitridosilicates, the field of nitridogallates is more unexplored but utilization of the lithium

flux technique may lead to a larger structural variety analogously to our experience with nitridosilicates.

## 2. Experimental

The synthesis of  $\text{LiBa}_5\text{GaN}_3\text{F}_5$  was carried out in Ta crucibles (30 mm length, 9.5 mm diameter, 0.5 mm wall thickness). Under argon atmosphere (glove box Unilab, MBraun), 0.35 mmol (22.8 mg)  $\text{NaN}_3$  (Acros, 99%), 0.138 mmol (9.6 mg) Ga (AluSuisse, 99.999%), 0.549 mmol (75.4 mg) Ba (Sigma Aldrich, 99.99%) and 0.027 mmol (5.7 mg)  $\text{EuF}_3$  (Sigma Aldrich, 99.99%) were mixed and filled into the Ta crucible. For the flux 2.174 mmol (50.0 mg) Na (Sigma Aldrich, 99.95%) and 0.145 mmol (1.0 mg) Li (Sigma Aldrich, 99.9%) were added. The Ta crucible was sealed under argon by arc welding. To protect the Ta crucible from oxidation, it was placed into a silica tube under argon atmosphere. In a tube furnace the crucible was heated to 760 °C with a rate of 50 °C h<sup>-1</sup>. The temperature was maintained for 48 h and then lowered with 3.7 °C h<sup>-1</sup> to 200 °C. Once the temperature reached 200 °C, the furnace was turned off and cooled down to room temperature. The Ta crucible was opened and Na was separated from the reaction products by evaporation at 320 °C under vacuum (0.1 Pa) for 18 h. From the inhomogeneous gray product, red needle-shaped single crystals (200–600 μm) were isolated (cf. Fig. 1), enclosed in glass capillaries and sealed under argon atmosphere. X-ray diffraction data were collected at room temperature with a STOE IPDS I

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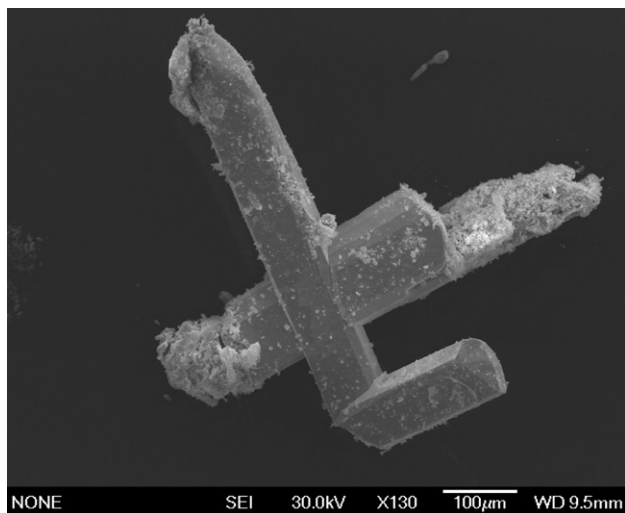


Fig. 1. SEM micrograph of  $\text{LiBa}_5\text{GaN}_3\text{F}_5$  crystals.

diffractometer. A numerical absorption correction using the programs XRED [12] and XSHAPE [13] was applied. The crystal structure was solved by using direct methods with SHELXS [14]. The refinement of the structure was carried out by the method of least-squares using SHELXL [14]. The chemical composition was confirmed by energy dispersive X-ray spectroscopy (EDX) using a JSM-6500F scanning microscope (Jeol) provided with a Si/Li EDX detector (Oxford Instruments, model 7418). Optical spectra of  $\text{LiBa}_5\text{GaN}_3\text{F}_5$  were measured with a modified microcrystal spectrophotometer CARY 17 (Spectra Services, ANU Canberra, Australia) [15–17]. Calculations of the band gap were carried out with the program package WIEN2K [18] utilizing the structural data from the single-crystal structure refinement.

Further details of the crystal structure investigations can be obtained from the Fachinformationzentrum Karlsruhe, 76344 Eggenstein–Leopoldshafen, Germany (Fax: +49-7247-808-666; E-Mail: [crysdta@fiz-karlsruhe.de](mailto:crysdta@fiz-karlsruhe.de)) on quoting the depository number CSD-421592.

### 3. Results and discussion

#### 3.1. Crystal structure

The crystal structure was solved and refined in orthorhombic space group  $Pnma$  (no. 62) with  $a = 15.456(3)$ ,  $b = 5.707(1)$  and  $c = 12.259(3)$  Å. The crystallographic data of  $\text{LiBa}_5\text{GaN}_3\text{F}_5$  is summarized in Table 1, the atomic coordinates and the isotropic displacement parameters are listed in Table 2.

In the crystal  $\text{LiBa}_5\text{GaN}_3\text{F}_5$  zigzag chains of vertex sharing  $\text{LiF}_6$  octahedrons running along [010] (cf. Fig. 2). Perpendicular to these chains, Ba atoms are arranged in layers. Likewise perpendicular to [010] “isolated” trigonal planar  $[\text{GaN}_3]^{6-}$  ions are found. The Ga–N bond-lengths range from 1.90 to 1.95 Å (Fig. 3) and agree well with the sum of the ionic radii [19–21] as well as with typical Ga–N distances (e.g.,  $\text{Sr}_4\text{GaN}_3\text{O}$ , Ga–N: 1.88–1.92 Å) [5]. Similar  $[\text{GaN}_3]^{6-}$  ions have been found in  $\text{Sr}_3\text{GaN}_3$ ,  $\text{Sr}_6\text{GaN}_5$  [2],  $\text{Sr}_4\text{GaN}_3\text{O}$  and  $\text{Sr}_4\text{GaN}_3(\text{CN}_2)$  [5]. The coordination sphere of the trigonal planar  $[\text{GaN}_3]^{6-}$  ions can be described as three-capped trigonal prisms of  $\text{Ba}^{2+}$  atoms. In  $\text{Sr}_3\text{GaN}_3$  [2] similar trigonal prism of  $\text{Sr}^{2+}$  atoms have been observed. The  $\text{N}^{3-}$  atoms are likewise surrounded in distorted octahedrons of five  $\text{Ba}^{2+}$  and one  $\text{Ga}^{3+}$  atom.

Table 1  
Crystallographic data of  $\text{LiBa}_5\text{GaN}_3\text{F}_5$ .

Formula	$\text{LiBa}_5\text{GaN}_3\text{F}_5$
Crystal system	orthorhombic
Space group	$Pnma$ (no. 62)
Lattice parameters (Å)	$a = 15.456(3)$ , $b = 5.707(1)$ , $c = 12.259(3)$
Cell volume (Å <sup>3</sup> )	1081.3(4)
Formula units per unit cell	4
Density (g cm <sup>-3</sup> )	5.531
$\mu$ (mm <sup>-1</sup> )	20.41
T (K)	293(2)
F(000)	1520
Profile range	$5.8 \leq 2\theta \leq 63.2$
Index ranges	$-20 \leq h \leq 18$ $-6 \leq k \leq 6$ $-15 \leq l \leq 15$
Independent reflections	1286 [ $R(\text{int}) = 0.072$ ]
Refined parameters	89
Goodness of fit	1.057
$R_1$ (all data); $R_1 (F^2 > 2\sigma(F^2))$	0.0230, 0.0212
$wR_2$ (all data); $wR_2 (F^2 > 2\sigma(F^2))$	0.0512, 0.0505
$\Delta\rho_{\text{max}}$ , $\Delta\rho_{\text{min}}$ (e Å <sup>-3</sup> )	1.56, -2.13

$\text{Li}^+$  does not directly coordinate to the  $[\text{GaN}_3]^{6-}$  ions but is surrounded by six  $\text{F}^-$  in a distorted octahedron. The Li–F distances are ranging between 1.97 and 1.98 Å for the equatorial F2 and F1 and 2.33–2.39 Å for the axial F3 and F4 (cf. Fig. 3). The latter ones exceed significantly the sum of the ionic radii (2.06 Å) [19] while the equatorial Li–F distances are slightly shorter than reported distances in other  $\text{LiF}_6$  octahedrons ( $\text{K}_2\text{LiAlF}_6$ , Li–F: 2.109 Å) [22,23]. In the second coordination sphere, the octahedron  $\text{LiF}_6$  is coordinated by eight  $\text{Ba}^{2+}$  in a cubic way, comparable with the cubic  $\text{Ca}^{2+}$ -coordination of  $\text{Ti}^{4+}$  in  $\text{CaTiO}_3$ . The coordination of  $\text{F}^-$  occurs in distorted octahedrons of five  $\text{Ba}^{2+}$  and one  $\text{Li}^+$ . Only F2 is coordinated by two  $\text{Li}^+$  and four  $\text{Ba}^{2+}$ . The atomic distances Ba–F and Li–F are mentioned above. The thermal displacement parameter  $U_{\text{iso}}^*/U_{\text{eq}}$  of F4 is considerably higher (cf. Table 3) in comparison with the other values for fluorine atoms. One reason for this observation may be the fact, that – considering the octahedron  $\text{LiF}_6$  – the F4 atom is a “free” one and does not connect to the next octahedrons. Additionally, the  $U_{11}$  value for F4 is almost three times higher than the values for the other  $\text{F}^-$ . The short distance between Ba4–F4 (2.83 Å) may be responsible for this observation. The F3 atom has an even shorter distance to Ba4 (2.76 Å) and is also a non-bridging one. Here we can also observe a little higher  $U_{\text{iso}}^*/U_{\text{eq}}$  value.

Table 2

Atomic coordinates and isotropic displacement parameters (Å<sup>2</sup>) of  $\text{LiBa}_5\text{GaN}_3\text{F}_5$ , standard deviations in parentheses.

Atom	Wyckoff position	x	y	z	$U_{\text{iso}}^*/U_{\text{eq}}$
Ba1	4c	0.43534 (2)	¼	0.68180 (2)	0.0092 (1)
Ba2	4c	0.51012 (2)	¾	0.87769 (3)	0.0090 (1)
Ba3	4c	0.34365 (2)	¾	0.47028 (3)	0.0089 (1)
Ba4	4c	0.24513 (2)	¾	0.77891 (3)	0.0100 (1)
Ba5	4c	0.16417 (2)	¼	0.63252 (2)	0.0109 (1)
Ga1	4c	0.35910 (4)	¼	0.92467 (4)	0.0076 (2)
F1	8d	0.4111 (2)	0.0121 (5)	0.3013 (2)	0.0151 (6)
F2	4a	½	0	½	0.0131 (7)
F3	4c	0.4075 (3)	¾	0.6854 (3)	0.0171 (8)
F4	4c	0.1779 (4)	¾	0.9936 (3)	0.027 (1)
N1	4c	0.4799 (4)	¼	0.8980 (4)	0.014 (2)
N2	4c	0.2813 (4)	¼	0.8020 (4)	0.013 (2)
N3	4c	0.3152 (4)	¼	1.0734 (5)	0.012 (2)
Li1	4c	0.459 (2)	¼	0.4020 (8)	0.026 (3)

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