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J Materiomics 2 (2016) 273-279

www.journals.elsevier.com/journal-of-materiomics/

# Synergistic effects of Lanthanum substitution on enhancing the thermoelectric properties of $\beta$ -Zn<sub>4</sub>Sb<sub>3</sub>

Tianhua Zou<sup>a</sup>, Wenjie Xie<sup>a,\*</sup>, Xiaoying Qin<sup>b</sup>, Menghan Zhou<sup>c</sup>, Marc Widenmeyer<sup>a</sup>, Jiangfeng Xu<sup>a</sup>, Jian He<sup>c</sup>, Anke Weidenkaff<sup>a,\*\*</sup>

<sup>a</sup> Institute for Materials Science, University of Stuttgart, 70569 Stuttgart, Germany

<sup>b</sup> Key Laboratory of Materials Physics, Institute of Solid State Physics, Chinese Academy of Sciences, 230031 Hefei, China <sup>c</sup> Department of Physics and Astronomy, Clemson University, Clemson, SC 29634-0978, USA

> Received 22 March 2016; revised 28 May 2016; accepted 2 June 2016 Available online 21 June 2016

### Abstract

A core challenge of thermoelectric research is decoupling the otherwise inversely interdependent properties electrical conductivity, thermopower, and lattice thermal conductivity offering synergistic effects. Herein, we present a systematic study in which we have substituted Lanthanum into the Zn-site in  $\beta$ -Zn<sub>4</sub>Sb<sub>3</sub> compound. We observed that La-substitution not only simultaneously enhances electrical conductivity and thermopower over a wide temperature range but also substantially reduces the lattice thermal conductivity. These synergistic effects of La-substitution are discussed in terms of the substitution-induced variation in carrier concentration and effective mass as well as stronger phonon scattering by point defects. As a result, a ~30% increase of the power factor and a ~23% reduction in the lattice thermal conductivity led to a state-of-the-art *ZT* ~1.3 at 648 K for the bulk sample of  $\beta$ -La<sub>0.01</sub>Zn<sub>3.99</sub>Sb<sub>3</sub>.

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Keywords: Thermoelectric; Band engineering; Rare earth; Phonon scattering

# 1. Introduction

In the wake of increasing energy demand and the environmental detriments of using fossil fuels, the requirement for alternative energy technologies has stimulated research of energy-related materials including thermoelectrics, the simplest technology for heat-to-electricity conversion. The conversion efficiency of a thermoelectric (TE) material is gauged by its dimensionless figure of merit,  $ZT = (\sigma S^2 / \kappa)T$ , where  $\sigma$ , S,  $\kappa$  are the electrical conductivity, thermopower, and total thermal conductivity, respectively.  $\kappa$  can be expressed as the sum of the carrier thermal conductivity  $\kappa_e$  and the lattice

Peer review under responsibility of The Chinese Ceramic Society.

thermal conductivity  $\kappa_L$ . A high *ZT* value requires a low  $\kappa_L$  and a high power factor  $PF = \sigma S^2$ . In recent decades, most advances in enhancing *ZT* values were made by reducing  $\kappa_L$  via nanostructuring [1,2], "phonon liquid" [3,4], all-scale hierarchical microstructures [5,6], anharmonicity [7], phonon localization due to random stacking [8], complex crystal structures [9], phase diagram approach [10] and alloying [11]. As  $\kappa_L$  of start-of-the-art TE materials approaches the "amorphous limit" [12,13], the room for further reduction of  $\kappa_L$ declines.

Now is the time to shift the focus of TE research to enhancing the *PF*. *PF* is a function of  $\sigma$  and *S*, both of which depend on the electron band structure and the scattering mechanisms. Attention should especially concentrate on *S*, because *PF* increases with the square of *S* but only proportional to  $\sigma$ . According to the Mott relation, *S* for a degenerate semiconductor can be expressed as:

http://dx.doi.org/10.1016/j.jmat.2016.06.002

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<sup>\*</sup> Corresponding author.

<sup>\*\*</sup> Corresponding author. E-mail addresses: xie@imw.uni-stuttgart.de (W. Xie), weidenkaff@imw. uni-stuttgart.de (A. Weidenkaff).

$$S = \frac{\pi^2 k_B^2 T}{3q} \left[ \frac{\partial \ln(\sigma(E))}{\partial E} \right]_{E=E_F}$$
  
=  $\frac{\pi^2 k_B^2 T}{3q} \left[ \frac{1}{p} \frac{\partial p(E)}{\partial E} + \frac{1}{\mu} \frac{\partial \mu(E)}{\partial E} \right]_{E=E_F}$  (1)

with the carrier mobility  $\mu(E) = q\tau/m^*$  in the Drude model, where q is the carrier charge, E the energy, p(E) and  $\mu(E)$  the energy dependent carrier concentration and mobility,  $k_B$  the Boltzmann constant,  $m^*$  the effective mass, and  $E_F$  the Fermi energy, respectively. In most cases where non-resonant scattering dominates, the relaxation time  $\tau$  follows a power law behavior  $\tau = \tau_0 E^{\lambda - 1/2}$ , where  $\lambda$  is the scattering parameter and  $\tau_0$  a constant. Eq. (1) indicates that S increases with increasing electron density of states (eDOS) near  $E_F$  [14]. However, due to the inverse inter-dependence of  $\sigma$  and S, improving S often impairs  $\sigma$ . Consequently, the simultaneous increase of  $\sigma$  and the S must result in synergistic effects. In this work, we show this by taking the example of La-substituted  $\beta$ -Zn<sub>4</sub>Sb<sub>3</sub>.

β-Zn<sub>4</sub>Sb<sub>3</sub> is a promising class of novel TE materials featuring a glasslike low  $κ_L$  and a relatively high *PF* in the temperature range of common applications [15]. Calculations predict that substitution with *f*-elements such as rare earth elements may introduce additional density of states near the Fermi level, thus, enhancing the thermopower. Indeed, substituting β-Zn<sub>4</sub>Sb<sub>3</sub> with Sm (6s<sup>2</sup>4f<sup>6</sup>), Pr (6s<sup>2</sup>4f<sup>3</sup>) and Gd (6s<sup>2</sup>4f<sup>6</sup>5d<sup>1</sup>) enhances the power factor [16–18]. This begs the question of the effect of substituting β-Zn<sub>4</sub>Sb<sub>3</sub> with La (6s<sup>2</sup>5d<sup>1</sup>), the only rare earth element without *f*-electrons. To date, studies on the effect of La-substitution on the TE properties of β-Zn<sub>4</sub>Sb<sub>3</sub> are scarce. The prime motivation of this work is, therefore, to reveal the role of *f*-electrons in rare earth element substituted β-Zn<sub>4</sub>Sb<sub>3</sub>.

We will demonstrate that La-substitution of  $\beta$ -Zn<sub>4</sub>Sb<sub>3</sub> has synergistic effects by simultaneously reducing the electrical resistivity due to a higher carrier concentration, enhancing the thermopower due to an enhanced effective mass, and reducing the lattice thermal conductivity due to point defect scattering of heat-carrying phonons. As a result, the *ZT* of  $\beta$ -La<sub>0.01</sub>Zn<sub>3.99</sub>Sb<sub>3</sub> reaches values of up to 1.3 at 648 K.

## 2. Experimental procedure

#### 2.1. Sample preparation

Polycrystalline  $\beta$ -(Zn<sub>1-x</sub>La<sub>x</sub>)<sub>4</sub>Sb<sub>3</sub> (x = 0, 0.005, 0.0075, 0.01, and 0.0125) samples were synthesized by melting highpurity elemental Zn (99.999%, powder), Sb (99.999%, powder) and La (99.999%, powder) according to the nominal composition. An excess of Zn of about 1% was added to compensate for the Zn loss during synthesis. The evacuated and sealed quartz glass tubes with the starting materials were heated at 1023 K for 12 h and then quenched in cold water. The as grown  $\beta$ -(Zn<sub>1-x</sub>La<sub>x</sub>)<sub>4</sub>Sb<sub>3</sub> (x = 0, 0.005, 0.0075, 0.01, and 0.0125) ingot was hand-ground into a fine powder and hot pressed at a pressure of 600 MPa in vacuum at 650 K for 1 h to get disc-shaped samples for TE property characterization.

## 2.2. Sample characterization

Sample phases were checked by X-ray powder diffraction on a Philips-X PERT PRO<sup>®</sup> diffractometer using  $Cu-K_{\alpha}$  radiation. Electrical resistivity  $\rho$  (=  $1/\sigma$ ) and thermopower S were measured on a ZEM-3 (ULVAC-RIKO®) in Helium atmosphere in the range of 300 K-650 K. Hall coefficients  $R_H$ were measured on a physical property measurement system (PPMS, Quantum Design<sup>®</sup>) at room temperature. The hole concentration p was estimated using the relationship p = 1/ $eR_{\rm H}$ , where e is the elementary charge. The Hall mobility  $\mu$ was calculated using the relationship  $\mu = \sigma R_{\rm H} = R_{\rm H}/\rho$ . The thermal diffusivity  $\alpha$  was measured with a NETZSCH LFA-457 ® apparatus in the range of 300 K-650 K in an argon atmosphere using cylinder disks (2-3 mm in thickness) coated with graphite. The total thermal conductivity  $\kappa$  was calculated according to  $\kappa = DC_{p}\alpha$ , where  $C_{p}$  is the isobaric specific thermal capacity measured on a differential scanning calorimeter (DSC, Perkin-Elmer<sup>®</sup>) in argon atmosphere and D the density measured by the Archimedes method.

#### 3. Results and discussion

#### 3.1. Phase characterization

XRD patterns of the  $\beta$ -(Zn<sub>1-x</sub>La<sub>x</sub>)<sub>4</sub>Sb<sub>3</sub> (x = 0, 0.0005, 0.0075, 0.01, and 0.0125) samples are shown in Fig. 1(a). All reflections can be indexed to  $\beta$ -Zn<sub>4</sub>Sb<sub>3</sub> (JCPDS #89-1969; space group  $R\overline{3}c$ ) and no secondary phase is detected. Upon La substitution, the refinement lattice constant *a* obviously increases (Fig. 1(b)) accompanied by an increase of the FWHM as demonstrated for the reflection at  $2\Theta \approx 25.3^{\circ}$  (Table 1). Both observations are consistent with a substitution of Zn<sup>2+</sup> with La<sup>3+</sup>. The host lattice increase originates from the larger ionic radius of La (ionic radius La<sup>3+</sup>: ~1.06 Å vs. ionic radius Zn<sup>2+</sup>: ~0.74 Å).

# 3.2. Thermoelectric properties

Fig. 2(a) shows the temperature dependence of the electrical conductivity ( $\sigma = 1/\rho$ ) of all  $\beta$ -(Zn<sub>1-x</sub>La<sub>x</sub>)<sub>4</sub>Sb<sub>3</sub> samples. All samples follow a similar pattern exhibiting a shallow minimum near 500 K, which is ascribed to the onset of the bipolar effect [15,19]. Importantly,  $\sigma$  increases with increasing La-substitution, which can be attributed to an increased carrier concentration (Table 1).

The temperature dependence of the thermopower *S* of  $\beta$ -(Zn<sub>1-x</sub>La<sub>x</sub>)<sub>4</sub>Sb<sub>3</sub> is displayed in Fig. 2(b). As with  $\sigma$ , all samples exhibit the same temperature dependence. *S* is positive in the whole temperature range indicating a *p*-type conduction effected by means of holes. Magnitude and temperature dependence of *S* are typical of degenerate semiconductors. In general, *S* increases linearly with increasing temperature in the range of 300 K-500 K before reaching a plateau between 500 K and 550 K indicating the onset of the bipolar effect.

We have made two interesting observations: (i) trivalent La (3+) substituting divalent Zn (2+) increased the carrier (hole)

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