



# A Green Biocompatible Fabrication of Highly Porous Functional Ceramics with High Strength and Controllable Pore Structures



Changlu Xu<sup>1</sup>, Haoran Liu<sup>1</sup>, Huilin Yang<sup>1,2</sup>, Lei Yang<sup>1,2,\*</sup>

<sup>1</sup> Orthopaedic Institute and Department of Orthopaedics, The First Affiliated Hospital, Soochow University, Suzhou 215006, China

<sup>2</sup> International Research Center for Translational Orthopaedics (IRCTO), Soochow University, Suzhou 215006, China

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A green biocompatible foaming method utilizing natural coconut oil and cornstarch was developed to fabricate highly porous functional ceramics with controllable strengths and pore structures. The porosity of Al<sub>2</sub>O<sub>3</sub> ceramics prepared via this method reached 79.6%–86.9% while these ceramics maintained high compressive strengths of 2.2–5.5 MPa. More importantly, porous Al<sub>2</sub>O<sub>3</sub> ceramic with a pore size gradient was also readily fabricated by casting serial layers of foams that were set for different time periods. The potential applications of porous Al<sub>2</sub>O<sub>3</sub> and HA ceramics fabricated by this green foaming method including scaffolds for oil cleaning and cell culture, respectively, were also demonstrated.

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## 1. Introduction

Porous ceramics with controllable porosity, mechanical property and pore structure are important and promising functional materials for a wide range of applications from tissue engineering, drug delivery, filtration to liquid or gas adsorption<sup>[1,2]</sup>. Fabrication methods of porous ceramics including freeze drying, leaching, foaming and sponge impregnation have been well established and widely used<sup>[3–5]</sup>. Among these methods, foaming technique is simple and efficient for the fabrication of large-scale porous structures but has a limited freedom in controlling pore parameters. In addition, typical foaming methods, however, use surfactant or modified ceramic particles to generate porous foam and rely on cross-linking agents or initiator to stabilize the foam<sup>[6,7]</sup>. Many of these surfactants, cross-linking agents or initiators require lengthy or toxic chemical processes to fabricate and are also risky and unsafe for biological or medical applications. Additionally, high-precision control of ceramic pore structures is challenging to achieve using foaming techniques. In these contexts, developing a green foaming method that can produce biocompatible, high-strength, high-porosity ceramics with controllable pore structures is desired.

This work aims to develop a green foaming method for fabricating highly porous ceramics with high biocompatibility, improved

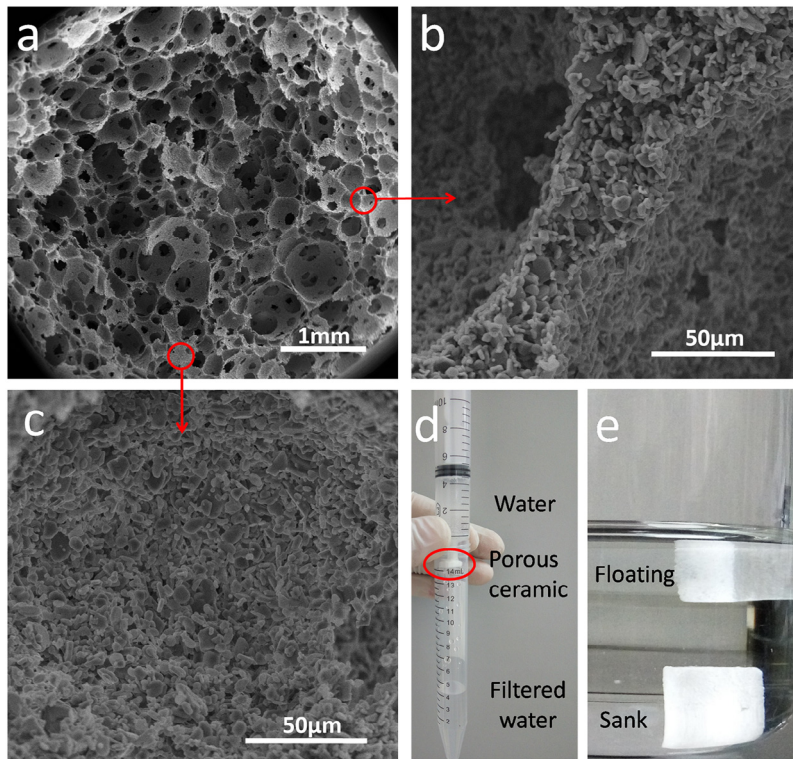
mechanical properties and adjustable pore structures. Here, a new foaming method based on natural coconut oil (cocamidopropyl betaine) and cornstarch as foaming and stabilizing agents, respectively, was first developed and a layer-by-layer casting method of fabricating complex pore structure including graded pores was also reported. Two examples of applying porous alumina (Al<sub>2</sub>O<sub>3</sub>) and hydroxyapatite (HA) ceramics (both are common bioceramics) for oil adsorption and cell culture, respectively, were also demonstrated.

## 2. Experimental

Commercial cornstarch and DI water were homogeneously mixed to obtain 10 wt% starch suspension in a beaker. Different amounts of Al<sub>2</sub>O<sub>3</sub> (Shanghai Gaoquan Chemical Engineering Co., Ltd) or HA (synthesized in house according to a method described elsewhere<sup>[8]</sup>) powder were added into the starch suspension and mixed uniformly to form slurries with varied solid contents. The slurry was heated to 90 °C in a water bath and 1 mL coconut oil (Heilongjiang Springhall Biological Technology Co., Ltd) was added. The slurry was agitated by an overhead stirrer until air bubbles fully infiltrated the slurry to form a thick froth and the beaker was then removed from the water bath to allow the froth to set. For fabricating ceramic with graded pores, slurries in different beakers were foamed and the froth was kept at 90 °C to allow bubbles to merge for different time periods. The froth in different beakers waited for varied merging times and was then casted layer by layer in a mold to obtain the foam with

\* Corresponding author. Fax: +86 512 67781165;

E-mail address: [leiy@suda.edu.cn](mailto:leiy@suda.edu.cn) (L. Yang).



**Fig. 1.** (a) SEM images of pore structure of  $\text{Al}_2\text{O}_3$  ceramics, and magnified images of pore ridge (b) and pore wall (c); (d) porous  $\text{Al}_2\text{O}_3$  ceramics having good permeability to water; and (e) porous  $\text{Al}_2\text{O}_3$  ceramics with (floating) and without (sank) a silicone layer outside the ceramic showing different floating status in ethanol.

graded pore sizes. All foams were set for 24 h and then dried into green bodies in ambient conditions. The green bodies of  $\text{Al}_2\text{O}_3$  or HA were sintered at 1550 °C and 1250 °C for 2 h to obtain porous ceramics, respectively.

Porosity was calculated by comparing theoretical and apparent densities of the porous ceramics<sup>[9]</sup>. Microstructure was characterized by scanning electron microscopy (SEM, FEI Quanta 250) and the uniaxial compression tests were performed on a mechanical tester (HY-1080) operating at a crosshead speed of 1 mm/min. Oil absorption capacity of porous  $\text{Al}_2\text{O}_3$  ceramic was qualitatively observed by placing a ceramic piece (1 cm × 1 cm × 0.5 cm) on top of peanut oil (0.3 mL) and the progress of oil absorption by the porous ceramic was photographed. For cell culture experiment, rat osteoblasts (MC3T3-E1) suspension were dribbled into a porous HA ceramic at a density of 10,000 cells per  $\text{cm}^3$  of ceramic, and the cell-loaded ceramic was cultured under standard conditions (37 °C, 5%  $\text{CO}_2$  and humidified air) for 24 h. The cell-loaded ceramic was then fixed by glutaraldehyde (2.5 wt%), dehydrated in a critical point dryer and observed by SEM.

### 3. Results and Discussion

Porous ceramics reported here consisted of a macro–micro porous architecture with different types of pores. SEM characterization of the porous  $\text{Al}_2\text{O}_3$  ceramics revealed primary large spherical pores with a narrow size distribution range of 400–800  $\mu\text{m}$  (Fig. 1(a)) and secondary interconnected pores with sizes of 100–300  $\mu\text{m}$  existing on the walls of primary pores. A close-up at the wall of the primary pores also showed a great number of submicron pores, which is probably attributed to the decomposition of starch during sintering (Fig. 1(b, c)).

The porous ceramics also exhibited high porosities of 79.6%–86.9%, which was inversely increased with respect to the solid contents of HA

and starch in the slurry (Table 1). High solid contents increased slurry density and viscosity and thus decreased foaming efficiency, causing lowered porosity<sup>[11]</sup>. Different shrink rates of the ceramic during sintering that resulted from the varied starch contents might also cause altered porosity. In general, the solid contents used in this work (28%–36%) revealed satisfied foaming efficiency, leading to the high porosity and macroporous structure mentioned above as well as high connectivity of pores. Owing to the interconnected pores and high porosity, the porous ceramic had high permeability to fluids like water and gas (Fig. 1(c)). Interestingly, Fig. 1(e) demonstrates that a porous ceramic covered by a silicone layer (to keep liquid from infiltrating the porous ceramic) could float on the ethanol while the ceramic without silicone layer sank immediately, again indicating the high porosity and connectivity of pores in the porous ceramic.

More importantly, compressive strengths of porous  $\text{Al}_2\text{O}_3$  ceramics remained high (5.5–2.5 MPa); even the porosity is >79%, reaching a maximum of  $5.5 \pm 1.4$  MPa at porosity of 79.6%. The porous ceramics produced here had significantly higher compressive strengths compared to the ceramics with similar porosity but fabricated by sponge replica method (Table 1)<sup>[10]</sup>. In general, small pores on the walls of primary large pores can greatly decrease the strength of porous ceramics<sup>[12]</sup>. The porous ceramics fabricated here, however,

**Table 1**  
Porosities and compressive strengths of porous  $\text{Al}_2\text{O}_3$  ceramics with different solid contents compared with the ceramics fabricated by sponge replica method<sup>\*[10]</sup>

Solid content of slurry (wt%)	Porosity (%)	Compressive strength (MPa)	Porosity from literature* (%)	Compressive strength from literature* (MPa)
28	86.9 ± 0.1	2.2 ± 0.3	86	0.19 ± 0.08
32	84.7 ± 0.7	3.8 ± 0.6	78	1.58 ± 0.07
36	79.6 ± 0.3	5.5 ± 1.4	71	4.52 ± 0.75

\* Data from Reference [10].

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