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RAPID COMMUNICATI	ON
From binary	y to multicomponent photoactive
layer: A pro	mising complementary strategy
to efficient	hybrid solar cells
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	Abstract
KEYWORDS	A strategy is demonstrated for fabrication of highly efficient hybrid solar cells based on the
KEYWORDS Solar cells; ZnO nanorod array;	polymer/nanoarrays with complementary multicomponents in photoactive layer, including a
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KEYWORDS Solar cells; ZnO nanorod array; Interfacial modification; Conjugated polymer *Corresponding author at: I Academy of Sciences, Hefei 23 Tel (fax: + 96 551 65503174	polymer/nanoarrays with complementary multicomponents in photoactive layer, including a scenario to controllably synthesize ternary ZnO/CdS/Sb ₂ S ₃ -core/shell/shell nanoarrays (ZCS-NAs) for a high open-circuit voltage (V_{oc}) and short-circuit current and an approach to dope amorphous polymer with lithium bis(trifluoromethanesulfonyl) amide at nanoscale for a remarkably improved fill factor. With the integrated benefits from the complementary multi-components having optimized nanoarray structure and doping concentration, an efficiency up to 5.01% under AM 1.5 illumination (100 mW/cm ²) is achieved in the polymer/ZCS-NA devices with poly(2-methoxy-5-(2-ethylhexyloxy)-1,4-phenylenevinylene) as the polymer. To the best of our knowledge, this is the highest efficiency in the polymer/nanoarray devices. It is found that the V_{oc} in the multicomponent solar cells is determined by the band level difference between ZnO core and polymer, and sufficient photo-excitation of the polymer is necessary for efficient photocurrent generation. The component effects on device performance are elucidated and a model concerning the effective polymer phase and illumination attenuation between nanorods is proposed for understanding the charge generation from polymer absorption in the multi-component solar cells.
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Introduction

Hybrid polymer-based solar cells (HPSCs) using a photoactive layer formed by a conjugated polymer as electron donor (D) and inorganic nanostructure as electron acceptor (A) have shown promises in pursuit of cost-effective photovoltaic devices 13 [1-6], due to their particular advantages of integrating the properties of organic polymer (good flexibility, light-weight, and 15 easy-film formation) and inorganic nanostructures (high electron mobility, high electron affinity, and good stability). In 17 principle, the polymer absorbs photons to generate excitons (bound electron-hole pairs); the photogenerated excitons dif-19 fuse to the D/A interface for dissociation into free charge carriers (electrons and holes) that are transported within D and 21 A components to the respective electrodes for photocurrent generation. The photoactive layer in HPSCs is often structured 23 into planar and bulk architectures. The D and A components are sequentially deposited on top of each other in the planer 25 architecture, providing bilayer heterojunction devices with a single and two-dimensional D/A interface. However, nanostruc-27 tured A component is dispersed in polymer matrix in the bulk architecture, resulting in bulk-heterojunction (BHJ) solar cells 29 with three-dimensionally distributed D/A interfaces. With the use of BHJ architecture, a high D/A interface area for exciton 31 dissociation is created, and the problem due to the limited exciton diffusion length $(L_{\rm D})$ in common conjugated polymers, 33 e.g., $L_D < 10$ nm in poly(2-methoxy-5-(2-ethylhexyloxy)-1,4phenylene vinylene) (MEH-PPV) [7-9] and poly(3-hexylthio-35 phene) (P3HT) [10,11], is addressed by nanoscale phase separation that results in a bicontinuous interpenetrating network 37 of D and A phases in photoactive layer. The power conversion efficiency (η) of around 5% has been achieved in the polymer/ 39 CdTe devices with combined bilayer and BHJ architectures [12] and the $P3HT/TiO_2$ bulk devices fabricated by infiltrating P3HTinto TiO₂/Sb₂S₃ coaxial nanowire arrays [13]. Seok and coworkers [14-16] have achieved the efficiencies of 5.13%, 6.18% and 6.3% in the BHJ devices based on Sb₂S₃-sensitized mesoporous TiO₂ films in combination with P3HT, poly(2,6-45 (4,4-bis-(2-ethylhexyl)-4H-cyclopenta [2,1-b;3,4-b']dithiophene)-alt-4,7(2,1,3-benzothiadiazole)) (PCPDTBT), and PCPDTB T-PCBM ([6,6]-phenyl-C61-butyric acid methyl ester) blend, respectively. However, the disordered electron transport formed by nanoparticles may cause a serious charge recombination; in particular, the nanomorphology formed by the simple 51 blend of polymer and nanoparticles will encounter a low thermal stability under illumination. 53

Replacing disordered A-phase pathways formed by nanoparticles with vertically aligned nanorod/nanowire arrays offers the way to prepare the HPSCs with an ideal BHJ architecture for long-term application [1-6,17,18]. The nanoarray devices have a high area and stabilized spatial distribution of D/A interfaces for exciton dissociation, and the straightforward and interdigitated nanochannels for electron transport. Zinc oxide nanorod/nanowire arrays (ZnO-NAs) are widely used in such aligned BHJ devices [17-22], due to their facile synthesis and environment-friendly characteristics [23-25]. Incorporating a

conjugated polymer into ZnO-NAs provides the polymer/ZnO-NA solar cells with binary components (i.e., polymer and ZnO) in photoactive layer. However, the η in the polymer/ZnO-NA devices is not high enough yet ($\eta = 0.2 - 0.5\%$). The efficiency of a solar cell is determined by the relationship $\eta = J_{sc} \times V_{oc} \times FF/$ $P_{\rm in}$, where $J_{\rm sc}$ is the short-circuit current density, $V_{\rm oc}$ is the open-circuit voltage, FF is the fill factor, and P_{in} is the incident light power density. The polymer/ZnO-NA devices with MEH-PPV and P3HT are suffering from a small V_{oc} (~0.1 to 0.5 V) and a narrow polymer absorption spectrum (400-600 nm) for a low J_{sc} [19-22].

The structural characteristics at the D/A interface governing charge generation and recombination are crucially important. Interfacial modification has been attempted to improve the performance in polymer/ZnO-NA devices. Modification of ZnO-NA with amphiphilic organic molecules can enhance the $J_{\rm sc}$ to a certain extent as a result of the improved compatibility between polymer and ZnO, but has a limited effect on increasing V_{oc} [26-29]. Moreover, modification of ZnO-NA with inorganic semiconductors can enhance V_{oc} up to above 0.8 V, but has an unsatisfactory improvement in J_{sc} [30-34]. Notably, these organic [28,29] or inorganic [32,33] modifications often do not improve remarkably the device FF (\sim 30% to 40%); in particular, the improved J_{sc} from those modifications is still limited to the scale of several mA/cm^2 . Reasonably, the binary nanoarrays (i.e., modifier and ZnO) and the resulting ternary photoactive layer (binary nanoarray and polymer) obtained from the modifications with an organic or inorganic modifier mainly improve either J_{sc} or V_{oc} . Therefore, it still remains challenging to get a significantly high V_{oc} and J_{sc} , as well as FF, in an individual device for a breakthrough at the efficiency of the polymer/ZnO-NA solar cells.

97 We previously coated the ZnO nanorods in ZnO-NA with a polycrystalline film of CdS guantum dots (QDs), and the 99 resulting heterostructured ZnO/CdS-core/shell nanorod array (ZC-NA) generated the HPSCs with a $V_{\rm oc}$ up to 0.82 V, for which 101 a not high efficiency of $\eta = 0.87\%$ is mainly due to a still rather low J_{sc} and FF [33]. Here we demonstrate, for the first time, a 103 strategy to fabricate the efficient HPSCs based on polymer/ ZnO-NA systems by using multicomponents with complemen-105 tary properties in photoactive layer. Our approach includes the controllable synthesis of ternary core/shell/shell nanoarrays 107 for a high $V_{\rm oc}$ and $J_{\rm sc}$, and the nanoscale doping of conjugated polymer for a remarkably improved FF. The chemical processes 109 for ternary nanoarray synthesis, the solar cell architecture and the charge transfer processes involved are depicted in 111 Figure 1. As illustrated in Figure 1a, the preformed ZnO nanorods in ZnO-NA are first coated with a polycrystalline film of 113 CdS-QDs as an inner shell via successive ion layer adsorption and reaction (SILAR) technique, producing a binary ZnO/CdS-115 core/shell nanorod array (i.e., ZC-NA); afterwards, a Sb₂S₃ outer shell is formed by depositing Sb₂S₃ nanocrystals on the 117 CdS shell through a two-step process that involves the chem-119 ical bath deposition (CBD) and thermally induced crystallization of amorphous Sb₂S₃, providing finally a ternary ZnO/CdS/Sb₂S₃-

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