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[Nano Energy \(](dx.doi.org/10.1016/j.nanoen.2014.09.028)IIII) I, III-III Available online at www.sciencedirect.com 63 1 nano energy **ScienceDirect** 3 65 5 67 journal homepage: www.elsevier.com/locate/nanoenergy 7 69 α 71 RAPID COMMUNICATION 11 73 From binary to multicomponent photoactive 13 75 layer: A promising complementary strategy 15 77 17 to efficient hybrid solar cells 79 19 81 Changwen Liu^{a, b}, Zeliang Qiu^{a, b}, Feng Li^c, Weili Meng^{a, b}, 21 83 a Wenjin Yue^{a,b}, Fapei Zhang^c, Qiquan Qiao^d, Mingtai Wang^{a,b,}* 23 85 25 87 ^aInstitute of Plasma Physics, Chinese Academy of Sciences, Hefei 230031, PR China Q2 ^bKey Lab of Novel Thin Film Solar Cells, Chinese Academy of Sciences, Hefei 230031, PR China 27 89 ^cHigh Magnetic Field Laboratory, Chinese Academy of Sciences, Hefei 230031, PR China ^dCenter for Advanced Photovoltaics, Department of Electrical Engineering and Computer Sciences, South Dakota 29 91 State University, Brookings, SD 57007, USA 31 93 Received 1 August 2014; received in revised form 2 September 2014; accepted 20 September 2014 33 95 35 97 **KEYWORDS** Abstract Solar cells; A strategy is demonstrated for fabrication of highly efficient hybrid solar cells based on the 99 37 ZnO nanorod array; polymer/nanoarrays with complementary multicomponents in photoactive layer, including a Interfacial modification; scenario to controllably synthesize ternary ZnO/CdS/Sb₂S₃-core/shell/shell nanoarrays (ZCS-39 101 Conjugated polymer NAs) for a high open-circuit voltage (V_{oc}) and short-circuit current and an approach to dope amorphous polymer with lithium bis(trifluoromethanesulfonyl) amide at nanoscale for a 103 41 remarkably improved fill factor. With the integrated benefits from the complementary multicomponents having optimized nanoarray structure and doping concentration, an efficiency up 105 43 to 5.01% under AM 1.5 illumination (100 mW/cm²) is achieved in the polymer/ZCS-NA devices with poly(2-methoxy-5-(2-ethylhexyloxy)-1,4-phenylenevinylene) as the polymer. To the best of 45 107 our knowledge, this is the highest efficiency in the polymer/nanoarray devices. It is found that the V_{oc} in the multicomponent solar cells is determined by the band level difference between 109 47 ZnO core and polymer, and sufficient photo-excitation of the polymer is necessary for efficient photocurrent generation. The component effects on device performance are elucidated and a 49 111 model concerning the effective polymer phase and illumination attenuation between nanorods is proposed for understanding the charge generation from polymer absorption in the multi-51 113 component solar cells. 53 115 55 117 *Corresponding author at: Institute of Plasma Physics, Chinese Academy of Sciences, Hefei 230031, PR China. 57 119 Tel./fax: +86 551 65593171. E-mail address: mtwang@ipp.ac.cn (M. Wang). 59 121 [http://dx.doi.org/10.1016/j.nanoen.2014.09.028](dx.doi.org/10.1016/j.nanoen.2014.09.028) 123 61 2211-2855/© [2014 Elsevier Ltd. All rights reserved.](dx.doi.org/10.1016/j.nanoen.2014.09.028)

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Introduction

Hybrid polymer-based solar cells (HPSCs) using a photoactive layer formed by a conjugated polymer as electron donor (D) and inorganic nanostructure as electron acceptor (A) have shown promises in pursuit of cost-effective photovoltaic devices [\[1](#page--1-0)-[6\],](#page--1-0) due to their particular advantages of integrating the properties of organic polymer (good flexibility, light-weight, and easy-film formation) and inorganic nanostructures (high electron mobility, high electron affinity, and good stability). In principle, the polymer absorbs photons to generate excitons (bound electron–hole pairs); the photogenerated excitons diffuse to the D/A interface for dissociation into free charge carriers (electrons and holes) that are transported within D and A components to the respective electrodes for photocurrent generation. The photoactive layer in HPSCs is often structured into planar and bulk architectures. The D and A components are sequentially deposited on top of each other in the planer architecture, providing bilayer heterojunction devices with a single and two-dimensional D/A interface. However, nanostructured A component is dispersed in polymer matrix in the bulk architecture, resulting in bulk-heterojunction (BHJ) solar cells with three-dimensionally distributed D/A interfaces. With the use of BHJ architecture, a high D/A interface area for exciton dissociation is created, and the problem due to the limited exciton diffusion length (L_D) in common conjugated polymers, e.g., $L_D < 10$ nm in poly(2-methoxy-5-(2-ethylhexyloxy)-1,4phenylene vinylene) (MEH-PPV) [\[7](#page--1-0)–[9\]](#page--1-0) and poly(3-hexylthio-phene) (P3HT) [\[10,11\],](#page--1-0) is addressed by nanoscale phase separation that results in a bicontinuous interpenetrating network of D and A phases in photoactive layer. The power conversion efficiency (η) of around 5% has been achieved in the polymer/ CdTe devices with combined bilayer and BHJ architectures [\[12\]](#page--1-0) and the P3HT/TiO₂ bulk devices fabricated by infiltrating P3HT into $TiO₂/Sb₂S₃$ coaxial nanowire arrays [\[13\].](#page--1-0) Seok and coworkers [\[14](#page--1-0)–[16\]](#page--1-0) have achieved the efficiencies of 5.13%, 6.18% and 6.3% in the BHJ devices based on Sb_2S_3 -sensitized mesoporous TiO₂ films in combination with P3HT, poly(2,6-(4,4-bis-(2-ethylhexyl)-4H-cyclopenta [2,1-b;3,4-b/]dithiophene)-alt-4,7(2,1,3-benzothiadiazole)) (PCPDTBT), and PCPDTB T-PCBM ([6,6]-phenyl-C61-butyric acid methyl ester) blend, respectively. However, the disordered electron transport formed by nanoparticles may cause a serious charge recombination; in particular, the nanomorphology formed by the simple blend of polymer and nanoparticles will encounter a low thermal stability under illumination. 9 11 13 15 17 19 21 23 25 27 29 31 33 35 37 39 41 43 45 47 49 51 53

Replacing disordered A-phase pathways formed by nanoparticles with vertically aligned nanorod/nanowire arrays offers the way to prepare the HPSCs with an ideal BHJ architecture for long-term application $[1-6,17,18]$ $[1-6,17,18]$. The nanoarray devices have a high area and stabilized spatial distribution of D/A interfaces for exciton dissociation, and the straightforward and interdigitated nanochannels for electron transport. Zinc oxide nanorod/nanowire arrays (ZnO–NAs) are widely used in such aligned BHJ devices [\[17](#page--1-0)–[22\],](#page--1-0) due to their facile synthesis and environment-friendly characteristics [\[23](#page--1-0)–[25\]](#page--1-0). Incorporating a

conjugated polymer into ZnO–NAs provides the polymer/ZnO– NA solar cells with binary components (i.e., polymer and ZnO) in photoactive layer. However, the η in the polymer/ZnO-NA devices is not high enough yet $(n=0.2-0.5%)$. The efficiency of a solar cell is determined by the relationship $\eta = J_{\rm sc} \times V_{\rm oc} \times FF/$ P_{in} , where J_{sc} is the short-circuit current density, V_{oc} is the open-circuit voltage, FF is the fill factor, and P_{in} is the incident light power density. The polymer/ZnO–NA devices with MEH-PPV and P3HT are suffering from a small $V_{\alpha c}$ (\sim 0.1 to 0.5 V) and a narrow polymer absorption spectrum $(400 - 600$ nm) for a low $J_{\rm sc}$ [\[19](#page--1-0)-[22\]](#page--1-0).

The structural characteristics at the D/A interface governing charge generation and recombination are crucially important. Interfacial modification has been attempted to improve the performance in polymer/ZnO–NA devices. Modification of ZnO–NA with amphiphilic organic molecules can enhance the $J_{\rm sc}$ to a certain extent as a result of the improved compatibility between polymer and ZnO, but has a limited effect on increasing V_{oc} [\[26](#page--1-0)-[29\]](#page--1-0). Moreover, modification of ZnO–NA with inorganic semiconductors can enhance V_{oc} up to above 0.8 V, but has an unsatisfactory improvement in $J_{\rm sc}$ [\[30](#page--1-0)-[34\]](#page--1-0). Notably, these organic [\[28,29\]](#page--1-0) or inorganic [\[32,33\]](#page--1-0) modifications often do not improve remarkably the device FF (\sim 30% to 40%); in particular, the improved $J_{\rm sc}$ from those modifications is still limited to the scale of several mA/cm². Reasonably, the binary nanoarrays (i.e., modifier and ZnO) and the resulting ternary photoactive layer (binary nanoarray and polymer) obtained from the modifications with an organic or inorganic modifier mainly improve either $J_{\rm sc}$ or $V_{\rm oc}$. Therefore, it still remains challenging to get a significantly high V_{oc} and J_{sc} , as well as FF, in an individual device for a breakthrough at the efficiency of the polymer/ZnO–NA solar cells.

We previously coated the ZnO nanorods in ZnO–NA with a polycrystalline film of CdS quantum dots (QDs), and the resulting heterostructured ZnO/CdS–core/shell nanorod array (ZC-NA) generated the HPSCs with a V_{oc} up to 0.82 V, for which a not high efficiency of $\eta = 0.87\%$ is mainly due to a still rather low $J_{\rm sc}$ and FF [\[33\].](#page--1-0) Here we demonstrate, for the first time, a strategy to fabricate the efficient HPSCs based on polymer/ ZnO–NA systems by using multicomponents with complementary properties in photoactive layer. Our approach includes the controllable synthesis of ternary core/shell/shell nanoarrays for a high V_{oc} and J_{sc} , and the nanoscale doping of conjugated polymer for a remarkably improved FF. The chemical processes for ternary nanoarray synthesis, the solar cell architecture and the charge transfer processes involved are depicted in [Figure 1](#page--1-0). As illustrated in [Figure 1](#page--1-0)a, the preformed ZnO nanorods in ZnO–NA are first coated with a polycrystalline film of CdS–QDs as an inner shell via successive ion layer adsorption and reaction (SILAR) technique, producing a binary ZnO/CdS– core/shell nanorod array (i.e., ZC-NA); afterwards, a Sb_2S_3 outer shell is formed by depositing Sb_2S_3 nanocrystals on the CdS shell through a two-step process that involves the chemical bath deposition (CBD) and thermally induced crystallization of amorphous Sb_2S_3 , providing finally a ternary $ZnO/CdS/Sb_2S_3$ -97 99 101 103 105 107 109 111 113 115 117 119

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