Computational Materials Science 104 (2015) 29-34

Contents lists available at ScienceDirect

Computational Materials Science

journal homepage: www.elsevier.com/locate/commatsci



Theoretical study of structural, mechanical, thermal and electronic properties of Ti₃B₄ with Ta₃B₄ structure under high pressure



Guoliang Wang^{a,b}, Yefei Li^{a,*}, Yimin Gao^a, Yonghong Cheng^c, Shengqiang Ma^a

^a State Key Laboratory for Mechanical Behavior of Materials, Xi'an Jiaotong University, Xi'an 710049, China ^b Xi'an Research Inst of Hi-Tech, Xi'an 710025, China

^c State Key Laboratory of Electric Insulation and Power Equipment, Xi'an Jiaotong University, Xi'an 710049, China

ARTICLE INFO

Article history: Received 15 September 2014 Received in revised form 21 January 2015 Accepted 16 March 2015 Available online 15 April 2015

Keywords: Structure Bonding Elastic behavior Heat capacity Thermal expansion

ABSTRACT

We present a systematic theoretical study on the structural, electronic, elastic, and thermal properties of Ti_3B_4 in Ta_3B_4 structure under pressure. Our calculations indicate that the volume of Ti_3B_4 cell decreases about 12% of primary structure and the *b* axis length is shorten by 5% when the applied pressure is 36 GPa. The peak heights of DOS and PDOS curves of Ti_3B_4 are lower slightly when under high pressure. The calculated elastic constants and elastic modulus are all found to increase linearly by increasing pressure; and the heat capacity at constant volume (C_V) as a function of pressure are evaluated, which is more sensitive to the temperature than to the pressure. Meanwhile, the elastic anisotropy, phonon dispersion and corresponding phonon density of states with pressure are also calculated and discussed. The average LTEC of Ti_3B_4 is evaluated to be $8.6 \times 10^{-6} \text{ K}^{-1}$ at room temperature, which may provide valuable datum for its thermal property investigation.

© 2015 Elsevier B.V. All rights reserved.

1. Introduction

Titanium borides present quite attractive advantages such as the high melting point, hardness, wear resistance, thermal shock resistance, and electrical conductivity [1]. These characteristics make them widely used in industrial areas such as cutting tools, crucibles, electrode materials and wear-resistant parts [2,3].

In Ti–B binary system, Murray et al. [4] treated the solution phases TiB (Pnma) and TiB₂ (P6/mmm) as the stoichiometric compounds. By different methods, the existence of stoichiometric compound Ti₃B₄ was also confirmed by many researchers [5–7]. This compound has an orthorhombic Ta₃B₄-type structure (Immm), and can form by a peritectic reaction between TiB₂ and metal liquid at very high temperature. However, the available literature data about peritectic reaction temperature is contradictory. In the publication [7] this reaction was assigned as 2333 K; while in the assessment [8] the reaction, Witusiewicz et al. [9] performed some key experiments by XRD and DTA, based on which they put forward the peritectic reaction temperature of 2412 K. The study of Ti–B binary system has been performed by many researchers [4–9], and much useful information has been achieved. However, up to now, some of their properties under pressure are still almost unknown. To our knowledge, there are only few published results about TiB₂ with AlB₂ structure under high pressure [10], and the published data for Ti₃B₄ ceramic under high pressure even do not exist. However, the study of the pressure behavior of hard materials is necessary. In our earlier work [11], the hardness of Ti₃B₄ was estimated to be as high as 33.1 GPa, which is just a little smaller than experimental datum of TiB₂ (35 GPa) [12]. Therefore, in present work, we will investigate the structural, electronic, mechanical, and lattice dynamical properties of Ti₃B₄ in Ta₃B₄ structure under high pressure.

2. Model and computational method

The crystal structure of Ti_3B_4 considered in this paper is shown in Fig. 1a, which consists of two alternative layers (layers A and B) in [010] direction. In large-scale image (2 × 2 in Fig. 1b), it can be clearly seen that both layers have the same atomic configuration, and layer B is a translation structure transformed from layer A by [0.500.5] fractional coordinate. Meanwhile, as illustrated in Fig. 1a, Ti_3B_4 could also be regarded as an intergrown structure consisting of two kinds of layered sub-structures. The middle sub-structure is similar to TiB_2 structure and the upper/lower

^{*} Corresponding author at: State Key Laboratory for Mechanical Behavior of Materials, School of Materials Science and Engineering, Xi'an Jiaotong University, 28 Xianning West Road, Xi'an, Shaanxi Province 710049, China. Tel./fax: +86 29 82665479.

E-mail address: yefeili@126.com (Y. Li).

[010] laver B uni [100] ub-structu [100 **Aiddle** Upper/lower sub-structure Middle sub-structure structure ower [001] [100] [100] [100 [001] [001 [010] [010] [010] [001] [100] (c) Extended middle sub-structure (1×2) (d) Extended upper/lower sub-structure (1×2) Ti₂B₄ structure (b) Extended layer A (2 \times 2) projected onto [010] direction projected onto [010] direction

Fig. 1. Schematic diagrams of the structures for the titanium boride cell: (a) Ti_3B_4 structure, (b) extended layer A (2 × 2), (c) extended middle sub-structure (1 × 2) projected onto [010] direction, and (d) extended upper/lower sub-structure (1 × 2) projected onto [010] direction.

sub-structure belongs to layered Ti–B sub-structure. We show the extended middle, upper/lower sub-structures and their projection on [010] direction in Fig. 1c and d, respectively. Both of the sub-structures adopt a layered hexagonal structure with alternating close-packed layers of graphite-like boron and titanium layers.

The present work was carried out by using first-principles method based on the density functional theory (DFT), as implemented CASTEP code [13,14]. The ultrasoft pseudo-potentials (USPPs) were used to represent the interactions between ionic core and valence electrons. The valence electrons of Ti and B are considered as 3s²3p⁶3d²4s², 2s²2p¹, respectively. A plane-wave basis set with cut-off energy of 350 eV was used. For the exchange and correlation terms in the electron-electron interaction, Generalized gradient approximation (GGA) exchange-correlation energy of PBE approach was used for calculations [15]. For k-space summation the $8 \times 8 \times 2$ for Ti₃B₄ with Monkhorst-Pack scheme in the first irreducible Brillouin zone [16] was used. The elastic constants were calculated by means of linear fitting of the stress-strain curves based on the Hooker's low [17]. The finite displacement method is applied to a supercell consisting of the geometrically optimized primitive lattice for calculating the phonon dispersion curves. The real space cutoff radius was set to 5 Å, which resulted in a supercell (Ti₂₄B₃₂) volume four times larger than the original unit cell. For thermal properties, the calculation method in details can be found in our past work [11]. In general, the commonly DFT gives the ground state total energy (E_{gs}) without considering the temperature effect. Then, the vibration free energy (F_{vib}) is calculated by phonon density of states at given volume using CASTEP code. We should note that for metallic Ti₃B₄ boride, the contribution of thermal electron must be considered for total Helmholtz free energy (F(V, T)). Finally, using data F(V, T) - V, the EOS at temperature *T* can be fitted; and the equilibrium volume V(T) at T is obtained. The volumetric thermal expansion coefficient β can be calculated through $V(T) = (1 + \beta T)V_0$, where V_0 represents the equilibrium volume at 0 K. The linear expansion coefficient and volumetric coefficient are related by $\alpha = \beta/3$.

3. Results and discussion

3.1. Pressure-induced structural changes

The optimized cell constants are in good agreement with the corresponding experimental values. The thermodynamically stable crystal structure requires the formation enthalpy to be negative. In this calculation, the obtained value is -0.94 eV/atom, which is similar to the assessed datum -1.03 eV/atom from Ma et al. [8]. The energy–volume values for the different lattice parameters (at constant b/a and c/a) were calculated using the full relaxed atomic position in each step. The calculated E(V, T) - V points were then fitted by means of Birch–Murnaghan's equation of state (Eq. (1)) at temperature *T*, as shown in Fig. 2a. It can be observed that as temperature increases, the total energy decreases continuously. The equilibrium volume of Ti₃B₄ increases with the raising temperature (Fig. 2b), which proves the nature of its thermal expansion and contraction. Meanwhile, the bulk modulus and its pressure derivative were estimated based on the same Murnaghan's EOS. Then, the pressure–volume relationship will be obtained, which will be discussed in next section.

$$E(V) = \frac{B_0 V_0}{B'_0} \left[\frac{1}{B'_0 - 1} \left(\frac{V}{V_0} \right)^{(1 - B'_0)} + \frac{V}{V_0} + \frac{B'_0}{1 - B'_0} \right] + E_0$$
(1)

In order to provide some pressure-induced structure information of Ti₃B₄, both the cell parameters and atomic positions were fully relaxed under pressure from 0 GPa to 36 GPa. The pressure dependence of relative lattice parameters a/a_0 , b/b_0 , c/c_0 , and V/V_0 were obtained and shown in Fig. 3a, where a_0 , b_0 , c_0 , and V_0 are the zero temperature/pressure cell lattice parameters, respectively. From Fig. 3a, all ratios decrease continuously with increasing pressure. The crystal structure along *b*-axis is slightly less compressible than that along the other axes. When the applied pressure is up to 36 GPa, the *b* axis length is shortened by about 5%, and finally the volume of Ti₃B₄ cell decreases about 12% of primary structure. The lattice compressibility may be closely related to the layered crystal structure of Ti₃B₄ along [010] direction; the strong chemical bonds exist intra-layer, which are stronger than the bonds between the layers. In Ti₃B₄ crystal cell, boron atoms form a layered sub-lattice by sharing the valence electrons through sp² hybridization; the Ti-B bond between each layer is the mixture of ironic-covalent interactions, which is relatively weaker than B–B bond [18–21]. These electronic characters are the main reason for the compressibility anisotropy. Fig. 3b illustrates the relative volumes (V/V_0) of Ti₃B₄ at different pressures and temperature, from which we can see that the relative volume in higher temperature is larger than that in lower temperature under the same pressure; as a given temperature, the relative volume decreases by increasing pressure. In other words, the volume expansion behavior is affected both by temperature and pressure; the former one has positive effect, while Download English Version:

https://daneshyari.com/en/article/1560226

Download Persian Version:

https://daneshyari.com/article/1560226

Daneshyari.com