



Deformation microstructures and tensile properties of an austenitic stainless steel subjected to multiple warm rolling

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ABSTRACT

The deformation microstructures of a 316L-type austenitic stainless steel subjected to multiple bar rolling to a total strain of 2 at temperatures of 773–1173 K and their effect on the mechanical properties at ambient and elevated temperatures were studied. The multiple warm rolling was accompanied by significant grain refinement. The finally evolved transverse grain size decreased from 3.4 μm to 0.85 μm with a decrease in the rolling temperature from 1173 K to 773 K. The warm rolled steel samples were characterized by significantly increased strength properties. The strengthening was studied by tensile tests at ambient and elevated temperatures. A decrease in the rolling temperature from 1173 K to 773 K increased the yield strength from 720 MPa to 945 MPa or from 395 MPa to 470 MPa at room temperature and 973 K, respectively. The strengthening obeyed Hall-Petch relationship with a grain size strengthening coefficient decreased from 410 MPa $\mu\text{m}^{0.5}$ at room temperature to 140 MPa $\mu\text{m}^{0.5}$ at 973 K.

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1. Introduction

Austenitic stainless steels are one of the most widely used structural materials because of their beneficial combination of mechanical, technological and functional properties [1]. A common disadvantage of austenitic stainless steels is their relatively low yield strengths of 200–400 MPa, which limits their usage in critical applications [2]. Generally, mechanical properties and, in particular, an yield strength of metallic materials depend significantly on their microstructures, which in turn can be controlled by thermo-mechanical treatments. Regarding the strength of structural steels and alloys, substantial strengthening can be obtained by structural refinement, which is achieved by a decrease in the grain size and can be expressed by Hall-Petch relationship [3–7].

Large strain deformations under conditions of warm to hot working, which are accompanied by the development of dynamic recrystallization (DRX), are powerful tool for processing the austenitic stainless steels with desirable microstructures [8–10]. The required grain size distribution can be developed directly during plastic working under appropriate conditions. Depending on deformation conditions, the austenitic stainless steels exhibit different DRX behaviors [11]. The discontinuous DRX takes place in

cyclic manner under hot working conditions, when the new grains repeatedly nucleate due to local migration, i. e. bulging, of the grain boundaries and then the recrystallizing nuclei grow at expense of work hardened surroundings [9]. On the other hand, the new grains evolve as a result of progressive increase in misorientations among the strain-induced subboundaries up to typical values of high-angle grain boundaries in the case of continuous DRX under warm working conditions [9]. The mean grain size that evolve at sufficiently large strains depends sensitively on the deformation conditions and can be expressed by power law functions of flow stress with grain size exponents of -0.7 and -0.4 for discontinuous and continuous dynamic recrystallization domains, respectively. These dependencies suggest that significant grain refinement should be expected during large strain deformation under high flow stresses, which can be realized under conditions of warm working at relatively low temperatures. However, in contrast to hot working, which has been widely using for steel processing, warm working is rarely involved in the modern thermo-mechanical techniques dealing with production of austenitic stainless steels.

The aim of the present paper is to report the results of our current investigations on the deformation microstructures and tensile properties of a 316L-type austenitic stainless steel subjected to caliber rolling under conditions of warm to hot working at temperatures of 0.45–0.7 of the melting point. The paper is particularly focused on the effect of rolling temperature on the

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yield strength of the steel at ambient and elevated temperatures.

2. Experimental

A 316L-type austenitic stainless steel, Fe – 0.04%C – 0.4%Si – 1.7%Mn – 17.3%Cr – 10.7%Ni – 2.0%Mo – 0.04%P – 0.05%S – 0.09%V – 0.04%Ti – 0.05%Nb – 0.4%Cu – 0.19%Co (all in mass%), was hot forged at 1373 K followed by air cooling. The starting material was characterized by an average grain size of 21 μm and an average dislocation density of $2.3 \times 10^{12} \text{ m}^{-2}$. The rod samples were heated in a muffle furnace to desired temperature in the range of 773–1173 K (0.45–0.7 Tm). Then, the multiple bar rolling was carried out with a pass strain of 0.25 to a total cumulative strain of 2 followed by water quenching. The steel rods were reheated in about 5 min to the processing temperature after each 0.5 strain increment.

Structural observations were performed on the sections parallel to the rolling axis, using a Quanta 600 FEG scanning electron microscope equipped with an electron back scattering diffraction pattern (EBSP) analyser incorporating an orientation imaging microscopy (OIM) system. The OIM images were subjected to clean up procedure setting a minimal confidence index of 0.1. The mean grain size was evaluated on the OIM micrographs as average distance between high-angle boundaries with misorientation of $\theta \geq 15^\circ$. The mechanical properties of processed samples were evaluated by means of isothermal tensile tests at ambient and elevated temperatures up to 973 K by using flat specimens with a gauge length of 12 mm and cross section of $3.0 \times 1.5 \text{ mm}^2$. The specimens were heated/held at a test temperature during 15 min using tensile device, which has been preheated to the desired temperature.

3. Results and discussion

3.1. Deformation microstructures

Typical microstructures that developed by multiple warm bar rolling to a total strain of 2 at different temperatures ranging from 773 to 1173 K are shown in Fig. 1. The starting microstructure is also shown in Fig. 1a for a reference. The developed deformation microstructures depend substantially on the rolling temperature. The microstructure that evolves during rolling at the lowest studied temperature of 773 K within the studied domain is characterized by an elongation of original grains along the rolling axis and corresponding axial alignment of strain-induced grain boundaries (Fig. 1b). As a result, the developed microstructure consists of highly elongated grains with the mean transverse grain size of 0.85 μm . The presence of well developed spatial sub-boundary net, some portions in which exhibit high-angle misorientations and look like incomplete grain boundaries in the grain interiors, is indicative of continuous DRX as the main mechanism responsible for the microstructure evolution at 773 K. An increase in the rolling temperature to 973 K results in an increase in the transverse grain size to 1.2 μm that developed at a total strain of 2 (Fig. 1c). Similar to the microstructure developed by rolling at 773 K, the deformation microstructure evolved at 973 K consists of highly elongated grains, which are interleaved with chains of the fine grains. Also, there are many separate incomplete grain boundaries of deformation origin in the grain interiors that is typical of continuous DRX [12,13], although the frequently corrugated grain boundaries suggest a possibility of partial contribution of discontinuous DRX to the development of new fine grains at boundaries of elongated grains. The microstructure that develops during rolling at 1173 K is clearly distinguished by rather large

grains; the transverse grain size comprises 3.4 μm (Fig. 1d). In contrast to the lamellar-type microstructures developed during rolling at lower temperatures, the steel sample subjected to rolling at 1173 K is characterized by almost equiaxed grains with sparse subboundaries in their interiors. Such equiaxed grains can result from discontinuous dynamic and post-dynamic recrystallization during interpass preheating to high temperature of 1173 K. Therefore, the discontinuous recrystallization is considered as the main mechanism of microstructure evolution during rolling at the highest studied temperature of 1173 K.

Since the flow stress can be expressed by a power law function of temperature-compensated strain rate, $Z = \dot{\epsilon} \exp(-Q/RT)$, where $\dot{\epsilon}$ is the strain rate, Q is the activation energy, R and T are the universal gas constant and temperature, respectively [14,15], the mean grain size (D) developed by DRX at sufficiently large strains should also obey power law functions of Z. The following relations have been obtained for discontinuous and continuous DRX domains, respectively, $D \sim Z^{-0.4}$ and $D \sim Z^{-0.1}$ [8,11]. The relationship between the DRX grain size and the temperature-compensated strain rate is shown in Fig. 2 for the present steel along with some previously reported results [15–20]. The value of Z around 10^{12} s^{-1} has been reported as a separation point between discontinuous and continuous DRX [8,20]. Namely, the discontinuous DRX takes place during hot working at $Z < 10^{12} \text{ s}^{-1}$, whereas continuous DRX develops at $Z > 10^{12} \text{ s}^{-1}$. This conclusion has been drawn after studying a dispersed strengthened stainless steel [20,21]. The present results indicate that the DRX behavior may be alternated by the phase content. The present 316L-type steel is characterized by lowered number of dispersed particles. As a result, the grain size of this steel is larger and transition from discontinuous to continuous DRX occurs at higher Z than for particle-bearing steel. It can be concluded, therefore, that rolling of the present steel at 1173 K corresponds to hot working conditions, when the flow stress depends remarkably on the temperature and strain rate. In contrast, the rolling temperatures of 773–973 K correspond to continuous DRX domain, which is characterized by a quite weak temperature/strain rate dependence of the flow stress.

3.2. Mechanical properties

Representative engineering stress-strain curves obtained by tensile tests at different temperatures for the 316L-type austenitic stainless steel subjected to warm to hot rolling to a total strain of 2 are shown in Fig. 3; and the main tensile properties of the steel at various temperatures are listed in Table 1. Rolling at temperatures of 773–1173 K resulted in remarkable strengthening of the steel; and the strength increment increases with a decrease in the rolling temperature. The room temperature yield strength varies from 720 MPa to 945 MPa depending on the rolling temperature within the studied temperature range. Correspondingly, the strengthening is accompanied by a degradation of plasticity. The total elongation at room temperature decreases to about 10% as the rolling temperature decreases to 773 K. Note that uniform elongation in this case does not exceed a couple of percents. Such tensile behavior is associated with the effect of the rolling temperature on the hardening ability during subsequent tensile tests. The steel sample subjected to rolling at relatively high temperature of 1173 K exhibits large strain hardening, which expands after yielding to 10% of elongation. In contrast, the tensile flow stress sharply increases to its maximum at about 2% of elongation followed by a necking and apparent softening until failure for the samples rolled at 773–973 K. The tensile strength of the steel decreases with an increase in the test temperature. It should be noted that this strength decrease with temperature is not accompanied by any increase in plasticity. Instead, the total and uniform elongations decrease with increasing the test temperature

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