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Electronic structures and mechanical properties of iron borides from first principles



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1. Introduction

Iron borides have attracted much attention, not only because of the potential to serve as a hardening agent in steels [1] or as hard protective coatings [2,3], but also because of the possibility of finding brand new stable Fe-B phases with appealing properties. Strikingly, Kolmogorov et al. [4] used advanced compound prediction methods to identify two previously unknown structures FeB₄ (space group Pnnm) and FeB₂ (space group Pnma). Moreover, the former was predicted to have the necessary features to exhibit phonon mediated superconductivity with a T_c of 15–20 K while the latter was shown to be the first metal diboride semiconductor. By examining their thermodynamic factors that could promote the formation, FeB₄ and FeB₂ were stabilized further under pressure [5]. Subsequently, Gou et al. [6] successfully synthesized the novel FeB₄ at pressure above 8 GPa and high temperatures. They not only confirmed FeB_4 to exhibit bulk superconductivity below 2.9 K, but also reported that it should belong to the group of superhard materials ($H_{\nu} > 40$ GPa), thus bridging the gap between the superhardness and superconductivity community.

Generally, superhard materials such as diamond and cubic boron nitride (*c*BN) are insulators or semiconductors with covalent bonds. Hardness depends strongly on plastic deformation, which brings about electron-pair bonds being broken and remade in a covalent crystal. Breaking an electron-pair bond means that two

ABSTRACT

The structural properties, mechanical behaviors and electronic structures of FeB_4 and FeB_2 have been studied systematically by first-principles calculations considering the strong correlation effect. Our results show that FeB_4 is incompressible and hard, but the recently reported superhard feature [Phys. Rev. Lett. 111 (2013) 157002] is not supported by the present calculations. Interestingly, we find that FeB_2 rivals FeB_4 in hardness. By analyzing their crystal geometries, band structures and density of states, we elucidate the underlying origins of the related physical properties.

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electrons become energetically excited from the valence band to the conduction band, so the intrinsic hardness of a single crystal is often related to its electronic structure, particularly to the band gap [7]. However, the measured nanoindentation hardness of FeB₄ was reported to be 62 GPa and its microhardness measurement with a load of 20 N gave the value of Vickers hardness ranging from 43 to 70 GPa [6]. All values in this range are exceptionally high, not only exceeding that of other hard metal-based compounds including OsB₂ (16.8–23.5 GPa), ReB₂ (26.6–48 GPa), and WB₃ (28.1–43.3 GPa), but also that of rivaling cBN (45–50 GPa) [8–15]. It therefore is a bit surprising that a superconducting metallic system is so hard. Because of the technical difficulties, no adequately sized single crystals of FeB₄ may be obtained, and thus its mechanical properties need to be confirmed theoretically.

In order to fully clarify mechanical properties of FeB₄, the theoretical calculations that can provide further details are highly desirable. So far, only two theoretical reports on the mechanical properties of FeB₄ were available [16,17], but their predicted hardness values (24.2 GPa, 25.1 GPa) are much lower than the measured data (43–70 GPa). As we know, Fe is one of the 3*d* transition metals with strong correlations. Although their calculations provide useful information for the understanding of mechanical characters of FeB₄, they neglect the strong correlations. Thus, the effect of strong correlations on mechanical properties of FeB₄ is unknown. Furthermore, less attention has been paid to the band structure of FeB₄, which is critical to understanding the related physical properties. On the other hand, FeB₂ is currently raising great expectations for superhard materials since it combines the semiconducting feature, high stability and

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compact structure with the prediction of a very large bulk modulus (311 GPa) [5], but its mechanical properties and electronic structures are not fully explored. In the present work, we first perform systematic first-principles calculations to study mechanical properties of FeB₄ and FeB₂. Our calculations indicate that FeB₄ is incompressible and hard, but do not support the experimentally measured superhardness [6]. It is found that the hardness of FeB₂ matches that of FeB₄. Then, the structural and electronic origins of the related physical properties of FeB₄ and FeB₂ are elucidated.

Calculations on FeB₄ and FeB₂ were performed using the projector augmented wave (PAW) method and the Vienna *ab initio* simulation package (VASP) [18]. We have conducted very careful tests for the convergence of calculated results with respect to the *k*-points number and the cutoff energy. A large cutoff energy of 500 eV was adopted and the dense Monkhorst–Pack *k*-points meshes were generated with the $12 \times 10 \times 14$ and $10 \times 12 \times 14$ grids for FeB₄ and FeB₂, respectively. We allowed spin polarization and did not find the existence of local magnetic moments on Fe atoms of FeB₄ and FeB₂. To compare the effect of strong correlations on their mechanical properties and electronic structures, we have used two approximations, the generalized gradient approximation (GGA) [19] and the GGA together with Hubbard model (GGA+*U*) [20] with effective Hubbard *U* (after subtraction of the exchange parameter *J*) of 4 eV.

Structural optimizations and total-energy calculations for FeB₄ and FeB₂ have been performed. The equilibrium volume (V_0) , bulk modulus (B_0) and its pressure derivative (B_0) are obtained by the least-square fit of calculated volume-energy sets to the Birch-Murnaghan equation of states [21]. The complete set of elastic constants and other mechanical properties (shear modulus, Young's modulus, and Poisson's ratio) are evaluated through our developed methods, which have been demonstrated to be very good in providing accurate and reliable predictions of mechanical properties of various systems [22-24]. The obtained results are summarized in Table 1. First, let us look at EOSs of FeB₄ and FeB₂. The lattice constants and bulk moduli with the GGA+U calculations are slightly smaller than those with GGA within an error of 2%. For FeB₄, the bulk moduli for the cases with GGA and GGA+Uare 268 GPa and 263 GPa, respectively. Both values accord with the experimental value (252 GPa) [6], but are smaller than those of WB_3 (295 GPa) [13] and ReB_2 (356 GPa) [10]. Interestingly, the bulk modulus of FeB₂ (311-314 GPa) is larger than that of FeB₄ and WB₃, even rivals that of ReB₂. To further compare the incompressibility of FeB₄, FeB₂, WB₃ and ReB₂ under pressure, their volume compressions as a function of pressure are presented in Fig. 1(a). We can explicitly see the order of incompressibility from low to high: $FeB_4 < WB_3 < FeB_2 < ReB_2$, although they are all incompressible materials. Then, we discuss the elastic constants of FeB₄ and FeB₂. It is noticed that the maximum difference between our GGA and GGA+U calculations totals to up to 18.6%, thus the effect of strong correlations on the mechanical properties should be considered. Large values of FeB₄ (C_{11} =429 GPa, C_{22} =790 GPa, C_{33} = 487 GPa) and FeB₂ (C_{11} =599 GPa, C_{22} =730 GPa, C_{33} =579 GPa) demonstrate that both phases possess high linear incompressibility. However, there is substantial anisotropy in the compressibility of FeB₄ and FeB₂. Their anisotropic behaviors can be observed from their axial compressions as function of pressure, as shown in Fig. 1(b). It is obvious that the *c* axis of FeB₄ is the most incompressible while the *a* axis of FeB₄ is the most compressible. At the same time, the shear properties of FeB₄ (C_{44} =220 GPa, C_{55} =171 GPa, C_{66} =235 GPa) and FeB₂ (C_{44} =275 GPa, C_{55} = 199 GPa, C_{66} =230 GPa) also exhibit much anisotropy, although they have high shear stiffness.

For FeB₄, its shear and Young's moduli (G=207 GPa, E=495 GPa) are much larger than those of Fe₂B (G=143 GPa, E=355 GPa) and FeB (G=157 GPa, E=399 GPa) [5], but are much smaller than those of WB₃ (G=252 GPa, E=588 GPa) [13] and ReB₂ (G=293 GPa, E=691 GPa) [10], respectively. On the contrary, the Poisson's ratio of FeB_4 (0.20) is smaller than that of Fe_2B (0.24) and FeB (0.27) [5], but is larger than that of WB₃ (0.168) [13] and ReB₂ (0.181) [10]. These results suggest that FeB₄ is superior to Fe₂B and FeB from the aspect of mechanical properties, but is inferior to WB₃ and ReB₂. According to the recently proposed hardness model [25], we have further estimated the Vickers hardness H_{ν} of FeB₄ to be 25.5 GPa from the GGA calculation. This value is close to the previous predictions (24.2 GPa, and 25.1 GPa) [16,17] but largely deviates from the experimentally reported value (43–70 GPa) [6]. Considering the strong correlation effect, our GGA+U calculations have presented a large hardness value of 31.2 GPa, which is still much smaller than the measured one. Therefore, FeB₄ is an incompressible and hard material but not a superhard one, at odds with the conclusion of Gou et al. [6].



Fig. 1. (Color online) (a) Calculated volume compression V/V_0 and (b) axis compression a/a_0 , b/b_0 , c/c_0 of FeB₄ and FeB₂ as a function of pressure.

Table 1

Calculated equilibrium lattice parameters a_0 (Å), b_0 (Å), c_0 (Å), bulk modulus B_0 (GPa), its pressure derivative B_0' , elastic constants C_{ij} (GPa), shear modulus G (GPa), Young's modulus E (GPa) and Poisson's ratio v for FeB₄ and FeB₂.

	<i>a</i> ₀	b ₀	<i>c</i> ₀	Bo	<i>B</i> ₀ ′	C ₁₁	C ₂₂	C ₃₃	C ₁₂	C ₁₃	C ₂₃	C ₄₄	C ₅₅	C ₆₆	G	Е	ν
FeB ₄ GGA GGA+U	4.517 4.500	5.285 5.265	3.007 2.998	268 263	4.02 3.91	409 429	768 790	451 487	161 131	161 137	152 130	216 220	154 171	222 235	189 207	462 495	0.22 0.20
FeB ₂ GGA GGA+U	4.815 4.772	4.805 4.775	3.738 3.698	314 311	4.14 4.10	592 599	723 730	574 579	185 179	160 154	136 129	271 275	198 199	221 230	230 235	555 564	0.21 0.19

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