



Effect of HfB_2 on microstructure and mechanical properties of ZrB_2 –SiC-based composites

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ABSTRACT

In this work, ZrB_2 –SiC-based composites with different additives were sintered in different temperature (1600, 1700, 1800 and 1900), time (4, 8, 12 and 16) and pressure (10, 20, 30 and 40 MPa). The effect of HfB_2 (0, 5, 10, 15 vol.%) on ZrB_2 -based ceramics was investigated. Microstructure evaluation and phases were done by SEM and EDAX. Effect of SPS conditions (temperature, time and pressure) on microstructure was investigated. The results showed that Hf and Zr diffuse in each other and (Zr, Hf) B_2 solid solution will be formed. In addition, it was concluded that by increasing temperature and time, solid solution formation increases. Pressure has little effect on diffusion. Also, MoSi_2 and ZrB_2 form (Zr, Mo) B_2 solid solution. Finally it is cleared that HfB_2 increases open porosity percent slightly. Finally, it was observed that HfB_2 has negative effect on densification while improves flexural strength due to (Zr, Hf) B_2 solid solution.

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1. Introduction

IV–V-group transition-metal diborides and carbides possess melting temperatures among the highest known, i.e. above 3000 °C. Among them diborides such as ZrB_2 and HfB_2 have unique combinations of mechanical and physical properties, including high hardness and elastic modulus, low electrical resistivity, and resistance to chemical attack. As a result, this material has been proposed for a variety of structural applications at room and elevated temperature including armor, cutting tools, molten metal containment, steel processing, and electrodes. Accordingly lower density and cost of ZrB_2 rather than HfB_2 , ZrB_2 is considered to be a candidate for use as leading edges and propulsion components in hypersonic aerospace vehicles and advanced reusable atmospheric reentry vehicles [1–3].

To improve mechanical properties of ZrB_2 different additives such as metals, Cu, Ni, Fe, Mo, Nb [4], nitrides, AlN, HfN, ZrN [5,6] Si_3N_4 , carbides, SiC and ZrC [2,7] and disilicides, MoSi_2 , ZrSi_2 , and TaSi_2 [5,6,8] were applied. SiC reinforcement of ZrB_2 is known to increase flexural strength, fracture toughness and oxidation resistance and so was chosen as additive in this study. Although additives, such as Ni, nitrides and disilicides, could reduce the densification temperature of ZrB_2 , the strength of the resulting composites is also degraded at high temperature, as a result

of softening of the inter-granular amorphous phase [2]. As an example, the strength of ZrB_2 containing ~3 vol.% Ni has been shown to drop by about 60% between 800 and 1000 [9].

Among the different additives which are mentioned above HfB_2 has positive effect on mechanical properties in high temperature as well as low temperature. Until now effect of it on microstructure and densification was not investigated. The aim of this work is to study and discuss it. Also, MoSi_2 is used as additives due to its good effect on densification. Effect of MoSi_2 on mechanical properties is not expressed clearly.

2. Experimental

The aim of this project is to reach composite with the best densification and mechanical properties (such as flexural strength, hardness and fracture toughness). So, all important factors such as SiC, C_f , MoSi_2 , HfB_2 and ZrC content, milling time of C_f and SPS parameters such as temperature, time and pressure were chosen according to previous studies [2, 10–18]. Four levels were selected for each of them. The used factors and levels are presented in Table 1.

To investigate all factors and levels, $4^9 = 262,144$ samples are needed in which manufacturing of them is impossible due to very high cost and time. So, an optimization strategy is necessary to evaluate all of these factors. Taguchi method is the best opportunity to eliminate variations during the design of experiment. In this study, L_{32} orthogonal array was chosen by Qualitek-4 software. With respect to the L_{32} orthogonal array, conditions in preparing each sample will be as shown in Table 2. The powders corresponding to 32 different compositions,

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Table 1

Process factors and their levels used in the experiments.

Test	FAC.1 SiC vol.%	FAC.2 C _f vol.%	FAC.3 Milling time h	FAC.4 MoSi ₂ vol.%	FAC.5 HfB ₂ vol.%	FAC.6 ZrC vol.%	FAC.7 Temperature C	FAC.8 Pressure MPa	FAC.9 Time min
Lev.1	5	0	0	0	0	0	1600	10	4
Lev.2	10	2.5	2.5	2	5	5	1700	20	8
Lev.3	15	5	5	4	10	10	1800	30	12
Lev.4	20	7.5	7.5	6	15	15	1900	40	16

according to Table 2, were mixed by wet ball-milling at 200 rpm for 3 h in a zirconia's bottle, using zirconia's balls and ethanol as media. The mixtures were then dried. The powder mixture was put into graphite die lines with graphite foil which has an inner diameter of 50 mm and sintered using SPS apparatus (SPS-20T-10, China). The sintering was performed at different temperatures (between 1600 °C and 1900 °C), different pressures (between 10 MPa and 40 MPa) and different holding times (between 4 min and 16 min) based on Taguchi design (Table 2) in vacuum. Final sintered specimen size was 50 mm diameter pellets with different thicknesses of 2.8 mm and 6.5 mm related to their compositions and SPS conditions. After removing the surface layer from the obtained disk by grinding, the bulk density was measured according to ASTM C 373-88. Flexural strengths were measured in three-point bending using a fully articulated test fixture according to ASTM Standard C1161 for type-B bars (45 mm by 3 mm by 4 mm). The phase composition was determined by X-ray diffraction (XRD, Siemens, D500) using Cu Ka radiation on polished cross-sectioned composite. Scanning electron microscopy (Sigma/VP, Zeiss) was performed to observe the microstructures of the composites equipped with backscattered electron imaging (BSE).

3. Results and discussion

3.1. Microstructure

SEM images of some composites are shown in Fig. 1. To identify phases in microstructure EDAX was applied. Fig. 2 shows EDAX analysis of composites 11 and 22. It is clear that the light phases (D, E) which are shown with arrows, have higher Hf content. Also, it can be observed that C and B phases have the most Si and Zr rather than other phases respectively. In composite 22, gray dark phase has the most Zr (spectrum 2) and approximately the light phase has the most Hf (spectrum 1). Since determining accurate content of B and C is impossible, line as well as point analysis was applied. Line analysis of composite 22 is shown in Fig. 3.

With respect to point and line analysis (Figs. 2 and 3), it appears that black grains are SiC due to high Si content. Gray dark grains are pure ZrB₂ because of their high Zr content and free of Hf. Totally, by investigating accurately microstructure of all composites, five phases with different color, light, approximately light, gray light, gray dark and black can be distinguished. As explained above, gray dark and black phases

Table 2

Conditions preparing for each sample.

Test	FAC.1 SiC vol.%	FAC.2 C _f vol.%	FAC.3 M.t (h)	FAC.4 MoSi ₂ vol.%	FAC.5 HfB ₂ vol.%	FAC.6 ZrC vol.%	FAC.7 Temperature C	FAC.8 Pressure MPa	FAC.9 Time min
1	5	0	0	0	0	0	1600	10	4
2	5	2.5	2.5	2	5	5	1700	20	8
3	5	5	5	4	10	10	1800	30	12
4	5	7.5	7.5	6	15	15	1900	40	16
5	10	0	0	2	5	10	1800	40	16
6	10	2.5	2.5	0	0	15	1900	30	12
7	10	5	5	6	15	0	1600	20	8
8	10	7.5	7.5	4	10	5	1700	10	4
9	15	0	2.5	4	15	0	1700	30	16
10	15	2.5	0	6	10	5	1600	40	12
11	15	5	7.5	0	5	10	1900	10	8
12	15	7.5	5	2	0	15	1800	20	4
13	20	0	2.5	6	10	10	1900	20	4
14	20	2.5	0	4	15	15	1800	10	8
15	20	5	7.5	2	0	0	1700	40	12
16	20	7.5	5	0	5	5	1600	30	16
17	5	0	7.5	0	15	5	1800	20	12
18	5	2.5	5	2	10	0	1900	10	16
19	5	5	2.5	4	5	15	1600	40	4
20	5	7.5	0	6	0	10	1700	30	8
21	10	0	7.5	2	10	15	1600	30	8
22	10	2.5	5	0	15	10	1700	40	4
23	10	5	2.5	6	0	5	1800	10	16
24	10	7.5	0	4	5	0	1900	20	12
25	15	0	5	4	0	5	1900	40	8
26	15	2.5	7.5	6	5	0	1800	30	4
27	15	5	0	0	10	15	1700	20	16
28	15	7.5	2.5	2	15	10	1600	10	12
29	20	0	5	6	5	15	1700	10	12
30	20	2.5	7.5	4	0	10	1600	20	16
31	20	5	0	2	15	5	1900	30	4
32	20	7.5	2.5	0	10	0	1800	40	8

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