



Microwave dielectric properties of low loss $\text{Li}_2(\text{Mg}_{0.95}\text{A}_{0.05})_3\text{TiO}_6$ ($\text{A} = \text{Ca}^{2+}, \text{Ni}^{2+}, \text{Zn}^{2+}, \text{Mn}^{2+}$) ceramics system

ABSTRACT

Keywords:

Low loss
Ceramic
Microwave dielectric properties
 $\text{Li}_2\text{Mg}_3\text{TiO}_6$

Ultra low loss microwave dielectric materials of $\text{Li}_2(\text{Mg}_{0.95}\text{A}_{0.05})_3\text{TiO}_6$ ($\text{A} = \text{Ca}^{2+}, \text{Ni}^{2+}, \text{Zn}^{2+}, \text{Mn}^{2+}$) ceramics were investigated through conventional solid-state reaction method. The effects of different bivalent A^{2+} ($\text{A}^{2+} = \text{Ca}^{2+}, \text{Ni}^{2+}, \text{Zn}^{2+}, \text{Mn}^{2+}$) substitution for Mg on the phase composition, microstructure and microwave dielectric properties were systematically discussed. The XRD patterns indicated that the main peaks belonged to $\text{Li}_2\text{Mg}_3\text{TiO}_6$, and little impurities were formed. We have found that Zn-substitution for Mg, which could increase its $Q \times f$ value and lower the $|\tau_f|$ compared to that of pure $\text{Li}_2\text{Mg}_3\text{TiO}_6$, had a significant effect on the $Q \times f$ and τ_f of the ceramics. $\text{Li}_2(\text{Mg}_{0.95}\text{A}_{0.05})_3\text{TiO}_6$ ceramics specimens with $\text{A} = \text{Zn}^{2+}$ sintered at 1275 °C for 6h exhibited excellent microwave dielectric properties of $\epsilon_r \sim 14.6$, $Q \times f \sim 158,000\text{GHz}$ (at 9.11GHz) and a near-zero $\tau_f \sim -3.2 \text{ ppm}/^\circ\text{C}$.

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1. Introduction

With the rapid development of mobile communications, microwave dielectric ceramics play an important role in multilayer or chip devices and attract much attention. These ceramics must fulfil a high quality factor ($Q \times f$), an appropriate dielectric constant (ϵ_r) and a near-zero temperature coefficient of resonant frequency (τ_f) [1,2]. However, many ceramics cannot meet these requirements simultaneously. Therefore, new material with excellent properties is still a hot issue for modern industry.

In previous works, the lithium based oxide ceramics such as $\text{Li}_2\text{MgTi}_3\text{O}_8$, $\text{Li}_2\text{MgTiO}_4$ and $\text{Li}_2\text{Mg}_3\text{BO}_6$ ($\text{B} = \text{Ti}, \text{Sn}, \text{Zr}$) possess excellent microwave dielectric properties [3–5]. George and Sebastian firstly reported that the $\text{Li}_2\text{MgTi}_3\text{O}_8$ ceramics was produced by reaction-sintering method and had a good microwave dielectric properties of $\epsilon_r \sim 27.2$, $Q \times f \sim 42000\text{GHz}$ and $\tau_f \sim 3.2 \text{ ppm}/^\circ\text{C}$ [3]. According to Su et al. [6], with the partial replacement of Mg^{2+} by Zn^{2+} , the $\text{Li}_2(\text{Mg}_{0.94}\text{Zn}_{0.06})\text{Ti}_3\text{O}_8$ ceramics exhibited good microwave dielectric properties of $\epsilon_r \sim 27.1$, $Q \times f \sim 44,800 \text{ GHz}$ and $\tau_f \sim 1.9 \text{ ppm}/^\circ\text{C}$. $\text{Li}_2\text{Mg}_3\text{TiO}_6$ ceramics was reported to exhibit excellent microwave dielectric properties ($\epsilon_r \sim 15.2$, $Q \times f \sim 152,000\text{GHz}$, $\tau_f \sim -39 \text{ ppm}/^\circ\text{C}$) when sintered at 1280 °C for 6h [5]. Afterwards, in order to meet the requirements for LTCC, the 4 wt% LiF addition was used to lower the sintering temperature of $\text{Li}_2\text{Mg}_3\text{TiO}_6$ ceramics and achieved a good dielectric properties of $\epsilon_r \sim 16.2$, $Q \times f \sim 131,000 \text{ GHz}$ and $\tau_f \sim -44 \text{ ppm}/^\circ\text{C}$ when sintered at 950 °C [7]. However, there were few work about the effects of different divalent ions substitution on microwave dielectric properties of $\text{Li}_2(\text{Mg}_{0.95}\text{A}_{0.05})_3\text{TiO}_6$ ($\text{A} = \text{Ca}^{2+}, \text{Ni}^{2+}, \text{Zn}^{2+}, \text{Mn}^{2+}$) ceramics.

In this paper, the effect on the microwave dielectric properties of $\text{Li}_2\text{Mg}_3\text{TiO}_6$ ceramics resulting from isovalent substitution of A

($\text{A} = \text{Ca}^{2+}, \text{Ni}^{2+}, \text{Zn}^{2+}, \text{Mn}^{2+}$) at the Mg site was firstly investigated. In addition, the X-ray diffraction (XRD) pattern and the scanning electron microscopy (SEM) analysis were used to analyze the microstructures.

2. Experimental procedure

$\text{Li}_2(\text{Mg}_{0.95}\text{A}_{0.05})_3\text{TiO}_6$ ($\text{A} = \text{Ca}^{2+}, \text{Ni}^{2+}, \text{Zn}^{2+}, \text{Mn}^{2+}$) ceramics were prepared using high-purity oxide powders of Li_2CO_3 (99%), MgO (99.9%), CaCO_3 (99.9%), ZnO (99.9%), NiO (99%), MnCO_3 (99%) and TiO_2 (99.9%). The raw materials were mixed according to the formula of $\text{Li}_2(\text{Mg}_{0.95}\text{A}_{0.05})_3\text{TiO}_6$ ($\text{A} = \text{Ca}^{2+}, \text{Ni}^{2+}, \text{Zn}^{2+}, \text{Mn}^{2+}$). The mixed powders were milled with ZrO_2 balls for 8h in distilled water. All mixtures were dried and calcined at 1000 °C for 4h. The calcined powders were remilled with ZrO_2 balls for 8h in distilled water, and then dried. These mixtures were mixed together with 6 wt% paraffin as a binder. Afterwards, the granulated powders were pressed into disks with 10mm in diameter and about 5 mm in height. Finally, these cylinder were sintered at 1225–1300 °C for 6 h in air.

The crystalline phases of the sintered samples were investigated by X-ray diffraction (XRD, Rigaku D/max 2550 PC, Tokyo, Japan) with Cu K α radiation generated at 40kV and 40mA. The microstructure of the ceramic surfaces were performed and analyzed by a scanning electron microscopy (SEM, ZEISS MERLIN Compact, Germany). The microwave dielectric properties of sintered specimen were measured by a network analyzer (N5234A, Agilent Co, America) in the frequency range of 8–13 GHz. The dielectric constants were measured by exciting the TE_{01} resonant mode of dielectric resonator as suggested by Hakki-Coleman [8]. The unload quality factors were measured using TE_{01} mode by the cavity method [9].

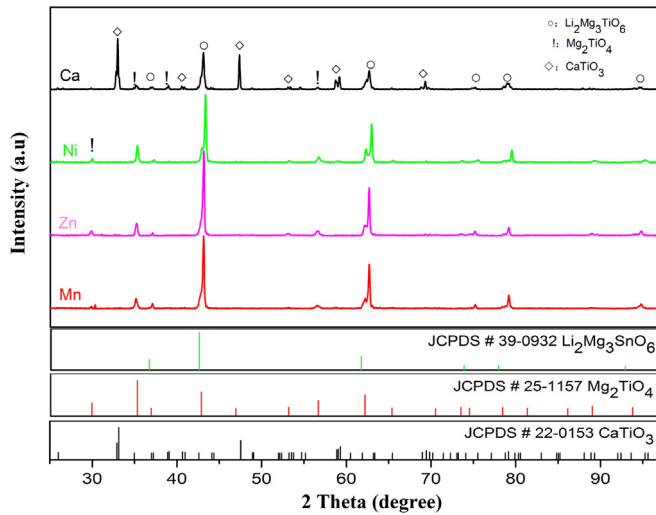


Fig. 1. The X-ray diffraction patterns of $\text{Li}_2(\text{Mg}_{0.95}\text{A}_{0.05})_3\text{TiO}_6$ ($\text{A} = \text{Ca}^{2+}, \text{Ni}^{2+}, \text{Zn}^{2+}, \text{Mn}^{2+}$) ceramics sintered at 1275 °C.

The temperature coefficient of resonant frequency (τ_f) was measured in the temperature range from 25 °C to 85 °C and was calculated by the following formula:

$$\tau_f = \frac{f_{85} - f_{25}}{f_{25}(85 - 25)} \times 10^6 (\text{ppm}/^\circ\text{C}) \quad (1)$$

where f_{85} and f_{25} were the resonant frequencies at 85 °C and 25 °C.

3. Results and discussions

The XRD patterns of $\text{Li}_2(\text{Mg}_{0.95}\text{A}_{0.05})_3\text{TiO}_6$ ($\text{A} = \text{Ca}^{2+}, \text{Ni}^{2+}, \text{Zn}^{2+}, \text{Mn}^{2+}$) ceramics sintered at 1275 °C for 6h are shown in Fig. 1. All samples exhibited the $\text{Li}_2\text{Mg}_3\text{SnO}_6$ -like (JCPDS#39-0932) cubic phase with ordered rock salt structure. The main peaks of $\text{Li}_2\text{Mg}_3\text{TiO}_6$ shifted slightly to the higher angle than $\text{Li}_2\text{Mg}_3\text{SnO}_6$ phase due to the smaller ionic radius of Ti^{4+} ($R = 0.605 \text{ \AA}$, CN = 6) than that of Sn^{4+} ($R = 0.69 \text{ \AA}$, CN = 6) [5]. Besides, little impurity peaks defined as Mg_2TiO_4 were formed because lithium was easily volatile when the sintering temperature was higher than 1000 °C [10]. Specially, the peaks of CaTiO_3 were found in $\text{Li}_2(\text{Mg}_{0.95}\text{Ca}_{0.05})_3\text{TiO}_6$. Similar phenomena were reported in CaCO_3 -doped $\text{Li}_2\text{MgTi}_3\text{O}_8$ and $\text{Li}_2\text{ZnTi}_3\text{O}_8$ ceramics systems [11].

Fig. 2 presents the SEM photographs of $\text{Li}_2(\text{Mg}_{0.95}\text{A}_{0.05})_3\text{TiO}_6$ -

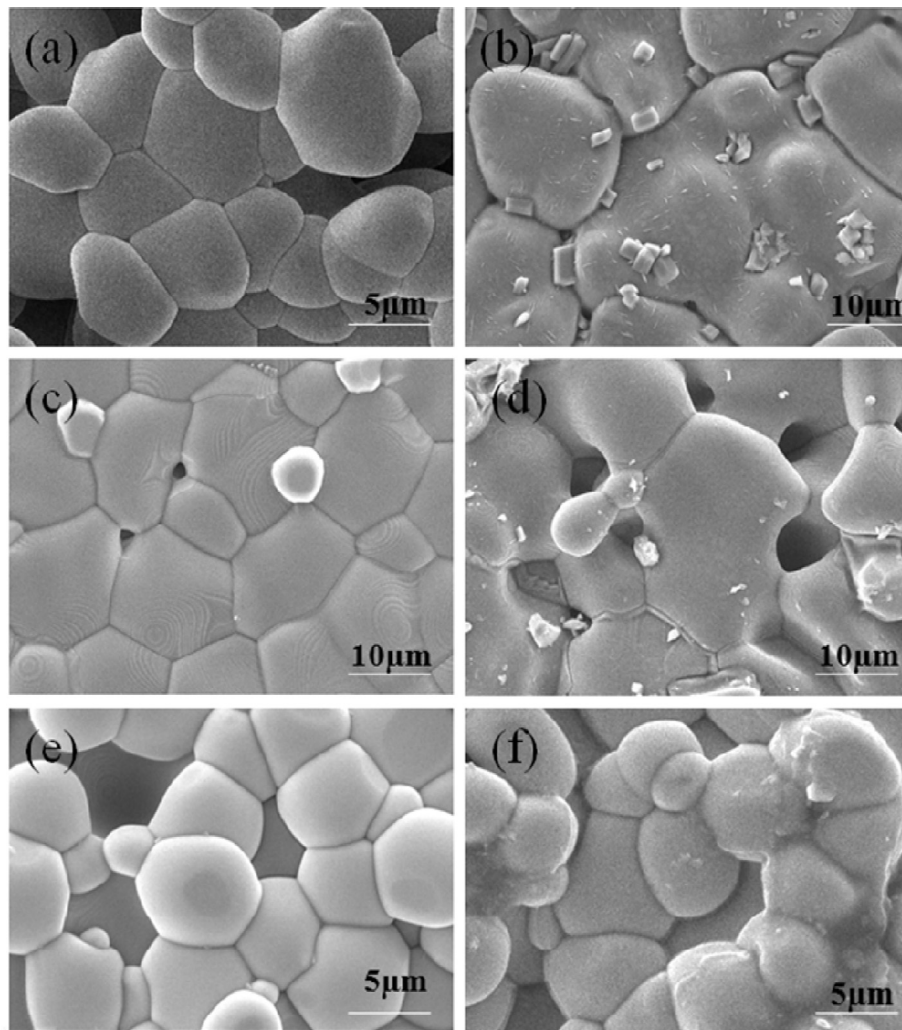


Fig. 2. The surface microstructural photographs of $\text{Li}_2(\text{Mg}_{0.95}\text{A}_{0.05})_3\text{TiO}_6$ ($\text{A} = \text{Ca}^{2+}, \text{Ni}^{2+}, \text{Zn}^{2+}, \text{Mn}^{2+}$) ceramics: (a) $\text{A} = \text{Ca}$, (b) $\text{A} = \text{Ni}$, (c) $\text{A} = \text{Zn}$, (d) $\text{A} = \text{Mn}$ sintered at 1275 °C, and $\text{Li}_2(\text{Mg}_{0.95}\text{Ca}_{0.05})_3\text{TiO}_6$ ceramics sintered at (e) 1250 °C and (f) 1300 °C.

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