#### Journal of Alloys and Compounds 680 (2016) 359-365

Contents lists available at ScienceDirect



Journal of Alloys and Compounds

journal homepage: http://www.elsevier.com/locate/jalcom

# Structure, magnetic and magnetothermal properties of the non-stoichiometric $ErCo_2Mn_x$ alloys



CrossMark

ALLOYS AND COMPOUNDS

攬

### imou\* NV Muchailou AA Inichou DD Torontou VC Couileo

E.G. Gerasimov<sup>\*</sup>, N.V. Mushnikov, A.A. Inishev, P.B. Terentev, V.S. Gaviko

Institute of Metal Physics UB RAS, S. Kovalevskaya str. 18, 620990, Ekaterinburg, Russia

#### ARTICLE INFO

Article history: Received 25 February 2016 Received in revised form 12 April 2016 Accepted 13 April 2016 Available online 26 April 2016

Keywords: Rare-earth intermetallics Magnetic ordering Exchange interactions Heat capacity Magnetocaloric effect

#### ABSTRACT

The structure, magnetic and magnetothermal properties and heat capacity of the  $\text{ErCo}_2\text{Mn}_x$  ( $0 \le x \le 1.4$ ) non-stoichiometric alloys have been studied. It is shown that the alloys with  $x \le 0.8$  are almost single-phase with the MgCu<sub>2</sub>-type Laves-phase structure, while the alloys for  $x \ge 1.0$  contain considerable amount of the cubic Th<sub>6</sub>Mn<sub>23</sub>-type phase. Alloying with manganese leads to a substantial increase in the exchange interactions. For the binary  $\text{ErCo}_2$  compound, the magnetization shows a first-order magnetic phase transition with the Curie temperature  $T_C = 35$  K. For all the Mn-containing alloys studied, the transition is found to be the second-order type; the Curie temperature in  $\text{ErCo}_2\text{Mn}_x$  sharply growths with increasing *x* and reaches its maximum value 212 K for the  $\text{ErCo}_2\text{Mn}_{0.6}$  alloy. Using the magnetization data and the results of the heat capacity measurements, we estimated the magnetocaloric effect for these compounds. The obtained results are discussed on the assumption of a strong variation of the 3*d*-metal magnetic moments upon alloying with Mn.

© 2016 Elsevier B.V. All rights reserved.

#### 1. Introduction

Intermetallic *R*Co<sub>2</sub> compounds (*R* is a rare earth metal, Y, or Sc) with the Laves phase structure of the MgCu<sub>2</sub>-type (space group  $Fd\overline{3}m$ ) are of considerable interest in view of the instability of the Co magnetic moment and itinerant electron metamagnetism of the d electrons of cobalt [1]. For binary RCo<sub>2</sub> compounds with nonmagnetic R elements (R = Sc, Y, Lu), the Stoner criterion for the 3d band ferromagnetism is very close to be fulfilled. Such compounds are exchange-enhanced Pauli paramagnets and show a metamagnetic transition in high magnetic fields [2]. The ground magnetic state of the cobalt sublattice changes to ferromagnetic under the influence of a molecular field produced by the rare earth atoms with a nonzero spin moment. Alternatively, the ferromagnetic state is formed as a result of modification of the d-band structure in quasi-binary  $R(Co_{1-x}M_x)_2$  compounds in which Co atoms are partially substituted by atoms of other elements M = AI, Ga, Mn, Si, Cr etc. [1-6]. An intensive search for materials for magnetothermal applications gives an additional interest to the studies of magnetic properties of the RCo<sub>2</sub> compounds, since the magnetocaloric effect can be rather large in these compounds

\* Corresponding author. E-mail address: gerasimov@imp.uran.ru (E.G. Gerasimov).

#### [7-10].

Recently, Wang et al. [11] discovered a series of novel compounds with a general formula  $RNi_2Mn$  (R = Tb, Dy, Ho, Er). These compounds crystallize in the MgCu<sub>2</sub>-type structure, though the atomic ratio of the rare earth to 3d-metal component is 1–3, at which the rhombohedral PuNi<sub>3</sub>-type structure is usually formed in binary RM<sub>3</sub> alloys. The RNi<sub>2</sub>Mn compounds are ferrimagnets and have a much higher temperature of magnetic ordering than the binary compounds RNi2 and RMn2. Later, it was reported [12] that the cobalt-containing alloys HoCo2Mn and ErCo2Mn with the 1:3 stoichiometry also possess the MgCu<sub>2</sub>-type structure. X-ray and neutron diffraction studies showed that the Mn atoms in Lavesphase structure of the RNi<sub>2</sub>Mn compounds partially occupy both the Ni (16d) and rare earth (8a) sites. Detailed studies of the TbNi<sub>2</sub>Mn alloys showed that the structure type depends on the conditions of the alloy preparation. As-cast arc-melted alloys show up the MgCu<sub>2</sub> structure [13], while in the samples annealed at 870 °C for several hours, the cubic AuBe<sub>5</sub>-type structure is formed [14]

Bearing in mind that both  $RNi_2$  and  $RNi_2Mn$  have the same structure, we studied magnetic properties and structure of the non-stoichiometric  $RNi_2Mn_x$  ( $0 \le x \le 1.5$ ) systems for R = Tb, Dy, Gd [15–17]. We have found that the MgCu<sub>2</sub>-type structure persists up to the manganese content x = 0.4 for R = Gd and up to x = 1.25 for R = Tb. The Curie temperature is a non-monotonous function of Mn

concentration which has a maximum far below the composition x = 1. Thus, the  $RM_2Mn_x$  (M = Co, Ni) systems give the opportunity to gradually modify magnetic properties of the Laves phase compounds by changing in a wide range the manganese content. In present paper, in order to clarify the effect of alloying with Mn on the structure and magnetic and magnetothermal properties of intermetallic Laves-phase compounds of rare earth elements with cobalt, we studied a series of non-stoichiometric  $ErCo_2Mn_x$  alloys for different  $x \leq 1.4$ .

#### 2. Experimental details

The ingots of the  $\text{ErCo}_2\text{Mn}_x$  alloys for x = 0, 0.2, 0.4, 0.6, 0.8, 1.0, 1.2, and 1.4 were prepared by induction melting of the constituents Er (99.9%), Co (99.99%), and Mn (99.9%) in alumina crucibles in argon atmosphere. To compensate for evaporating of volatile constituents, the excess of 1 wt% of Er and 3 wt% of Mn was added to the working mixture. In order to obtain the equilibrium phase state, the ingots were annealed at 900°C for five days.

Structural and magnetic studies were performed at the Centre of Collective Use of the Institute of Metal Physics UB RAS. X-ray diffraction analysis was carried out on powdered samples with the particle size  $30-50 \ \mu m$  using a DRON-type diffractometer with Cr K $\alpha$  radiation at room temperature. The X-ray diffraction patterns were interpreted with the use of the PowderCell 2.4 program.

For magnetic measurements, we used bulk samples of a spherical form. The magnetization was measured with the Quantum Design PPMS apparatus in magnetic fields up to 9 T in the temperature range 2–320 K. The temperature dependences of the initial ac magnetic susceptibility  $\chi_{a\geq}$  were measured with a system of compensated pick-up coils in a sinusoidal alternate magnetic field with the frequency of 80 Hz and amplitude 300 A/m. The heat capacity was measured with the low-temperature adiabatic calorimeter.

#### 3. Results and discussion

Fig. 1 shows X-ray diffraction patterns of the  $\text{ErCo}_2\text{Mn}_x$  alloys. For the compositions with  $x \le 0.4$ , the X-ray diffraction patterns can be well described on the assumption of the only cubic Laves phase (space group  $Fd\overline{3}m$ ). In the alloys with the Mn content  $x \ge 0.6$ , in addition to the Laves phase C15, we observe reflections that belong to a phase with the cubic Th<sub>6</sub>Mn<sub>23</sub>-type structure (space group  $Fm\overline{3}m$ ). The intensities of these reflections are, however, very weak for x = 0.6 and 0.8. According to our quantitative analysis, the volume fraction of the Th<sub>6</sub>Mn<sub>23</sub>-type phase does not exceed 10% for these compositions. For the alloys with  $x \ge 1.0$ , its amount increases and the ErCo<sub>2</sub>Mn<sub>14</sub> sample contains 63% of the Th<sub>6</sub>Mn<sub>23</sub>-type phase (see Fig. 2 and Table 1).

The lattice parameter of the parent ErCo<sub>2</sub> compound amounts to a = 0.7160, which is consistent with the previously reported values. The Mn-containing alloys are characterized by a slightly lower value of the lattice parameter that is virtually independent of the manganese content (Fig. 2). Such a behavior reflects the interplay of two opposite factors. Substitution of Co atoms at the 16d sites by Mn atoms with a larger metallic radius tends to increase the lattice volume. On the contrary, partial substitution of Er atoms by Mn at the 8*a* sites tends to decrease the lattice size. For comparison, for the *R*Ni<sub>2</sub>Mn<sub>x</sub> systems, this competition leads to a non-monotonous concentration dependence of the lattice parameter with a maximum at x = 0.4-0.6 [16,17]. The secondary phase formed in  $RNi_2Mn_x$  alloys at large x values is the PuNi<sub>3</sub>-type phase. It was surprising to discover the Th<sub>6</sub>Mn<sub>23</sub>-type phase in the RCo<sub>2</sub>Mn<sub>x</sub> alloys, since it does not form in the binary R-Co system. The lattice parameter of the Th<sub>6</sub>Mn<sub>23</sub> phase found in our alloy is considerably



**Fig. 1.** X-ray diffraction patterns of the  $ErCo_2Mn_x$  alloys with different *x*. Bragg peak positions are indicated by the markers for MgCu<sub>2</sub>-type structure (line bars), Th<sub>6</sub>Mn<sub>23</sub>-type structure (solid triangles) and Mn (open circles).



**Fig. 2.** Concentration dependences of the lattice parameter and the amount of the  $MgCu_2$ -type phase in  $ErCo_2Mn_x$  alloys.

smaller than that of the binary  $\text{Er}_6\text{Mn}_{23}$  compound (a = 1.2285 nm). Therefore, the phase is expected to have the composition  $\text{Er}_6(\text{Mn}_{1-x}\text{Co}_x)_{23}$  and contain both manganese and cobalt.

Thus, we succeed to obtain the samples with the amount of more than 90% of Laves phase in  $\text{ErCo}_2\text{Mn}_x$  alloys for  $x \le 0.8$  only. It should be noted that in Ref. [12] the authors reported on the  $\text{ErCo}_2\text{Mn}$  compound as a single MgCu<sub>2</sub> phase. However, the X-ray diffraction patterns presented in Ref. [12] contain reflections of the Th<sub>6</sub>Mn<sub>23</sub> phase the amount of which has not been estimated. The  $\text{ErCo}_2\text{Mn}$  alloy studied in Ref. [12] was prepared by arc melting with the 6 wt% excess of Mn and annealed at 800 °C for one week. The Mn solubility limit can depend to some extent on the alloying conditions, as well as the temperature and time of the heat

Download English Version:

## https://daneshyari.com/en/article/1605369

Download Persian Version:

https://daneshyari.com/article/1605369

Daneshyari.com