



First-principles study of the nickel–silicon binary compounds under pressure



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ABSTRACT

The effects of high pressure on structural, stable, elastic, thermodynamic properties and electronic structures of Ni–Si binary compounds (i.e. β -Ni₃Si, γ -Ni₃₁Si₁₂, δ -Ni₂Si, θ -Ni₂Si, ϵ -Ni₃Si₂, ϵ -NiSi and α -NiSi₂) have been firstly studied by first-principles calculation based on density functional theory method within generalized gradient approximation. The calculated lattice parameters of the seven compounds at zero pressure and zero temperature agree well with the available experimental values and previous theoretical data. The values of V/V_0 decrease with pressure going up to 50 GPa and the rate of change decrease gradually. The lattice parameters of NiSi₂ are the most sensitive to external pressure change. Ni₃Si, Ni₃₁Si₁₂, Ni₂Si (δ) and Ni₃Si₂ are mechanically stable by estimating stability criteria with pressure ranging from 0 to 50 GPa. But Ni₂Si (θ), NiSi and NiSi₂ are not mechanically stable with pressure up to 10, 40 and 30 GPa, respectively. The calculated results of bulk modulus B , shear modulus G and Young's modulus E illustrate pressure can improve the hardness of Ni₃Si, Ni₃₁Si₁₂, Ni₂Si (δ) and Ni₃Si₂ compounds. In addition, ratio of shear modulus to bulk modulus G/B shows that all the considered compounds are ductile materials except NiSi. The Debye temperature Θ_D of Ni₃Si, Ni₃₁Si₁₂, Ni₂Si (δ) and Ni₃Si₂ can be improved with the increase of pressure, so the thermal conductivity can also be improved. Finally, the pressure-dependent behavior of total density of states is analyzed to explore the effect of physical origin of pressure on the structural, elastic and thermodynamic properties of the seven Ni–Si binary compounds.

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1. Introduction

The nickel-silicide binary alloys play a significant role in modern electronic device technology due to their remarkable physical and chemical properties as well as low electric resistivity [1,2]. Nickel silicides are promising high temperature electronic materials, which are widely used as interconnectors, ohmic contact, and gate materials for integrated circuits [3,4]. Therefore, investigation on the nickel-silicide systems has drawn great academic field attention.

The nickel–silicon systems have been studied firstly by Guertler and Tammann [5] in 1906. Then, the nickel–silicon systems have been widely investigated by using the method of theoretical calculation and experiments. Mey [6], Nash and Nash [7], Lindholm and Sundman [8], and Du and Schuster [9] have studied the Ni–Si phase diagram by carrying out the thermodynamic analysis based on experimental data. Lee et al. [10], Foggiano

et al. [11], Tous et al. [12], Guihard et al. [13], Tomita et al. [14] and Beregovsky et al. [15] also have done some related research for better understanding the formation mechanism of the nickel–silicon systems. The nanometre-sized nickel-silicide systems, Ni₂Si, NiSi and NiSi₂, have been researched by energy-loss near-edge structure and first-principles calculation [16]. Abhaya et al. [17] have investigated bulk nickel silicides (NiSi and NiSi₂) by means of the experimental positron lifetime and depth-resolved positron beam measurements. Moreover, Connétable and Thomas [18] have studied the nickel-silicide ordered phases using first-principles calculation. The phases diagram, lattice parameters, formation energy and electronic structures have been calculated and analyzed. However, all these literatures are conducted at zero pressure or at room temperature. Up to now, the effects of pressure on the structural, elastic, thermodynamic and electronic properties of the nickel-silicide binary systems have not been reported yet.

In this paper, we calculated the structural, elastic, thermodynamic and electronic properties of Ni₃Si, Ni₃₁Si₁₂, Ni₂Si, Ni₃Si₂, NiSi and NiSi₂ by first-principles calculation within the frame work of density function theory (DFT). The pressure ranges from 0 to

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50 GPa with a step of 10 GPa. We firstly present a systematic study on pressure dependences of the nickel-silicide binary compounds. The main focus is elucidated the effects of pressure on mechanical properties and phase transformation.

2. Computational methods

All the calculations in this work are performed by first-principles plane-wave ultrasoft pseudopotential method [19,20] within density functional theory (DFT) [21], which is implemented in Cambridge Serial Total Energy Package (CASTEP) code [22]. The generalized gradient approximation (GGA) method with the Perdew–Wang91 (PW91) parameterization [23] is applied for determining the exchange-correction function. The cutoff energy of atomic wave functions, E_{cut} , is set to be 440 eV. Atomic electronic configurations in this study are Ni $3d^8 4s^2$ and Si $3s^2 3p^2$. Brillouin zone integrations are performed by using a Monkhorst–Pack k point mesh and a grid of $8 \times 8 \times 8$, $5 \times 5 \times 2$, $6 \times 8 \times 4$, $8 \times 8 \times 6$, $2 \times 3 \times 4$, $5 \times 8 \times 5$ and $6 \times 6 \times 6$ k point are set to be the sampling of β -Ni₃Si, γ -Ni₃₁Si₁₂, δ -Ni₂Si, θ -Ni₂Si, ε -Ni₃Si₂, ε -NiSi and α -NiSi₂, respectively. All the structural optimizations and properties are under hydrostatic pressure which ranges from 0 to 50 GPa with a step of 10 GPa. The separation of the reciprocal space is around 0.01 Å and the self-consistent field (SCF) tolerance is set as 5.0×10^{-7} eV/atom.

3. Results and discussion

3.1. Structural properties

The crystallographic data and calculated values of lattice parameters a_0 , b_0 and c_0 at 0 GPa of Ni–Si binary compounds are tabulated in Table 1 below together with available experimental and theoretical data for comparison. From Table 1, we can find that the calculated values of lattice parameters are in good agreement with experimental values [7,24–29] and previously calculated values [18], which indicate that the proposed computational methodology in this work is available and the simulative results are dependable.

In order to study the structural change with pressure, variations of unit cell volume at an applied hydrostatic pressure in the range

of 0–50 GPa with a step of 10 GPa are calculated. The pressure–volume (P – V) curves can reveal the volume changes under external pressure. And it can be obtained by fitting the third order Birch–Murnaghan equation [30], as follows:

$$P = \frac{3}{2}B \left(\left(\frac{V}{V_0} \right)^{-7/3} - \left(\frac{V}{V_0} \right)^{-5/3} \right) \left(1 + \frac{3}{4}(B' - 4) \left(\left(\frac{V}{V_0} \right)^{-2/3} - 1 \right) \right) \quad (1)$$

where V is the unit cell volume at external pressure, V_0 is the unit cell volume at zero pressure. B and B' are the bulk modulus and its pressure derivative, respectively. The B and B' are usually expressed as:

$$B = -V \left(\frac{\partial P}{\partial V} \right)_{P=0} \quad \text{and} \quad B' = - \left(\frac{\partial B}{\partial P} \right)_{P=0} \quad (2)$$

The bulk modulus B and its pressure derivative B' under zero pressure of Ni–Si binary compounds are also listed in Table 1. The calculated B of Ni₃Si, NiSi and NiSi₂ accord well with previously reported [31–34]. Again, the present computation parameters and conditions selected are reliable. Unfortunately, no other available experimental or theoretical calculation values can offer us for comparison. Fig. 1 shows the pressure–volume (P – V) curves of Ni–Si binary compounds. We can easily find that the ratio of V/V_0 decrease with the increase of pressure for all the seven compounds. Moreover, the ratio of change of NiSi₂ is maximum while Ni₃Si is minimum, elucidating that the lattice parameters of NiSi₂ are the most sensitive to external pressure change while the effect on Ni₃Si is the lowest. However, it is difficult to compress the crystal when pressure enhanced, because the distance reduces and the repulsive interaction strengthens between atoms. By fitting the pressure–volume curves, the following functions for Ni₃Si, Ni₃₁Si₁₂, Ni₂Si (δ), Ni₂Si (θ), Ni₃Si₂, NiSi and NiSi₂ can be acquired, respectively:

$$V/V_0 = 0.99910 - 0.00417 \times P + 2.49567 \times 10^{-5} \times P^2 \quad (3)$$

$$V/V_0 = 0.99958 - 0.00432 \times P + 2.54526 \times 10^{-5} \times P^2 \quad (4)$$

$$V/V_0 = 0.99977 - 0.00443 \times P + 2.52719 \times 10^{-5} \times P^2 \quad (5)$$

$$V/V_0 = 0.99947 - 0.00439 \times P + 2.61347 \times 10^{-5} \times P^2 \quad (6)$$

Table 1

The structure data of Ni–Si binary systems under zero pressure, including lattice parameters a_0 , b_0 and c_0 (Å), bulk modulus B (GPa) and its pressure derivative B' .

Phase	Composition (at.% Si)	Space group	Pearson symbol	Species	Lattice parameters (Å)			B (GPa)	B'
					a_0	b_0	c_0		
β Ni ₃ Si	25.0	$Pm\bar{3}m$	cP4	This work	3.517	3.517	3.517	214.07	4.72
				Experiment	3.506 [24]	3.506	3.506	201.7 [31]	–
				Calculation	3.516 [18]	3.516	3.516	213.9 [32]	–
γ Ni ₃₁ Si ₁₂	27.9	$P321$	hP43	This work	6.691	6.691	12.328	173.70	4.30
				Experiment	6.671 [7]	6.671	12.228	–	–
				Calculation	6.668 [18]	6.668	12.319	–	–
δ Ni ₂ Si	33.3	$Pnma$	oP12	This work	5.059	3.730	7.080	189.64	3.84
				Experiment	5.009 [25]	3.732	7.066	–	–
				Calculation	5.079 [18]	3.700	7.069	–	–
θ Ni ₂ Si	33.3	$P6_322$	hP6	This work	3.915	3.915	5.020	203.06	4.33
				Experiment	3.805 [26]	3.805	4.809	–	–
				Calculation	3.916 [18]	3.916	4.992	–	–
ε Ni ₃ Si ₂	40.0	$Cmc2_1$	oS80	This work	12.278	10.854	6.937	182.96	3.66
				Experiment	12.229 [27]	10.805	6.924	–	–
				Calculation	12.229 [18]	10.798	6.924	–	–
ε NiSi	50	$Pnma$	oP8	This work	5.151	3.372	5.656	155.26	3.25
				Experiment	5.177 [28]	3.325	5.616	–	–
				Calculation	5.165 [18]	3.378	5.621	168.0 [33]	–
α NiSi ₂	66.7	$Fm\bar{3}m$	cF12	This work	5.446	5.446	5.446	139.23	4.33
				Experiment	5.406 [29]	5.406	5.406	–	–
				Calculation	5.470 [18]	5.470	5.470	160.0 [34]	–

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